

**2019 Spring**

# **“Phase Equilibria *in* Materials”**

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**Eun Soo Park**

**Office: 33-313**

**Telephone: 880-7221**

**Email: [espark@snu.ac.kr](mailto:espark@snu.ac.kr)**

**Office hours: by an appointment**

- How is the reaction in three phase region among liquid,  $\alpha$  and  $\beta$ ?

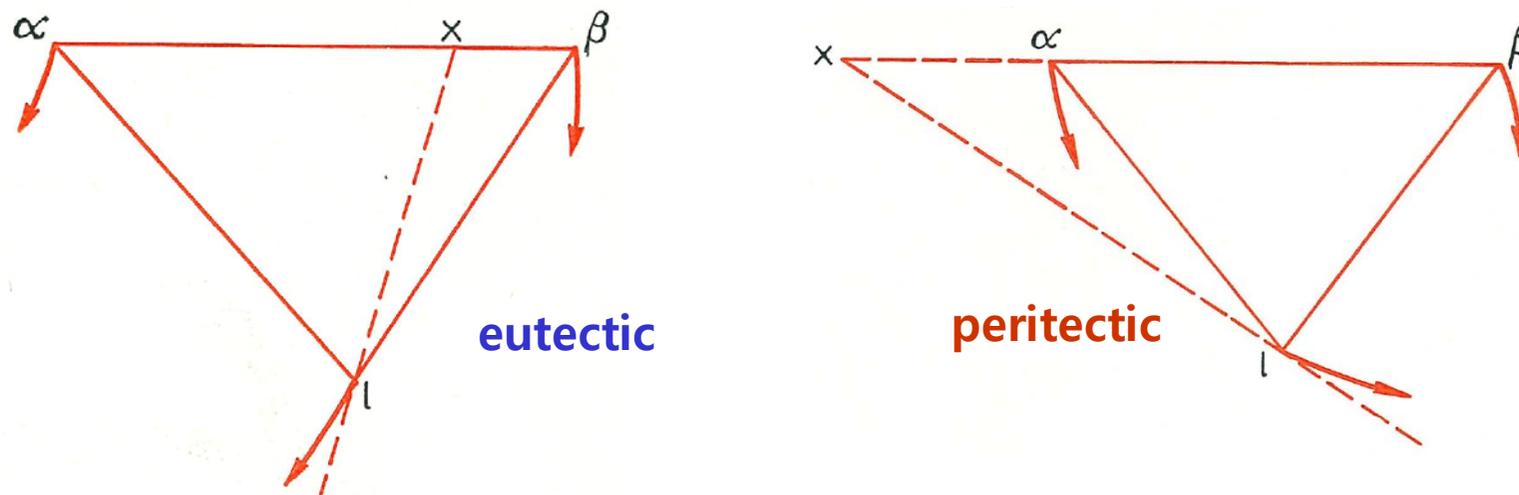


Fig. 149. Criteria for distinguishing eutectic and peritectic reactions in ternary three-phase equilibrium.

The **tangent to the liquid curve at a particular temperature** is extrapolated to meet the tie line connecting the  $\alpha$  and  $\beta$  phases.

- 1) If the extrapolated line intersected the  $\alpha\beta$  tie line, the equilibrium was considered to be **eutectic**
- 2) If it met the  $\alpha\beta$  tie line only when the latter was extrapolated, the equilibrium was considered to be **peritectic**.

Similarly, a **eutectoid reaction** could be distinguished from a **peritectoid** and a **monotectic** from a **syntectic**.



A small change in temperature,  $dT$ , causes a small change in the composition and amounts of each phase, but not of the alloy itself,

$$X_A, X_B, X_C = \text{constant}, \Delta X_A = 0, \text{ and } \Delta m_\alpha + \Delta m_\beta + \Delta m_l = 0$$

$$\Delta X_A = m_\alpha \cdot \Delta X_A^\alpha + m_\beta \cdot \Delta X_A^\beta + m_l \cdot \Delta X_A^l + X_A^\alpha \cdot \Delta m_\alpha + X_A^\beta \cdot \Delta m_\beta + X_A^l \cdot \Delta m_l = 0$$

1) Assumption,  $X_A^\beta = X_A^l > X_A^\alpha$   
 (New coordinate system)

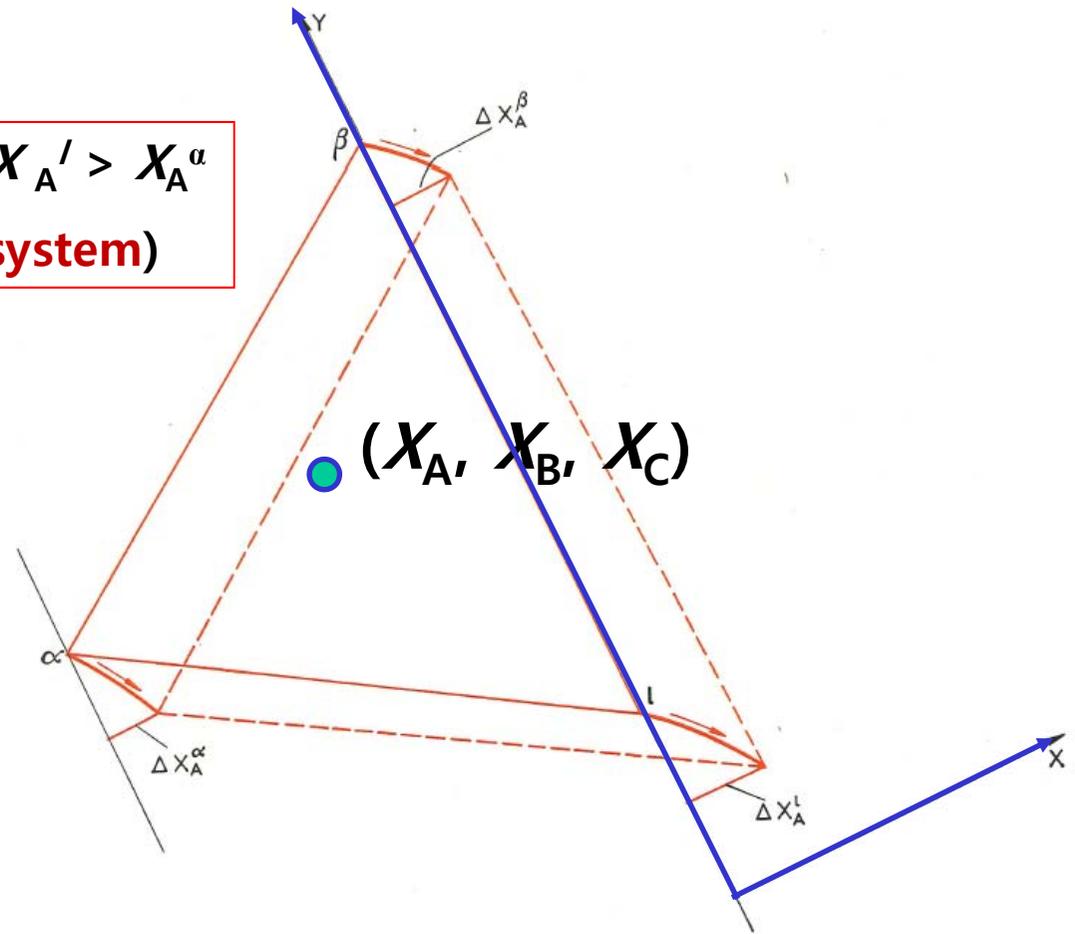


Fig. 150. Illustration of Hillert's criterion for distinguishing eutectic and peritectic reaction in ternary three-phase equilibrium; ———, equilibrium at  $T$ ; - - - - -, equilibrium at  $T-dT$ .

To simplify the calculation,

**Assumption,  $X_A^\beta = X_A^l > X_A^\alpha$  (New coordinate system)**

$$\Delta m_\beta + \Delta m_l = -\Delta m_\alpha$$

$$\Delta X_A = m_\alpha \cdot \Delta X_A^\alpha + m_\beta \cdot \Delta X_A^\beta + m_l \cdot \Delta X_A^l + X_A^\alpha \cdot \Delta m_\alpha + X_A^\beta \cdot \Delta m_\beta + X_A^l \cdot \Delta m_l = 0$$

$$-X_A^\alpha \Delta m_\alpha - X_A^\beta \Delta m_\beta - X_A^l \Delta m_l = m_\alpha \Delta X_A^\alpha + m_\beta \Delta X_A^\beta + m_l \Delta X_A^l$$

$$\Delta m_\alpha (X_A^\beta - X_A^\alpha) = m_\alpha \Delta X_A^\alpha + m_\beta \Delta X_A^\beta + m_l \Delta X_A^l$$

Sign	Assumption	Sign
$\Delta m_\alpha (X_A^\beta - X_A^\alpha)$	$X_A^\beta = X_A^l > X_A^\alpha$	$m_\alpha \Delta X_A^\alpha + m_\beta \Delta X_A^\beta + m_l \Delta X_A^l$
$\Delta m_\beta (X_A^\alpha - X_A^\beta)$	$X_A^\alpha = X_A^l > X_A^\beta$	$m_\alpha \Delta X_A^\alpha + m_\beta \Delta X_A^\beta + m_l \Delta X_A^l$
$\Delta m_l (X_A^\alpha - X_A^l)$	$X_A^\alpha = X_A^\beta > X_A^l$	$m_\alpha \Delta X_A^\alpha + m_\beta \Delta X_A^\beta + m_l \Delta X_A^l$

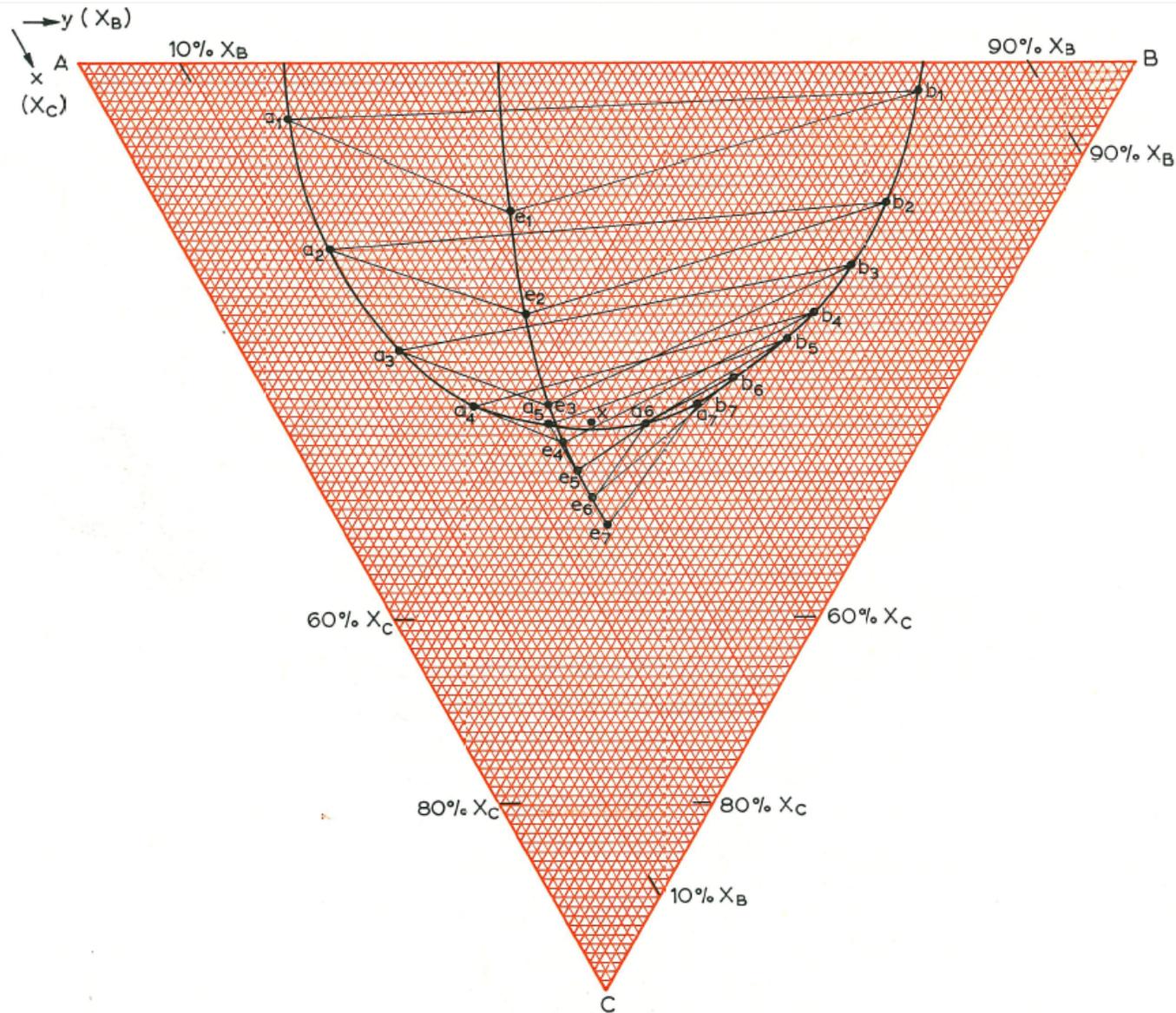
here,  $\Delta m_\alpha$  : change of  $\alpha$  phase fraction with  $\Delta T$

$\Delta m_\alpha$	$\Delta m_\beta$	$\Delta m_l$		
+	+	-	$l \rightarrow \alpha + \beta$	eutectic
+	-	-	$l + \beta \rightarrow \alpha$	peritectic
-	+	-	$l + \alpha \rightarrow \beta$	peritectic

Hillert's criterion indicates that the relative amounts of the  $\alpha$ ,  $\beta$  and liquid phases (the average alloy composition) are of importance in determining the type of reaction.

### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

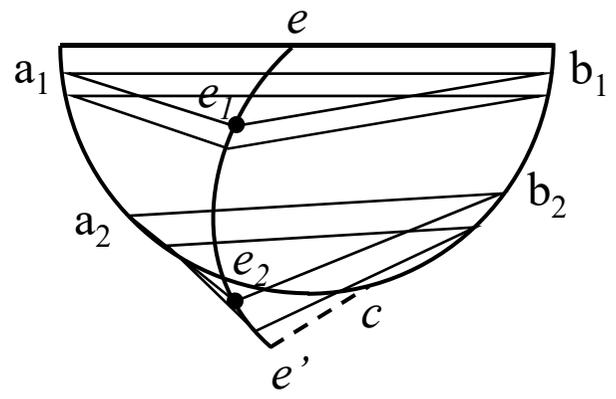
- three phase regions  $a_1e_1b_1, a_2e_2b_2, \dots, a_7e_7(b_7)$  projected on the concentration triangle.



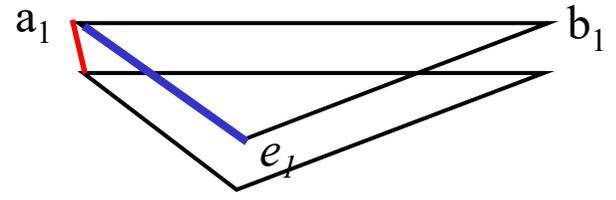
To determine whether the reaction is always a monovariant eutectic type, irrespective of alloy composition within the three-phase region, we apply Hillert's criterion to each pair of isotherms.

### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

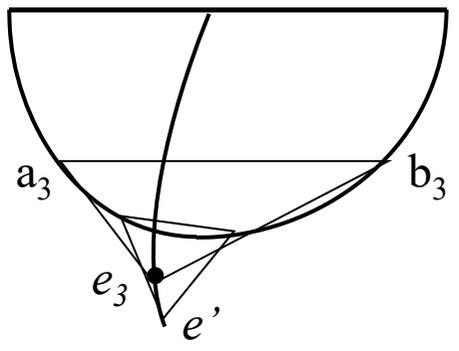
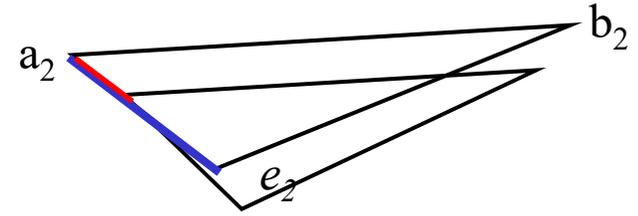
- Relative position of vertex in tie triangle with  $\Delta T$



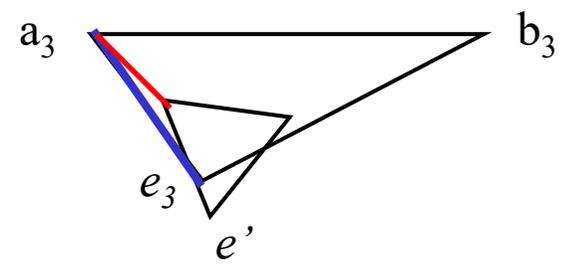
① Slope of tangent line at  $a_1 >$  slope of line  $a_1e_1$



② Slope of tangent line at  $a_2 =$  slope of line  $a_2e_2$



③ Slope of tangent line at  $a_3 <$  slope of line  $a_3e_3$



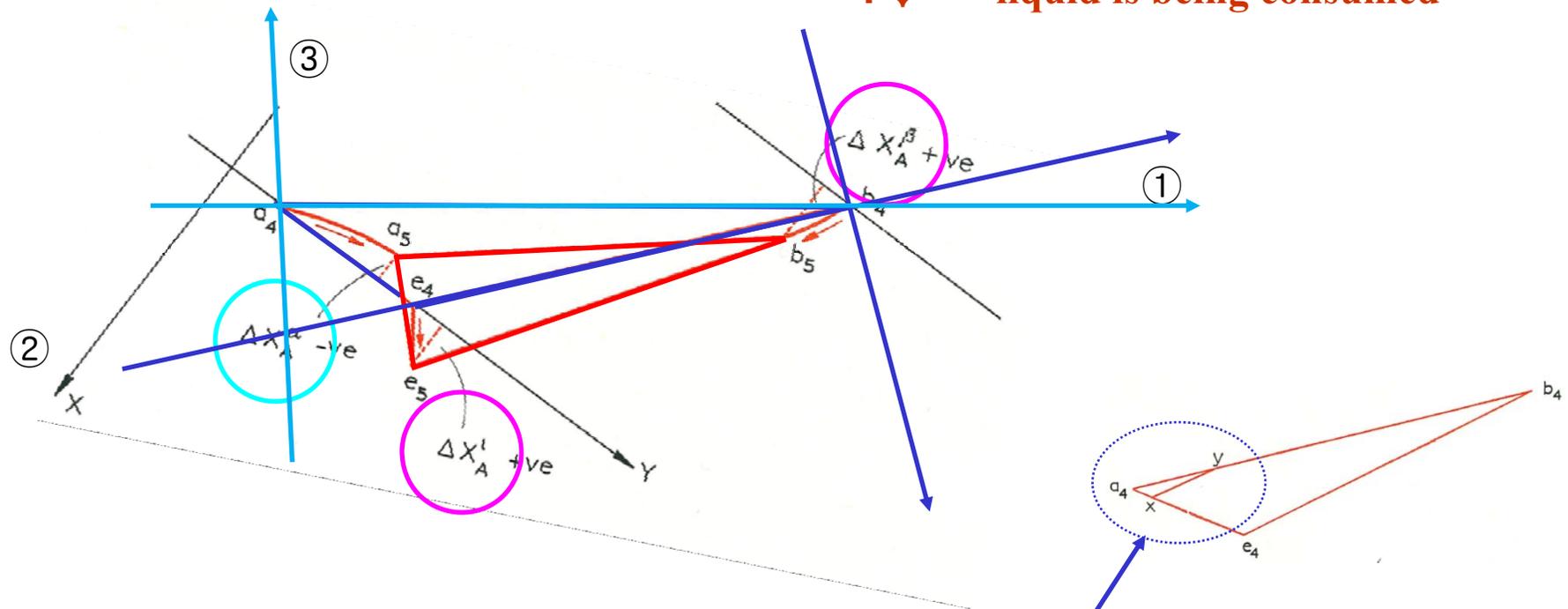
Consider tie triangle  $a_4e_4b_4$ , and  $a_5e_5b_5$

①  $X_A^\beta = X_A^l > X_A^\alpha$

③  $X_A^\alpha = X_A^\beta > X_A^l$

$$m_\alpha \Delta X_A^\alpha (+) + m_\beta \Delta X_A^\beta (+) + m_l \Delta X_A^l (+) \rightarrow \Delta m_\alpha (+) \Delta m_l (-) \rightarrow m_\alpha \Delta X_A^\alpha (-) + m_\beta \Delta X_A^\beta (-) + m_l \Delta X_A^l (-)$$

**T ↓ → liquid is being consumed**



②  $X_A^\alpha = X_A^l > X_A^\beta \quad \Delta m_\beta \rightarrow m_\alpha \Delta X_A^\alpha (-) + m_\beta \Delta X_A^\beta (+) + m_l \Delta X_A^l (+)$

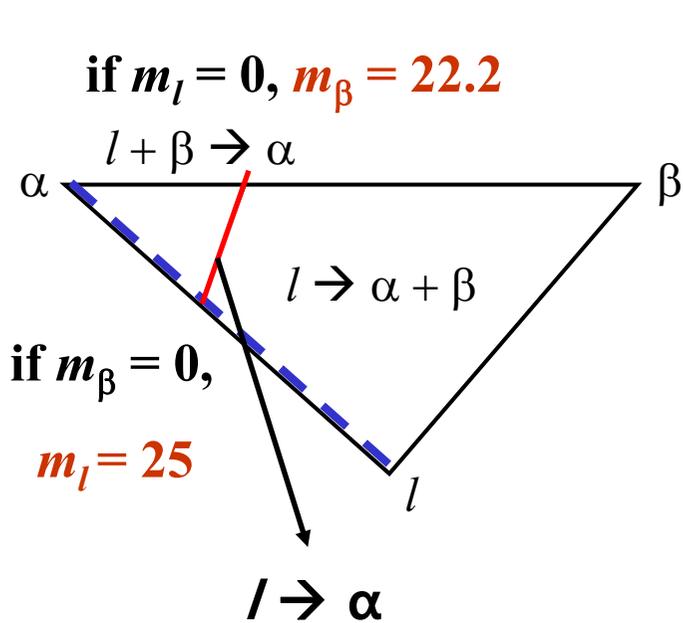
❖  $\Delta m_\alpha (+), \Delta m_l (-)$ ; if  $m_\alpha$  is very larger than  $m_\beta$  and  $m_l \rightarrow \Delta m_\beta (-) \rightarrow (l + \beta \rightarrow \alpha)$

**if  $m_\alpha$  is much smaller than  $m_\beta$  and  $m_l \rightarrow \Delta m_\beta (+) \rightarrow (l \rightarrow \alpha + \beta)$**  8

### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- How to decide the boundary btw eutectic & peritectic?

