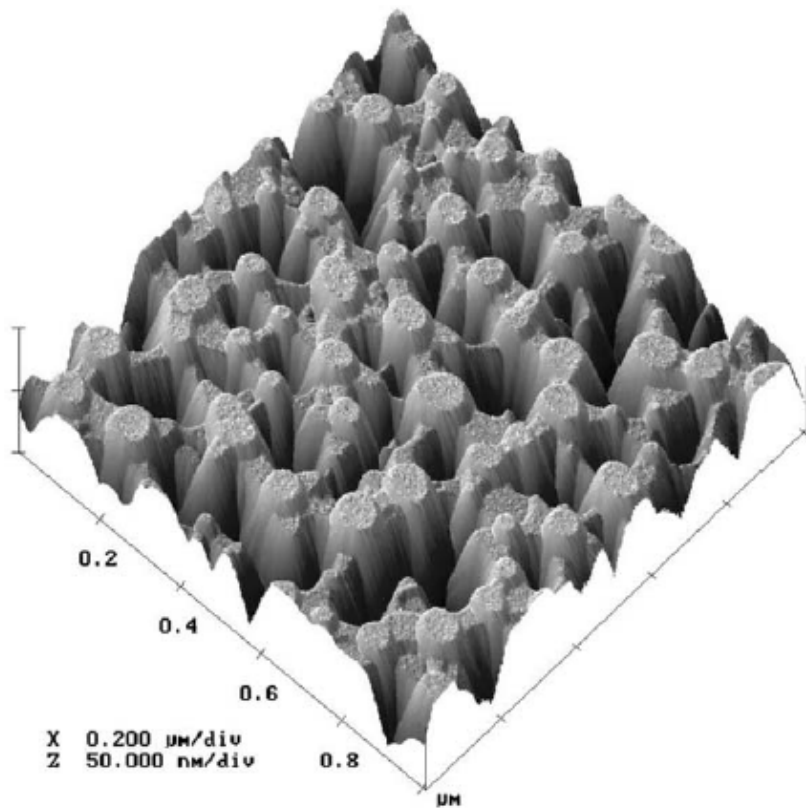


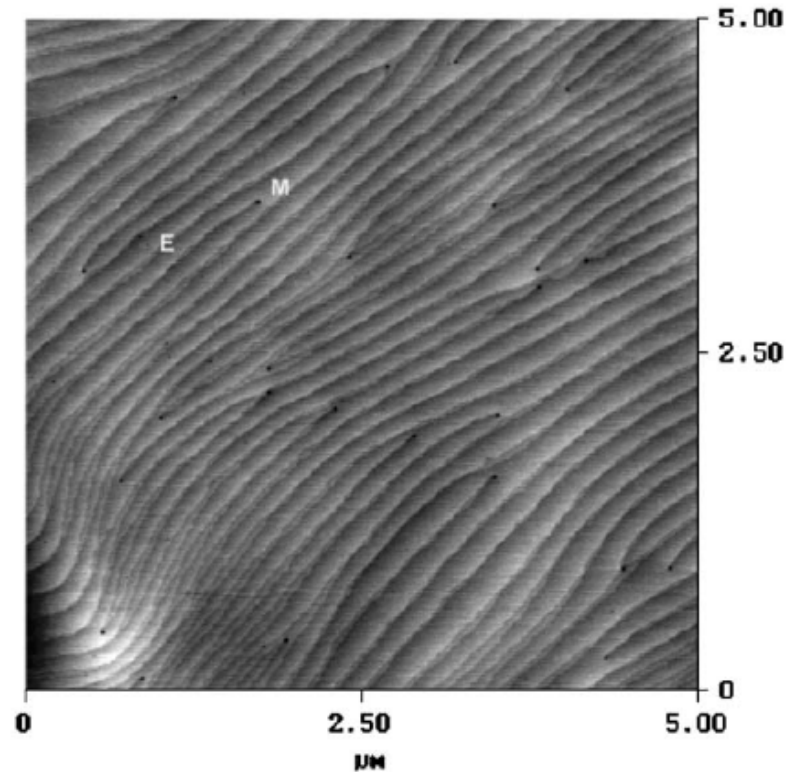
# 화합물 반도체 ( II-4 ) Heterostructure Growth

2007 / 가을 학기

# Reduced Dislocation Densities with $\text{SiH}_4$ Treatment



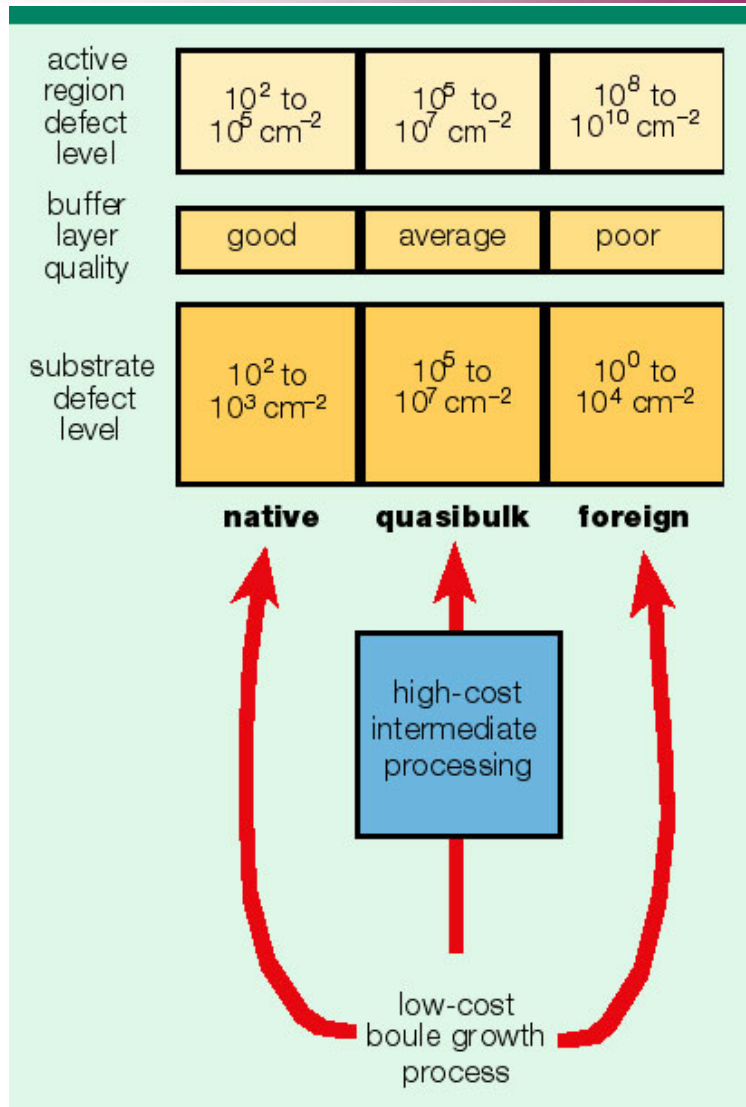
Etched surface with  $\text{SiH}_4$  treatment  
at  $1100^\circ\text{C}$  for 300s



TD densities  $\sim 1 \times 10^8 \text{ cm}^{-2}$   
(without  $\text{SiH}_4$  treatment,  $\sim 1 \times 10^9 \text{ cm}^{-2}$ )

Ref.: K. Pakula, et al., Journal of Crystal Growth 267, pp. 1–7, 2004

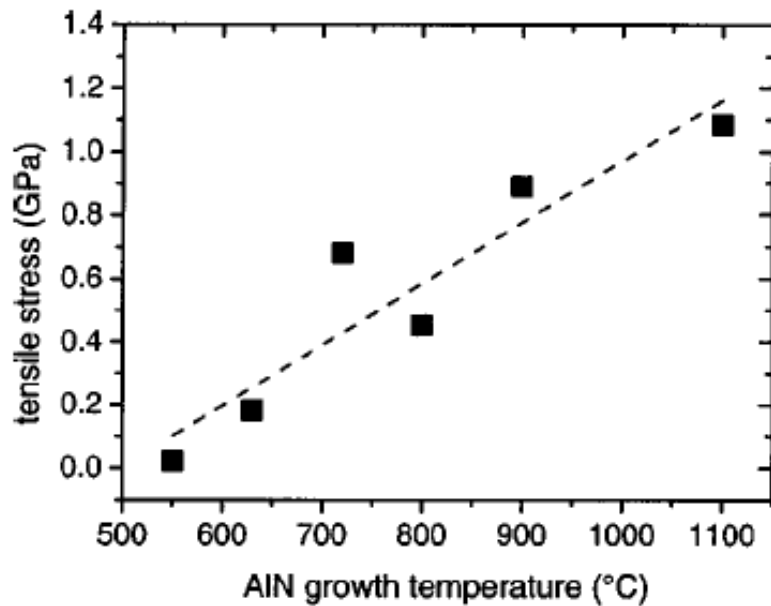
# Bulk GaN Growth



- Bulk GaN growth with pressures of 15,000 atm and temperatures of 1600°C → 10 mm in diameter with TD densities of 100 cm<sup>-2</sup> (commercialized by Topgan for research)
- 1~2 inch bulk AlN growth with sublimation recondensation process → appropriate for Al-rich AlGaN growth for DUV laser diodes (commercialized by Crystal IS)

Ref. : Compound Semiconductor Magazine, July, Oct. 2004

# Stress Reduction with Low-Temperature AlN Interlayer



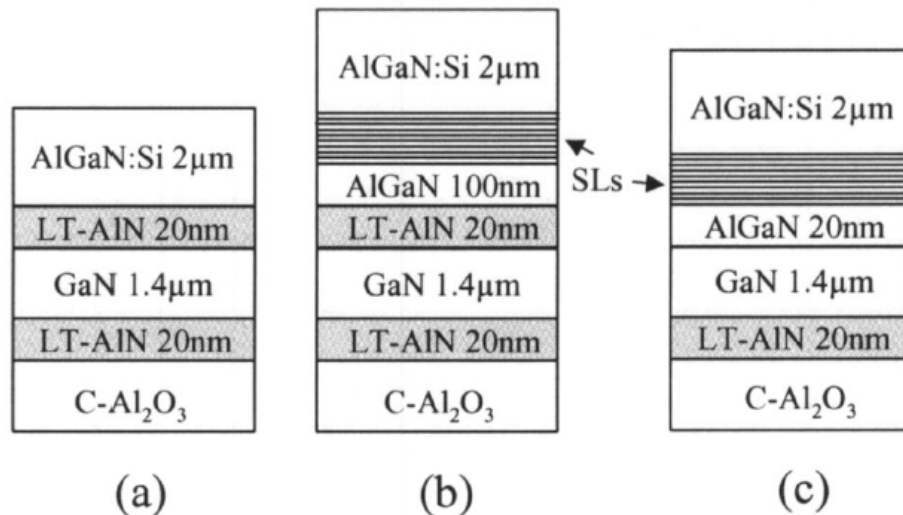
Average tensile stress of 1.3 $\mu$ m thick GaN layer grown on 12nm thick AlN buffer

- Relaxed AlN buffer at low temp growth

Sample	$T_{\text{AlN}}$ [°C]	Curvature radius [m]	Total stress [GPa]	$a\text{-AlGaN}$ [Å]	$a\text{-GaN}$ [Å]
A	630	14.7	-0.01	3.1653	3.1899
B	900	7.9	0.46	3.1665	3.1923
C	1145	2.9	1.13	3.1923	3.1923

