



# **Part II Electrical Properties of Materials**

Chap. 7 Electrical Conduction in Metals and Alloys
Chap. 8 Semiconductors
Chap. 9 Electrical Properties of Polymers, Ceramics, Dielectrics, and Amorphous Materials





(a)

Figure 9.1. (a) Polyethylene. (b) Polyvinylchloride. (The dashed enclosures mark the repeat unit. Polyethylene is frequently depicted as two  $CH_2$  repeat units for historical reasons.



Polymers consist of (macro)molecules which are long and chainlike. Several atoms combine and form a specific building block, called a monomer, and thousands of monomer combine to a polymer.

(b)

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# 9.1 Conducting Polymers and Organic Metals



The binding force between individual atoms within a chain

covalent(mostly) + ionic in nature

The binding force between macromolecules:

Figure 9.2. Simplified representation of a semicrystalline polymer (folded-chain Van der Walls type(weak) model).

Polyacetylene: the simplest conjugated organic polymer; the prototype of a conducting polymer



Figure 9.3. Theoretical isomers of polyacetylene (a) *cis*-transoidal isomer, (b) *trans*-transoidal isomer. Polyacetylene is synthesized as cis-(CH)<sub>x</sub> and is then isomerized into the *trans*-configuration by heating it at 150°C for a few minutes.



Trans-polyacetylene is obtained as silvery flexible film that has a conductivity comparable to that of silicon.

# 9.1 Conducting Polymers and Organic Metals







Figure 9.5. Calculated band structure of trans-(CH)<sub>x</sub> for different carbon–carbon bond lengths: (a) uniform (1.39 Å); (b) weakly alternating (C=C, 1.36 Å; C–C, 1.43 Å); and (c) strongly alternating (C=C, 1.34 Å; C–C, 1.54 Å). Note the band gaps at Y as bond alternation occurs. Reprinted with permission from P.M. Grant and I.P. Batra, *Solid State Comm.* **29**, 225 (1979).

# 9.1 Conducting Polymers and Organic Metals



Figure 9.6. Conductivity change of polyacetylene as a result of doping.

Conductivity increases by doping in polymerbased semiconductor



Figure 9.7. A broken symmetry in polyacetylene creates a *soliton*. (An *antisoliton* is the mirror image of a soliton.)

"Soliton" is a structural distortion in a conjugated polymer and is generated when a single bond meets another single bond.



# 9.2 Ionic Conduction



Figure 9.8. Schematic representation of a potential barrier, which an ion ( $\bullet$ ) has to overcome to exchange its site with a vacancy ( $\Box$ ). (a) Without an external electric field; (b) with an external electric field. d = distance between two adjacent, equivalent lattice sites; Q = activation energy.

The ionic conduction is caused by the movement of some charged ions which hop from lattice site to lattice site under the influence of electric field.

$$\sigma_{\rm ion} = N_{\rm ion} e \mu_{\rm ion}$$

.....

 $N_{\rm ion}$  depends on vacancy concentration in the crystal





## 9.2 Ionic Conduction



Figure 9.9. Schematic representation of  $\ln \sigma$  versus 1/T for Na<sup>+</sup> ions in sodium chloride. (Arrhenius plot.)



## 9.2 Ionic Conduction



Whenever vacant lattice site is created, an overall **charge neutrality** needs to be maintained.

Both a cation and anion are removed from a lattice (**Shottky defect**)

Formation of vacancy- interstitial pair (Frenkel defect)

Figure 9.10. Schematic representation of a  $\{100\}$  plane of an ionic crystal having the NaCl structure. The diffusion of a cation into a cation vacancy is shown. Also depicted is the creation of a cation vacancy when replacing a Na<sup>+</sup> ion with a Mg<sup>2+</sup> ion.