Reactions in the tie triangle  $a_4e_4b_4$ , along boundary,  $\beta$  plays no role  $\rightarrow l = \alpha$



$$X_A^\alpha = X_A^l > X_A^\beta$$

$$\Delta m_\beta (X_A^\alpha - X_A^\beta) = m_\alpha \Delta X_A^\alpha + m_\beta \Delta X_A^\beta + m_l \Delta X_A^l$$

$$(\Delta X_A^\alpha = -1, \Delta X_A^\beta = 3.5, \Delta X_A^l = 3) \text{ (next page)}$$

$$-m_\alpha + 3.5m_\beta + 3m_l = 0 \quad (m_\alpha + m_\beta + m_l = 100)$$

$$-100 + 4.5m_\beta + 4m_l = 0$$

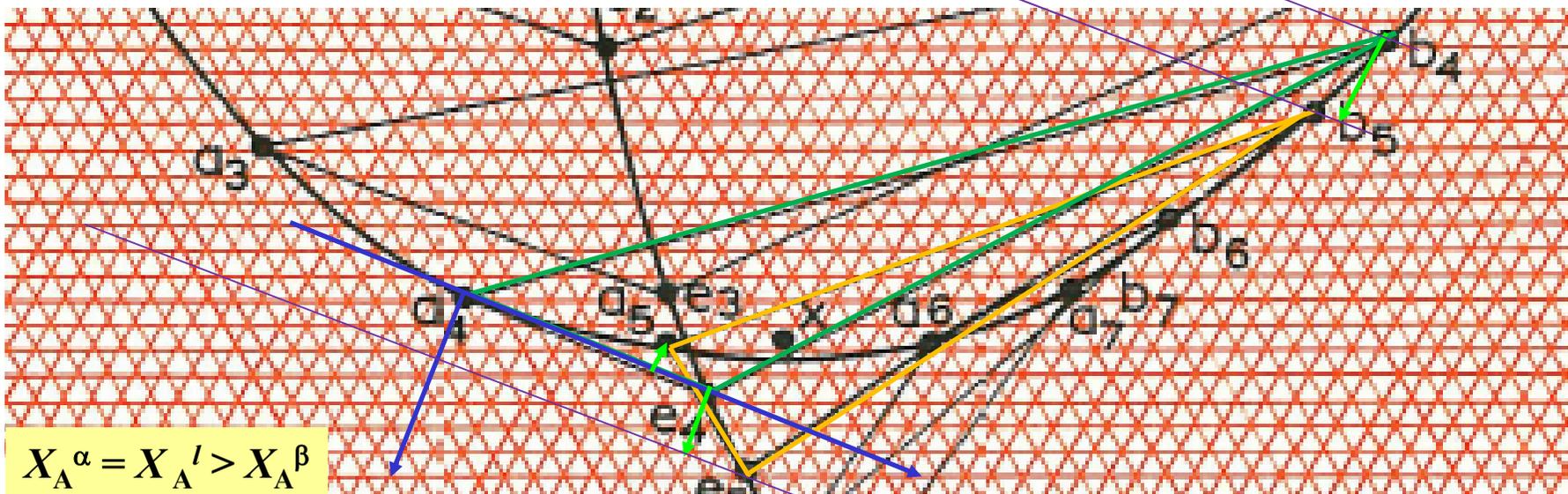
$$\text{if } m_\beta = 0, m_l = 25$$

$$\text{if } m_l = 0, m_\beta = 22.2$$

Initially, peritectic region confined the  $\alpha$  corner.

Consideration of three-phase triangles at lower temperatures will indicate that the peritectic region sweeps round from the  $\alpha$  corner towards the  $\beta$  and liquid corners.

- three phase regions  $a_1e_1b_1, a_2e_2b_2, \dots, a_7e_7(b_7)$  projected on the concentration triangle.



The boundary line can be determined by measuring  $\Delta X_A^\alpha, \Delta X_A^\beta,$  and  $\Delta X_A^l$ .

In Fig. 151,

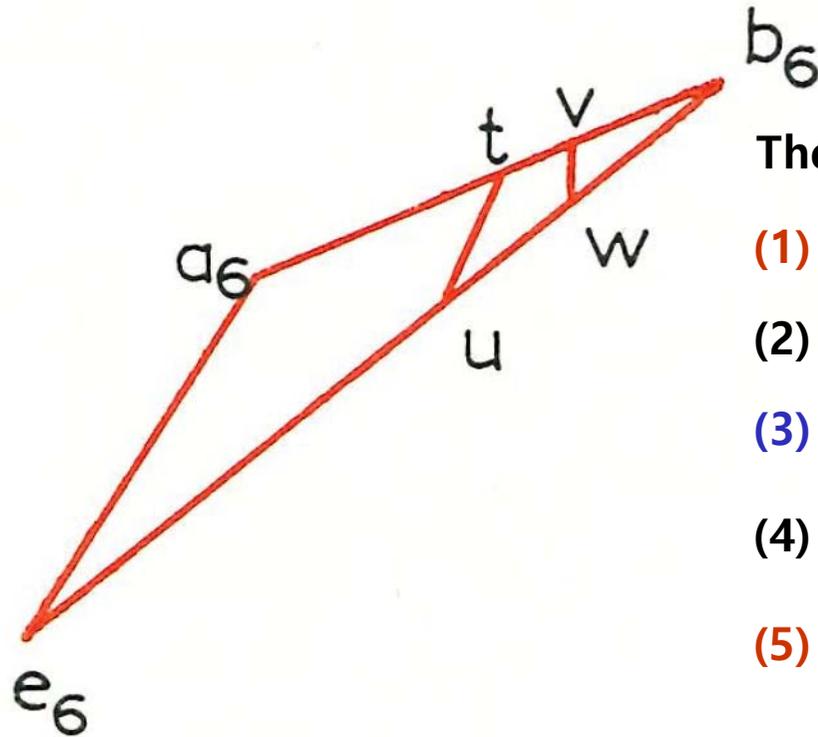
$\Delta X_A^\alpha = -1, \Delta X_A^\beta = -3.5$  and  $\Delta X_A^l = 3$  units.

Figure 151

Table 2

	$X_B,$	$X_C$		$X_B,$	$X_C$		$X_B,$	$X_C$
$e_1$	33,	16	$a_1$	17,	6	$b_1$	78,	3
$e_2$	29,	27	$a_2$	14,	20	$b_2$	69,	15
$e_3$	26,	37	$a_3$	15,	31	$b_3$	62,	22
$e_4$	25.3,	41	$a_4$	19,	37	$b_4$	56,	27
$e_5$	25,	44	$a_5$	25,	39	$b_5$	52,	30
$e_6$	25,	47	$a_6$	34,	39	$b_6$	45,	34
$e_7$	25,	50	$a_7(b_7)$	40,	37			

Monovariant  $\beta$  curve coincides with the  $l\alpha$  tie line between isotherms  $a_5e_5b_5$  and  $a_6e_6b_6$   
 → **Second peritectic reaction** area appears at the  $\beta$  corner of the three-phase triangle



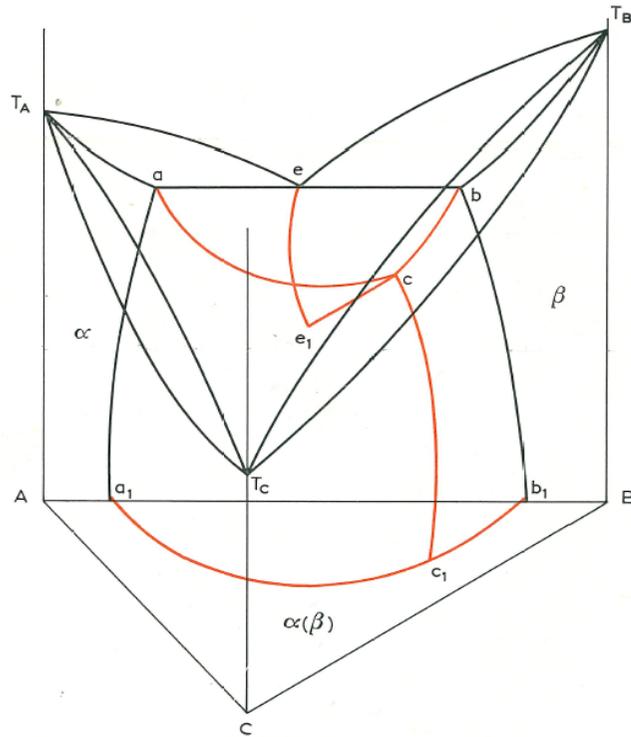
The tie triangle  $a_6e_6b_6$ , reaction equilibria

- (1) **Peritectic  $l + \beta \leftrightarrow \alpha$  in area  $a_6e_6ut$**
- (2) **Two-phase  $l \leftrightarrow \alpha$  along line  $tu$**
- (3) **eutectic  $l \leftrightarrow \alpha + \beta$  in area  $tuvw$**
- (4) **two-phase  $l \leftrightarrow \beta$  along line  $vw$**
- (5) **Peritectic  $l + \alpha \leftrightarrow \beta$  in area  $b_6vw$**

To summarise, the three-phase reaction is **initially eutectic** for all alloys until the temperature of the three phase triangle  $a_4e_4b_4$ , is reached. From that temperature until the end of the three-phase reaction at the tie line  $e_7a_7(b_7)$ , the reaction type is dependent on **the alloy composition within the sequence of the three-phase triangles.**

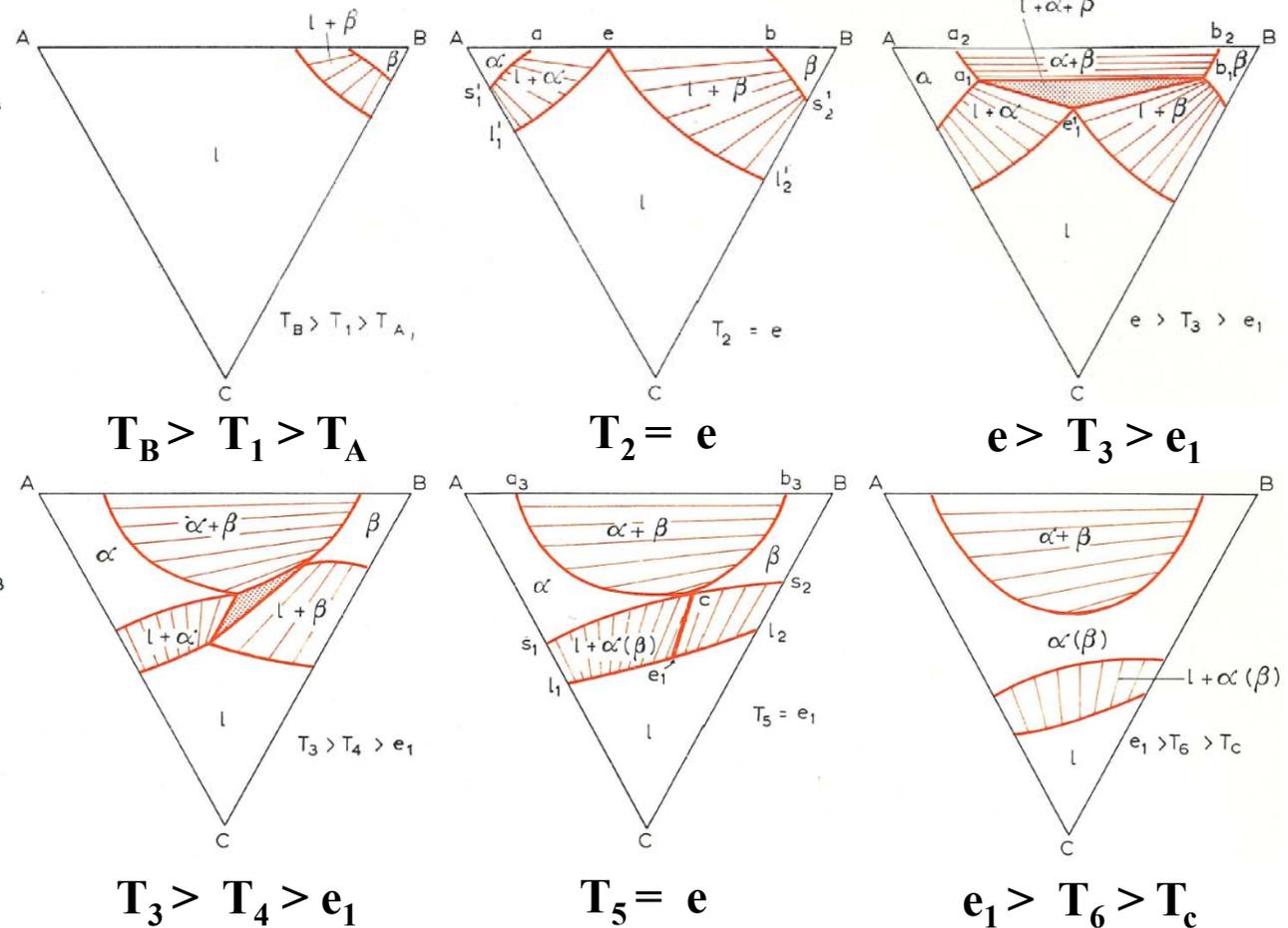
# 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- Isothermal section



A-B: eutectic

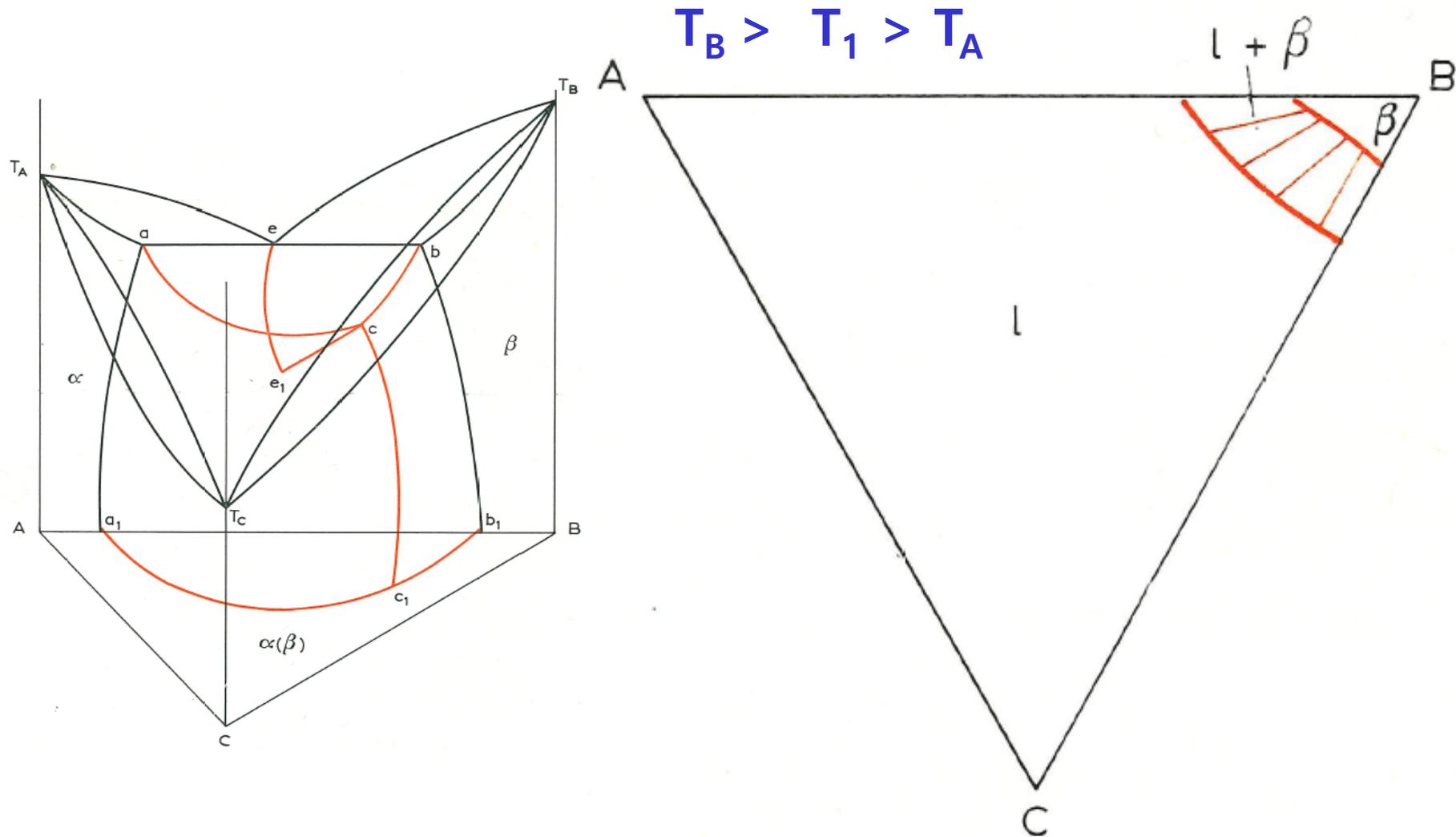
A-C/B-C: continuous series  
of solid soln