Ref.: J. Blasing, et al., Appl. Phys. Lett., pp. 2722–2724, 7 October 2002

# AlGaN Grown on GaN with Various Interlayers

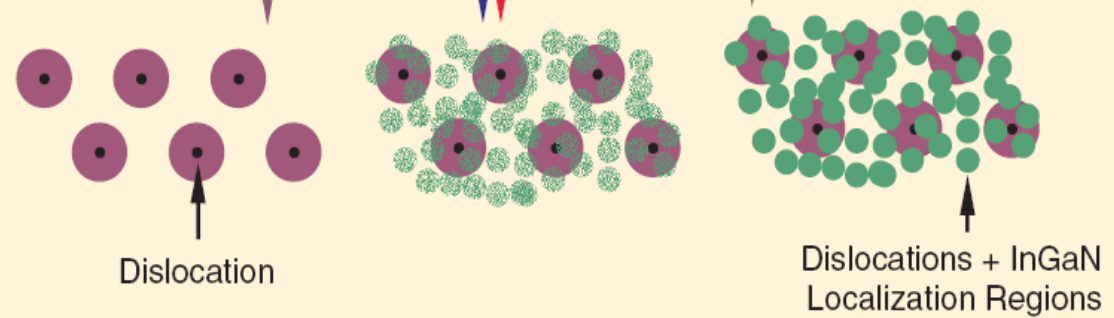
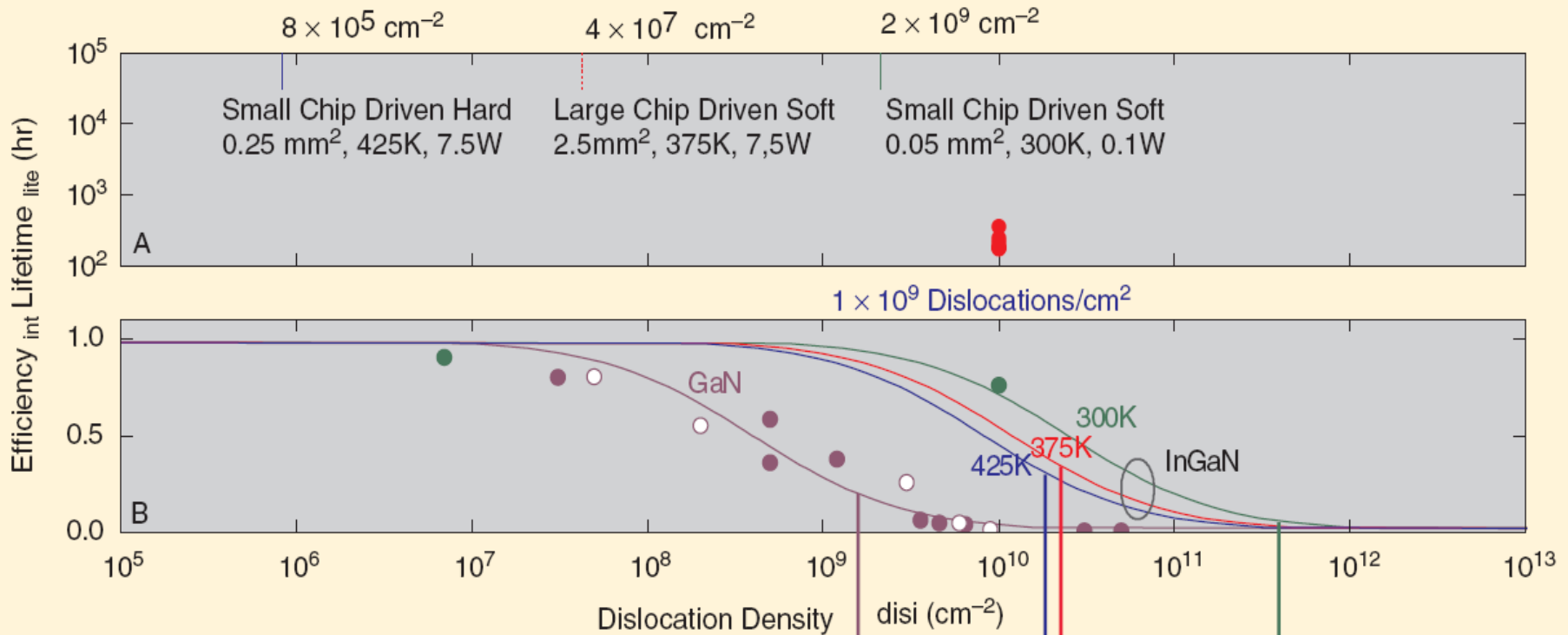


**Ten period  
AlN 4nm/AlGaN 36nm  
Superlattices  
(Al<sub>0.2</sub>GaN)**

Interlayer	Surface	FWHM in (0002) $\omega$ scan (arcmin)	FWHM in (2024) $\omega$ scan (arcmin)	Density of etch pits (cm <sup>-2</sup> )	Mobility (cm <sup>2</sup> /V s) and concentration (cm <sup>-3</sup> )
No	Crack network	9.5	14.6		
LT-AlN	Several cracks	12.4	18.2	$6 \times 10^9$	
LT-AlN and SLs	Crack free	12.1	16.9	$4 \times 10^9$	Mob.: 87, Con.: $3.0 \times 10^{18}$
SLs	Crack free	6.4	11.8	$2 \times 10^9$	Mob.: 161, Con.: $2.5 \times 10^{18}$
Directly on Sapphire	Crack free	14.6	23.3	$7 \times 10^9$	Mob.: 38, Con.: $2.2 \times 10^{18}$

**Ref.: Q. C. Chen, et al., Appl. Phys. Lett., pp. 4961–4963, 23 December 2002**

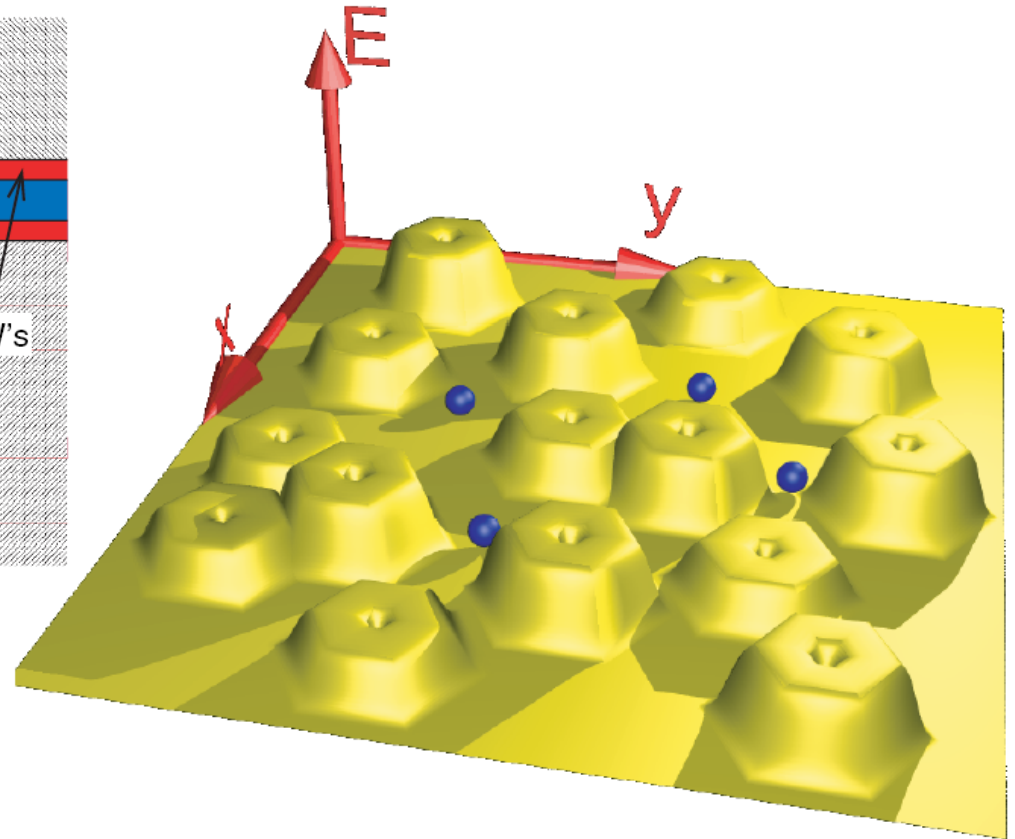
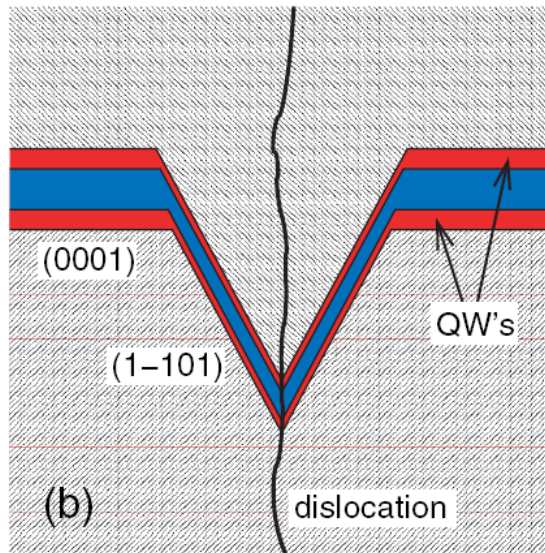
# Effects of Dislocations on Light Emission Efficiency



Ref. : Jeff Y. Tsao, "Solid State Lighting," IEEE Cir. & Dev. Magazine, pp. 28-37, May/June 2004

# *New Model for Suppression of Nonradiative Recombination*

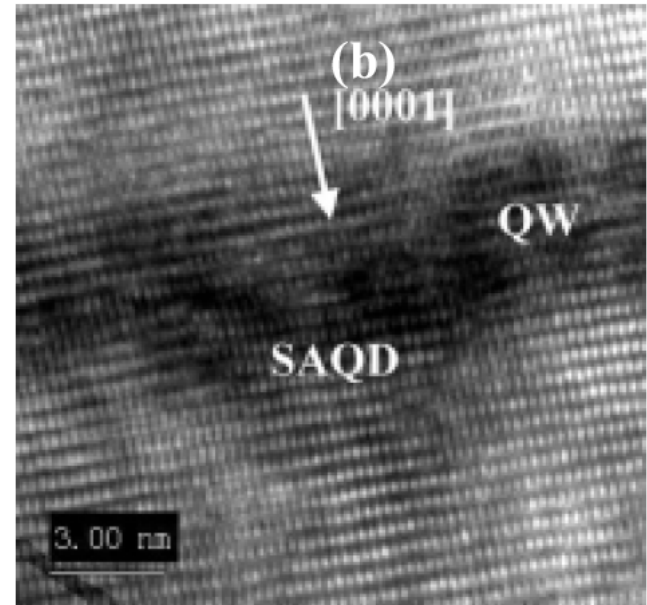
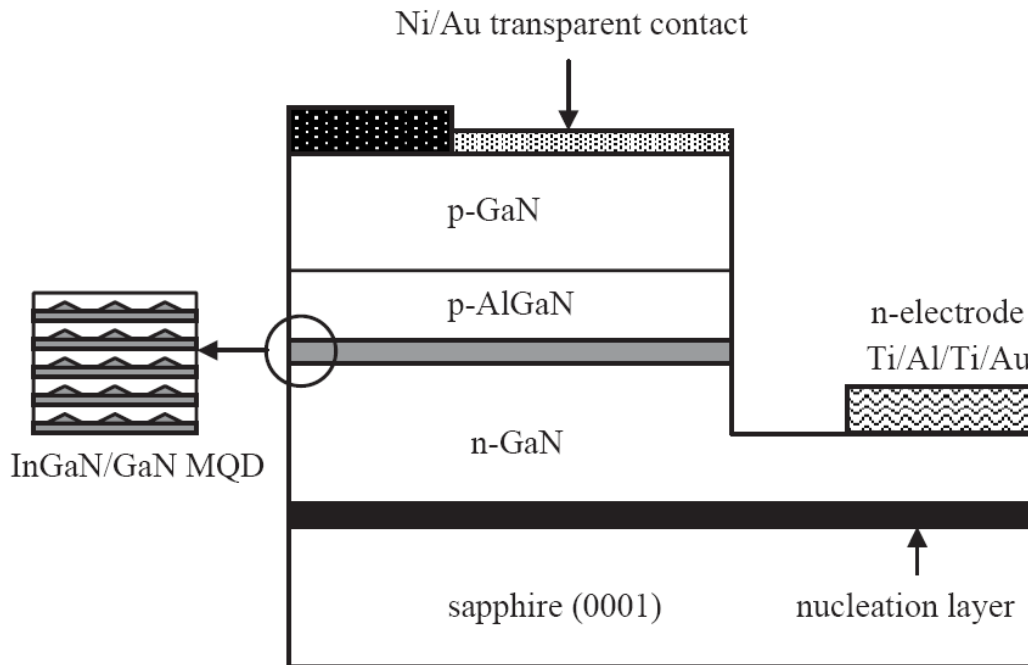
High resolution TEM  
Near-field micro-PL



- hexagonal V-shaped pits  
decorating the defects
- narrow sidewall QW  
⇒ large effective band gap  
⇒ suppressing nonradiative  
recombination

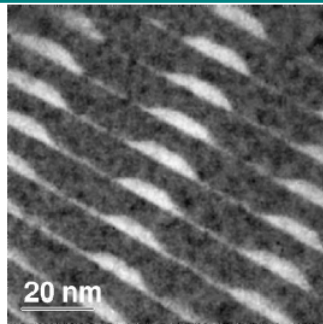
Ref) A. Hangleiter, et al., Phys. Rev. Lett. 95,  
127402, 2005