Metal oxide can be insulating, have metallic conduction properties, or be semiconducting : For understanding the mechanisms involved in metal oxides, their electronic configuration in the orbital (or band structure) should be considered. (Appendix 3. p.409)

- 1. TiO<sub>2</sub> (O :1 $s^2$  2 $s^2$ 2 $p^4$ , Ti: [Ar]3 $d^2$ 4 $s^2$ )
- Noble gas configuration, insulator
- insulator with wide band gap
- 2. TiO (O :1 $s^2$  2 $s^2$ 2 $p^4$ , Ti: [Ar]3 $d^2$ 4 $s^2$ ) : partially filled 3 $d^2$
- Metallic
- 3. ZnO (O :1 $s^2$  2 $s^2$ 2 $p^4$ , Zn: [Ar]3 $d^{10}4s^2$ )
- Insulator for stoichiometric
- *n*-type semiconductor for non-stoichiometric





4. SnO<sub>2</sub> (some times doped with  $In_2O_3$ )

- Transparent in the visible region and which is a reasonable conductor in the 1  $\Omega^{\text{-1}}\text{cm}^{\text{-1}}$  range

- Used in optoelectronics to provide electrical contacts without blocking the light from reaching a device: indium-tin-oxide (ITO)

### 5. NiO (O :1 $s^2 2s^2 2p^4$ , Ni: [Ar]3 $p^8 4s^2$ )

- -Insulator for stoichiometric
- *p*-type semiconductor for nonstoichiometric



# 9.4 Amorphous Materials (Metallic Glasses)



Figure 9.11. Two-dimensional schematic representation of a dense random packing of hard spheres (Bernal model).

Atomic structure of amorphous metals and alloy

Dense random packing of hard spheres model **(Bernal model)** 

In transition metal-metalloid compounds (such as Ni-P), small metalloid atoms are thought to occupy the holes which occur as a consequence of this packing (**Bernal-Polk model**).



Figure 9.12. Defects in crystalline and amorphous silicon. (a) Monovacancy in a crystalline semiconductor; (b) one and (c) two dangling bonds in a continuous random network of an amorphous semiconductor. (Note the deviations in the interatomic distances and bond angles.)

### **Defect in crystalline**

### **Defect in amorphous**

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# 9.4 Amorphous Materials (Metallic Glasses)



Figure 9.13. Schematic representation of the molecular orbital energy level diagram and the density of states curves for Zr–Cu clusters. The calculated density of states curves agree reasonably well with photoemission experiments.





### **Cluster Model**

A series of clusters were assumed which exhibit the symmetry of closed-packed lattice fcc (as Cu) and hcp (as for Zr).

Stronger binding forces between the atoms in covalently bound materials, the valence electrons are tightly bound, or localized  $\rightarrow$ the density of state for localized state extends into the "band gap"

# Electrical conductivity for amorphous semiconductors

$$\sigma_{\rm A} = N_{\rm A} e \mu_{\rm A}.$$

 $\sigma_{\rm A} = \sigma_0 \exp\left[-\left(\frac{Q_{\rm A}(T)}{k_{\rm B}T}\right)\right].$ 

# 9.4 Amorphous Materials (Metallic Glasses)

### 9.4.1 Xerography



Figure 9.15. Schematic representation of the electrophotography process. The individual steps are explained in the text.





# **9.5 Dielectric Properties**



### Capacitance, C

*the ability to store an electric charge, q per unit applied voltage, V.* 



$$C = \varepsilon_r \varepsilon_0 \frac{A}{L} \qquad \varepsilon_r = \frac{C}{C_{\text{vac}}}$$

 $\varepsilon_r$ : dielectric constant (unitless), or *relative permittivity*,

 $\epsilon_0$  : permittivity of empty space , 8.85  $\, \mathrm{x} \,$  10^{-12} F/m

Figure 9.16. Two metal plates, separated by a distance, L, can store electric energy after having been charged momentarily by a battery.