cf) Movie

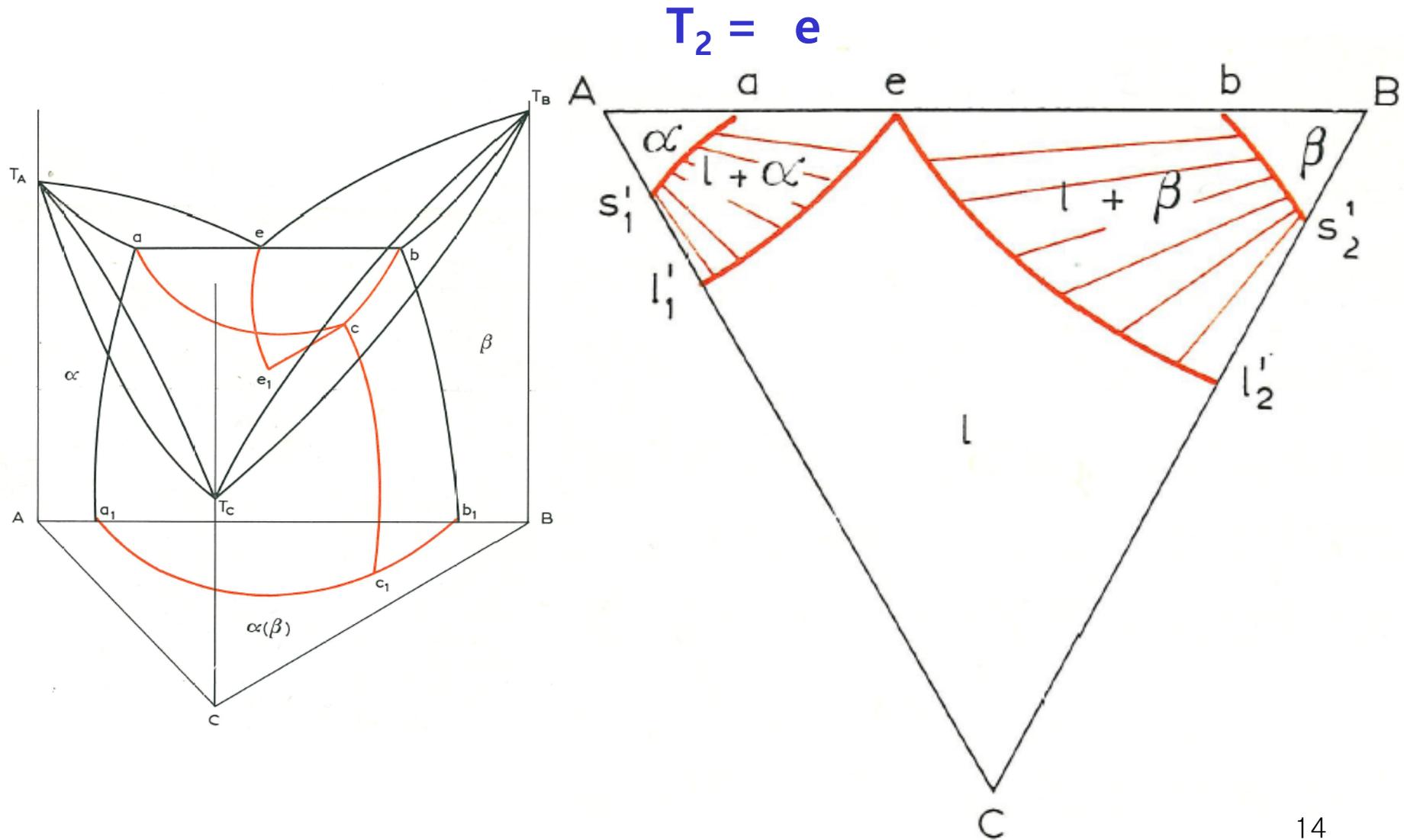
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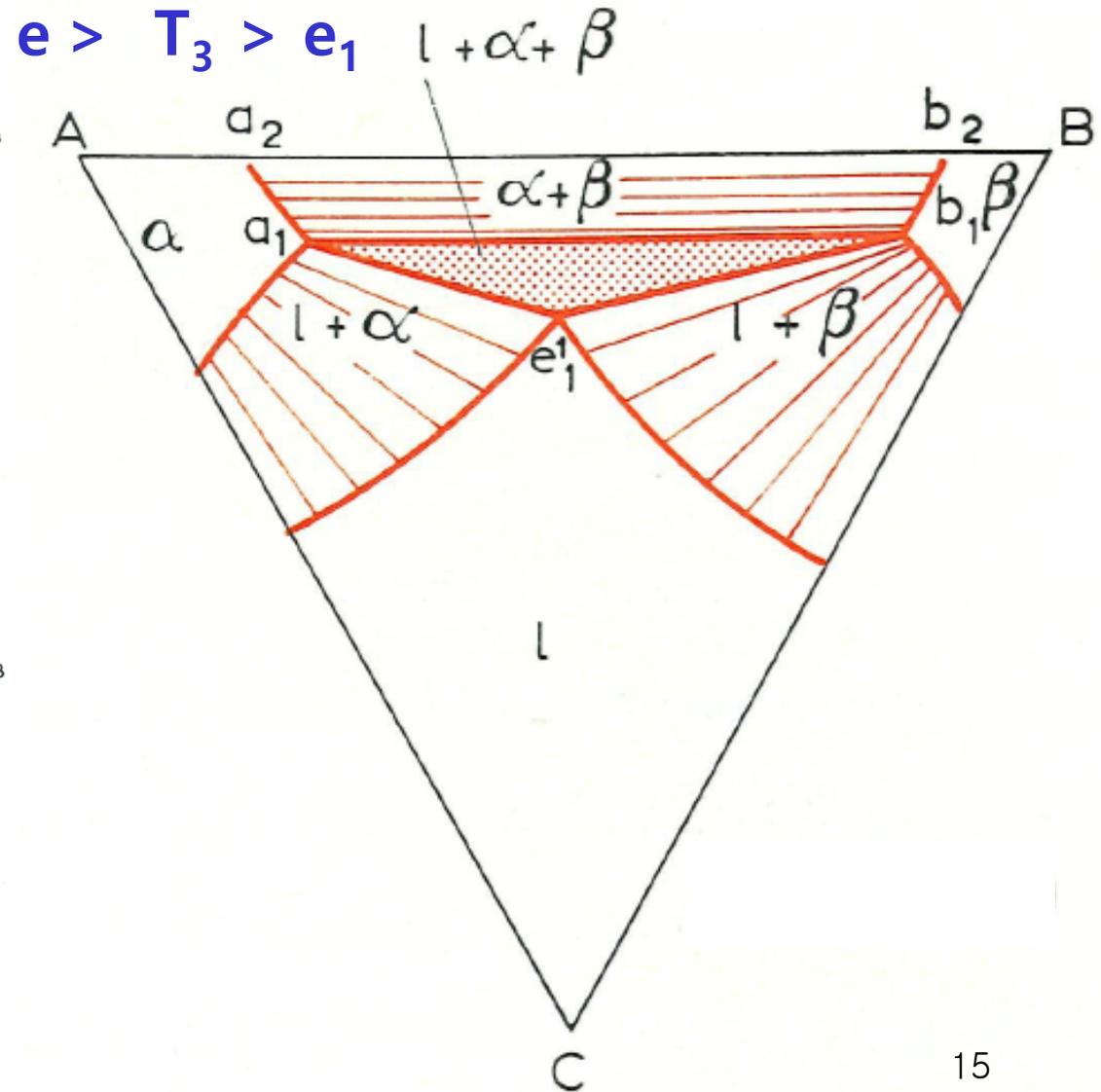
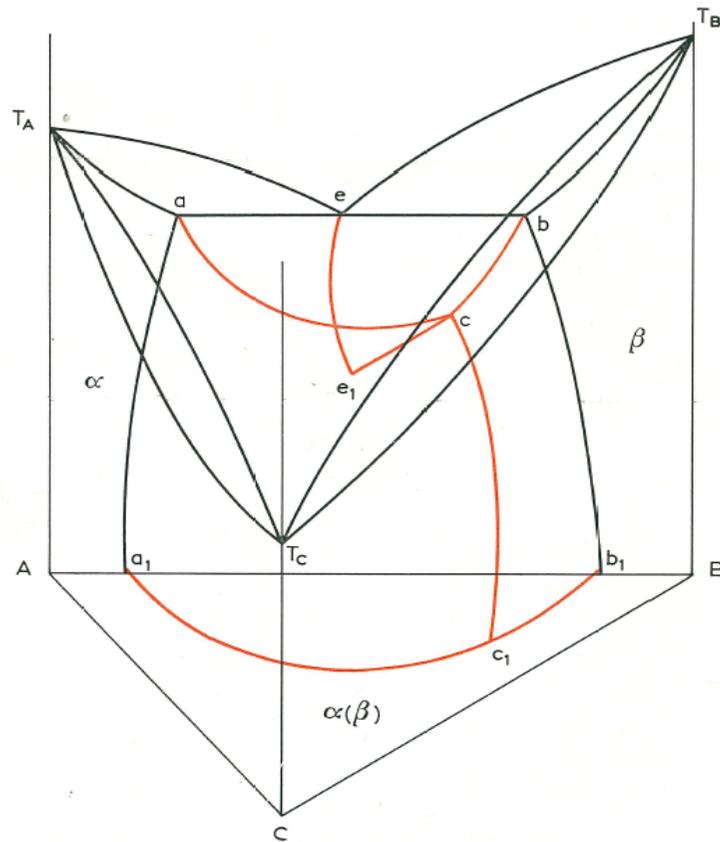
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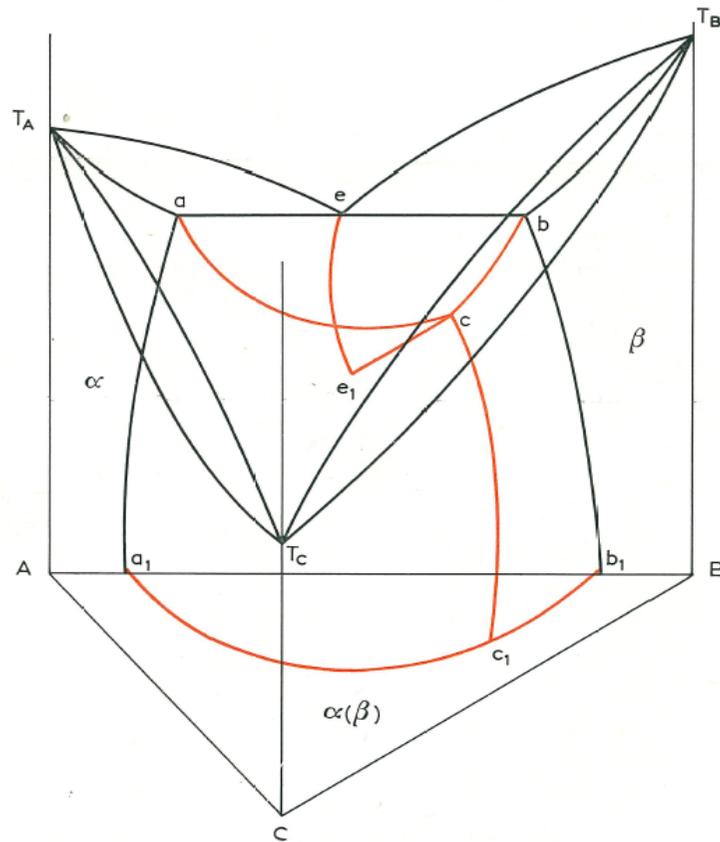
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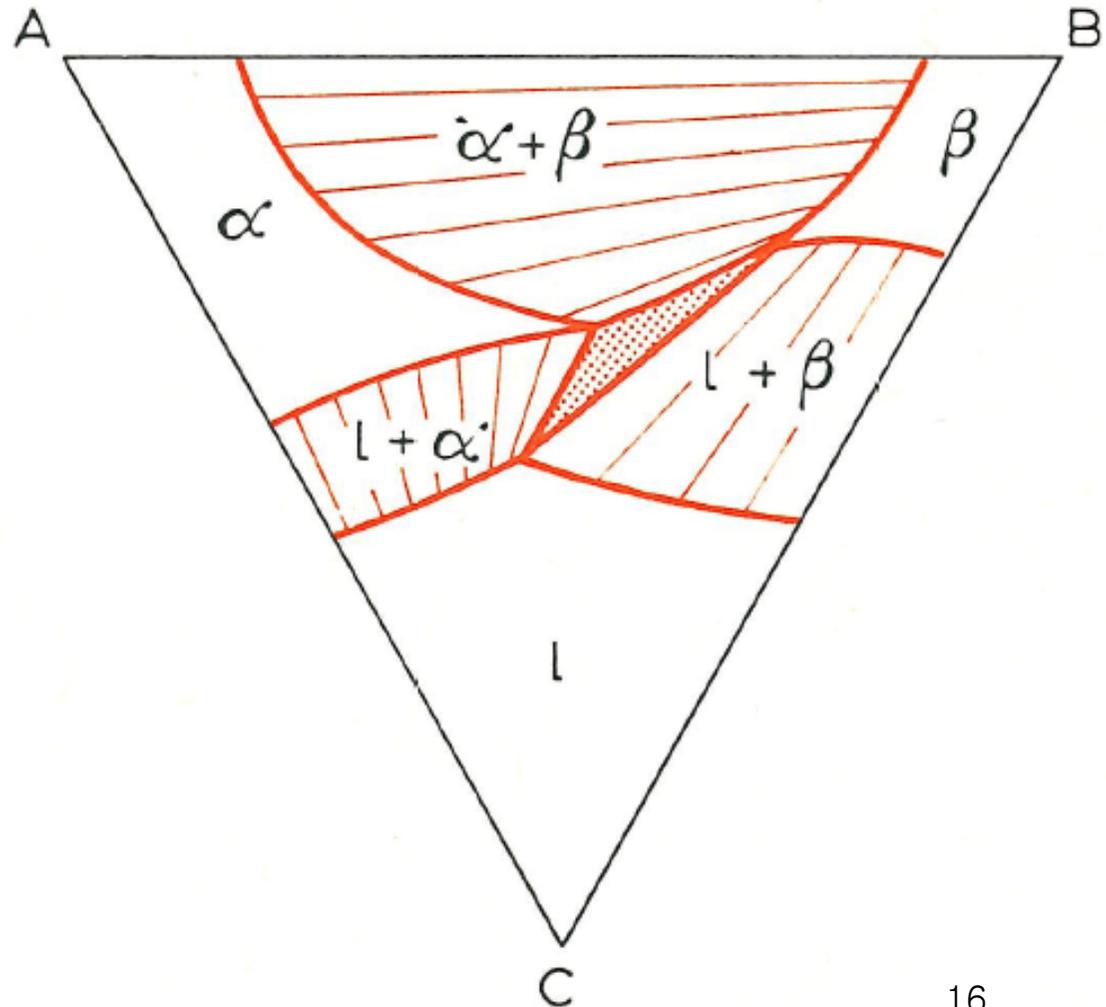


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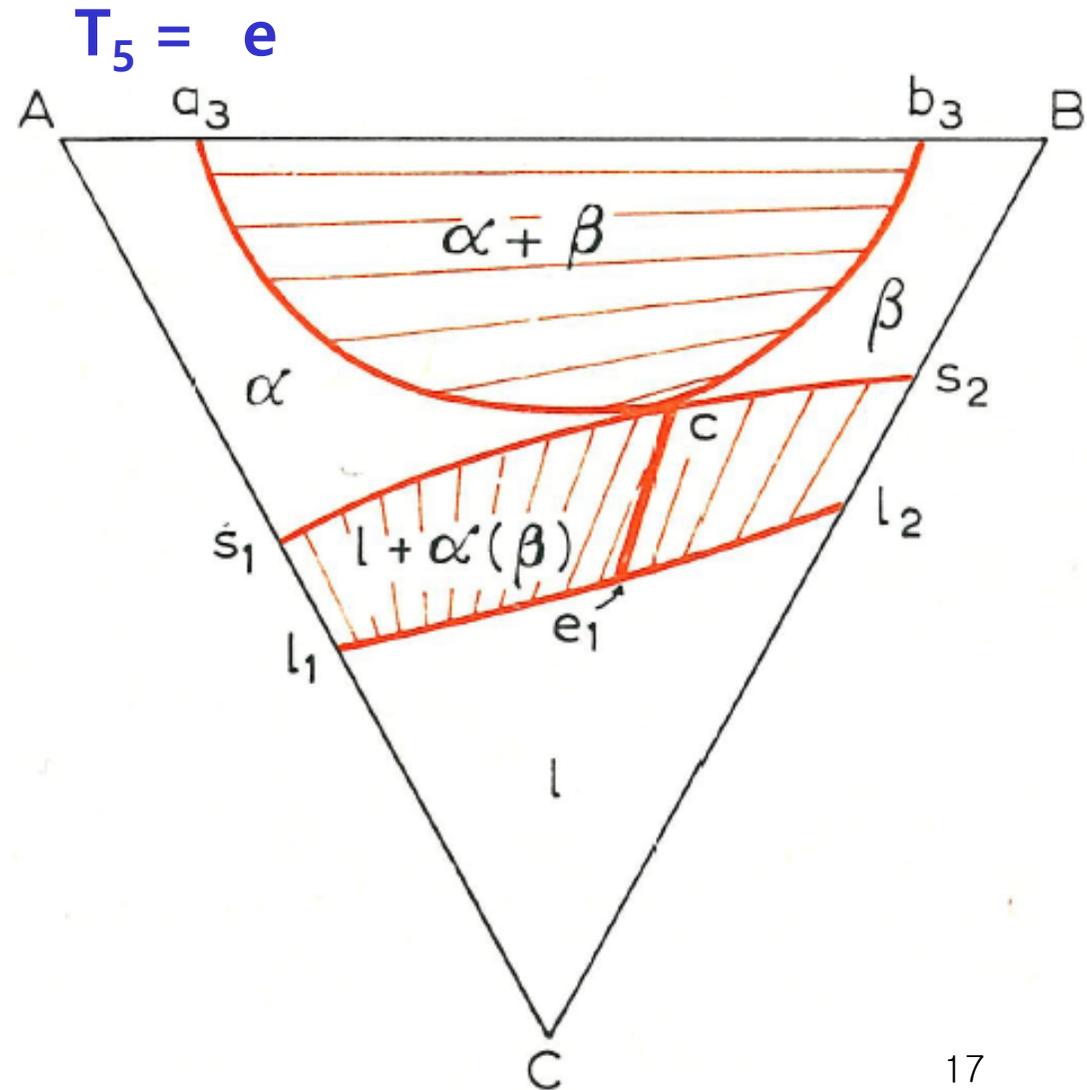
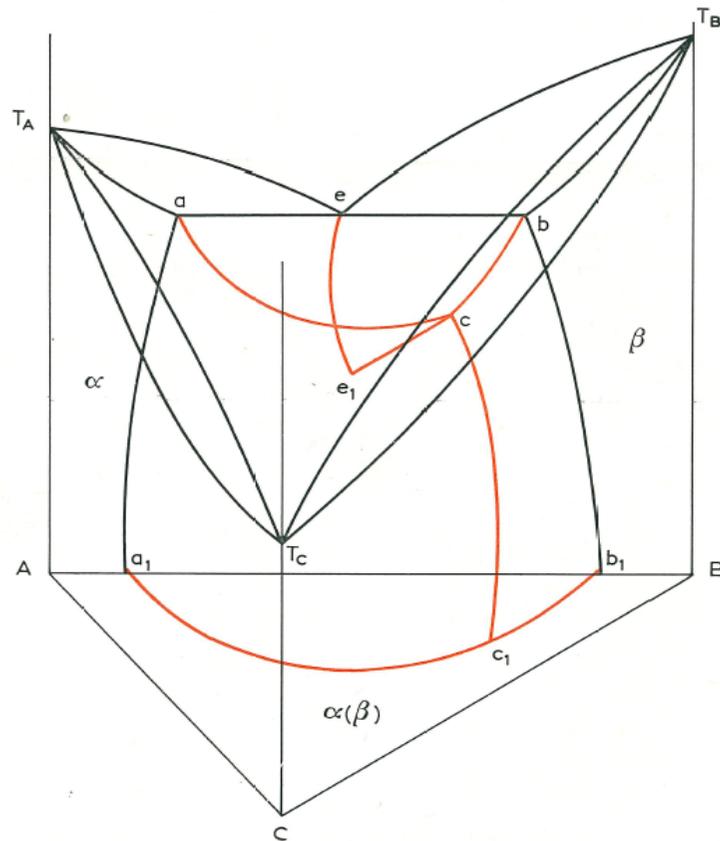


$$T_3 > T_4 > e_1$$



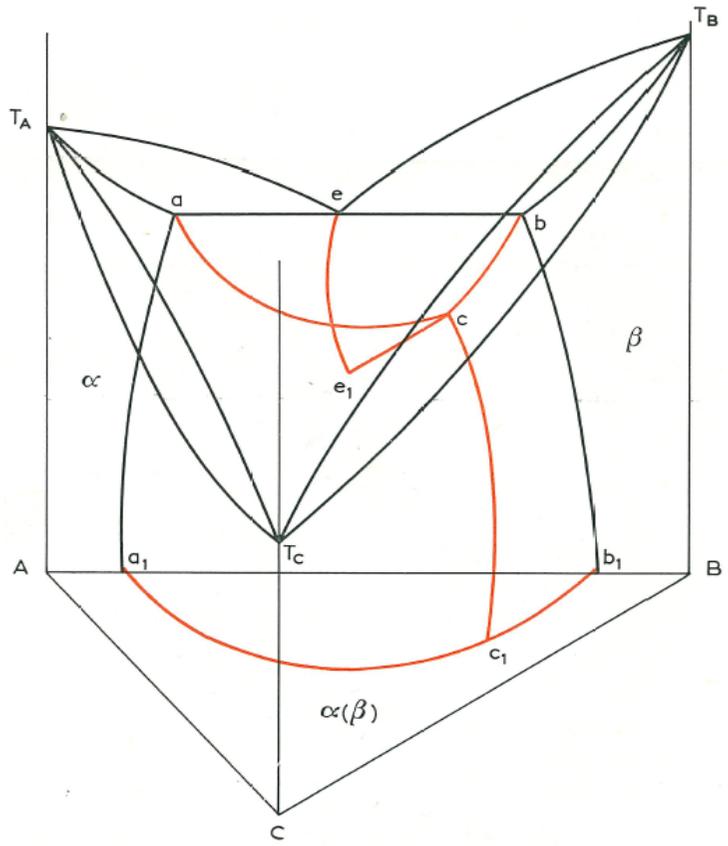
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- Isothermal section

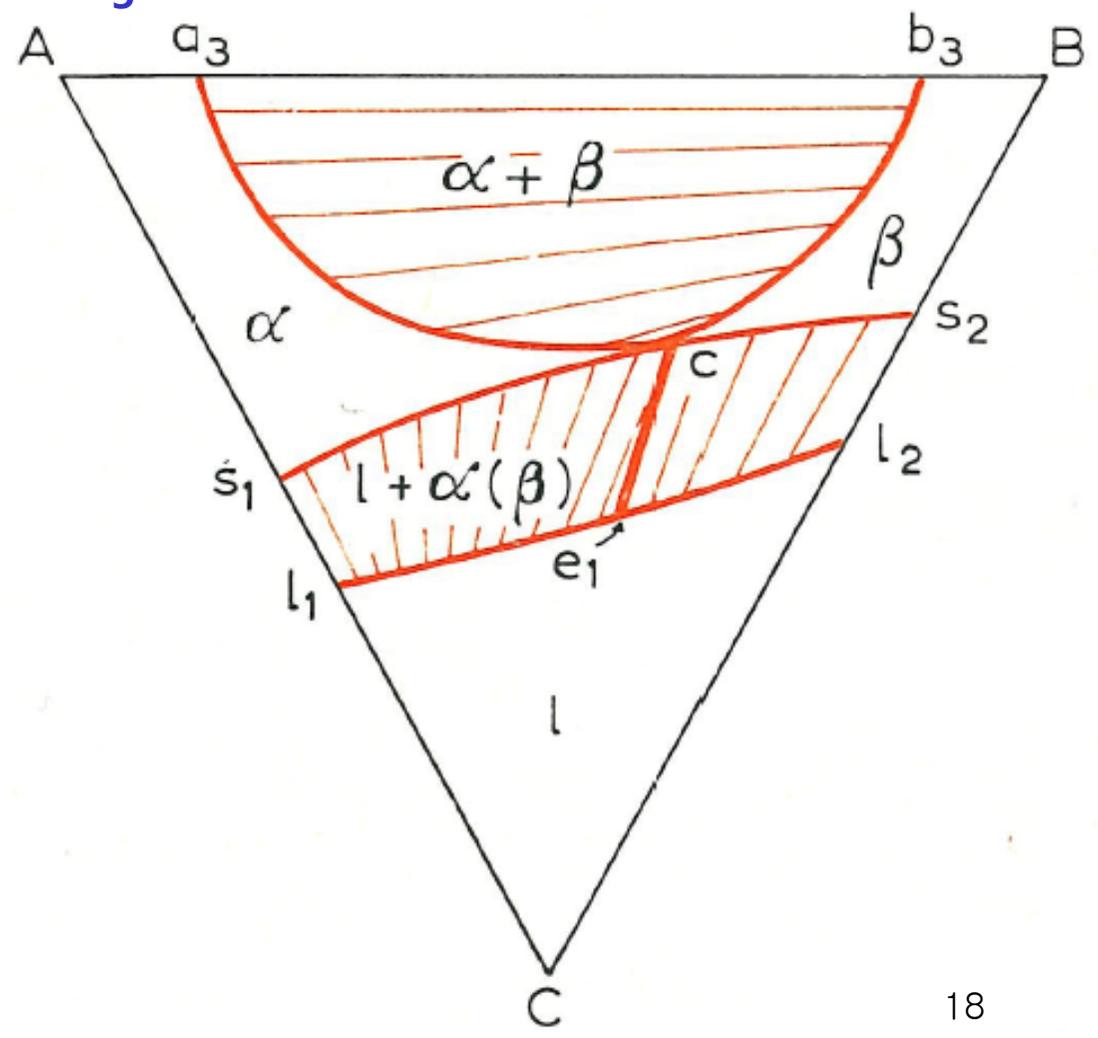


### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- Isothermal section



$T_5 = e$



## 9.2. THREE-PHASE EQUILIBRIUM

### ① Coalescence of miscibility gap and two phase region

- How we can have 3 phase equil.?

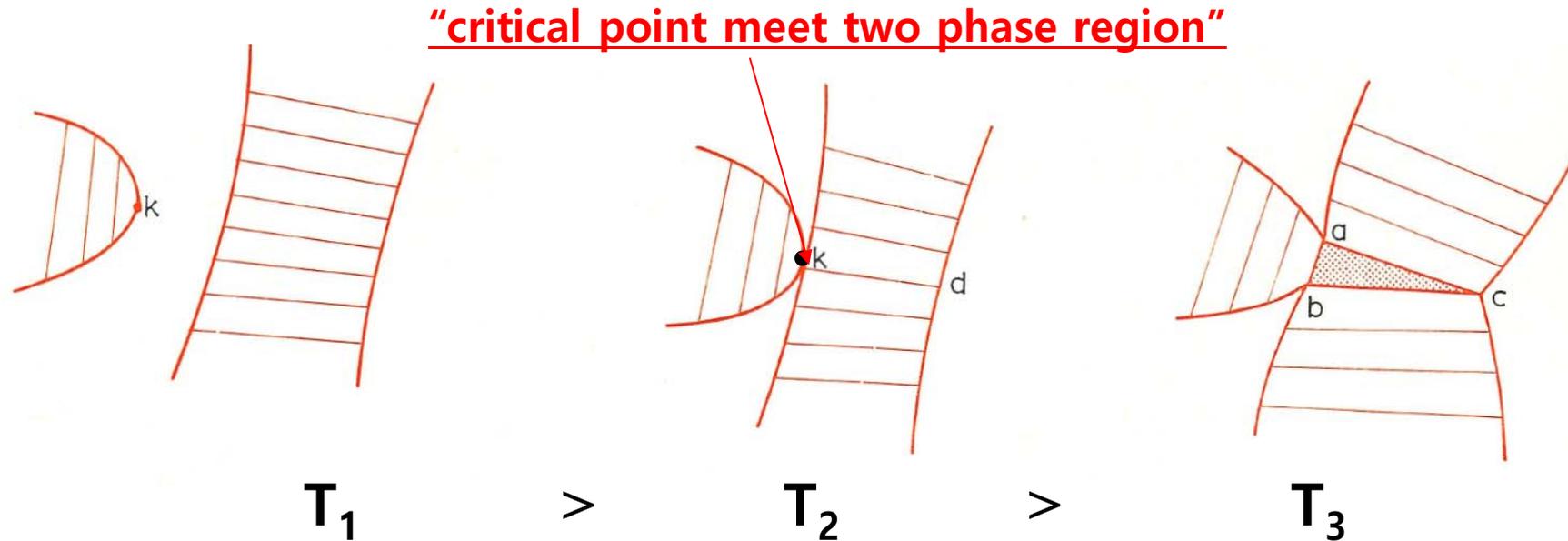


Fig. 136. Production of a ternary three-phase equilibrium by the coalescence of two two-phase regions