# InGaN Multi-Quantum Dot LED ( I )



- 2.4nm InGaN well/15nm thick GaN barrier

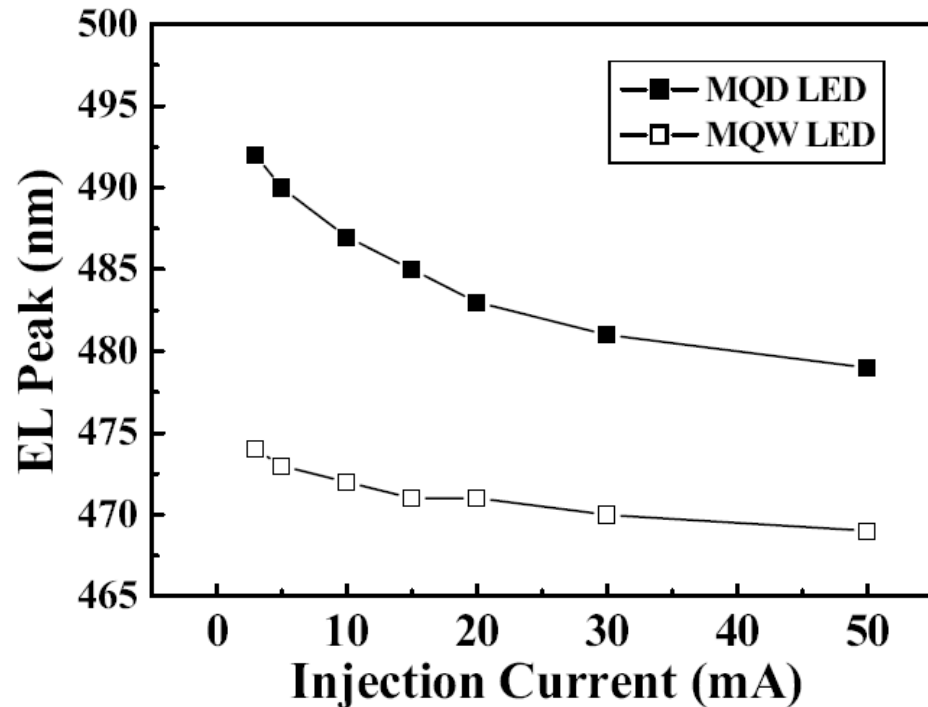
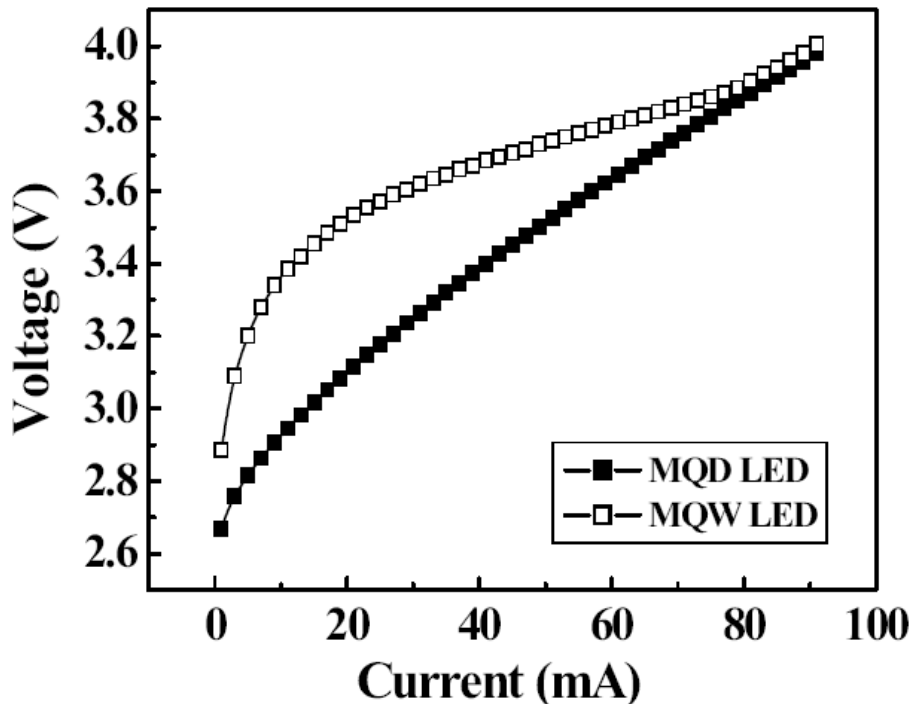
- GaN QDs embedded inside an AlN matrix (large stress)
- Plasma MBE-grown (APL 87, 203112 2005)



- interrupted growth to achieve InGaN dots-in-a-well structure
- typically a pyramidal dot with a 3nm height and a 10nm diameter
- QD density :  $10^{10} \sim 10^{11} \text{ cm}^{-2}$



## InGaN Multi-Quantum Dot LED ( II )

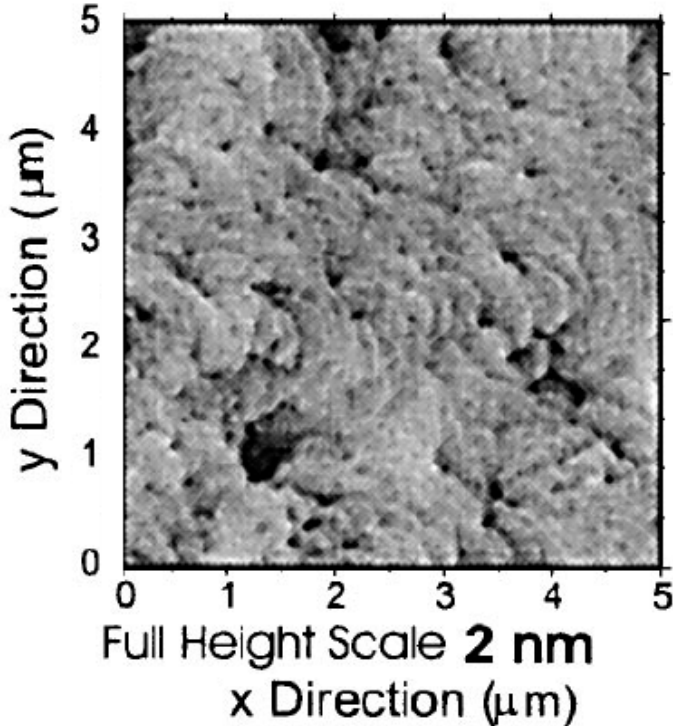


- Smaller forward voltage 3.1V in MQD LED ➔ cellar phone applications ( ? )
- Large EL blueshift in QD reveals that deep localization of excitons (or carriers) originates from QDs.

Ref) L. W. Ji, et al., Phys. Stat. Sol. (c) 1, No. 10, pp. 2405, 2004

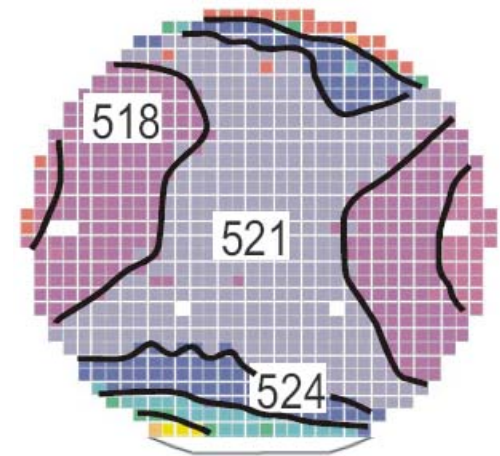
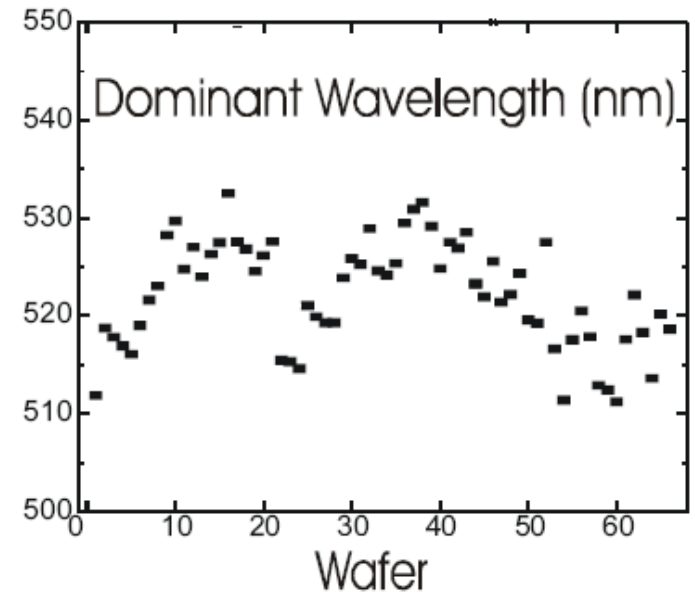
# High Performance Green LED with Smooth Surface

## Very Smooth MQW Active Regions (AFM Surface Morphology)



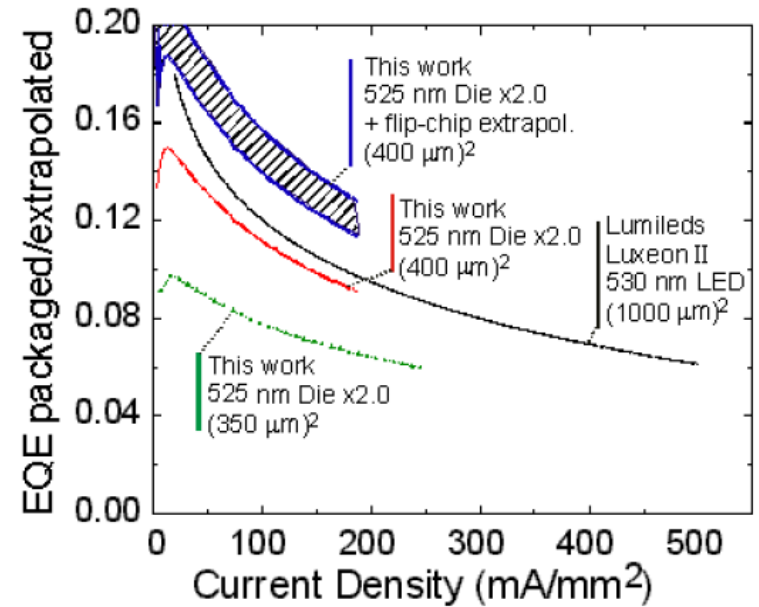
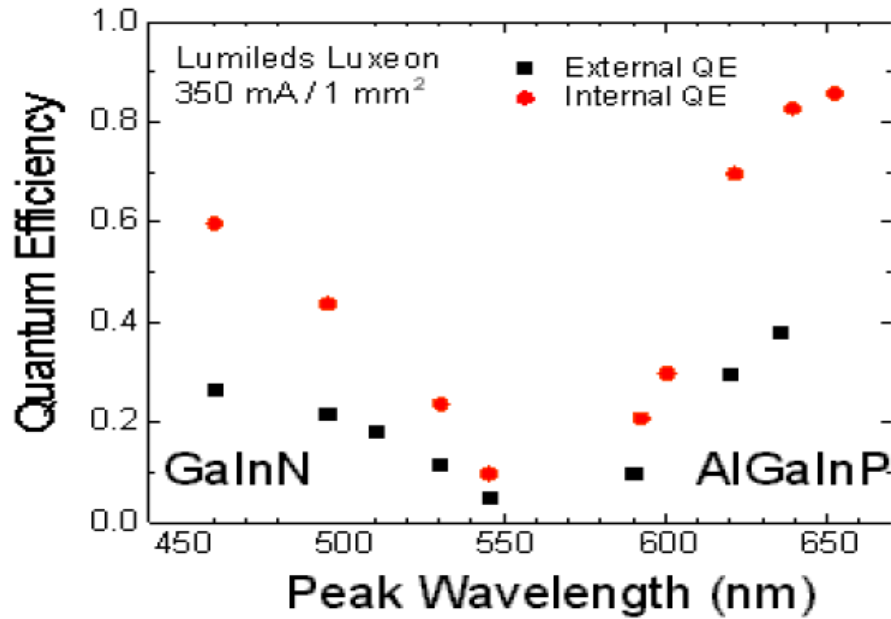
- high performance green LED (525nm)
- uniform & 98% high yield process ( $\lambda$ , power,  $V_F$ ,  $I_R$ )
- At 20mA,  $V_F = 3.2\text{V}$ , power=1.6mW in bare chip

- Small density of pits due to V defects ;  $3.6 \times 10^8 \text{cm}^{-2}$
- Very smooth surface morphology rms surface roughness ; 0.14nm