# **9.5 Dielectric Properties**

Table 9.1. DC dielectric constants of some materials	
6000 4000 700 170	Ferroelectric
$ \begin{array}{c} 81.1\\ 20\\ 11.8\\ 10.9\\ 8.5\\ 6.9\\ 6.0\\ 4.0\\ 4.0\\ 4.0\\ 4.0\\ 3.5\\ 3.0\\ 2.8\\ 2.3\\ 2.0\\ 1.000576\end{array} $	Dielectric
	$\begin{array}{c} \text{of some materials} \\ \hline 6000 \\ 4000 \\ 700 \\ 170 \\ \hline \\ 81.1 \\ 20 \\ 11.8 \\ 10.9 \\ 8.5 \\ 6.9 \\ 6.0 \\ 4.0 \\ 4.0 \\ 4.0 \\ 4.0 \\ 4.0 \\ 4.0 \\ 3.5 \\ 3.0 \\ 2.8 \\ 2.3 \\ 2.0 \\ 1.000576 \\ \end{array}$

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Figure 9.17. An atom is represented by a positively charged core and a surrounding, negatively charged, electron cloud (a) in equilibrium and (b) in an external electric field. (c) Schematic representation of an electric dipole as, for example, created by separation of the negative and positive charges by an electric field, as seen in (b).



 $D = \mathcal{E}_0 E + P$ 

#### Electric dipole moment

$$D = \varepsilon_r \varepsilon_0 E = \frac{q}{A}$$

**Polarization** : the process of dipole formation (or alignment of already existing dipoles) under the influence of an external electric field that has an electric field strength, *E* 

# 9.5 Dielectric Properties



Figure 9.18. Schematic representation of two capacitor plates between which a dielectric material is inserted. (a) Induction of electric dipoles of opposite charge. (b) Weakening of the electric field *within* the dielectric material [Eq. (9.13)]. (c) The direction of the polarization vector is from the negative induced charge to the positive induced charge see Fig. 9.17(b). (d) The dielectric displacement, D, within the dielectric material is the sum of  $\varepsilon_0 \mathscr{E}$  and P [Eq. (9.15)].



### due to polarization



Figure 9.19. Schematic representation of the polarization as a function of excitation frequency for different polarization mechanisms.

How quickly do the dipoles to reorient or to align under a rapidly changing electric filed (in alternating circuit)

Polarization mechanisms which can respond equally quick to an alternating electric field

### \*\*\*\*\*



### 9.6 Ferroelectricity, Piezoelectricity, and Electrostriction

### **Ferroelectric materials**



A spontaneous polarization without the presence of an external electric field :suitable for the manufacturing of small sized, highly efficient capacitors



### Hysteresis loop

 $P_S$ : saturation polarization  $P_r$ : remanent polarization  $E_C$ : coercive field

Figure 9.20. Schematic representation of a hysteresis loop for a *ferroelectric* material in an electric field. Compare to Figure 15.6.

## 9.6 Ferroelectricity, Piezoelectricity, and Electrostriction



Mechanism for spontaneous polarization



#### Tetragonal BaTiO<sub>3</sub>:

A large number of such dipole moment line up in a clusters (also called *domains*) In the virgin state, the polarization directions of the individual domains are randomly oriented: no net polarization

An external field orients the dipoles of favorably oriented domains parallel to *E*: those domains in which the dipoles are already nearly parallel to *E* at the expense of unfavorably oriented domains

Figure 9.21. Tetragonal crystal structure of barium titanate at room temperature. Note the upward displacement of the Ti<sup>4+</sup> ion in the center compared to the downward displacement of all surrounding O<sup>2-</sup> ions. a = 0.398 nm; c = 0.403 nm.



Spontaneous alignments of electric dipoles within a domain and random alignment of the dipole moments of several domains in a ferroelectric material

Figure 9.22. Schematic representation of spontaneous alignments of electric dipoles within a domain and random alignment of the dipole moments of several domains in a ferroelectric material such as  $BaTiO_3$ . Compare to Figure 15.9.