## 9.2. THREE-PHASE EQUILIBRIUM

### ② Coalescence of two two-phase region

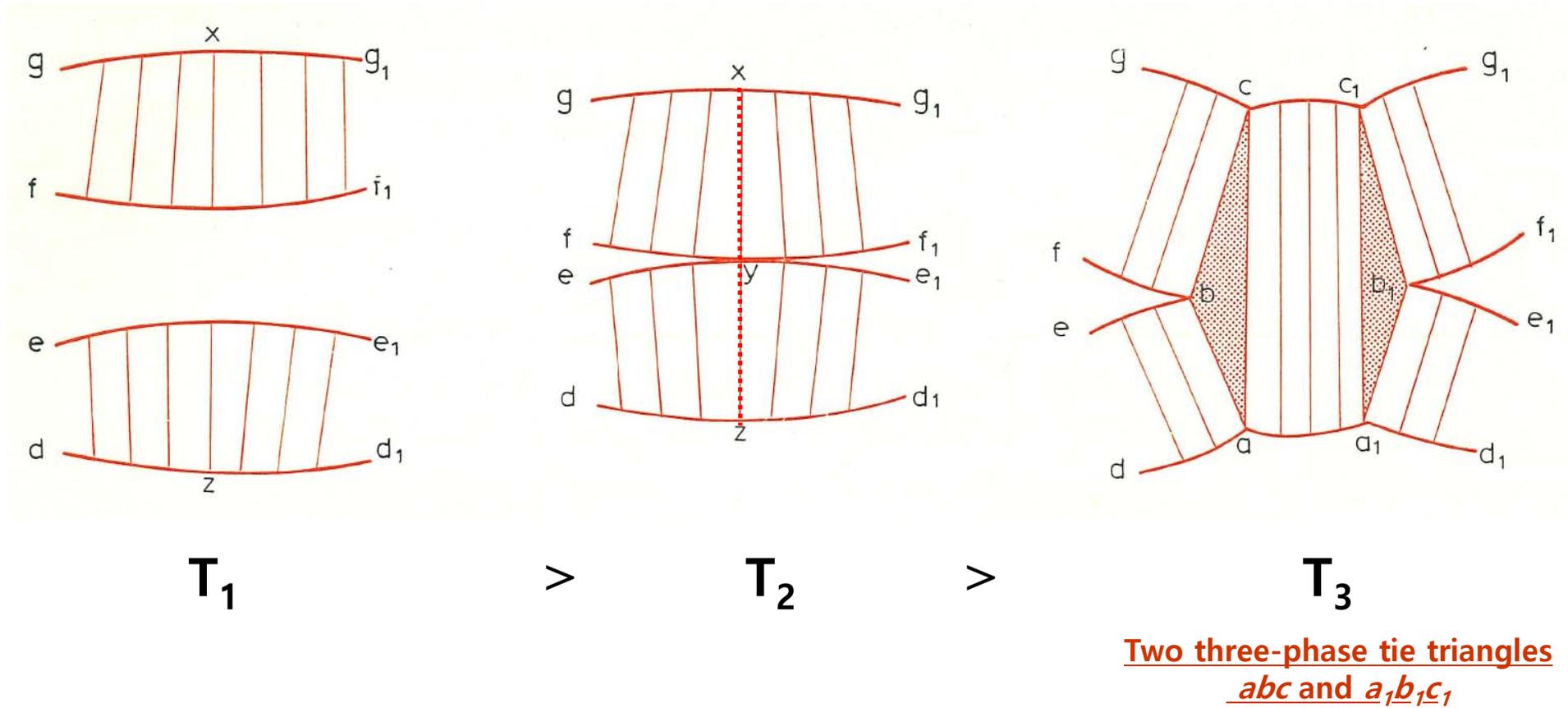
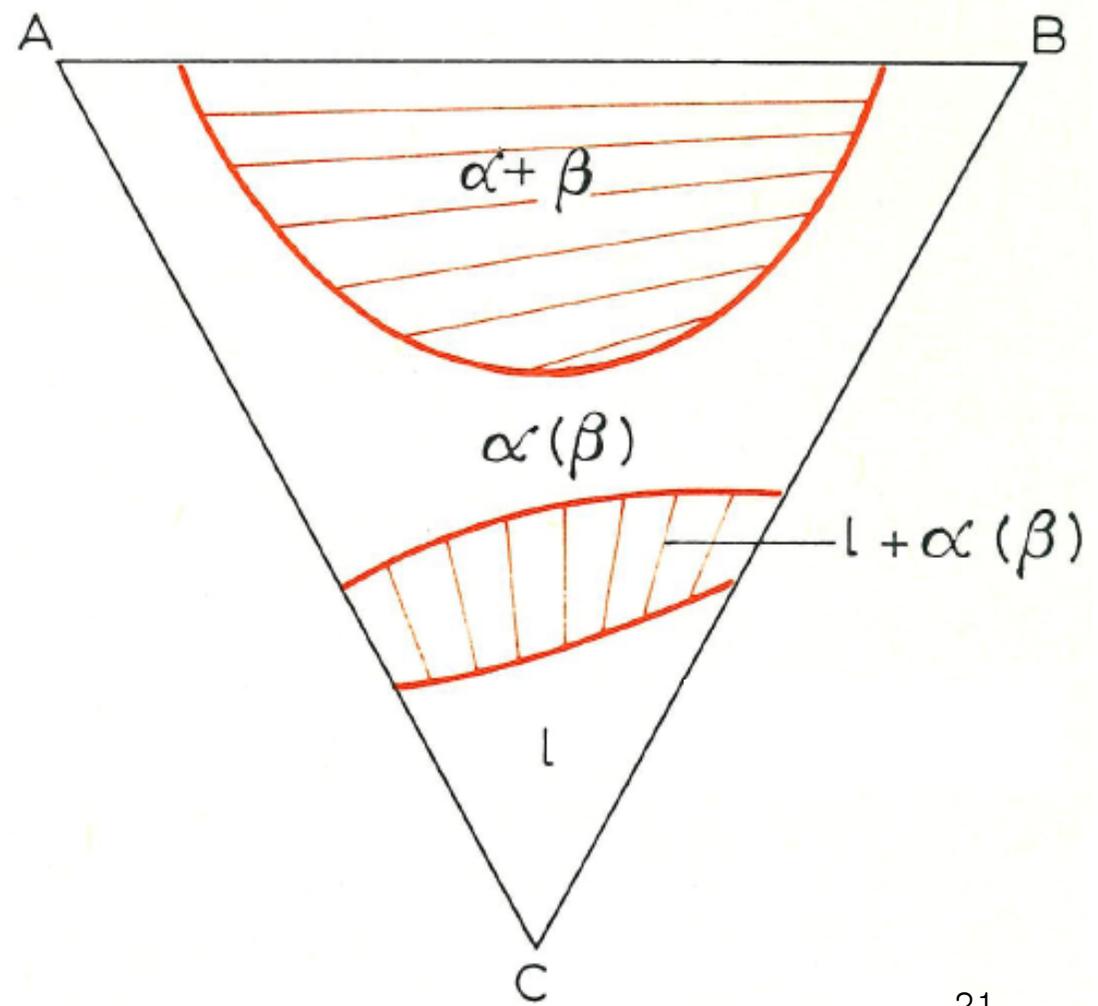
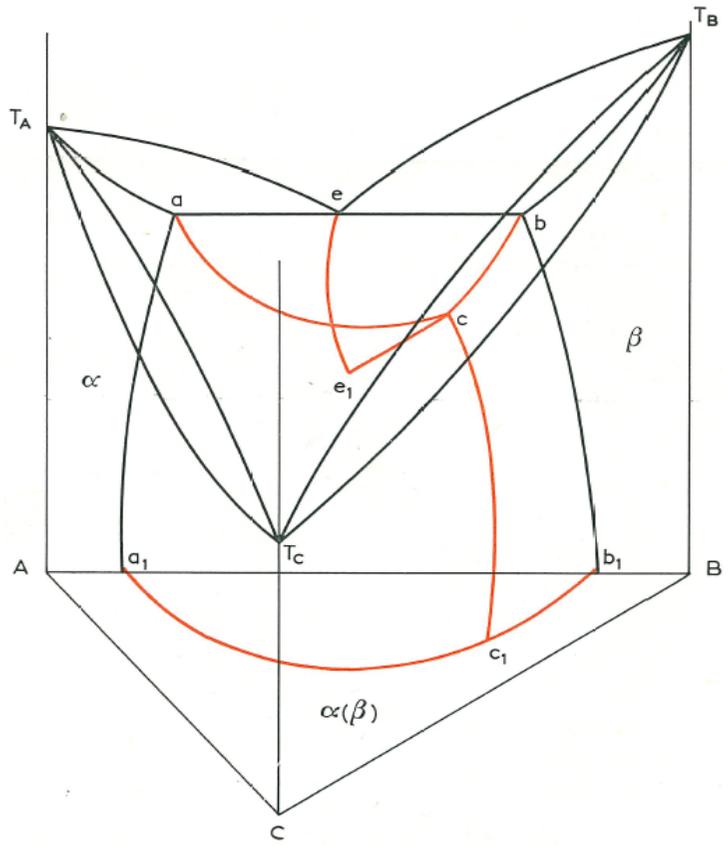


Fig. 138. Alternative method to Fig. 136 for the production of a ternary three-phase equilibrium by the coalescence of two two-phase regions

### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

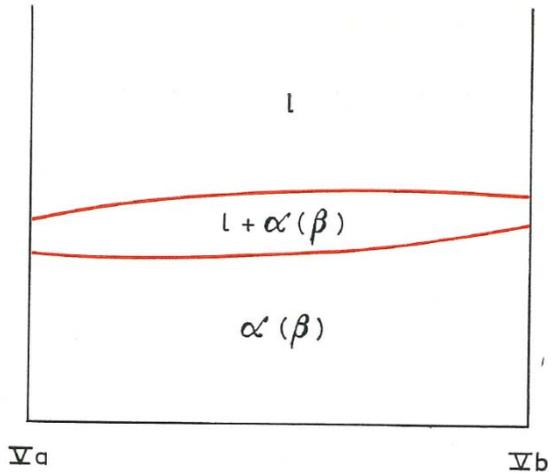
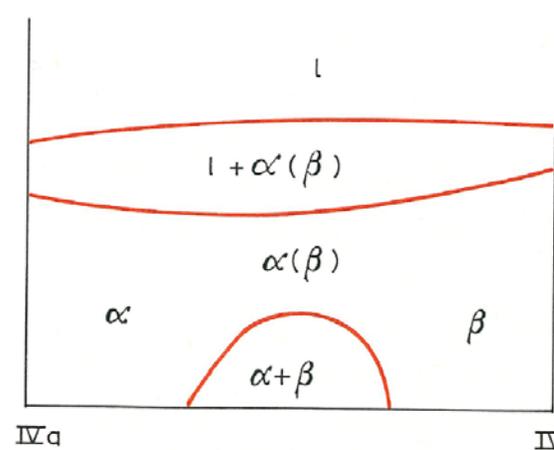
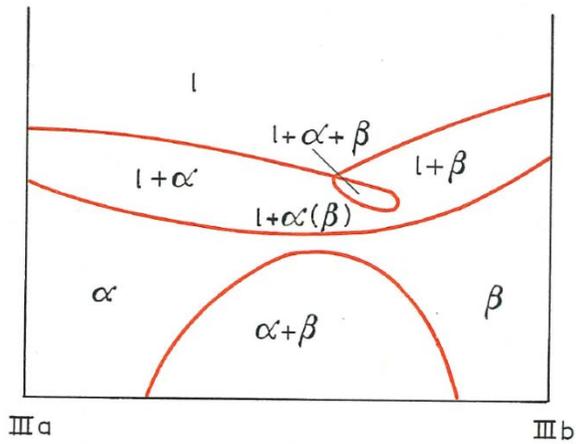
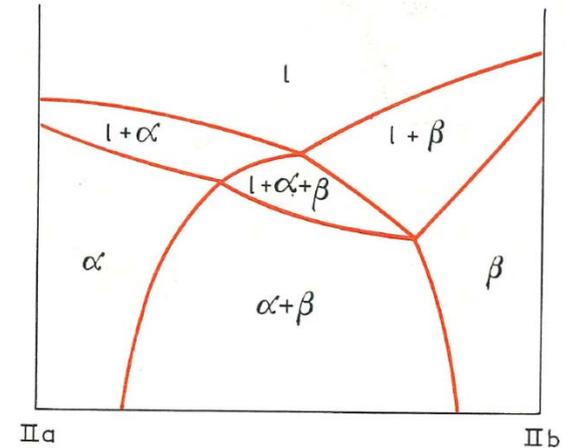
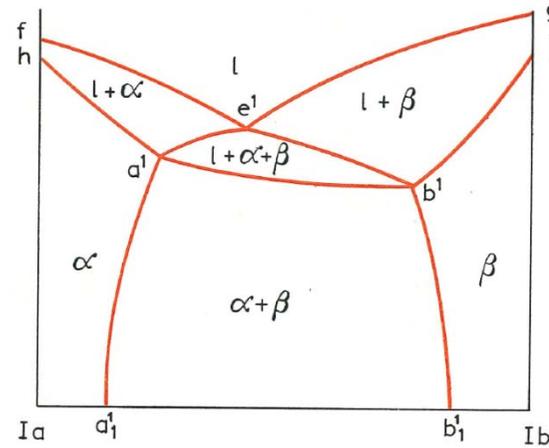
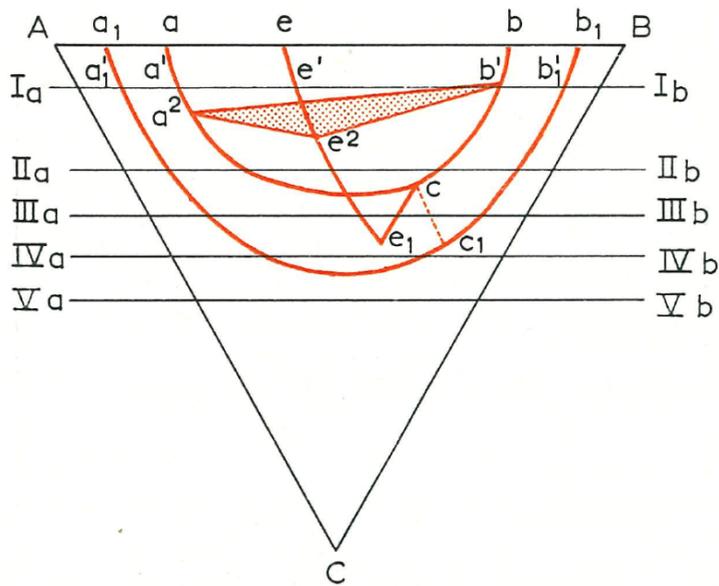
- Isothermal section

$$e_1 > T_6 > T_c$$



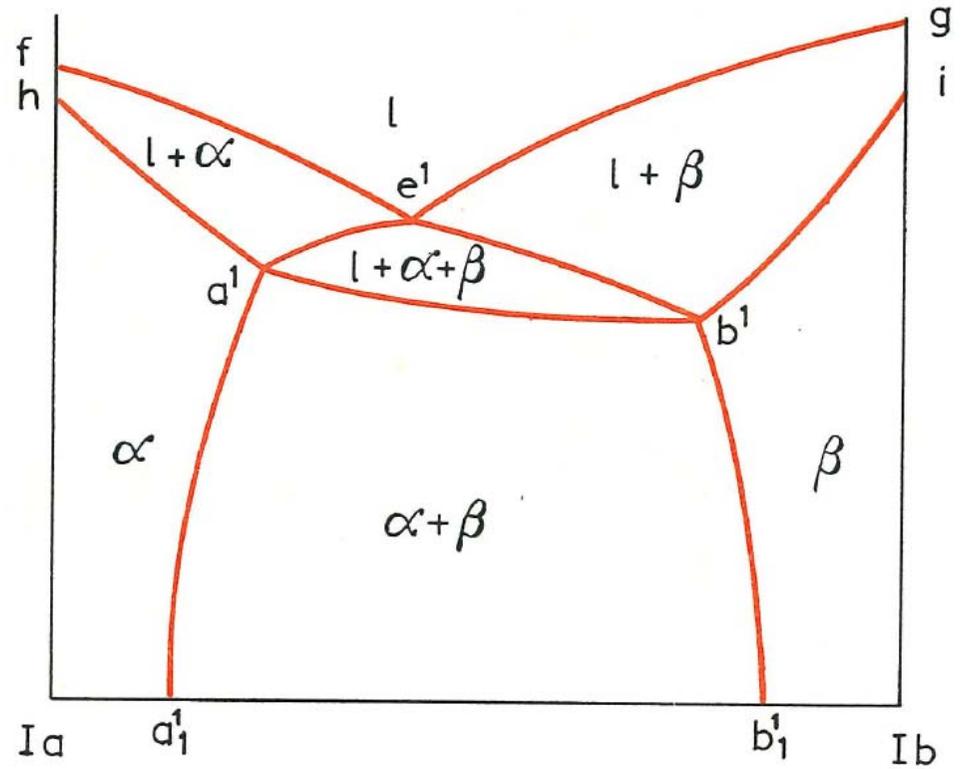
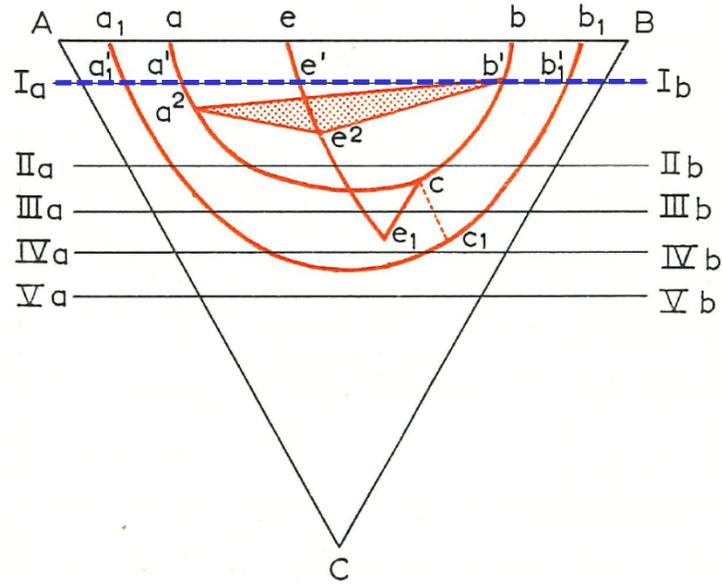
### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- Vertical section