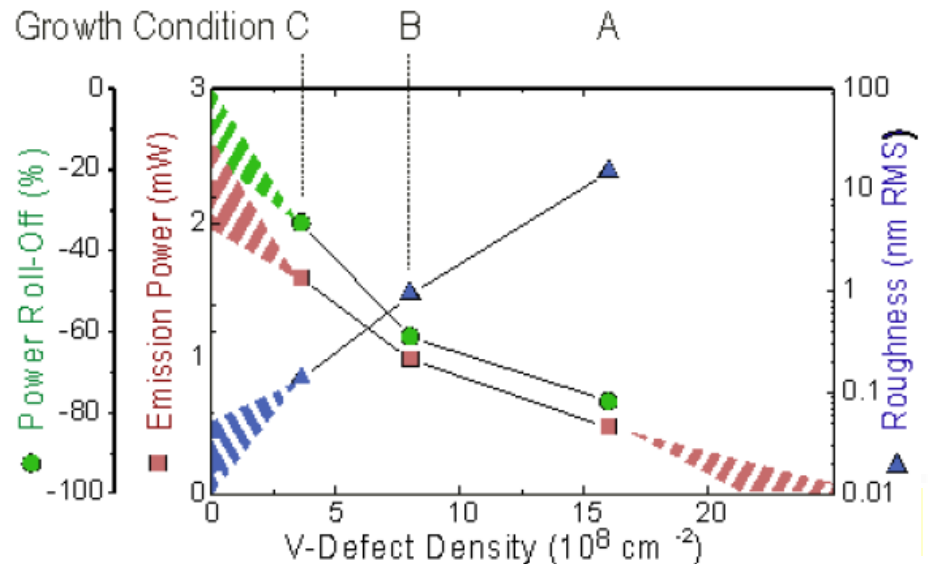


Ref : C. Wetzel, et al.,  
Phys. Stat. Sol. (c) 2 ,  
pp. 2871, 2005

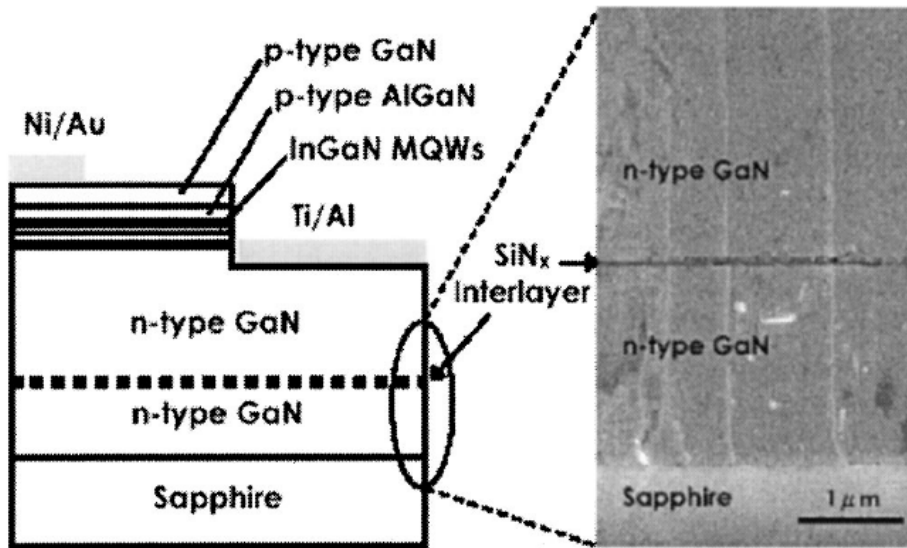
# High Power Green Light Emitting Diode



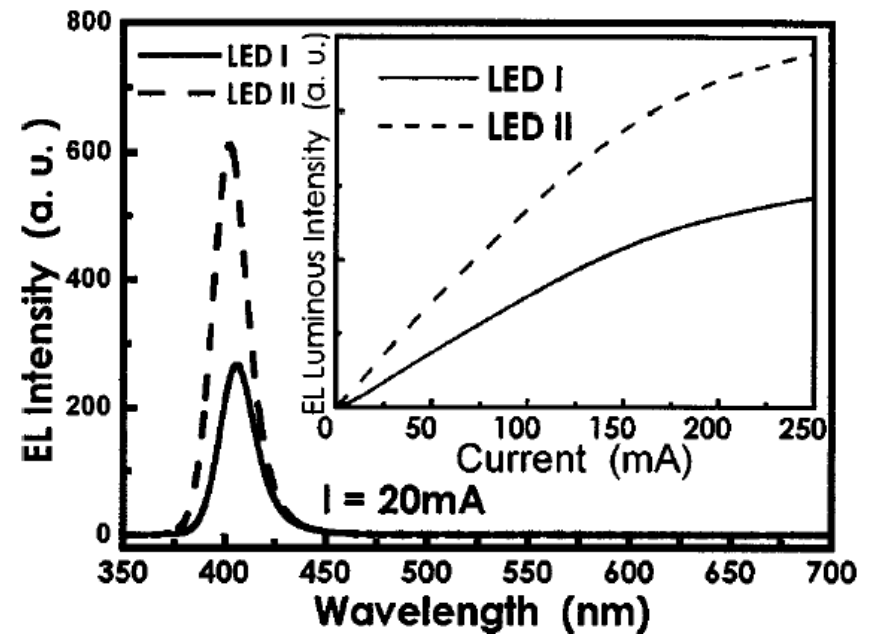
Ref : C. Wetzel, et al., MRS Internet J. Nitride Semicond. Res. 10, 2 (2005)



# Effects of Dislocations on LED Performance



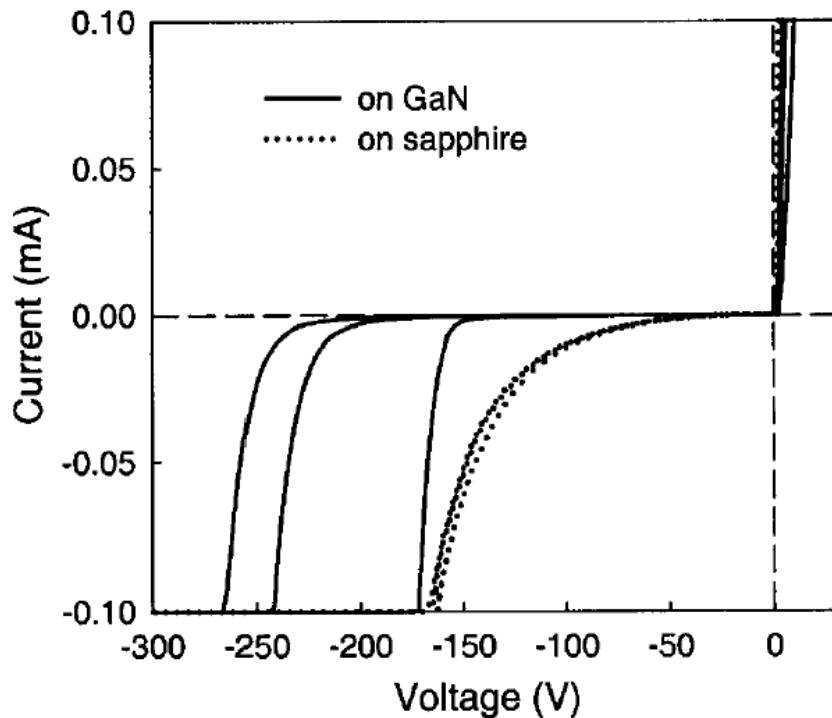
Ref. : Ru-Chin Tu, et al., Appl. Phys. Lett.  
pp. 3608-3610, 27 October 2003



# Effects of Dislocations on Reverse Leakage Current

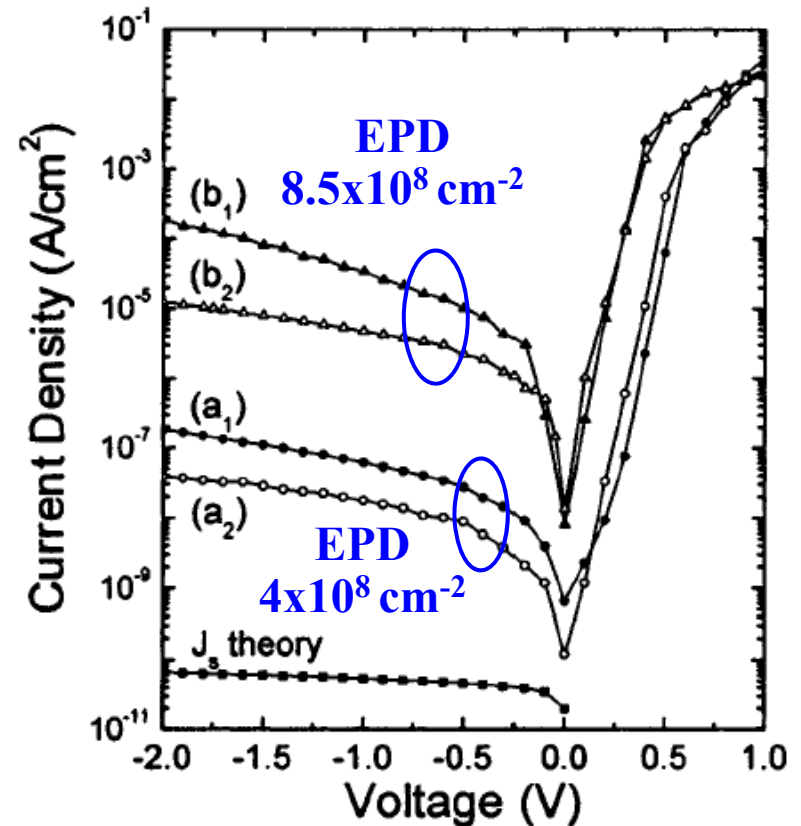
## GaN PIN Rectifier

( $4\mu\text{m } 1 \times 10^{17} \text{cm}^{-3} \text{N}^-$  Region)



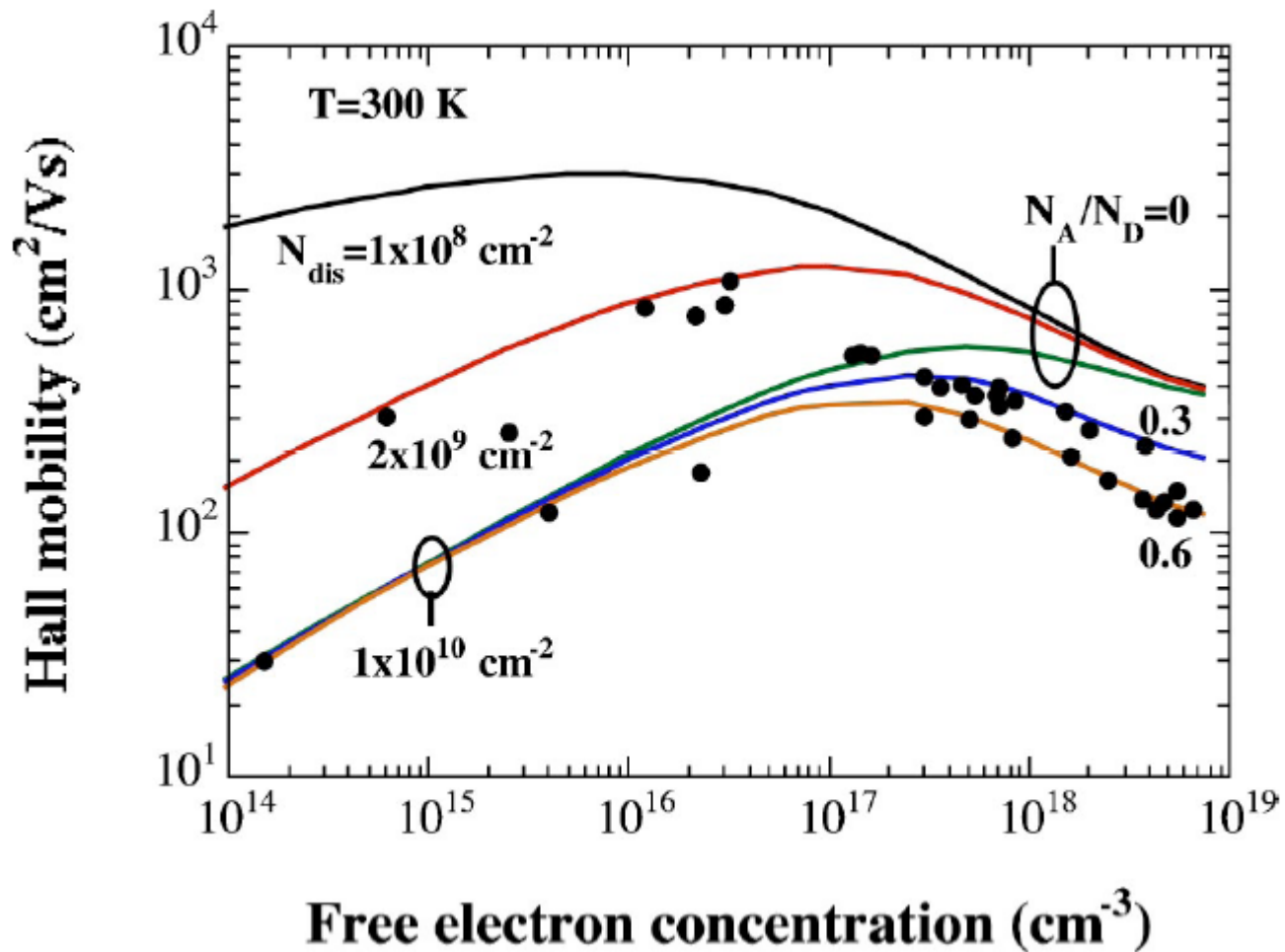
Ref. : X. A. Cao, et al., *Appl. Phys. Lett.*  
87, 053503, 2005

## Au/n-GaN Schottky diodes



Ref. : Y. Huang, et al., *J. Appl. Phys.*,  
pp. 5771-5775, 1 Nov. 2003

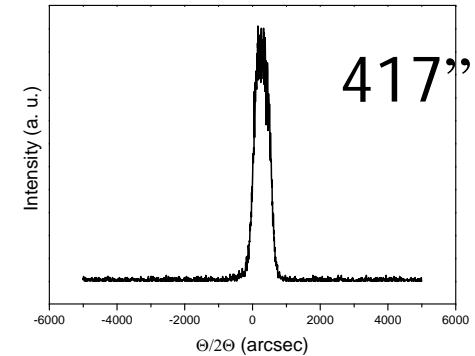
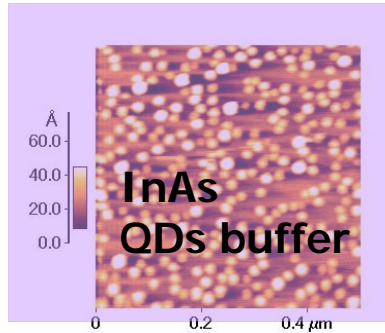
# Effects of Dislocations on Hall Mobility