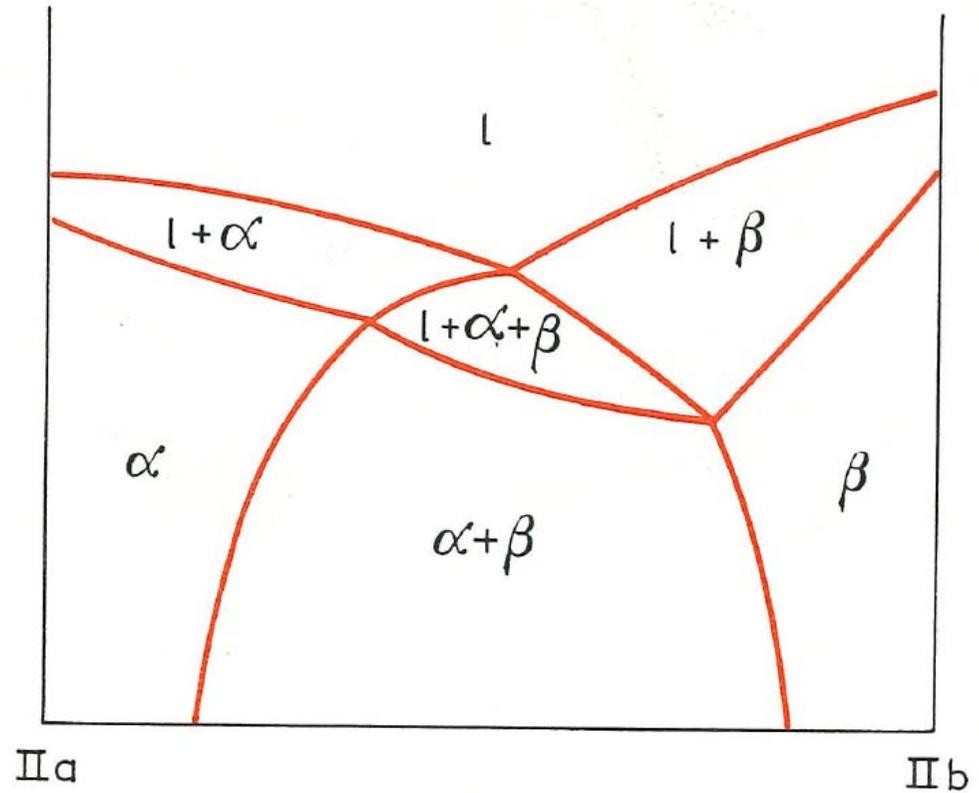
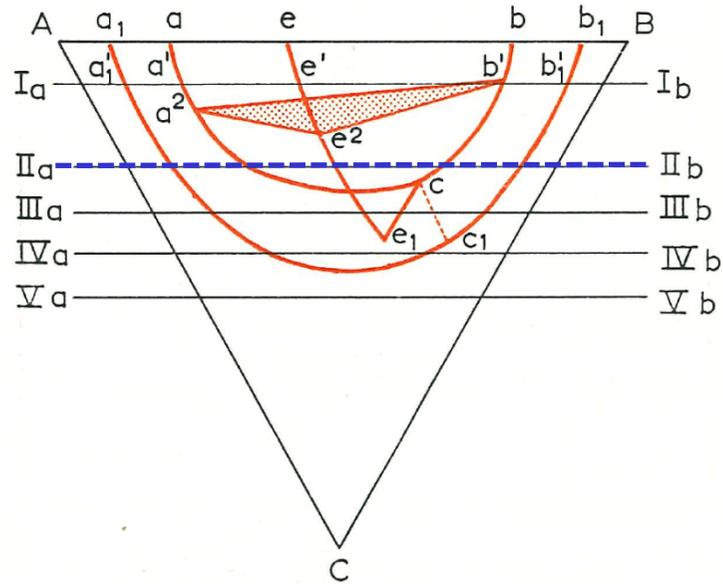
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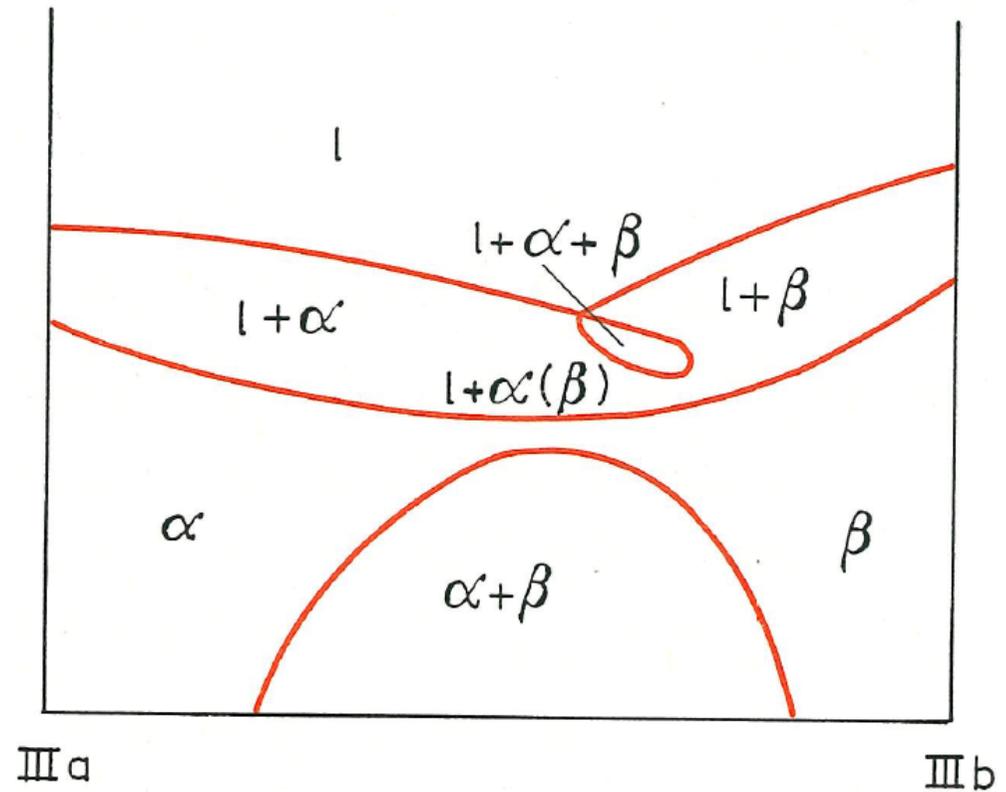
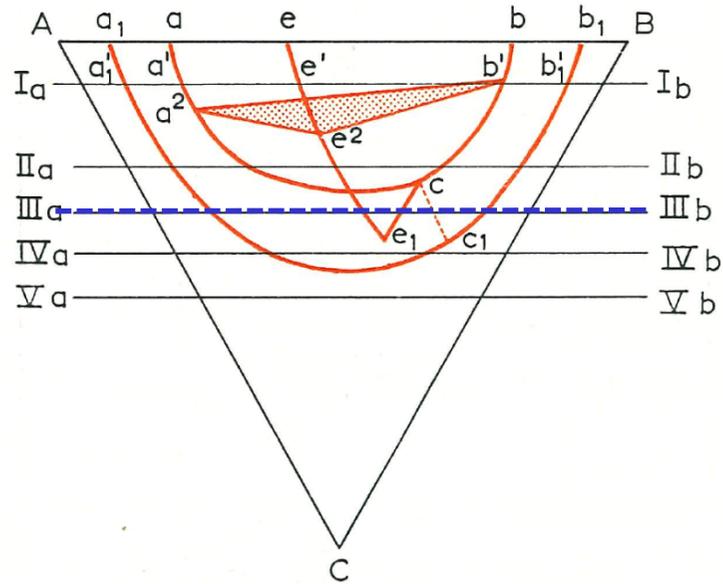
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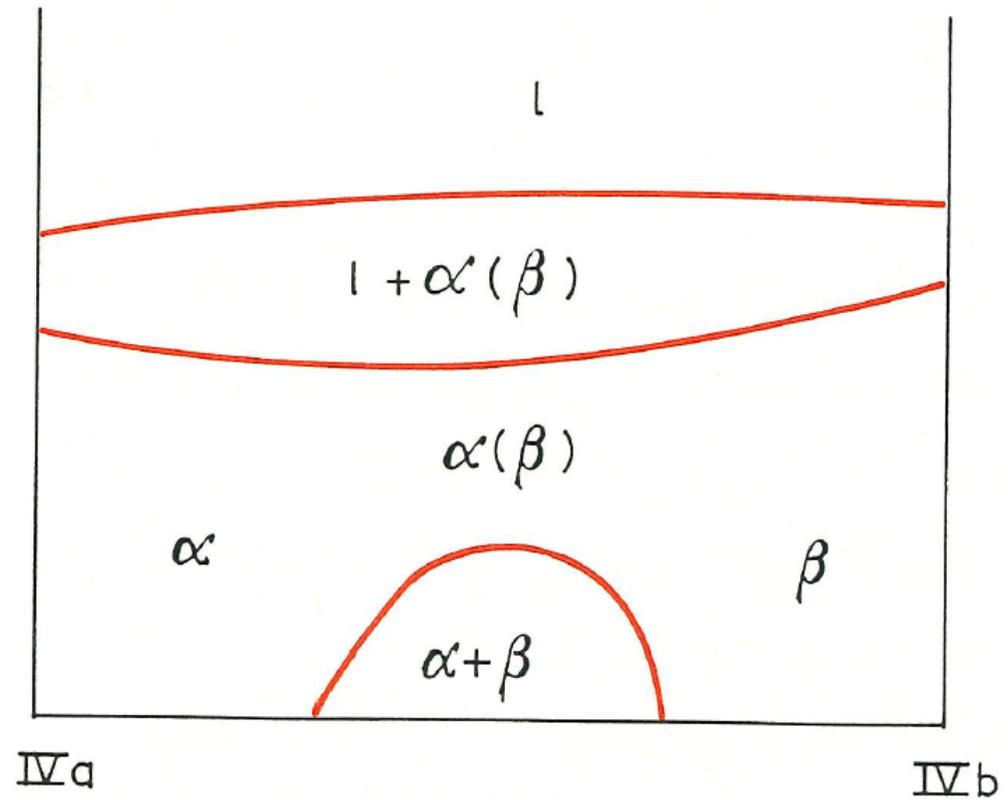
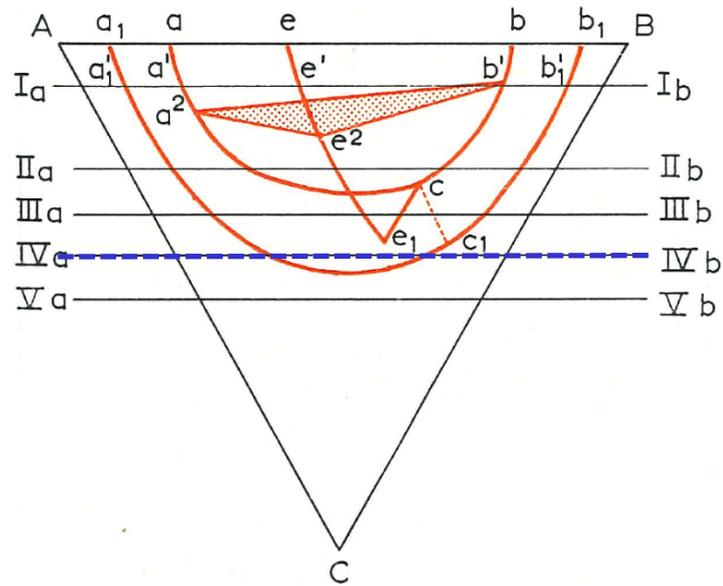
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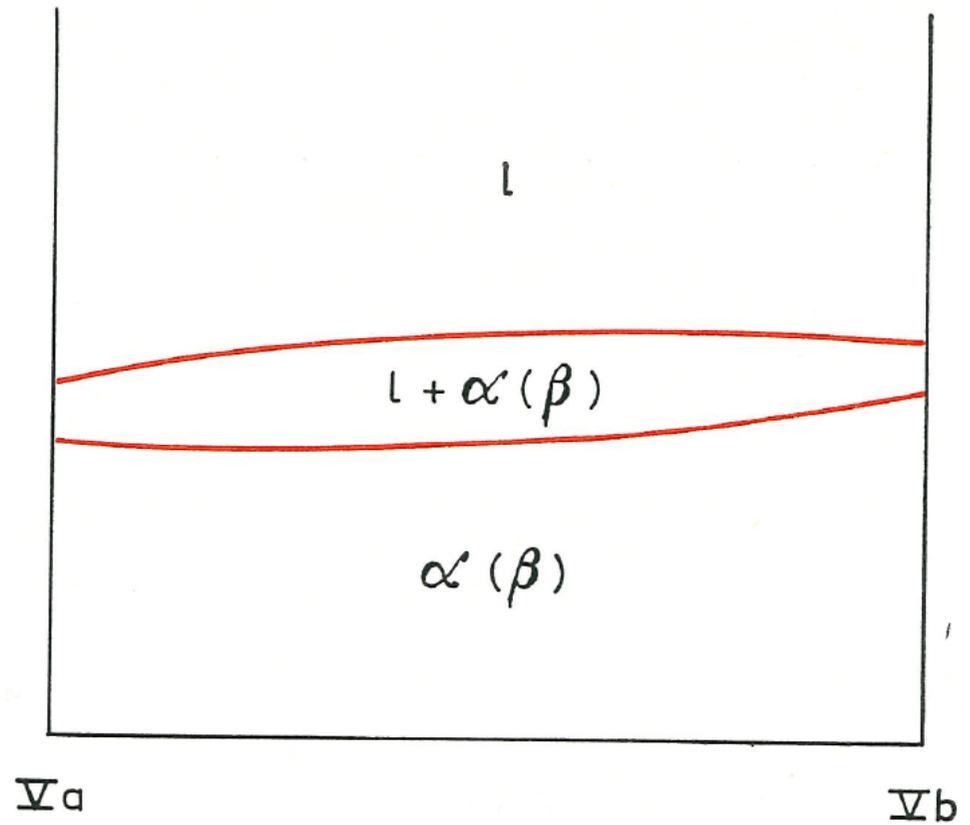
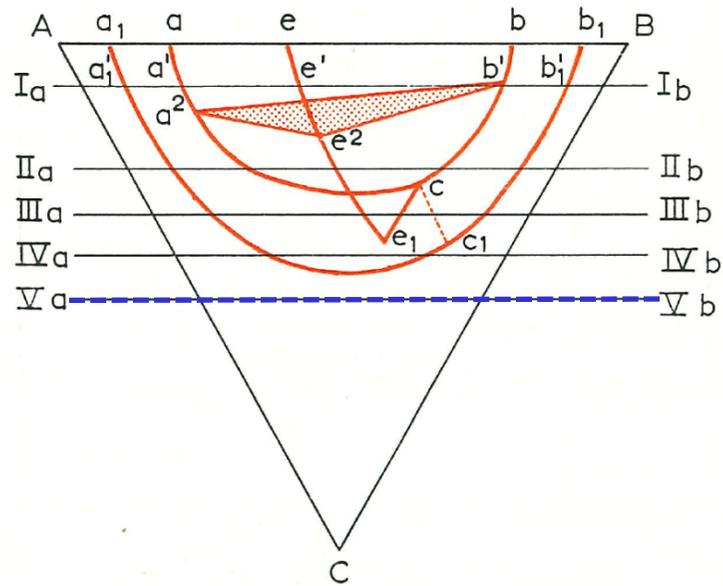
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- Vertical section



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- Vertical section

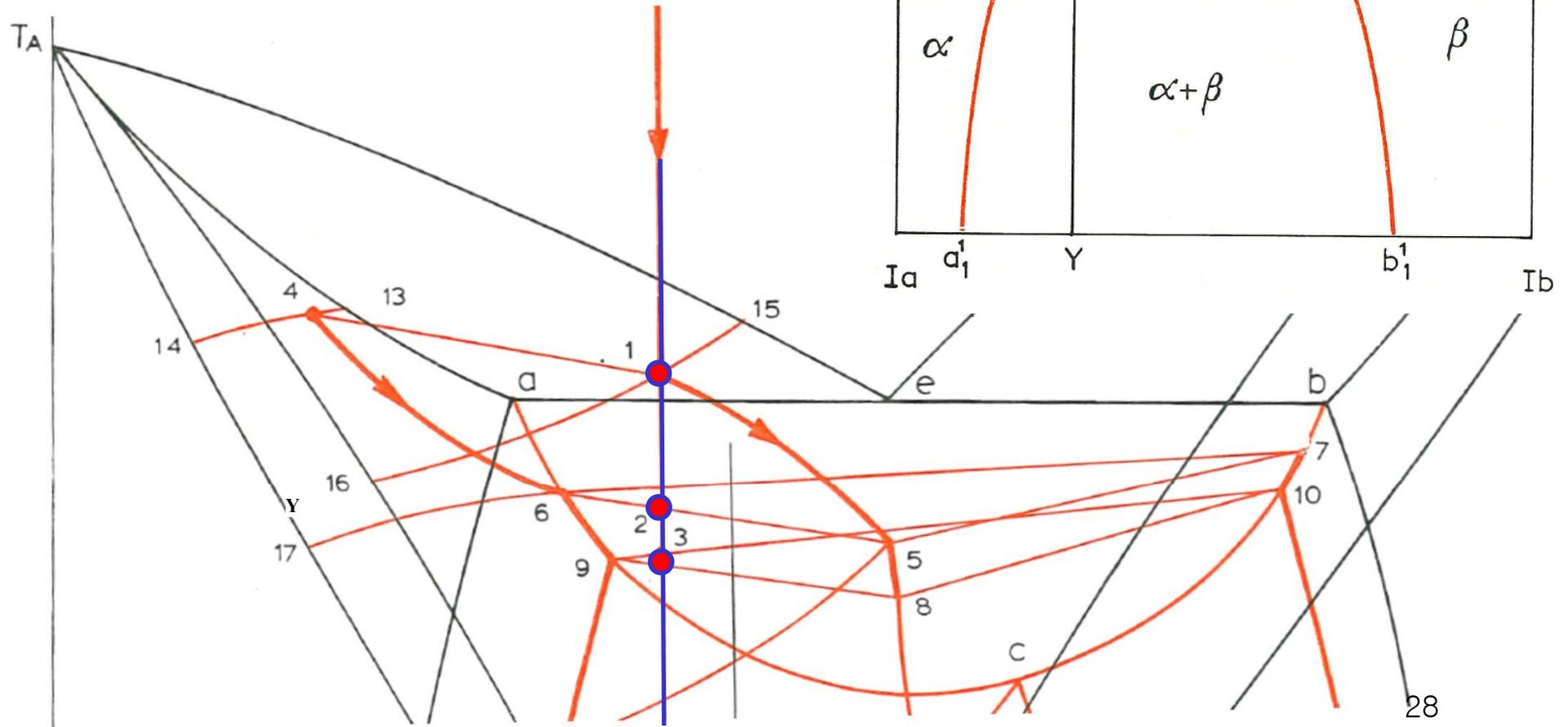
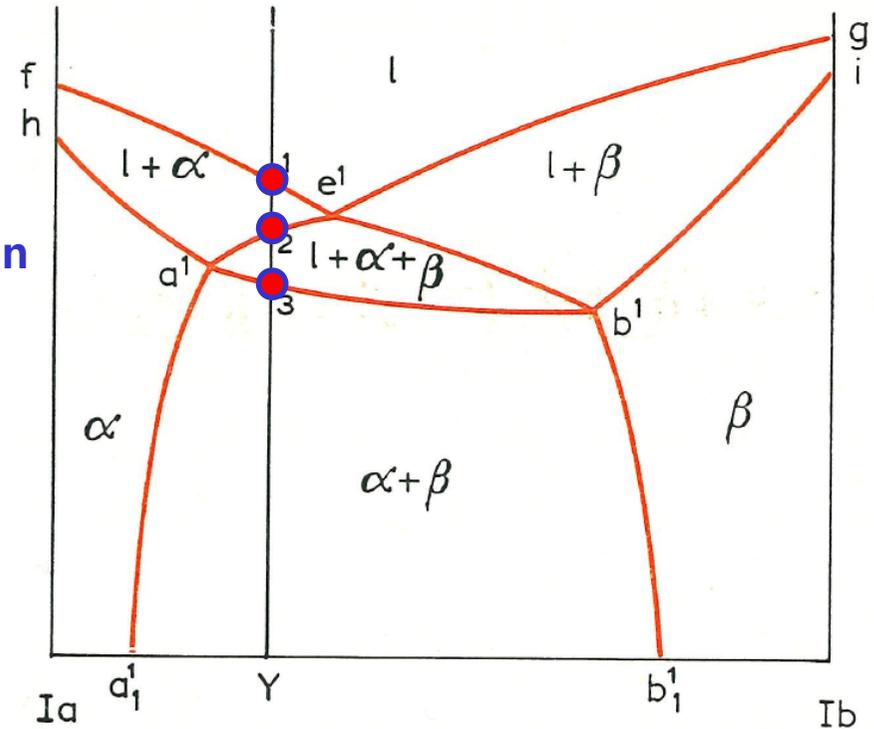


### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- Transformation on cooling

Y alloy:  $l \rightarrow l + \alpha \rightarrow l + \alpha + \beta \rightarrow \alpha + \beta$

A clear idea of the sequence of crystallization



> Point 1: 4 on the  $\alpha$  solidus surface

> Point 1- Point 2

\* 4→6 on the  $\alpha$  solidus surface

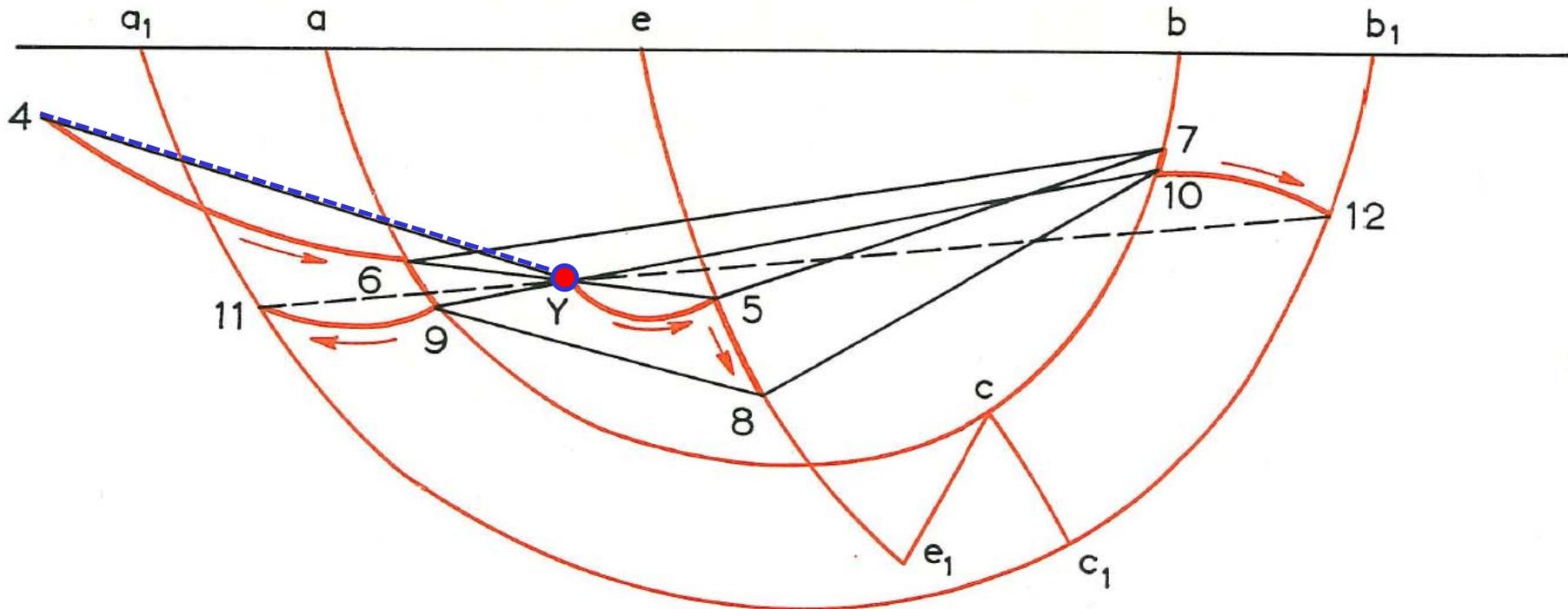
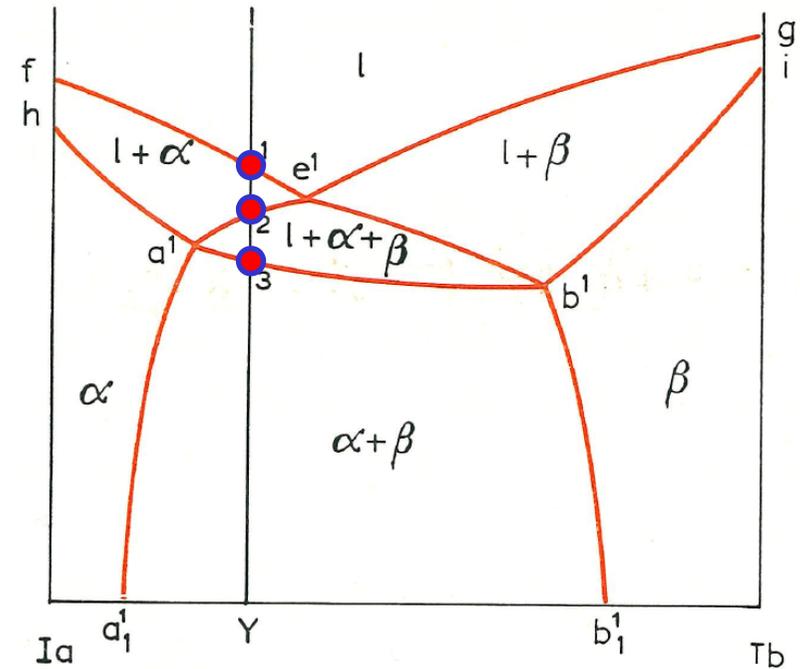
\* 1→5 on the  $\alpha$  liquidus surface

Three phase equilibrium l5,  $\alpha$ 6,  $\beta$ 7

\*  $\alpha$ : 6→9,  $\beta$ : 7→10, l: 5→8

> Point 3: on the tie line 9-10

> Point 3-Y:  $\alpha$ : 9→11,  $\beta$ : 10→12

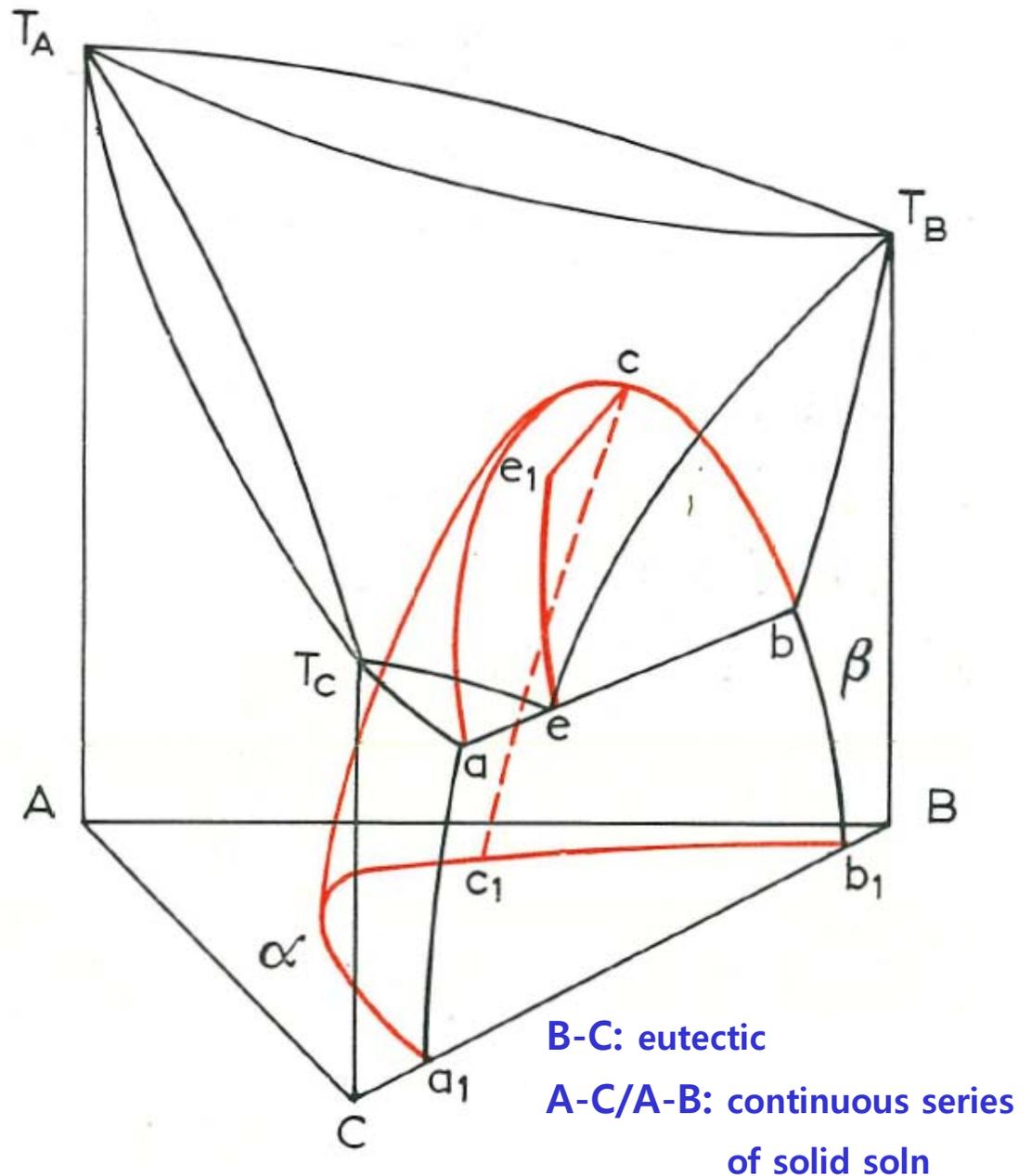
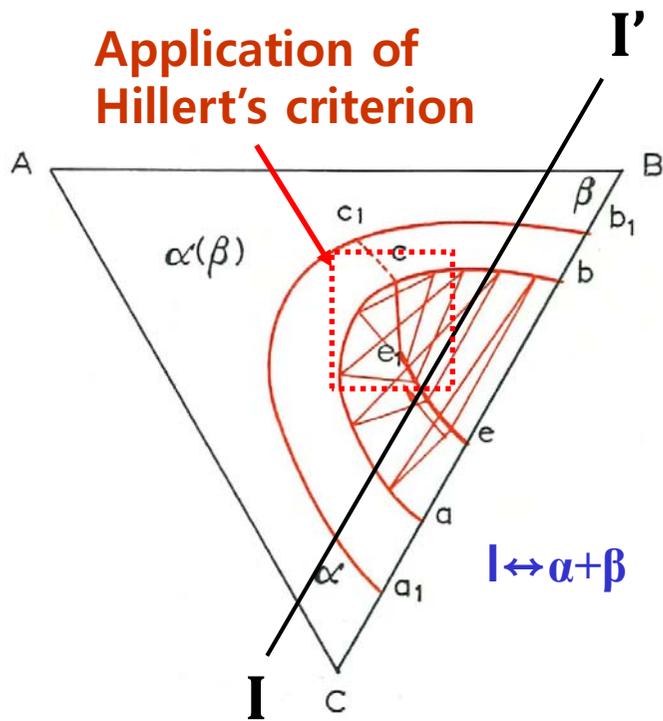