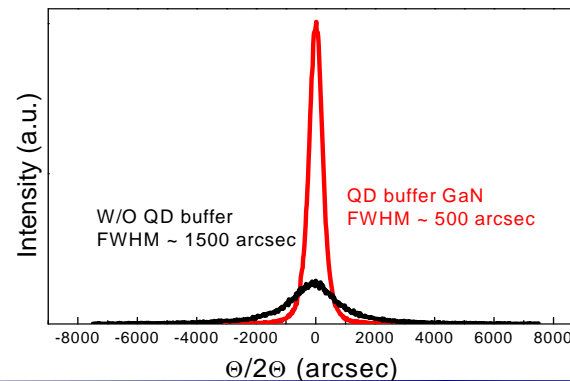
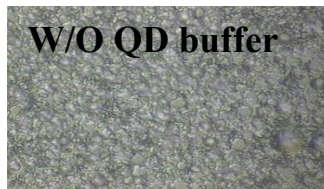
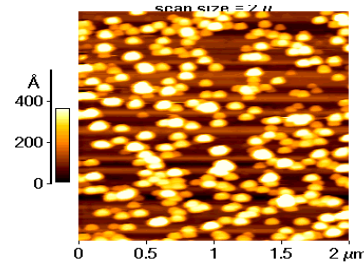
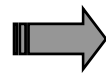
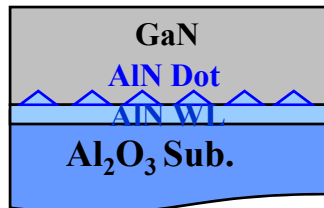
Ref. : M. N. Gurusinge, et al., Physical Review B 67, 235208, 2003

# Hetero-Epitaxy with the Defect-Free QD Buffer layer

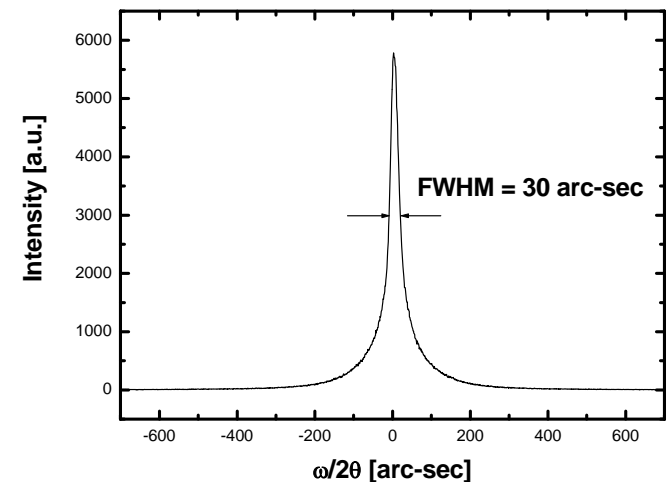
## GaSb epilayer on GaAs sub.



## GaN epilayer on Sapphire sub.



The recent GaN epilayer quality after further optimization



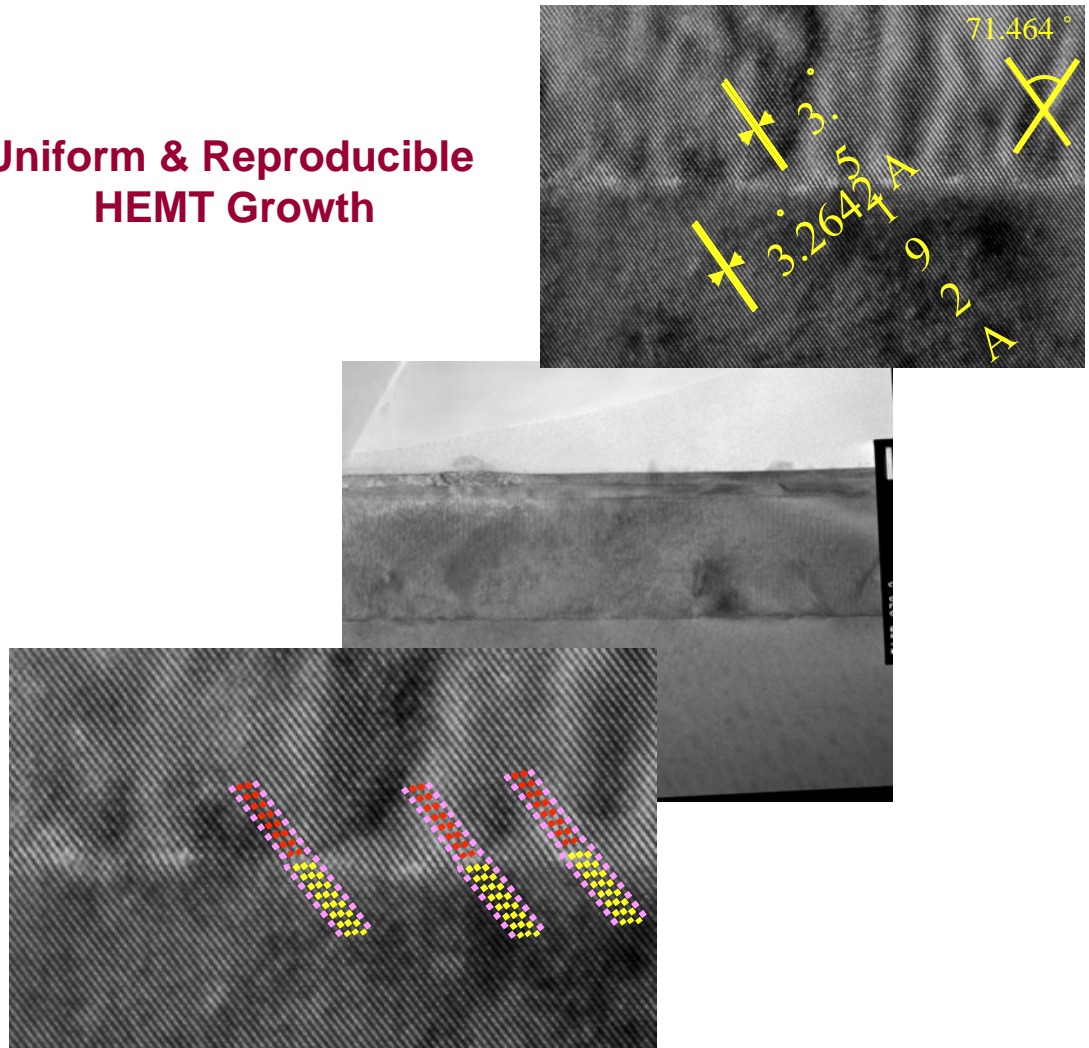
# Thin barrier InAs HEMT grown using InAs QD/GaSb Buffer

InAs QD	InAs	50Å
	Al <sub>0.2</sub> GaSb	100Å
	Δ-doping (5e17)	
	Al <sub>0.2</sub> GaSb	100Å
	InAs	100Å
	GaSb	0.5um
	GaAs	0.5um
SI-GaAs		

$$n_s = 3.33 \times 10^{12} \text{ cm}^{-2}$$

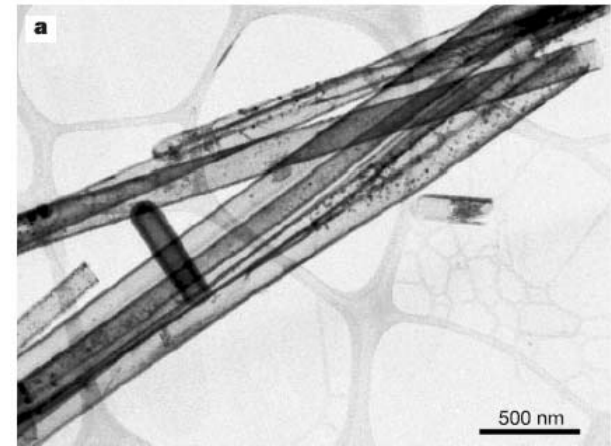
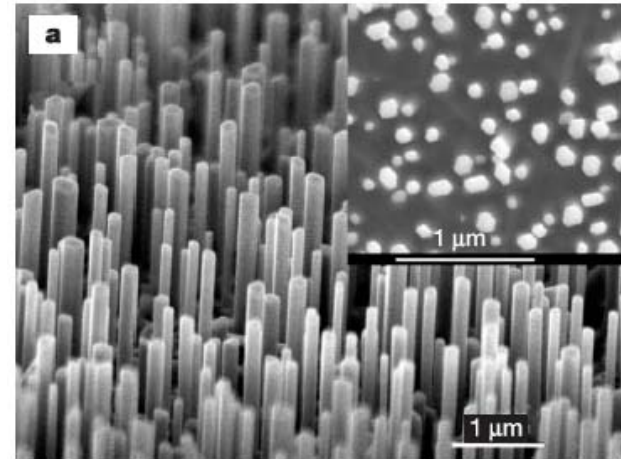
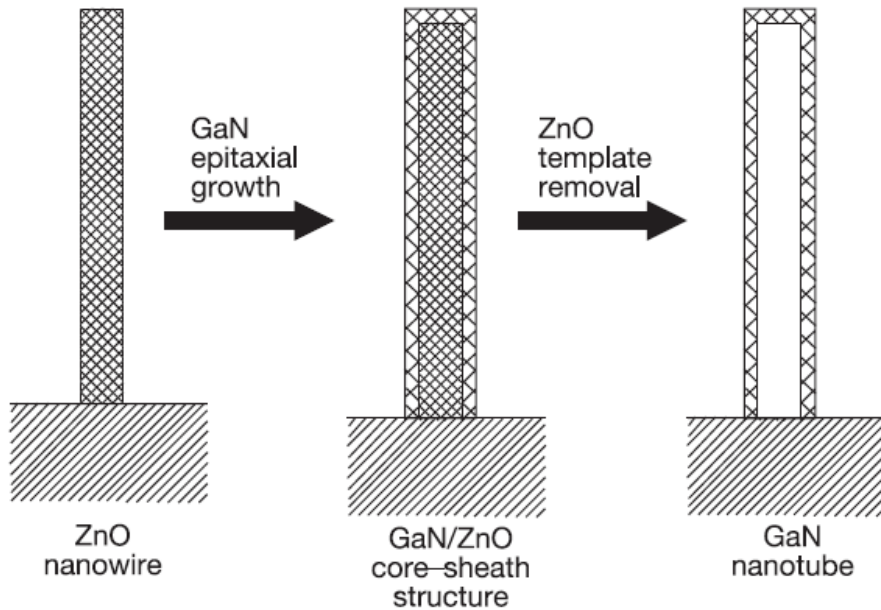
$$\mu = 8064 \text{ cm}^2/\text{V-s}$$

Uniform & Reproducible  
HEMT Growth



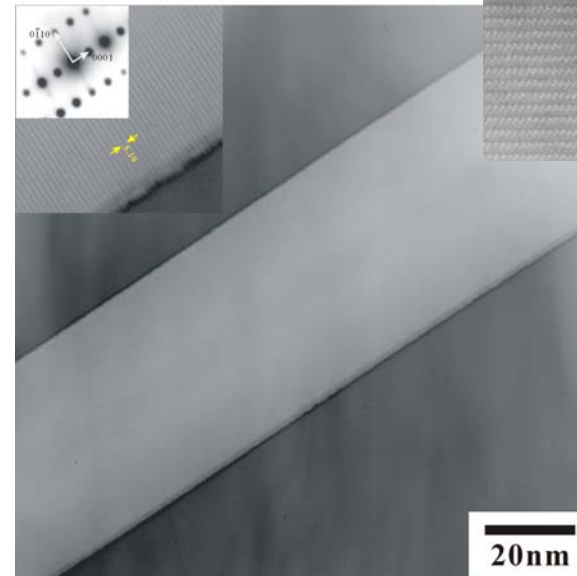
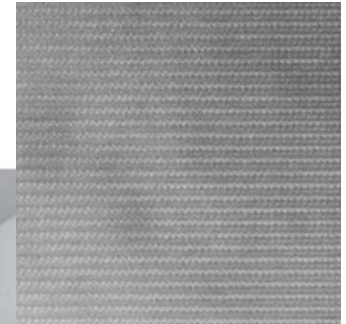
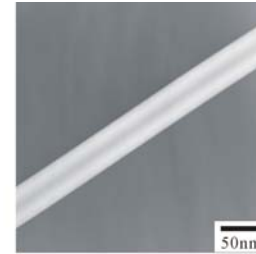
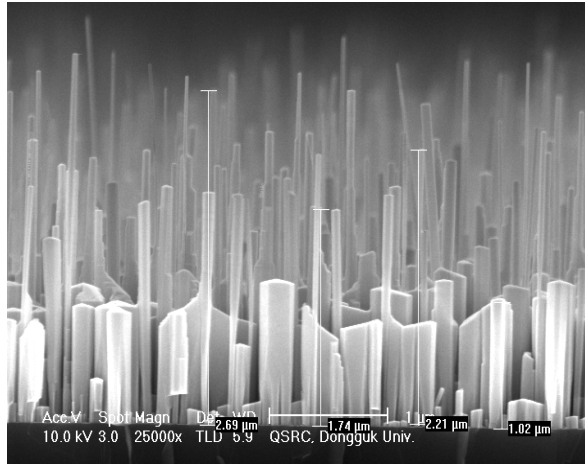


# GaN Nanotube



Ref: J. Goldberger, et al., *NATURE*,  
pp. 599-602, APRIL 2003

# Nano-rod Formation on Si substrate with no Catalysis



- Defect free material/structures are obtained
- Partially relaxed structure
- Diameter ranging from 5 nm~350 nm can be controlled :  $f_{Ga}/f_N$
- Feasible for vertical device structures



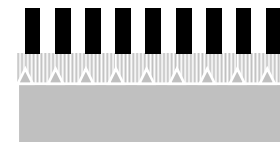
(I) QD Nucleation



(II) Columnar Growth

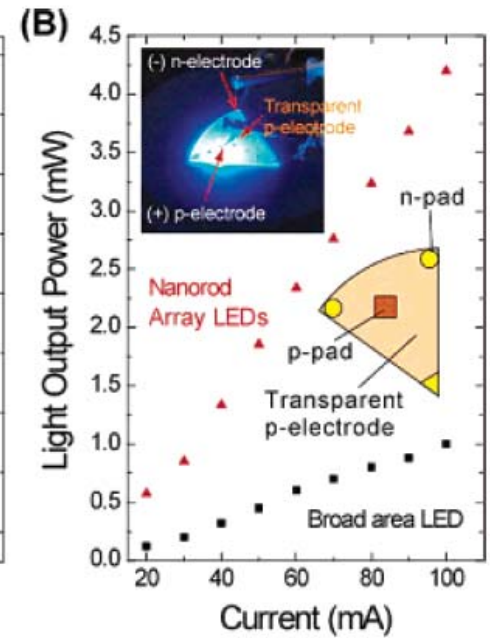
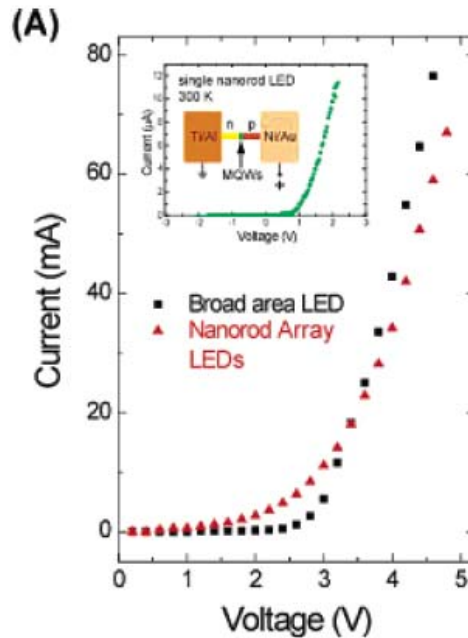
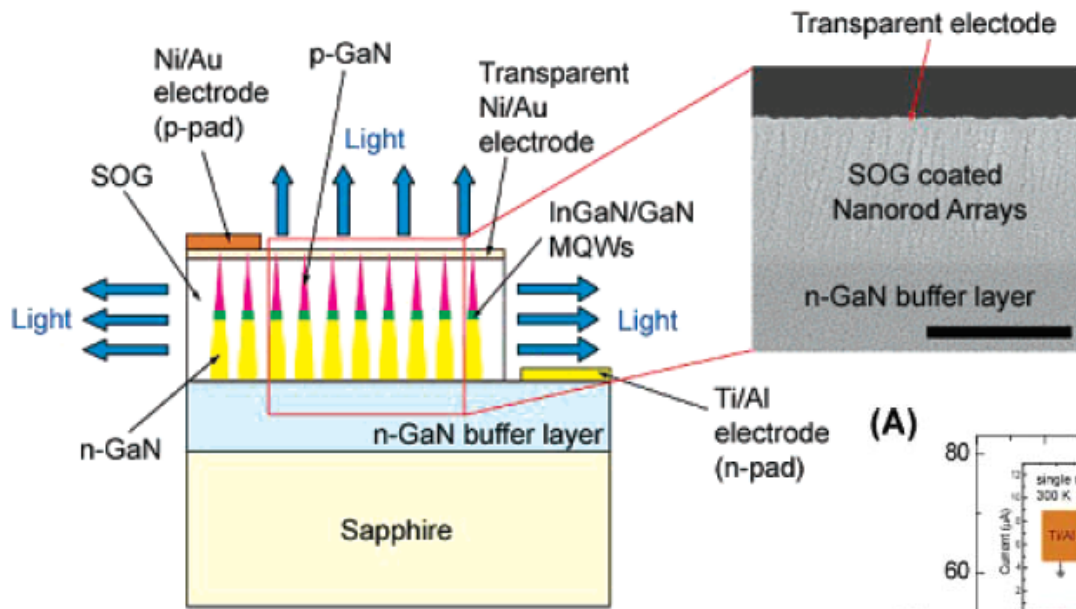


(III) Selective Nanorod Growth



(IV) Lateral Nanorod Growth

# High Brightness InGaN/GaN MQW Nanorod LED



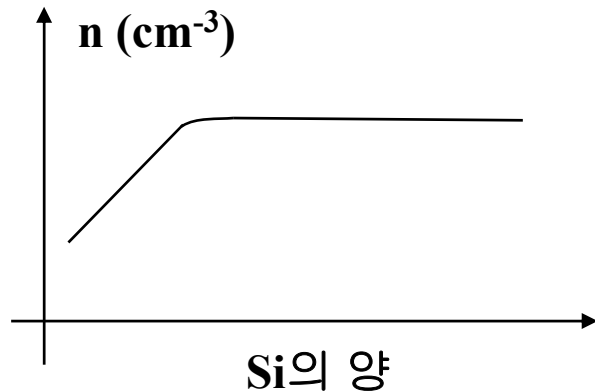
Ref: Hwa-Mok Kim, et al., NANO LETTERS, pp. 1059-1062, 2004

# GaAs ( InP ) 의 shallow impurities

	II	III	IV	V
	IV			
		Ga	n형	As
p형 불순물	Be		Si	S**
	Zn**			Se**
	Mg**		C	
				n형 불순물

\*\* elements; large diffusion constant  
→ thin epitaxy에 부적합

\* Si ; amphoteric impurities ( n-& p-type ) ; self-compensation



- MBE ; Si (n-type), Be (p-type)
- MOCVD ; Si (n-type), C (p-type)

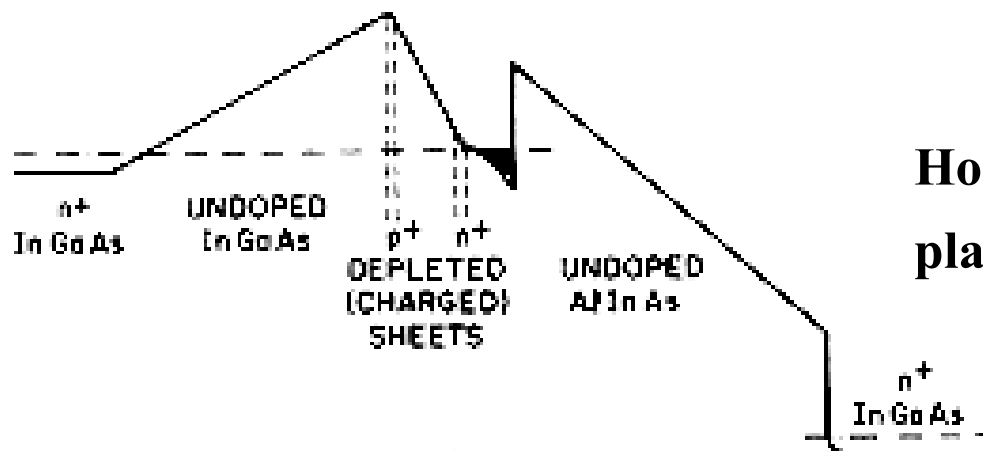
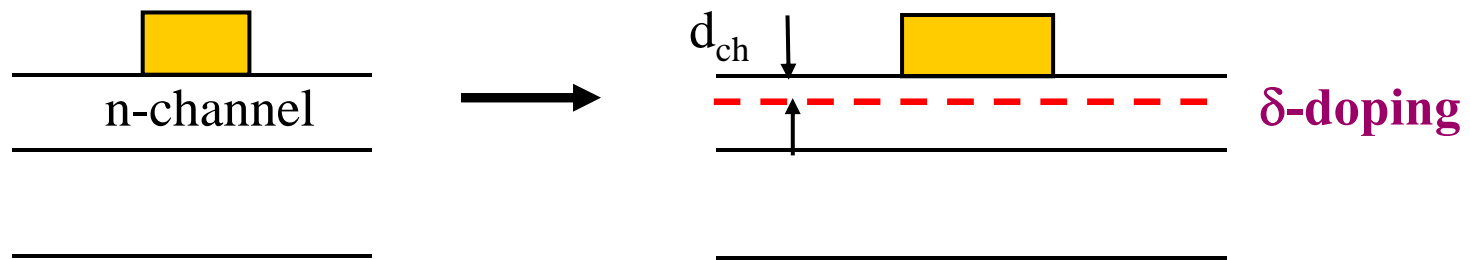
↓  
CCl<sub>4</sub> gas: 1. very small diffusion constant  
2. high doping

\* HBT에서는 very thin, high doped P-base 필요 → MOCVD & C (Kopin)

# *$\delta$ doping/Planar doping*

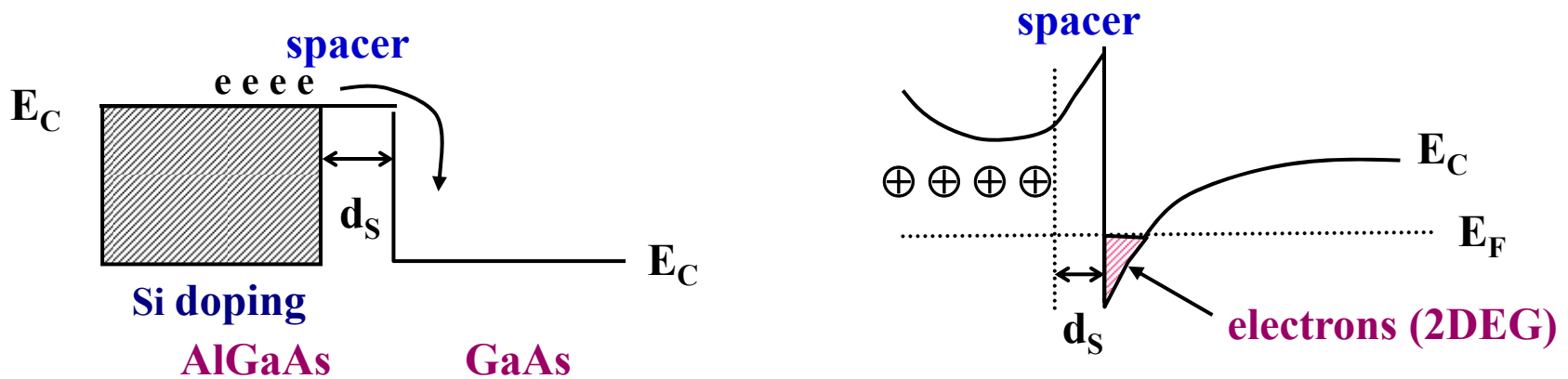
**$\delta$ -doping 형성** : 일정한 plane에 doping (planar doping)

- growth 를 중지한 상태에서 donor /acceptor 불순물을 도입.



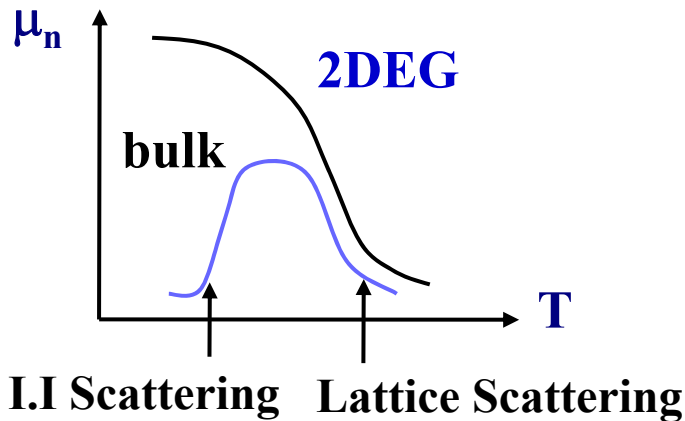
**Hot electron transistor with  
planar doped barrier emitter**