Projection of the solidification sequence for alloy Y on the concentration triangle

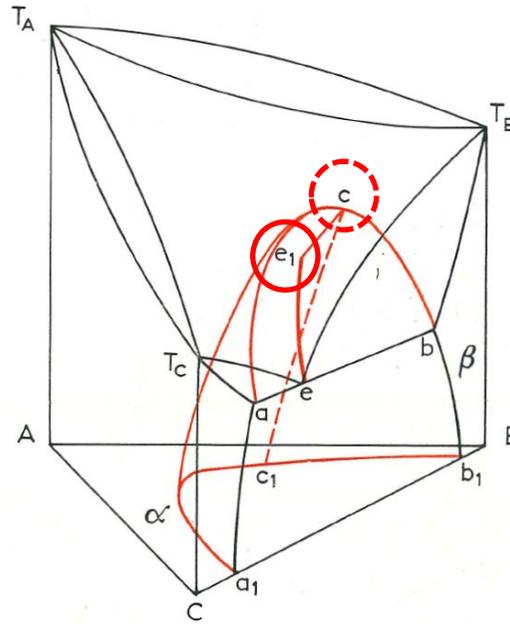
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- A maximum critical point

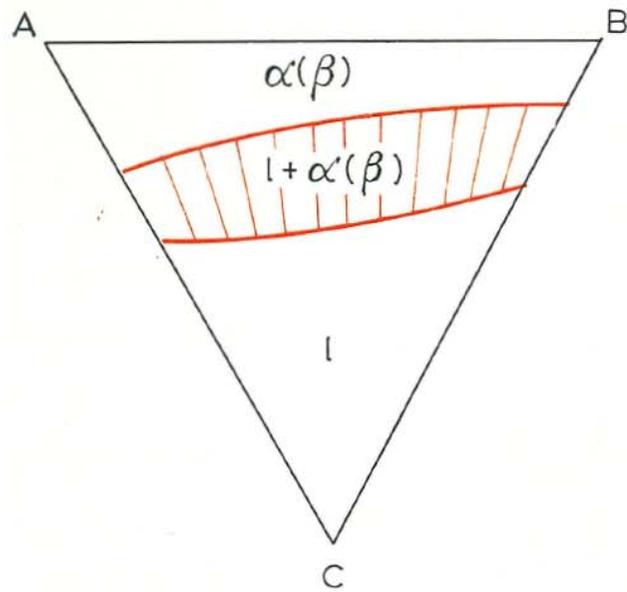
Basically, the reaction we can expect is eutectic reaction ( $l \rightarrow \alpha + \beta$ ). But, in reality, we can have eutectic and peritectic reaction depending on the relative amount of three phase.



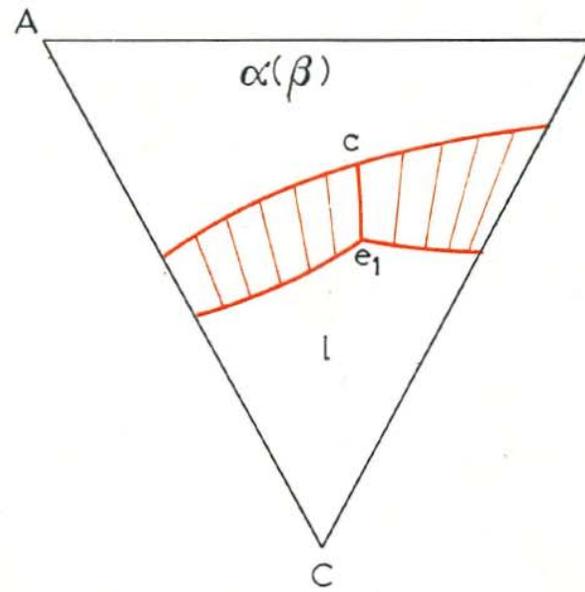
- A maximum critical point



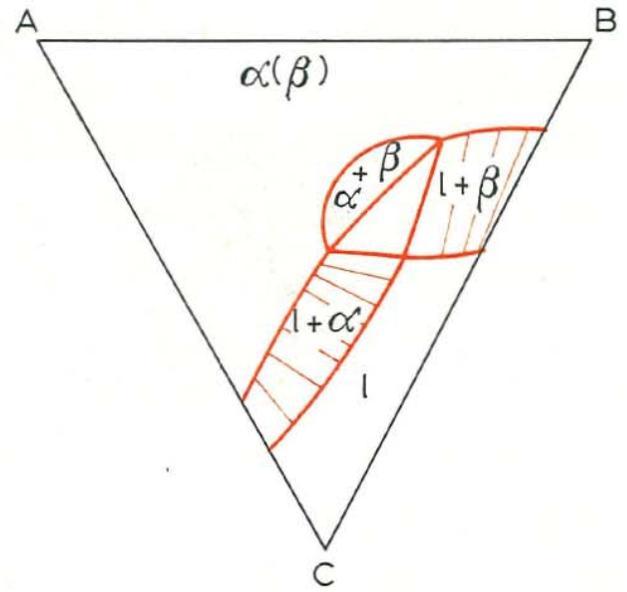
isothermal section



$T > e_1$

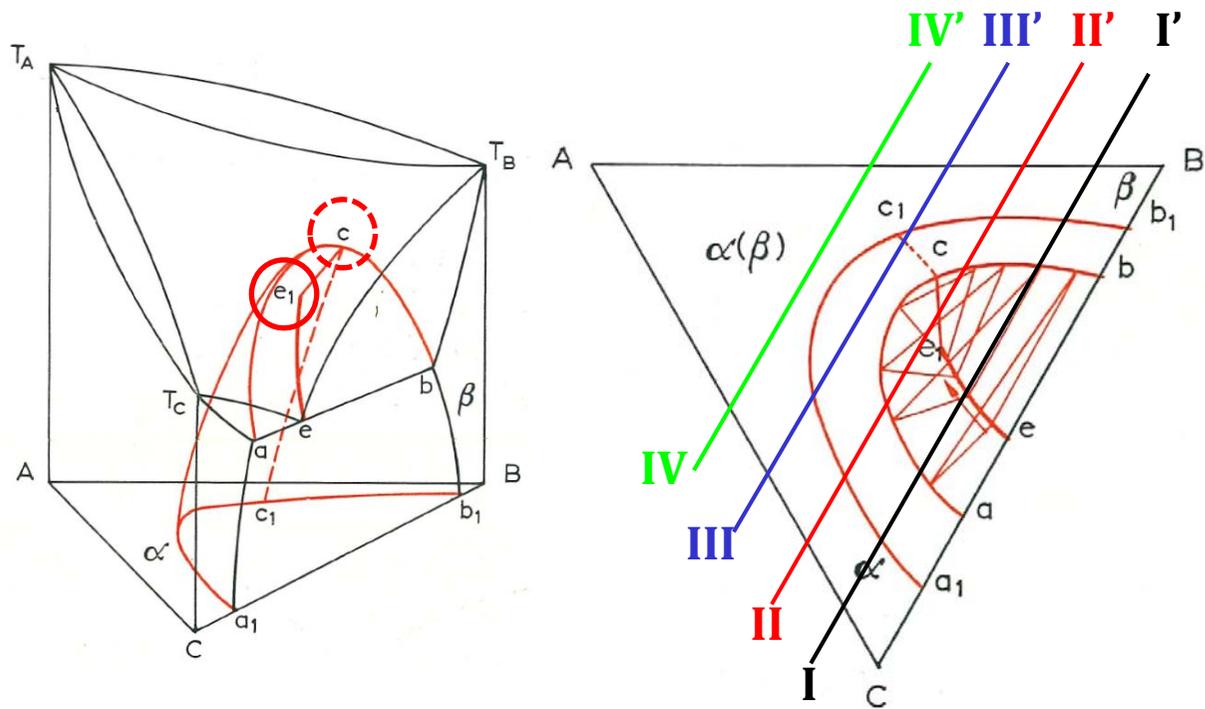


$T = e_1$

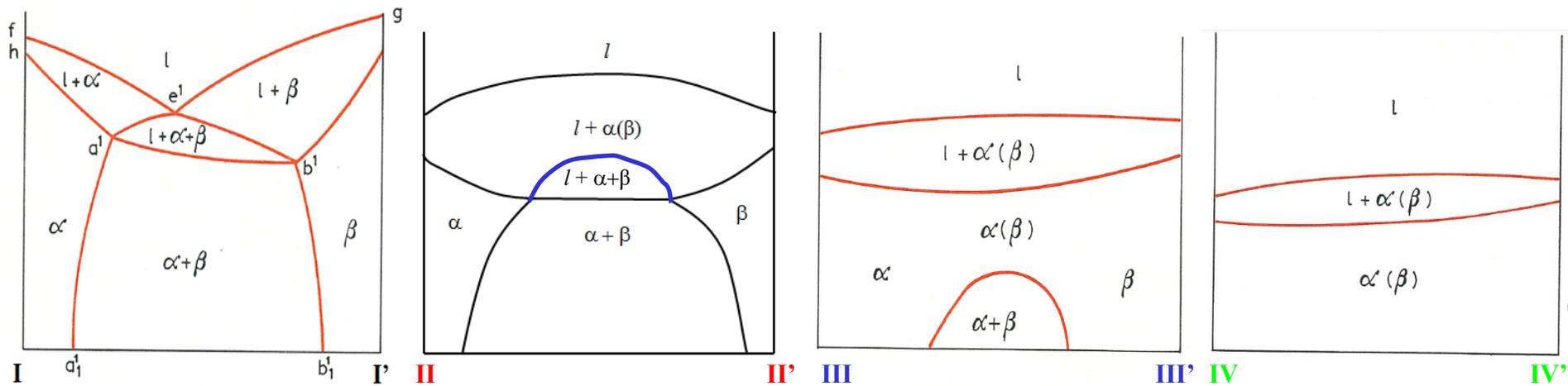


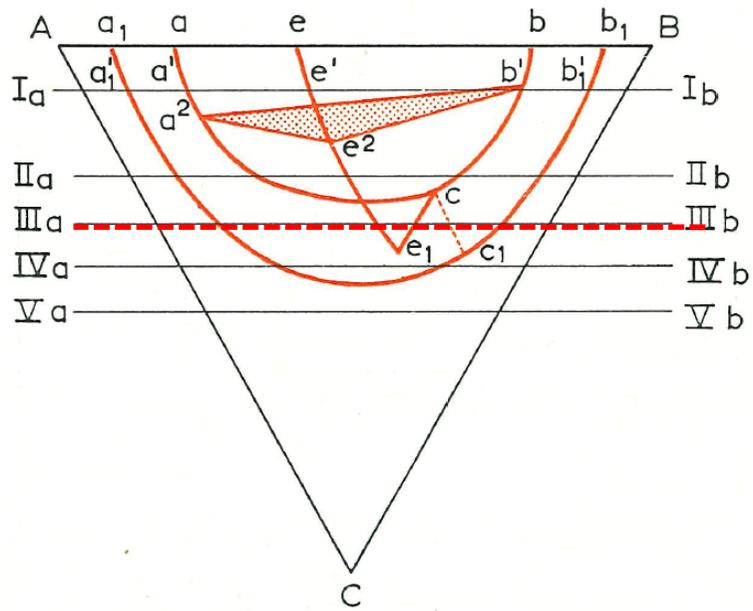
$T < e_1$

- A maximum critical point

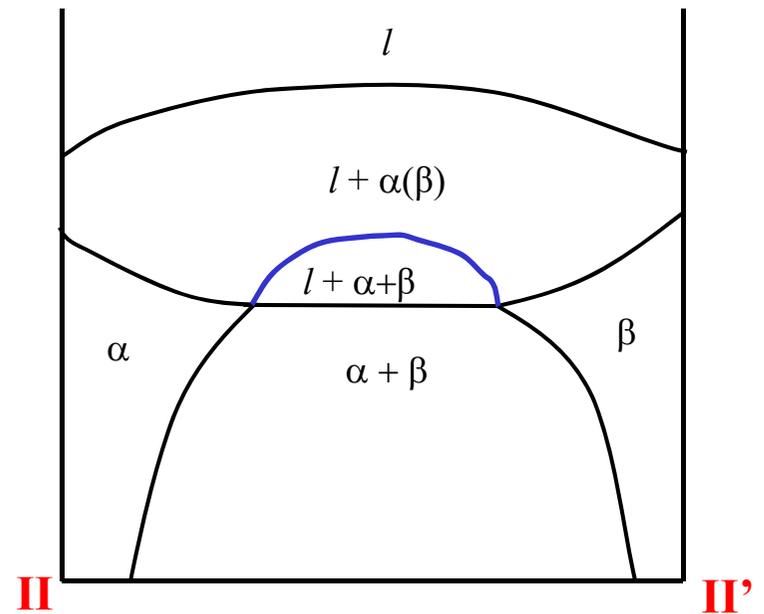
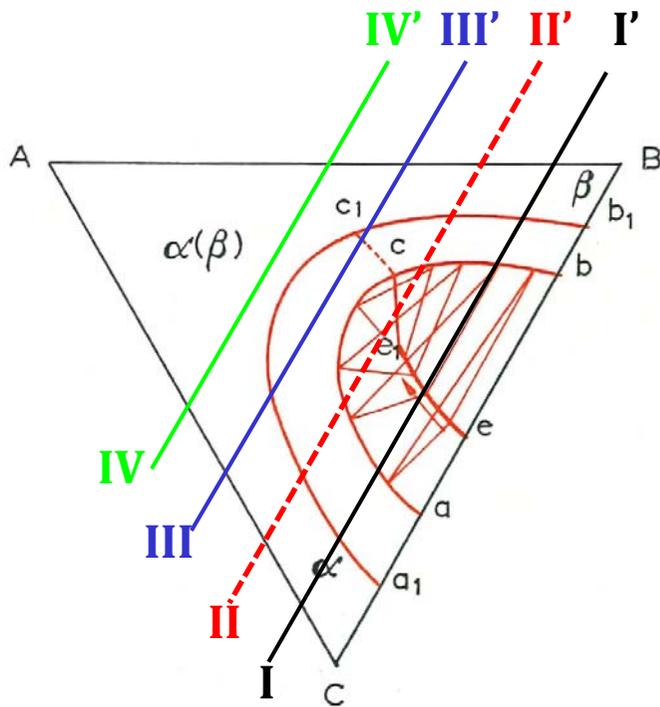
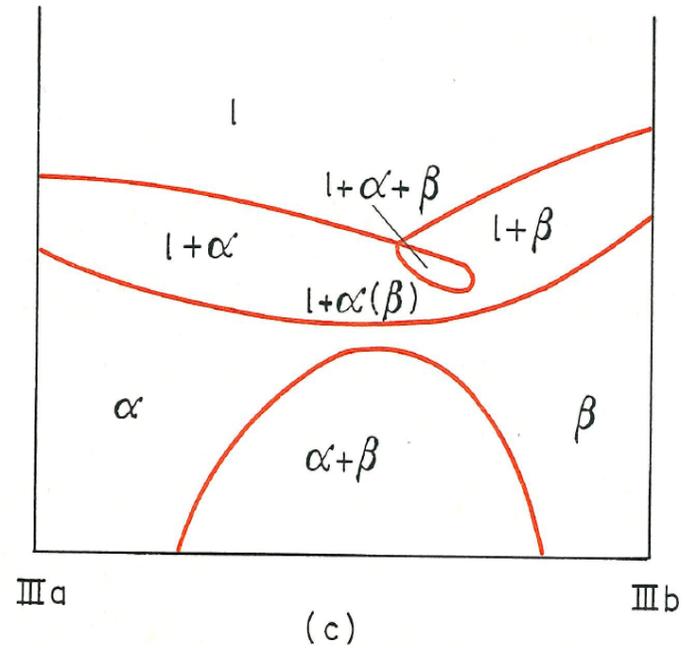


### Vertical section



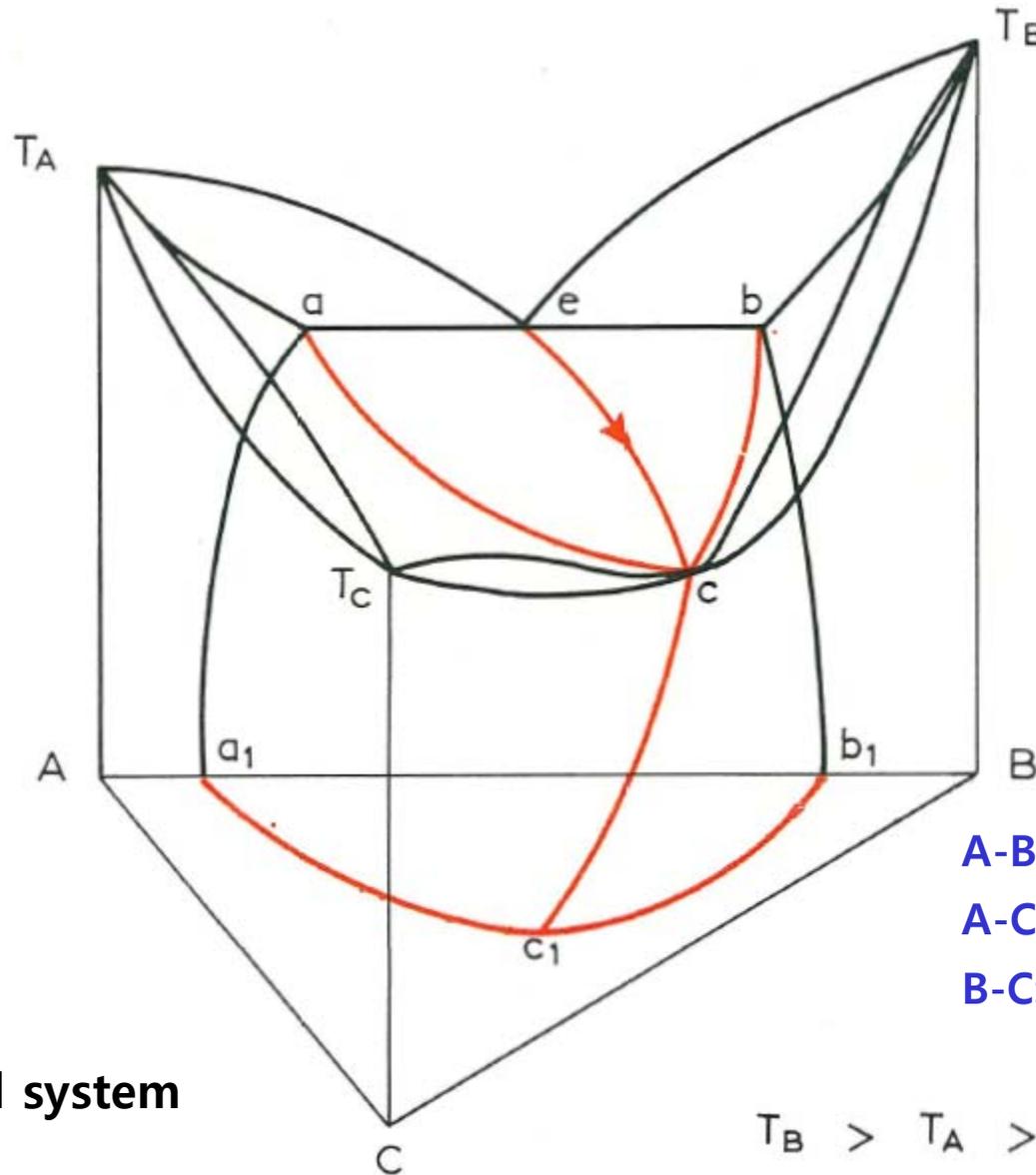


### Vertical section



### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- A limiting case: solubility gap just reaches one of the binary systems.