# Modulation Doping (I)



channel electron은 ionized-doner에 의한 impurity-scattering을 겪지 않음

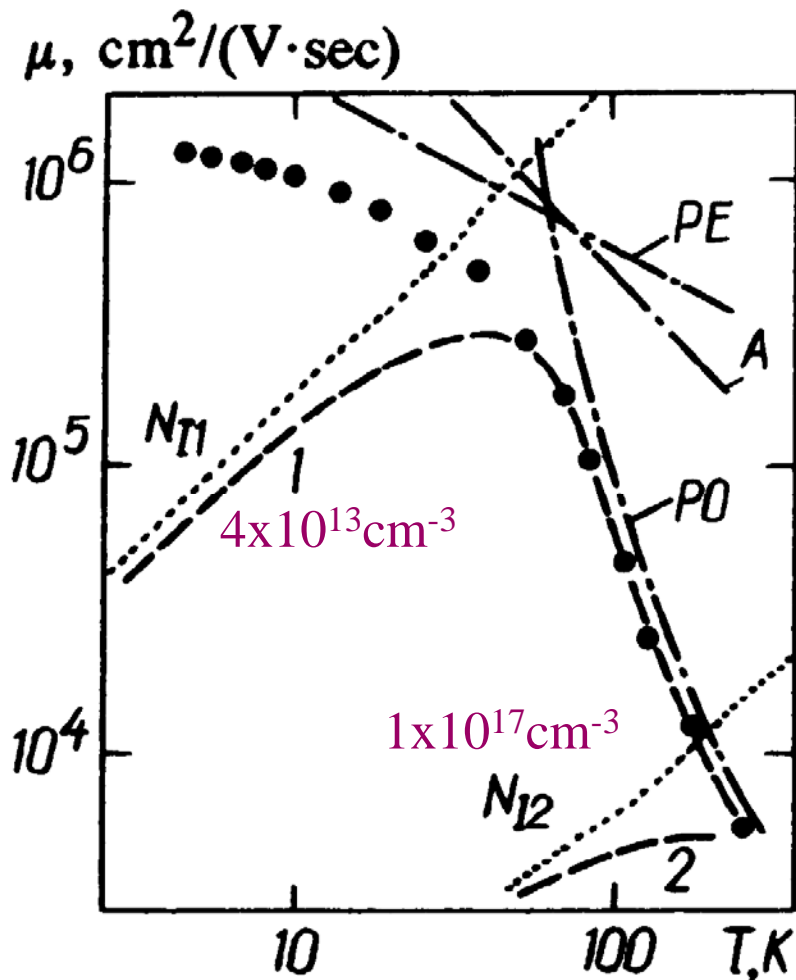
→  $\mu_n \uparrow$ ,  $V_{drift} \uparrow$  with modulation doped structure



$d_s \uparrow \rightarrow \mu_n \uparrow$  ; I-I scattering 감소  
 (예)  $d_s = 200 \text{ \AA}$  (77K)  $\mu_n \sim 150,000 \text{ cm}^2/\text{V}\cdot\text{s}$   
 (4K)  $\mu_n > 1,000,000 \text{ cm}^2/\text{V}\cdot\text{s}$

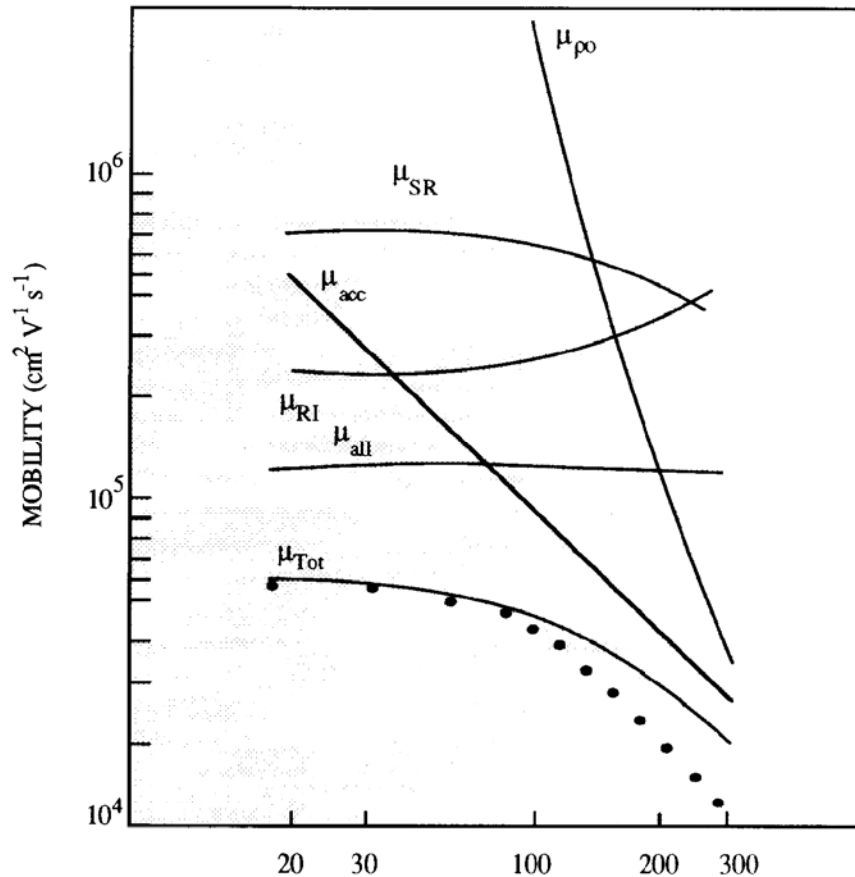
# Modulation Doping (II)

## GaAs/AlGaAs MD Structure



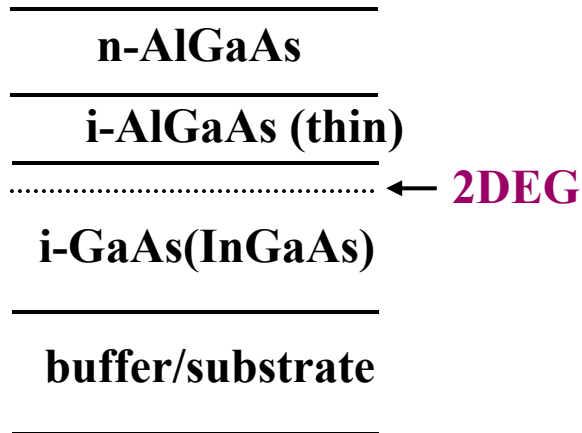
## InGaAs/InAlAs MD Structure

- alloy scattering limited



# MD Structure 의 여러 구조 (1)

## < Normal HEMT >



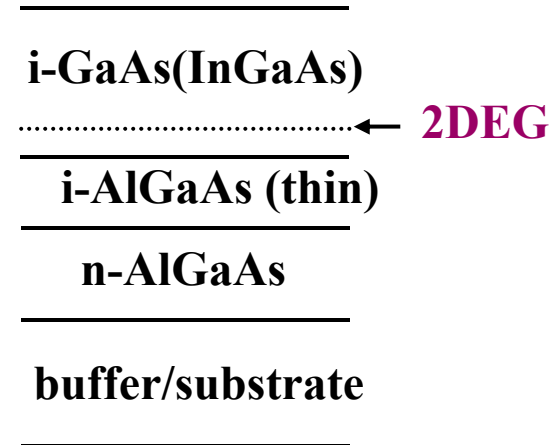
- large mobility

→ smallest sheet resistance

→ thermal noise 감소

(저잡음 소자, LNA)

## < Inverted HEMT >



- small mobility due to

1. rough AlGaAs surface

2. dopant (Si) diffusion

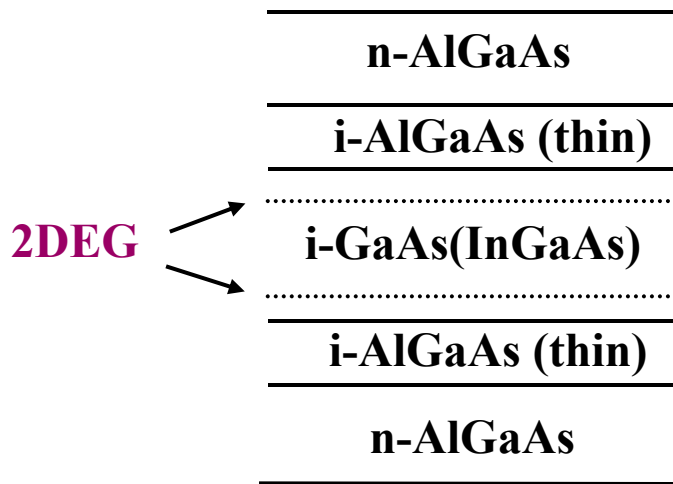
& segregation

-better ohmic contact



# MD Structure의 여러 구조 (2)

## < Double-Heterostructure HEMT >

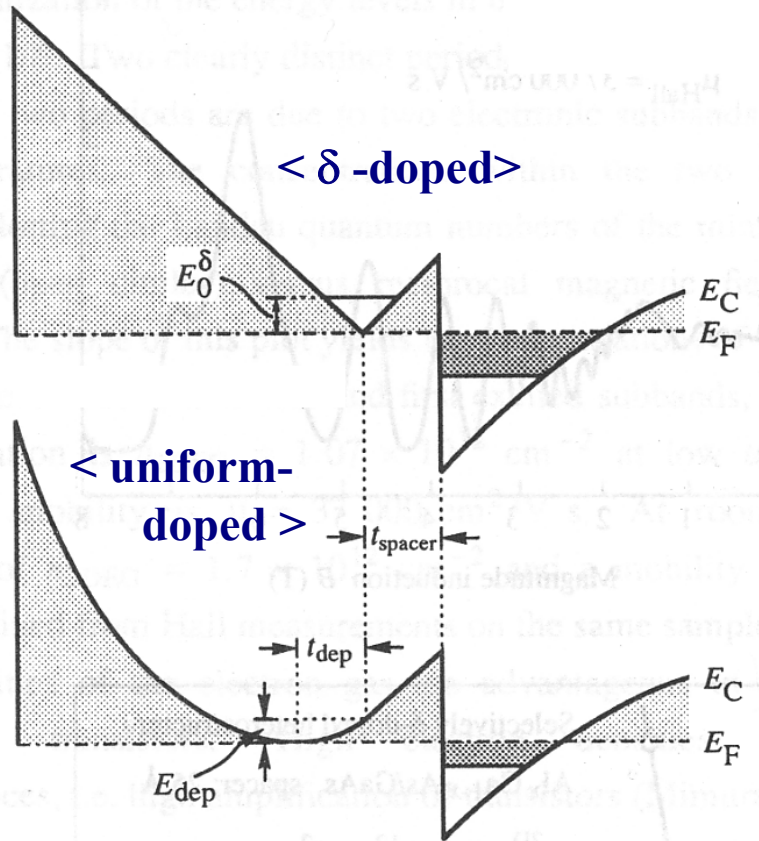


- largest 2DEG concentration ( $n_s$ )

→  $I_D = q n_s v_s W$  증가

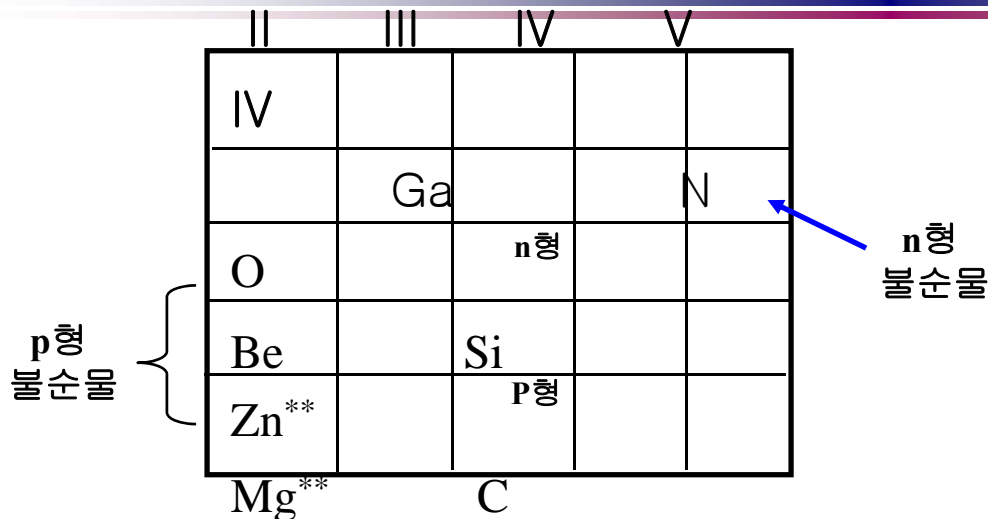
→ power device

## ➤ barrier의 doping 방법



- barrier의  $\delta$ -doping은 channel 전하량 ( $n_s$ )를 증가시킴.