A-B: eutectic

A-C: complete solid soln

B-C: complete solid soln  
with congruent minima

Au-Co-Pd system

$$T_B > T_A > T_C > e > c \quad 34$$

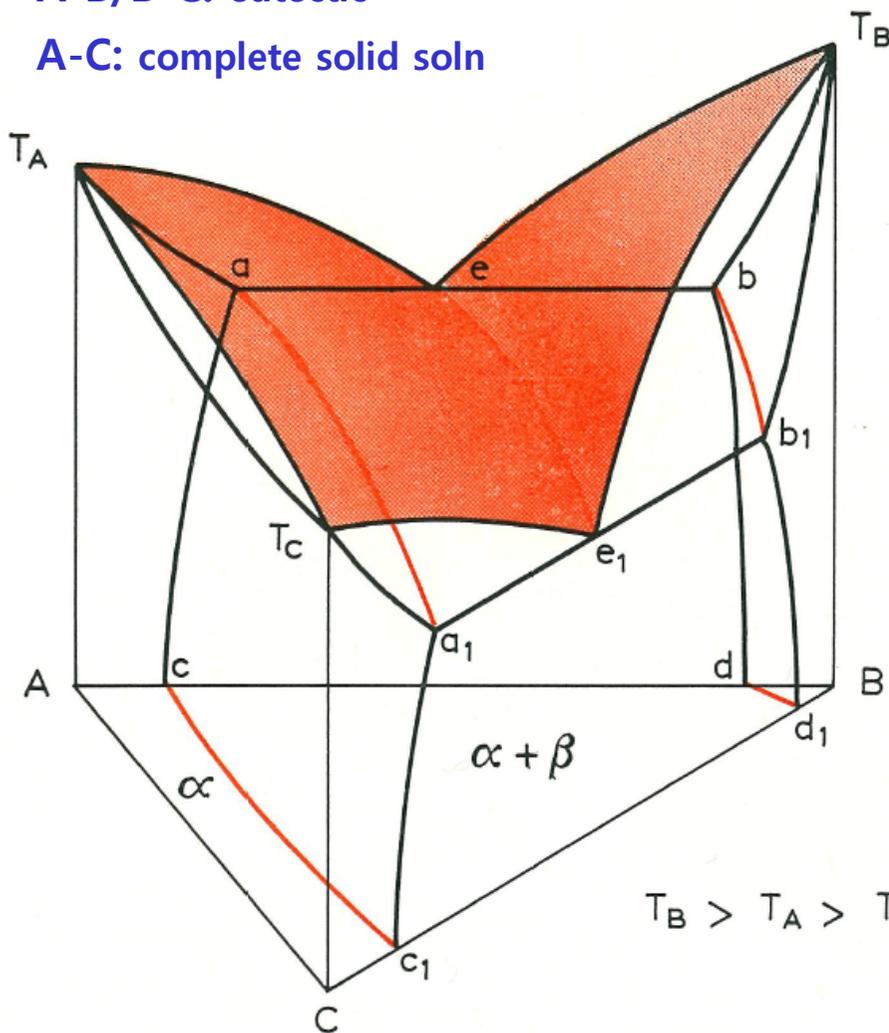
# 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- A limiting case

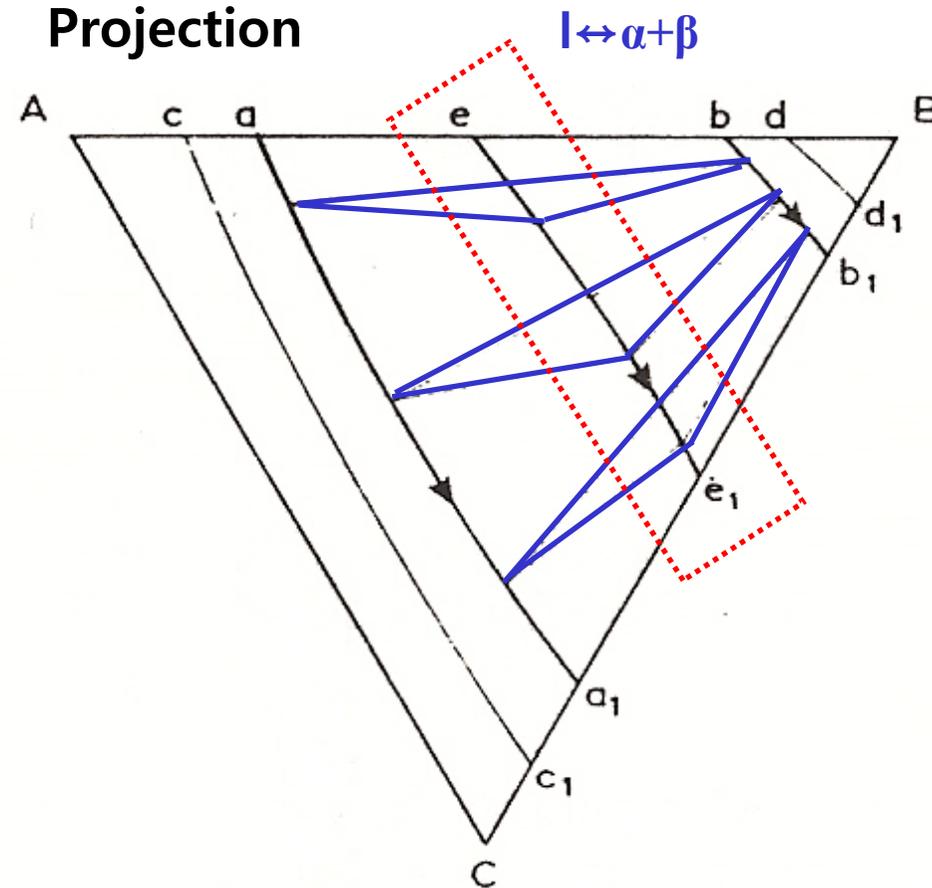
## Space model

A-B/B-C: eutectic

A-C: complete solid soln



## Projection



$$T_B > T_A > T_C > e > e_1$$

### 9.3. THREE-PHASE EQUILIBRIUM INVOLVING EUTECTIC REACTIONS

- A limiting case

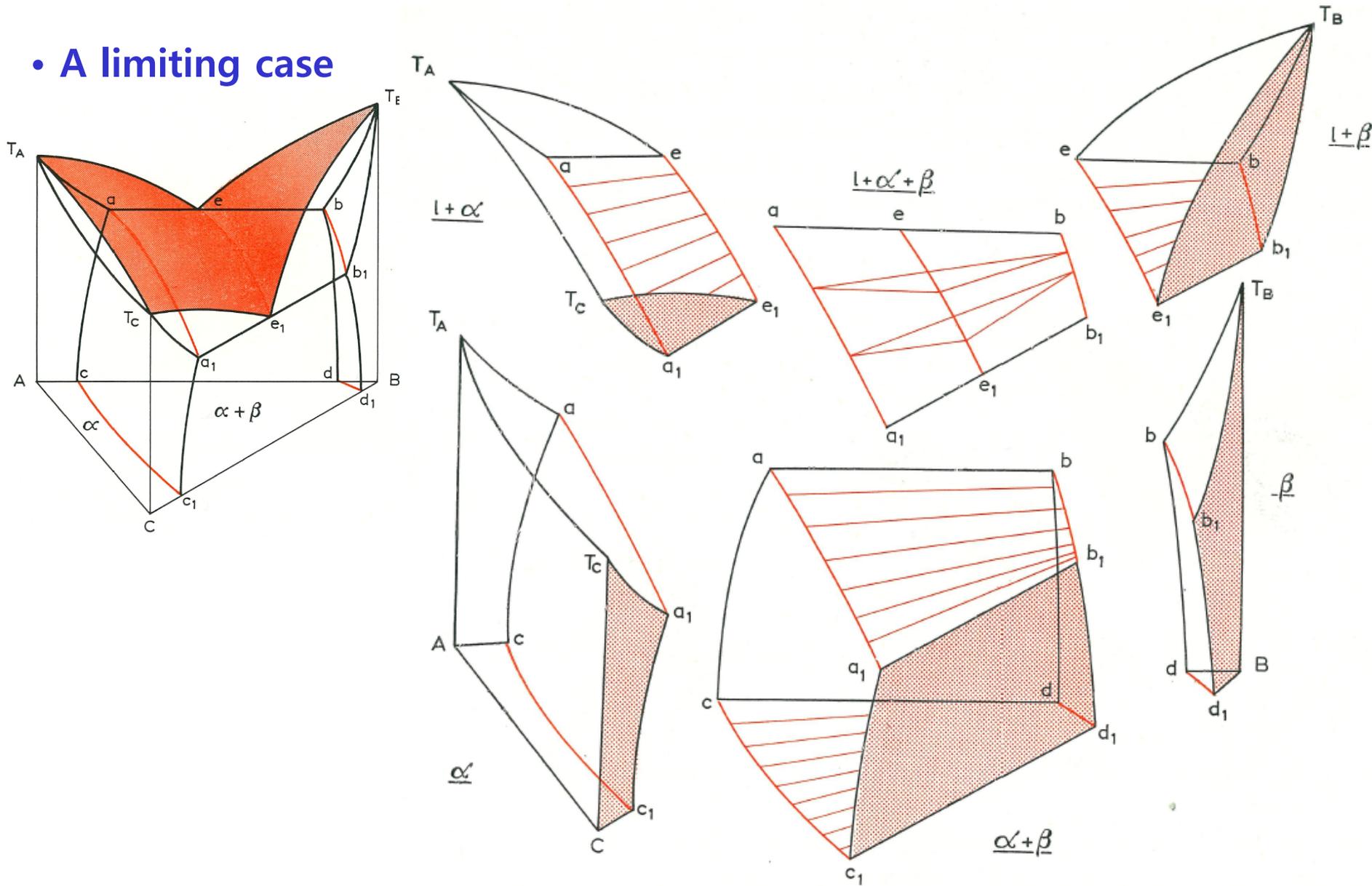
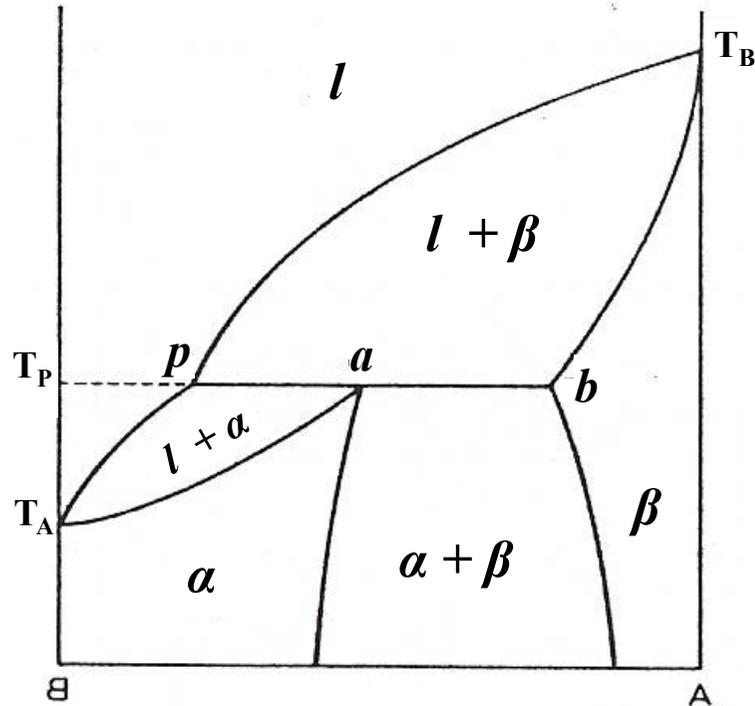


Fig. 165. The phase regions in Fig. 164a.

## 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

- A peritectic solubility gap in one binary system



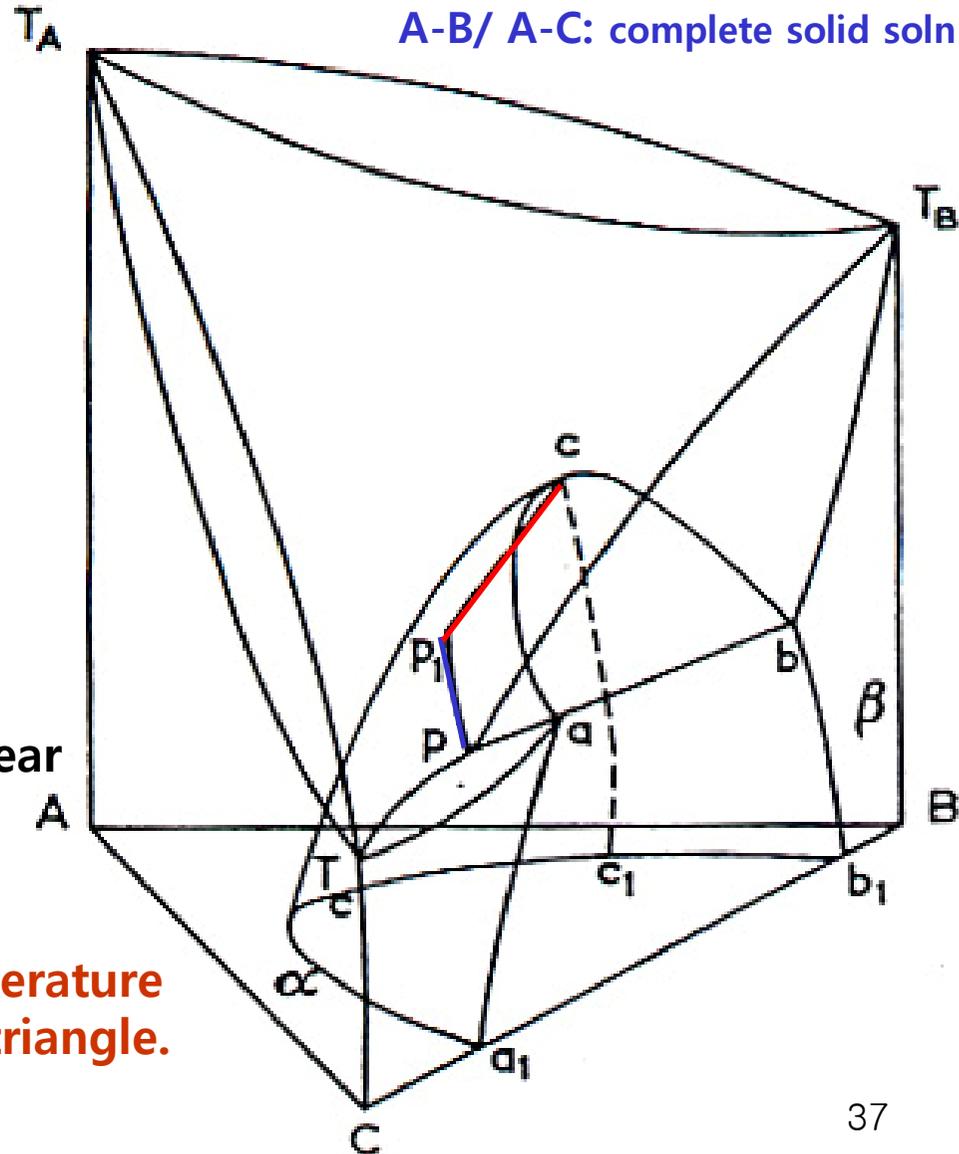
B-C: peritectic

A-B/ A-C: complete solid soln

: A minimum or a maximum may appear in the monovariant liquid curve.

PP<sub>1</sub>: monovariant curve for liquid

Points P<sub>1</sub> and c lie at the same temperature and the line P<sub>1</sub>c is a degenerate tie triangle.

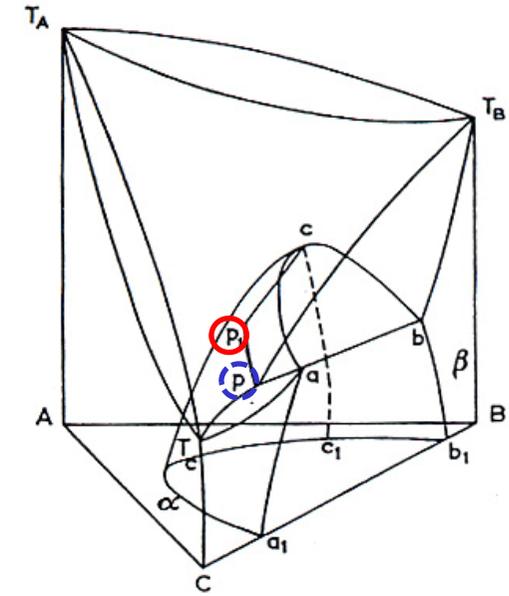


# 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

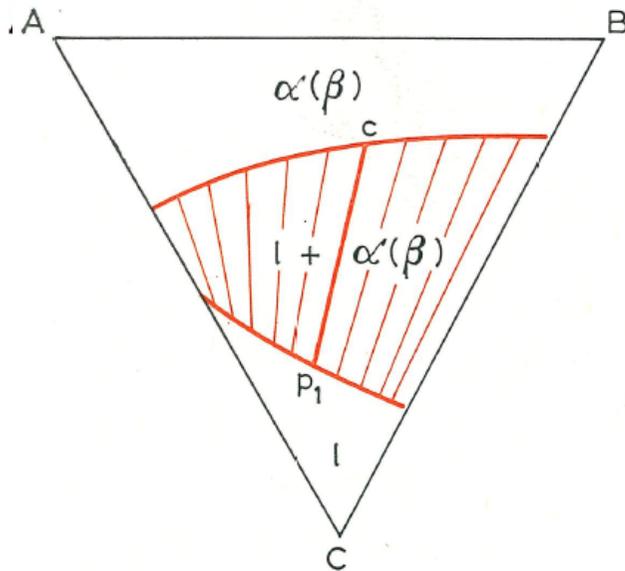
- A peritectic solubility gap in one binary system

$PP_1$ : monovariant curve for liquid

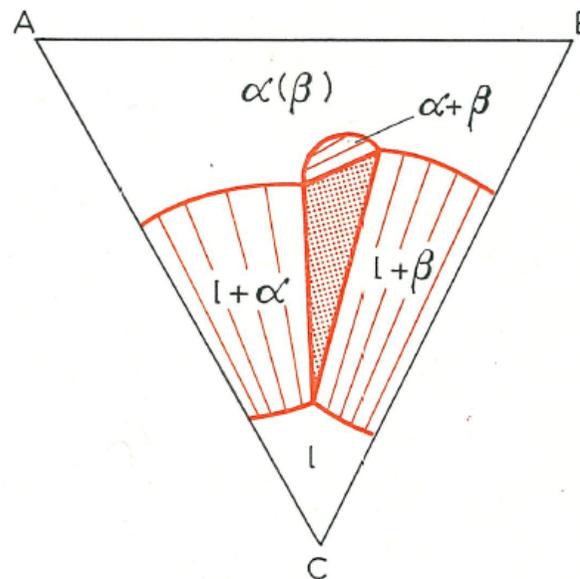
Points  $P_1$  and  $c$  lie at the same temperature and the line  $P_1c$  is a degenerate tie triangle.