# Various Impurities in GaN

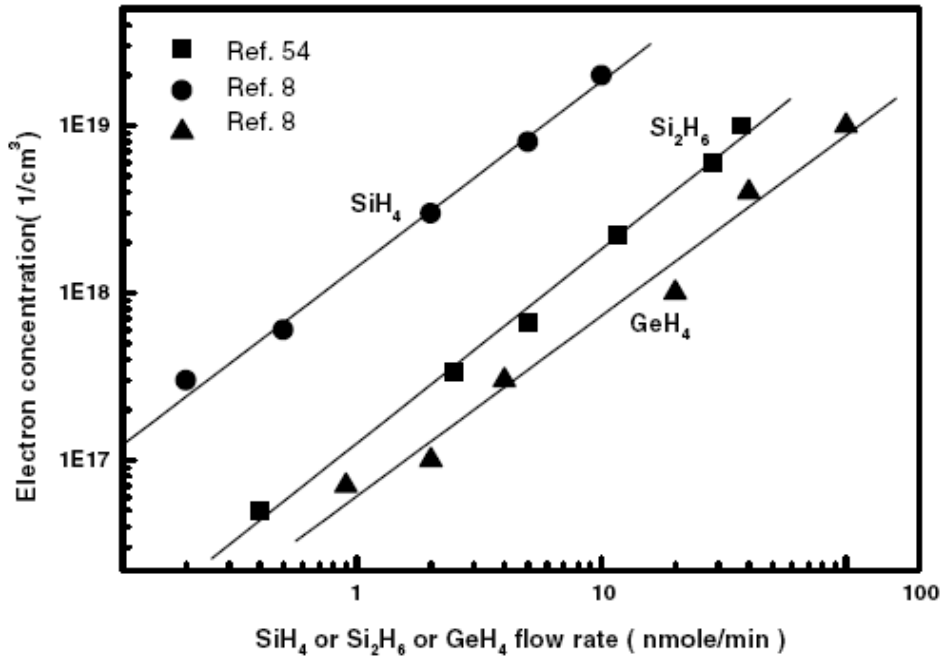


## Acceptor Ionization Energies ( $E_A$ , meV) in wurtzite(wz) and zinc-blende(bz) GaN

	Be	Mg	Ca	Zn	Cd	C	Si	Ge
$E_A$ (wz)	187	224	302	364	625	152	224	281
$E_{\text{expt}}$	90	209		328	550		224	
	160	224		340				
	250	250						
$E_A$ (bz)	183	220	297	357	620	143	220	276
$E_{\text{expt}}$		213						
		224						

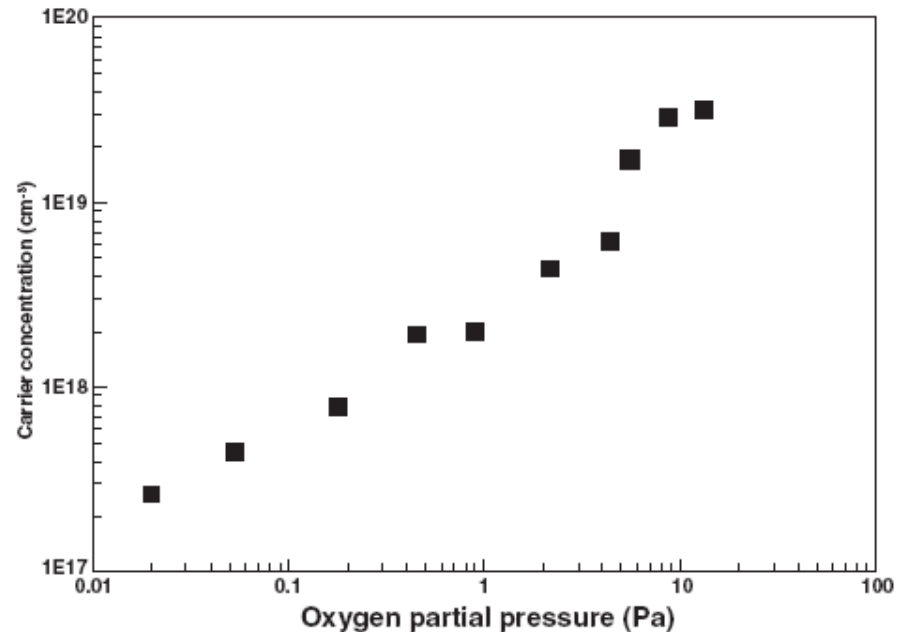
(Ref.) H. Wang et al, Phys. Rev. B 125212, 2001

# N-Doping of GaN in MOCVD Growth



(Ref.) J. K. Sheu, et al, J. Phys.:  
Condens.  
Matter 14 (2002) R657–R702

Oxygen in GaN  
– Shallow n-type impurity  
– 27meV Activation Energy



# *Ion Implantation & Annealing for Selective Doping*

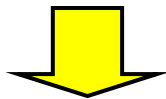
**Ion Implantation** – selective area doping  
– doping profile control



**Main Technology for Si**  
(even for GaAs & InP)

## **Problems in ion implantation for CS Technology**

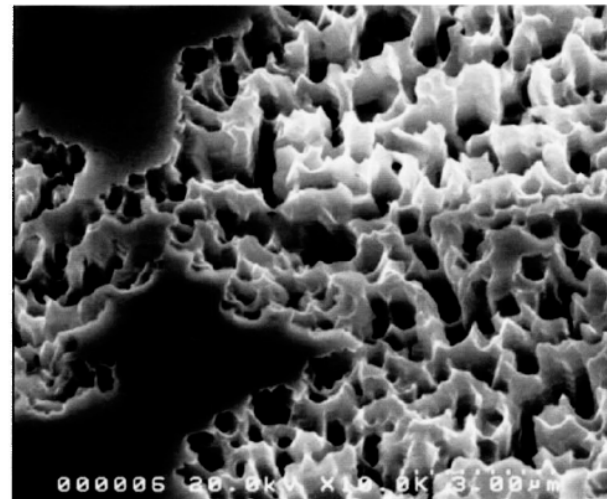
- 1. Damage Annealing**
- 2. Surface Degradation during  
high temperature annealing**
- 3. Furnace Annealing  
versus RTA**



**Various Encapsulant during  
High Temp. Annealing**

**$\text{Si}_3\text{N}_4$ ,  $\text{SiO}_2$ ,  $\text{AlN}$ , ·····**

**Surface Morphology of GaN  
after 30s RTA at 1200°C**



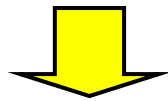
**3 μm**

# Si Implant and RTA with $\text{Si}_3\text{N}_4$ Encapsulant

(Ref.) S. Matsunaga, et al, J. Appl. Phys.,  
pp. 2461-2466, Mar. 2004

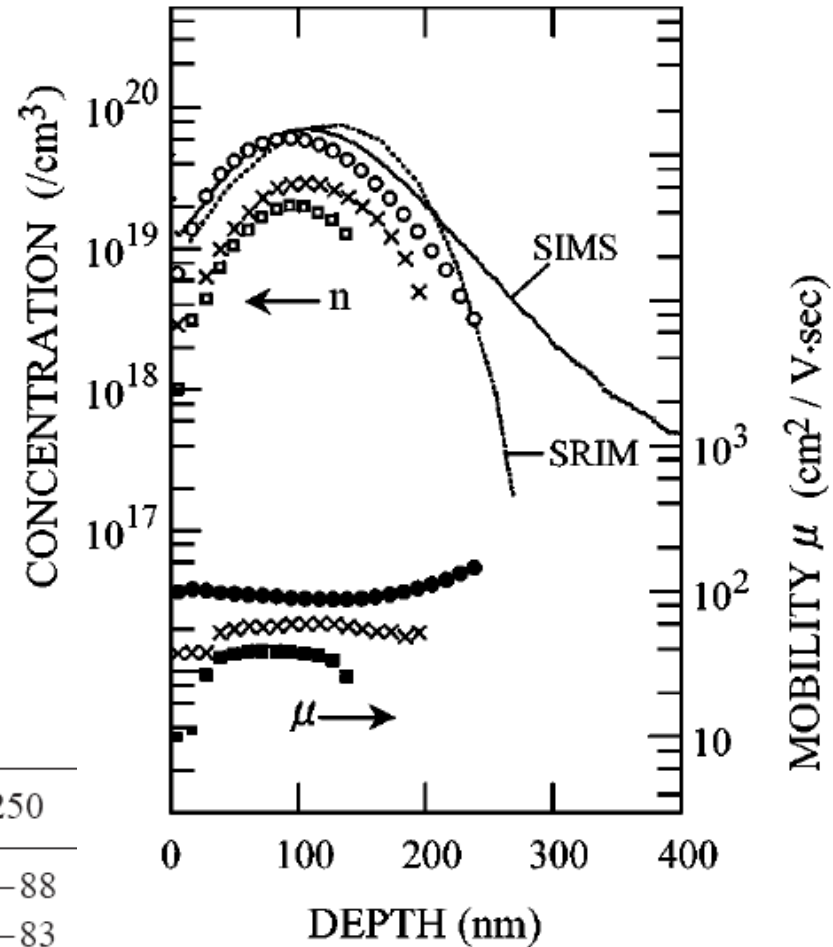
## Ion Implantation & Anneal Condition

- 150 keV,  $1 \times 10^{15}/\text{cm}^2$   $\text{Si}^+$  Ions implant (Room Temp. Implant)
- Substrate tilted  $7^\circ$  from the incident  $\text{Si}^+$  ion beam to minimize the channeling effect.
- 140-nm-thick  $\text{Si}_3\text{N}_4$  film as an encapsulant
- RTA with  $\text{N}_2$  flowing



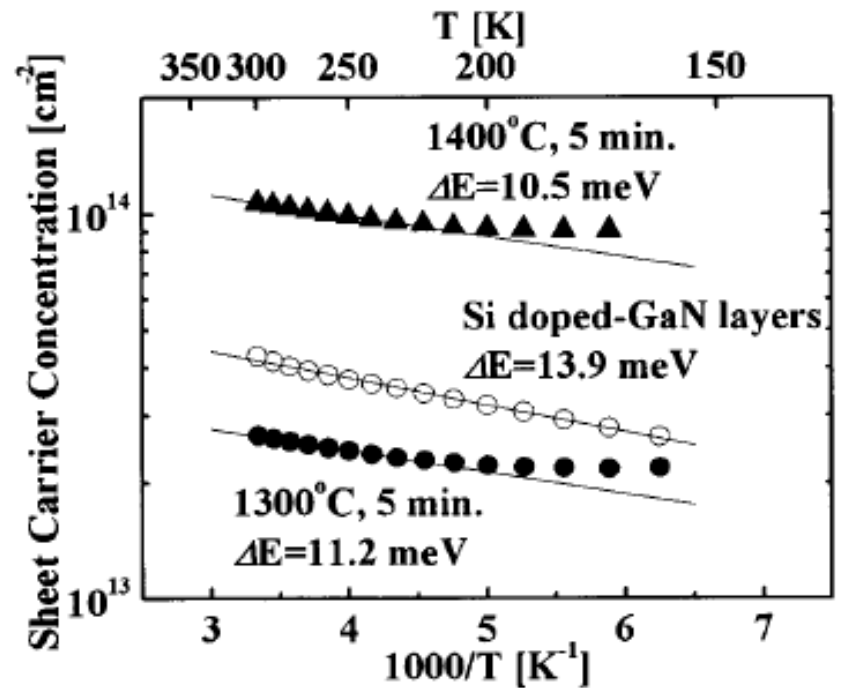
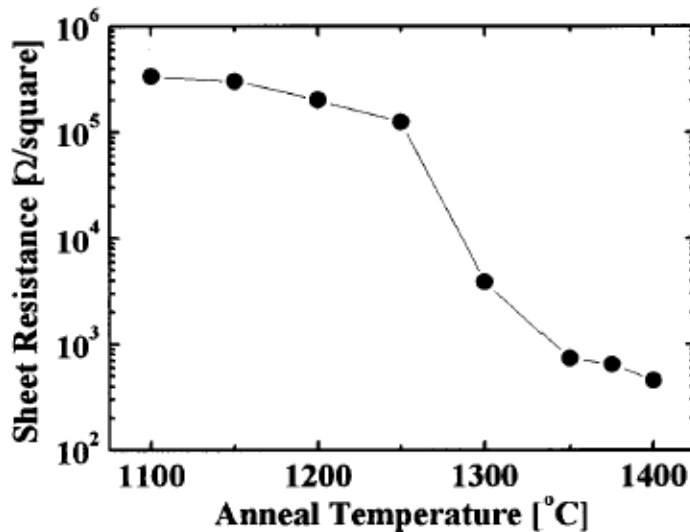
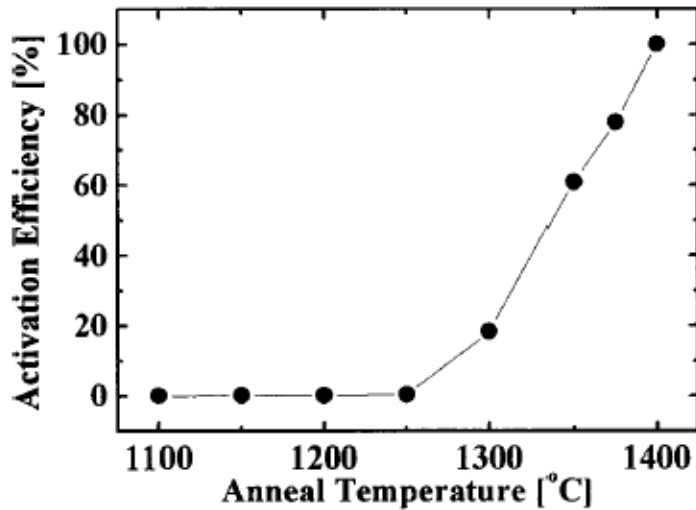
RTA temperature ( $^\circ\text{C}$ )

Properties	1150	1200	1250
$\rho_s$ (ohm/sq)	870–2200	240–370	82–88
$\mu_{\text{eff}}$ ( $\text{cm}^2/\text{V s}$ )	36–20	61–57	97–83
$N_s$ ( $\times 10^{14}/\text{cm}^2$ )	2.0–1.5	4.3–3.0	7.9–8.6



# Low Dose Si Implantation & Annealing

- $1 \times 10^{14}/\text{cm}^2$   $^{28}\text{Si}^+$  ions implant (Room Temp. Implant)
- 500 nm sputtered  $\text{SiO}_2$  capping
- annealed for 5 min under  $\text{N}_2$  ambient.



(Ref.) Y. Irokawa, et al, J. Appl. Phys. 97, 083505, 2005