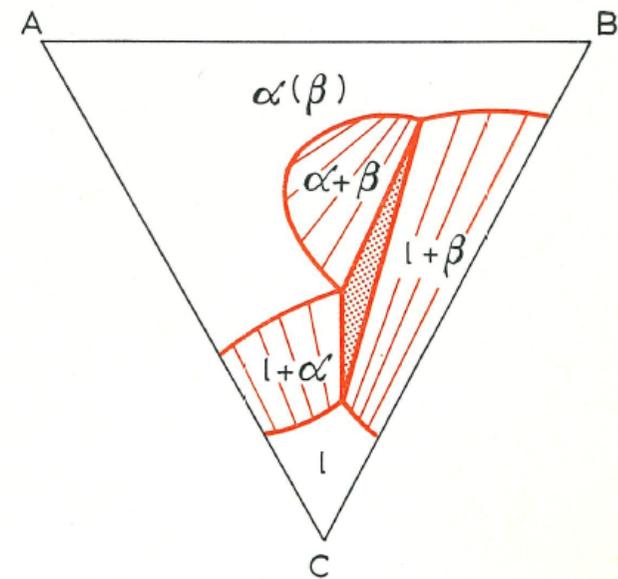
isothermal section



$$T = P_1$$



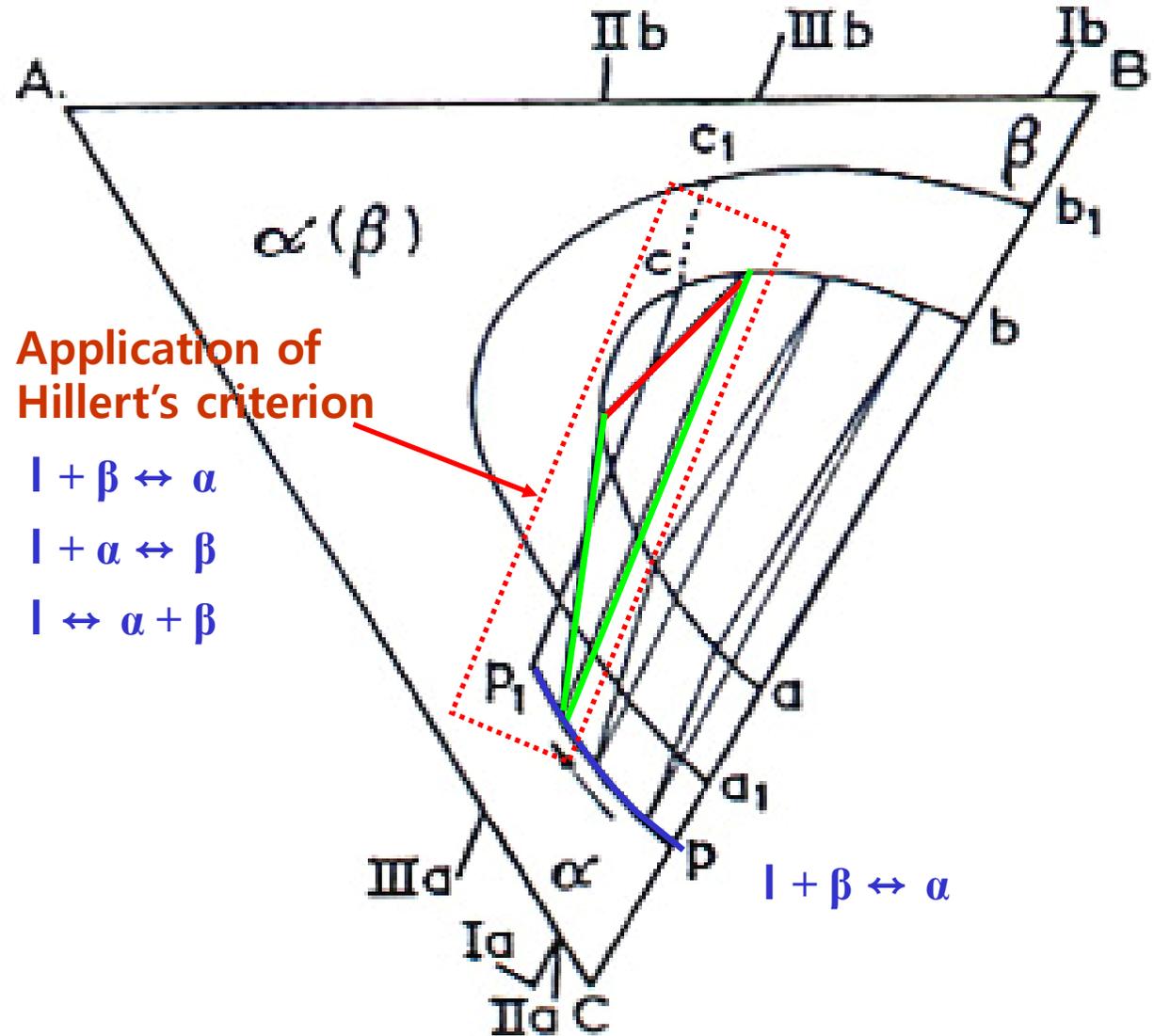
$$P_1 > T_1 > P$$



$$T_1 > T_2 > P$$

## 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

- A peritectic solubility gap in one binary system



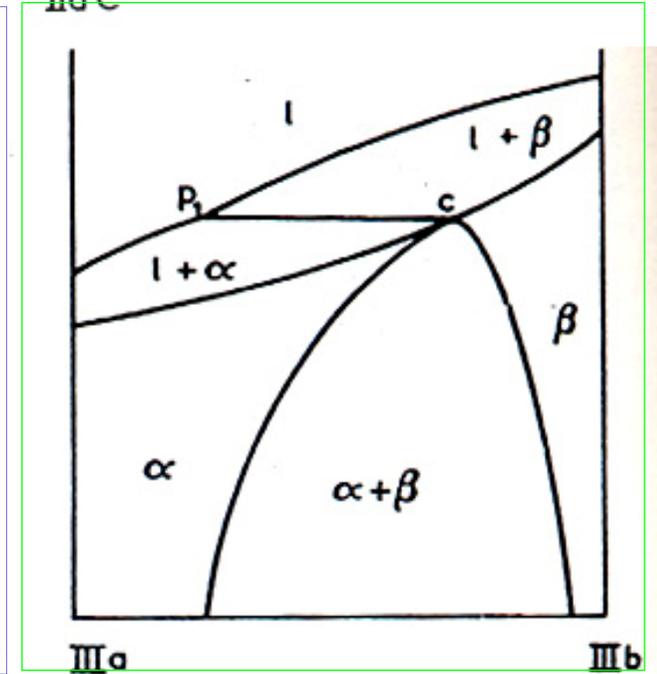
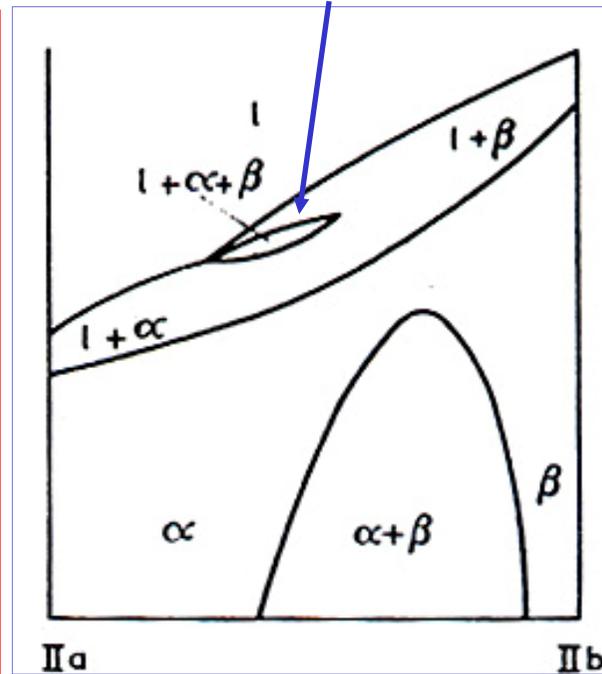
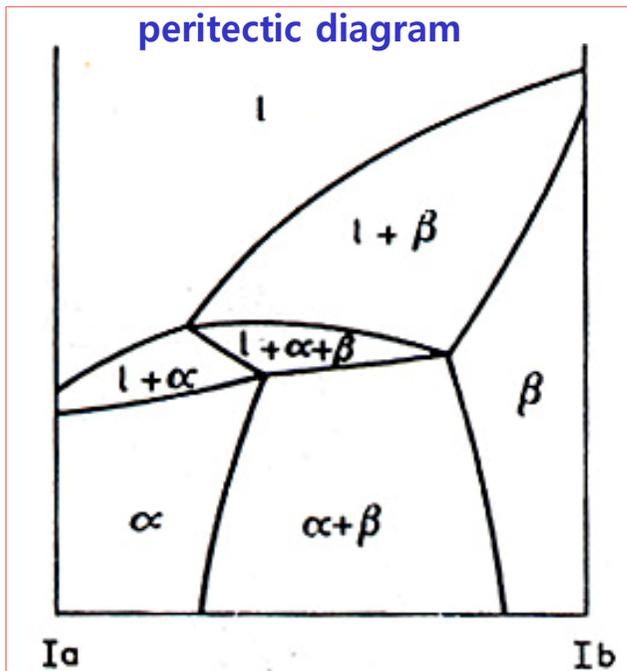
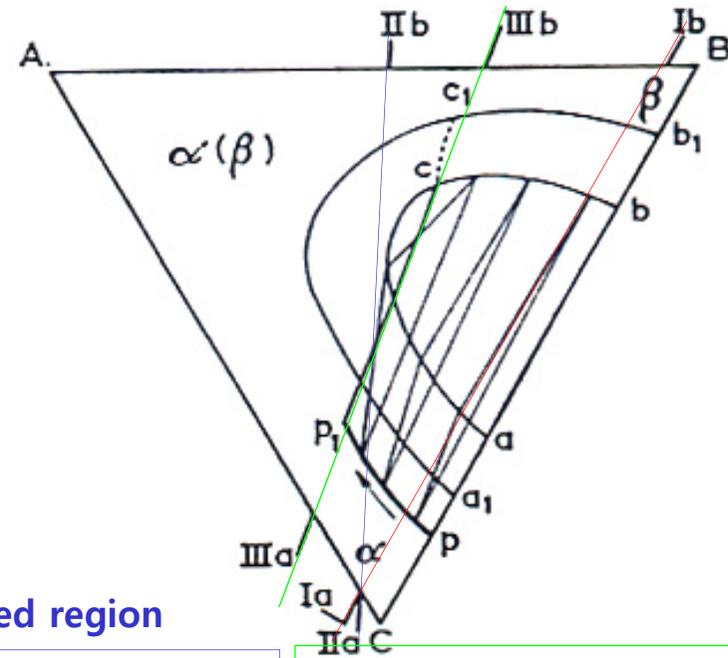
## 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

- A peritectic solubility gap in one binary system

<vertical section>

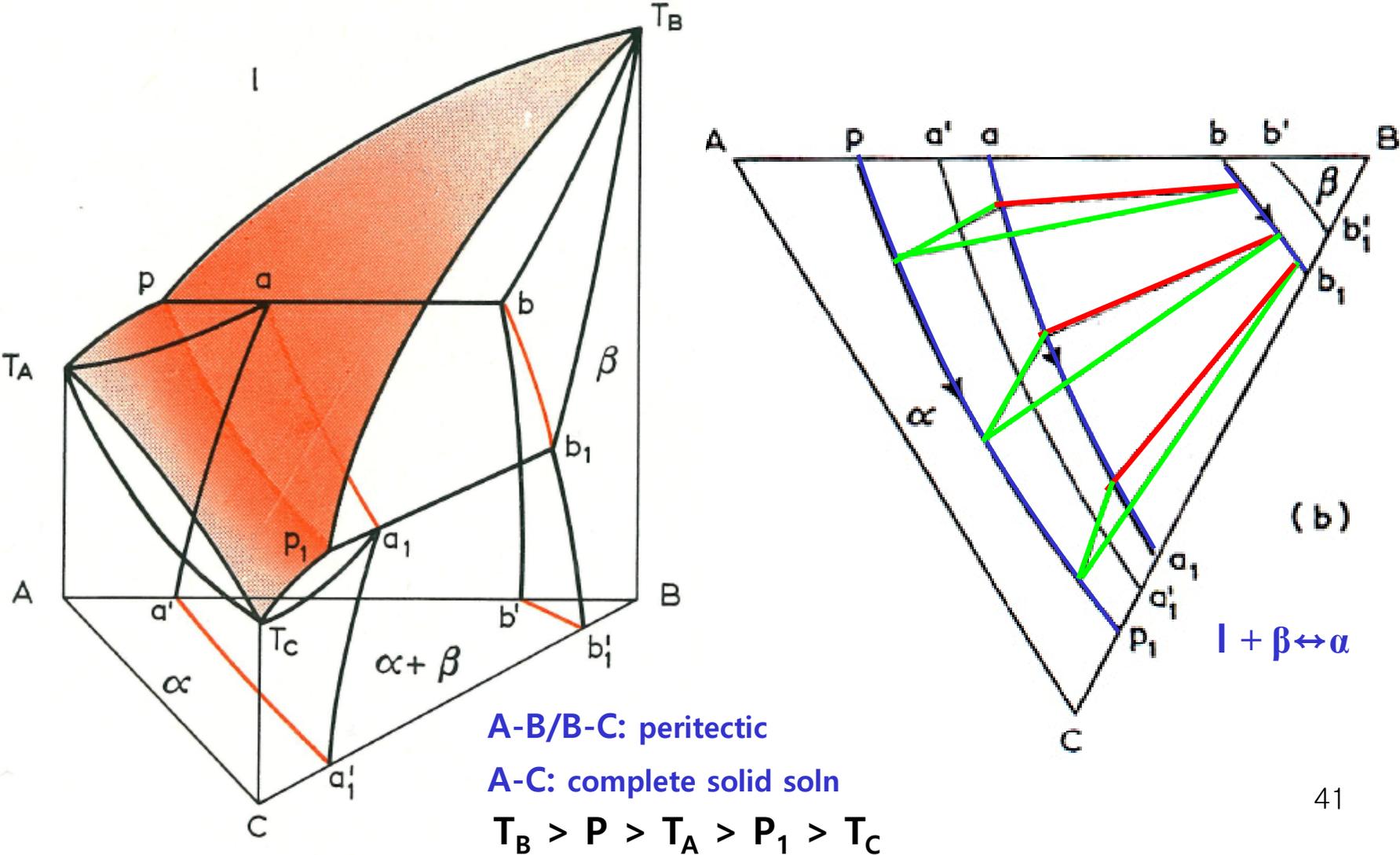
Similar to the binary peritectic diagram

loop shaped region



# 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

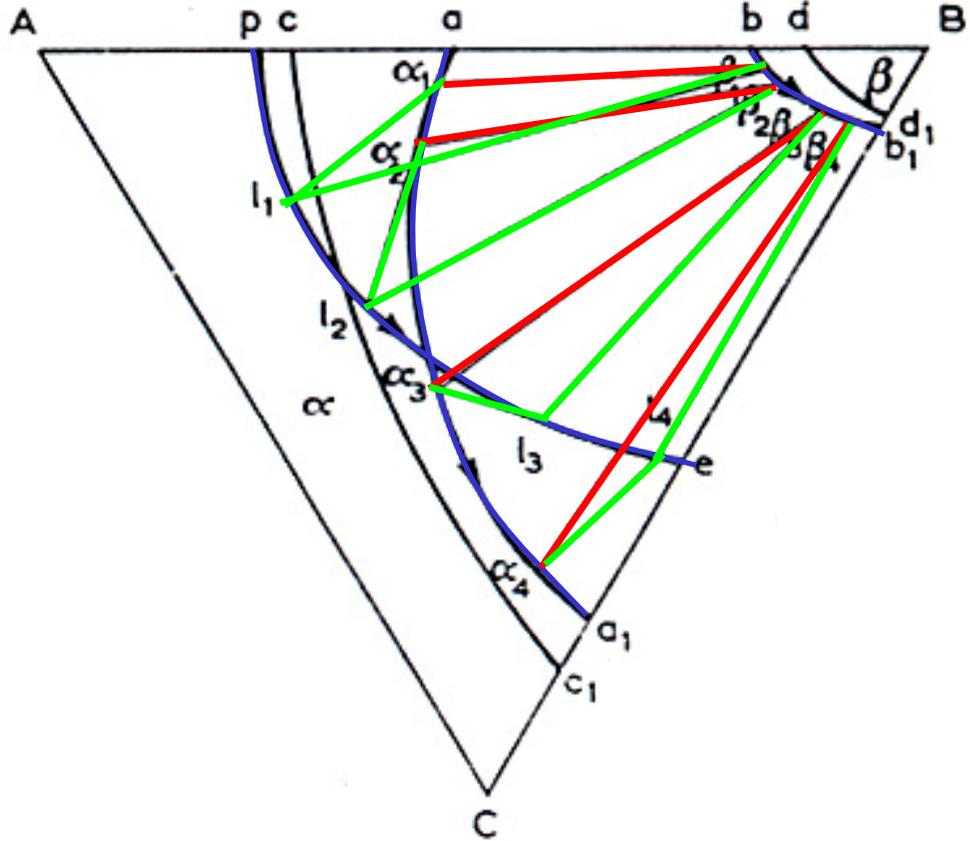
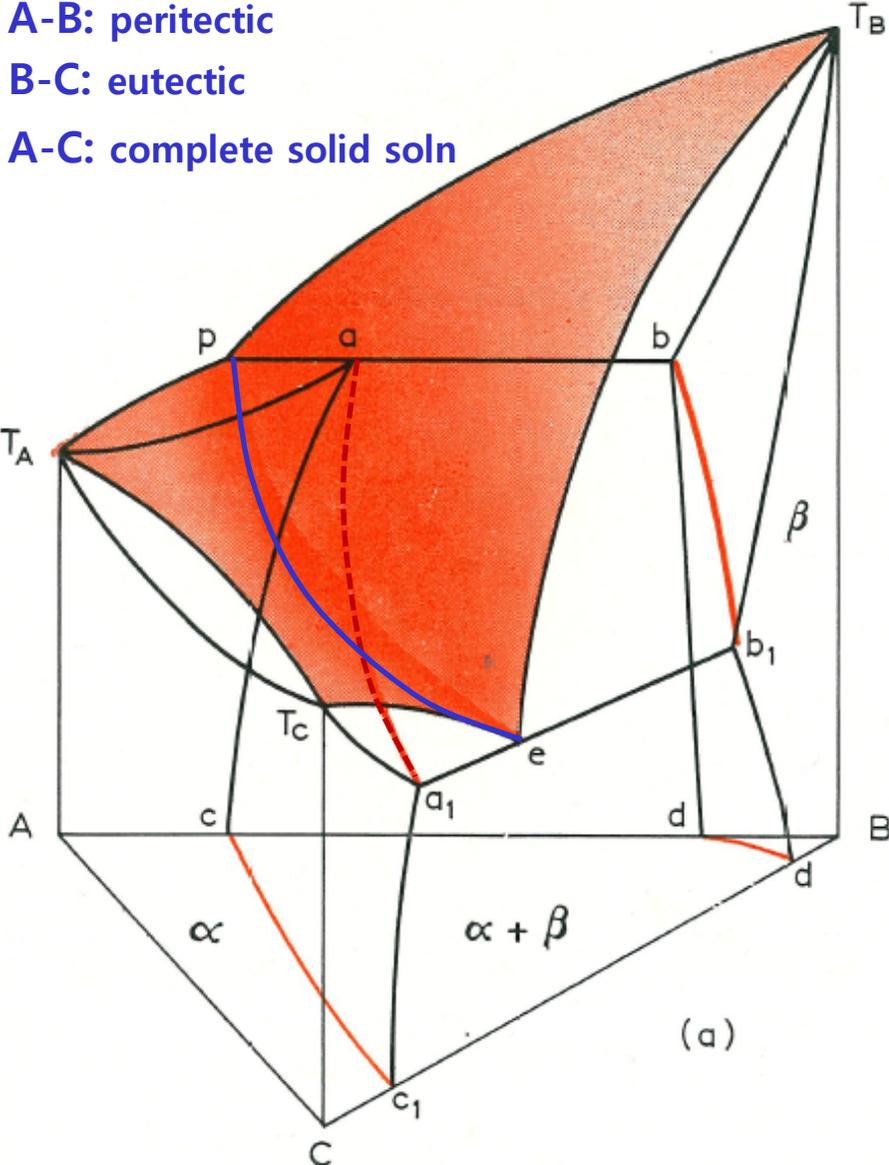
- A peritectic solubility gap in two binary system



# 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

- A transition from a binary eutectic to a binary peritectic reaction

A-B: peritectic  
 B-C: eutectic  
 A-C: complete solid soln

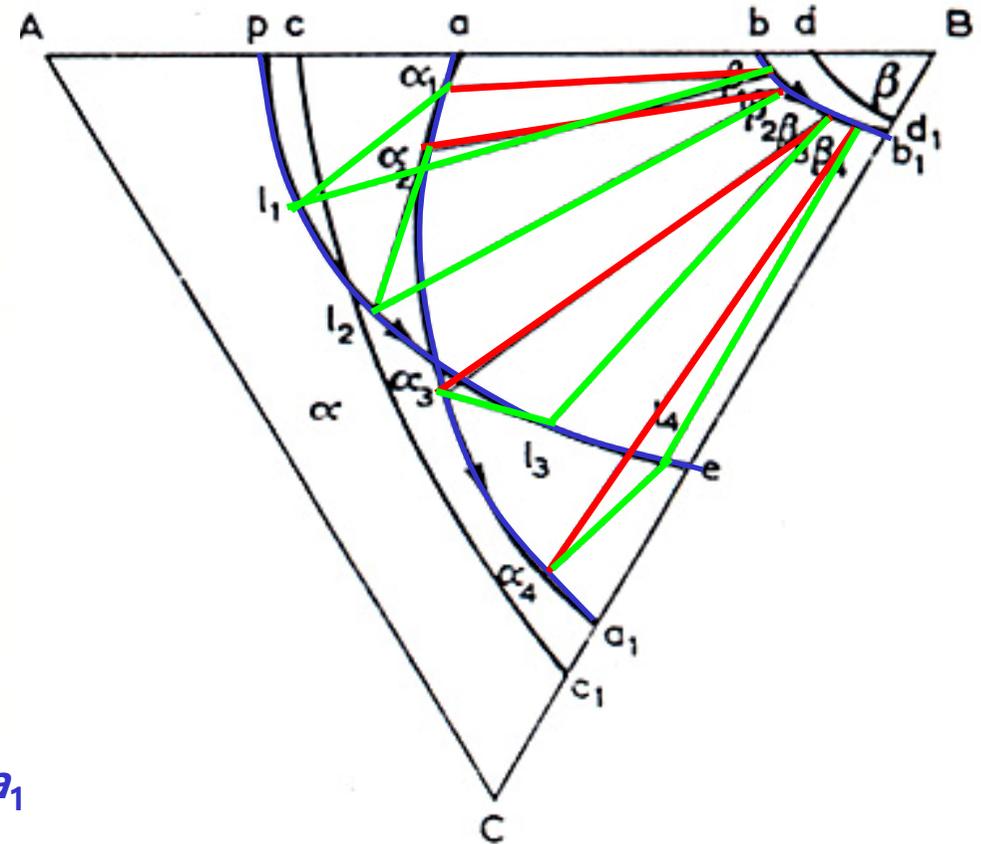
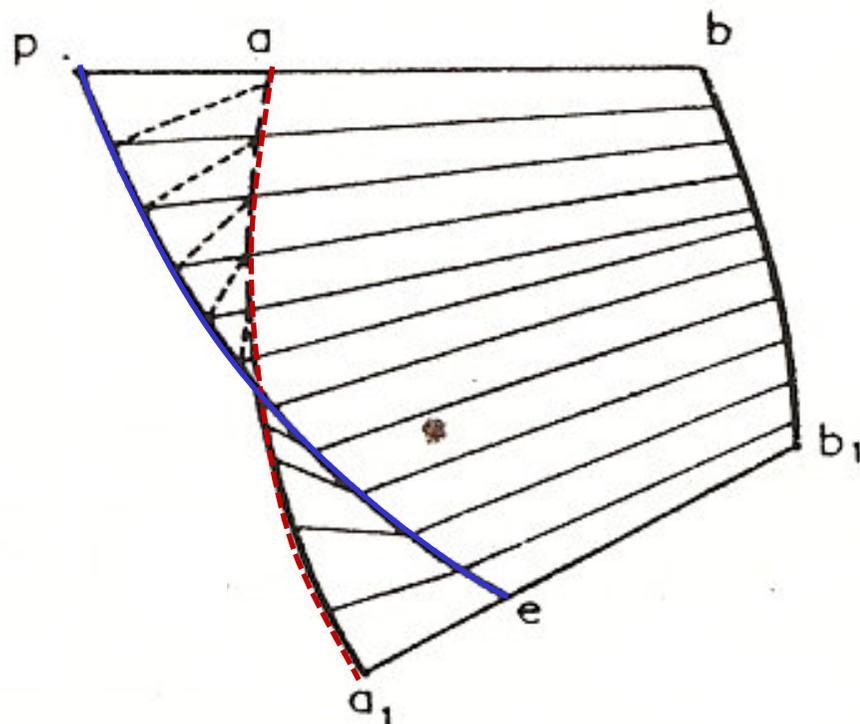


$$T_B > P > T_A > T_C > e$$

## 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

- A transition from a binary eutectic to a binary peritectic reaction

$$T_B > P > T_A > T_c > e$$



- curve  $pe$  always lies above curve  $aa_1$
- Tie lines are drawn on the  $l\beta$  and  $l\alpha$  surfaces only.
- By Hillert to show that the transition from a peritectic to a eutectic reaction does not occur at a unique temperature.

## 9.4. THREE-PHASE EQUILIBRIUM INVOLVING PERITECTIC REACTIONS

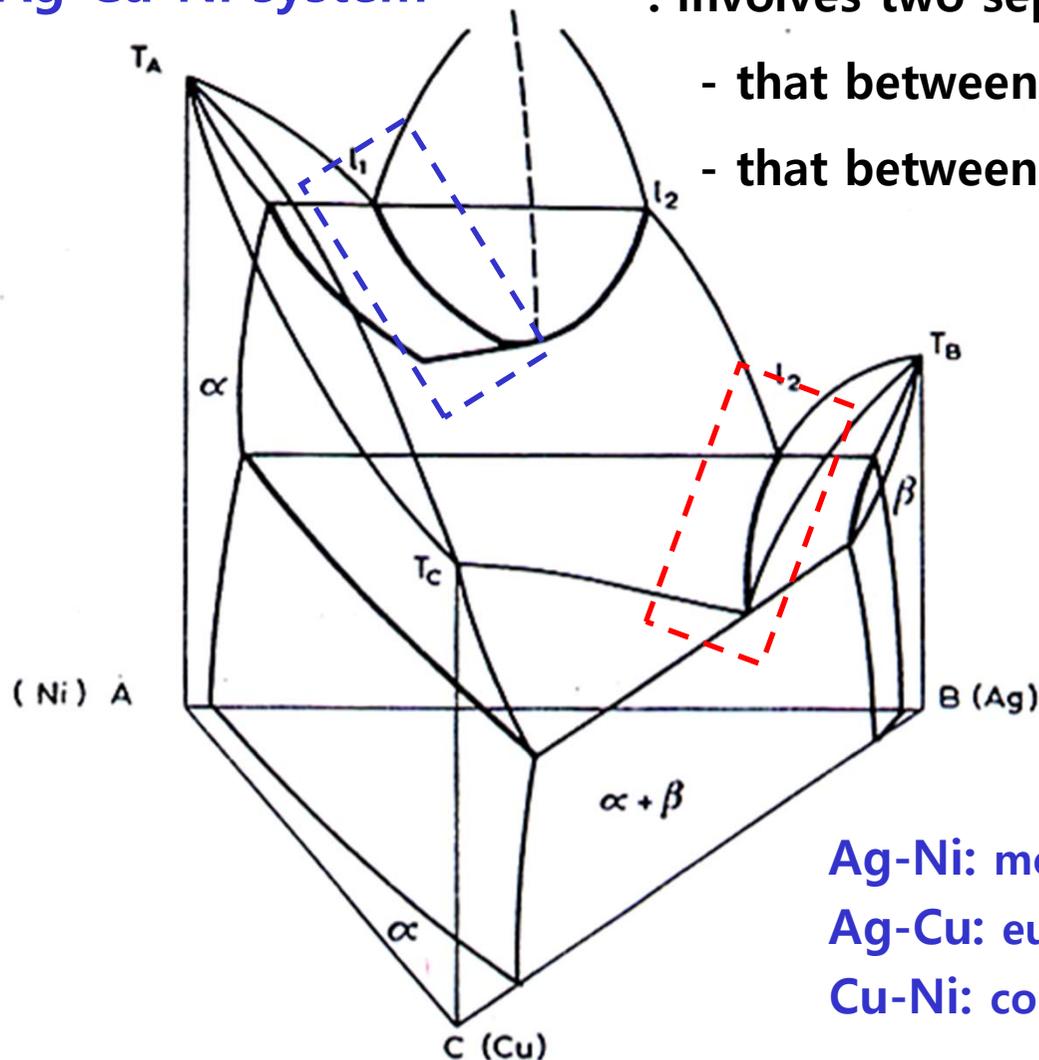
- Binary Monotectic, syntectic and metatectic reactions in combination with each other as well as with binary eutectic and peritectic reactions.

- **Ag-Cu-Ni system**

: involves two separate three phase equilibria

- that between  $\alpha$ ,  $l_1$  and  $l_2$ , and

- that between  $\alpha$ ,  $\beta$  and  $l_2$



Ag-Ni: monotectic

Ag-Cu: eutectic

Cu-Ni: complete solid soln