VI. Magnetism in Small Structures

(1) Introduction
(2) Exchange Coupling
(3) Nanostructured Magnetic Materials
(4) Surface and Thin Film Magnetism
(1) INTRODUCTION

Nanostructured Magnetic Materials

(a) Granular solids in which one or both phases are magnetic
   Single-domain crystalline particles in an amorphous matrix:
     - Fe-B-Si-Nb-Cu: Fe$_3$Si particles in a matrix of residual amorphous phase
     - Co-Nb-B: Co$_n$B particles in a nonmagnetic amorphous Nb-B-rich matrix,  - (Fe-Co)$_{80}$M$_9$C$_{10}$
   Composite nanostructured permanent magnets: α-Fe/Fe-Nd-B
   Granular GMR materials: Fe/Cu, Fe-Ag, Co-Cu

(b) Quasi-granular films made by heat treating multilayers of immiscible solids
   GMR films: [Fe-Ni/Ag]$_N$

(c) Nanograined layers grown on columnar films
   Thin film magnetic recording media: Co-Cr-Ta/Co-Cr

(d) Columnar thin films and engineered nanowires

(e) Acicular particulate recording media: Cobalt-modified ferric oxide media: Co$_2^+$ γ-Fe$_2$O$_3$

(f) Four different types of thin-film structures whose unique properties arise from one nanoscale dimension:
   Magnetic multilayers: [Fe/Cr]$_N$, Spin valves: Co/Cu/Ni-Fe, Spin switches: Ni-Fe/Cu/Ni-Fe
   Spin-tunnel junction: F-I-F tunnel junctions such as FeCo/Al$_2$O$_3$/Co
(1) INTRODUCTION

Key Factors understanding Magnetism of Nanostructured Materials

- Single-domain behavior
  If the magnetostatic energy of the single-domain state is lower than the energy required for creating a domain wall, nanostructured materials should be composed of a single domain below a certain dimension.

- Superparamagnetism
  Below a critical particle dimension, the remanent magnetization is no longer fixed in the direction dictated by particle shape or crystal anisotropy because of ambient thermal energy (a magnetic analog of Brownian motion) and the coercivity becomes zero because of thermal effects that are strong enough to spontaneously demagnetize particles. Typically below a radius of order 20 nm, magnetic particles become superparamagnetic.

- Exchange coupling: The coupling between spins on different sides of a physical interface
  It is critical to understand the magnetic behavior of nanocrystalline materials and the behavior of many thin-film magnetic structures and devices. It is also the origin of the surface anisotropy.

- Random anisotropy: Local magnetic anisotropy in amorphous alloys and nanstructured materials
  It is relevant in treating the variation in direction of magnetization from particle to particle.

- Surface anisotropy: The magnetic anisotropy at the film surface due to exchange coupling
  It is different from the bulk anisotropy and important to thin-film magnetism.
Single-domain particles between $r_o$ and $r_c$

$$r_c \approx 9 \frac{AKu}{\mu_0 Ms^2}^{1/2} \quad \text{(strong anisotropy)}$$

$$r_c \approx \sqrt{9\frac{A}{\mu_0 Ms^2} \left[\ln\left(\frac{2r_c}{a}\right)-1\right]} \quad \text{(weak anisotropy)}$$

$cf) r_o^{1 \text{ yr}} = (9kT/K_u)^{1/3}$
A critical film thickness, $t_c$ below which the single-domain state is stable

The total energy density,

$$f_{\text{total}} = 2(1.7 \sigma_{dw} \mu_0 M_s^2 t)^{1/2}/L$$

Free energy of the single-domain state

$$f_{ms}^{sd} \approx \left(\frac{Wt}{L^2}\right)[\ln\left(\frac{4L}{W + t}\right) - 1] \mu_0 M_s^2$$

Therefore,

$$t_c \approx 4\left(\frac{L}{W}\right)^2\left(\frac{1.7 \sigma_{dw}}{\mu_0 M_s^2}\right)[\ln\left(\frac{4L}{W + t}\right) - 1]^{-2}$$

Figure 8.19  (a) Comparison of the thickness dependence of the free energy density for the demagnetized state $f_{tot}$ [Eq. (8.23)] with the free energy density for the single domain state, $f_{ms}^{sd}$ [Eq. (8.24)]. Note the crossover below which the energy is lower for the single-domain state. (b) Variation of the critical thickness with the ratio $L/W$ for two different values of magnetization and $\sigma_{dw} = 0.1 \text{ mJ/m}^2$. 
(2) Exchange Coupling

Exchange Coupling?
- The coupling between spins on different sides of a physical interface, leading to a preferential relative orientation of two different magnetic materials
- Different from a long-range magnetostatic dipole coupling at the boundary between two materials
- The spins in magnetic particles or layers of different materials can be coupled with different strengths or even different signs depending on the nature of their exchange interaction and the thickness and properties of the material between them

Types of exchange coupling
- Ferromagnetic-Antiferromagnetic (F/A) Exchange Coupling
- Ferromagnetic-Ferromagnetic Coupling
- Oscillatory Exchange Coupling
(2) Exchange Coupling

- **Ferromagnetic-Antiferromagnetic (F/A) Exchange Coupling**

  - **M-H loops of fine Co particles (2r ≈ 20 nm)**
    - Displacement of the field axis is due to the existence of a CoO thin surface layer
      
      \[ \text{CoO: paramagnetic above } T_N (= -3 \degree \text{C}), \text{ antiferromagnetic below } T_N \]
    
    - Below \( T_N \), Co moments in CoO are ferromagnetically coupled in \{111\} planes with the Co moments in adjacent \{111\} planes antiparallel.
    
    - The magnetic moments in the CoO chose an axis of magnetization that minimizes their energy of interaction with the Co moment across the interface, leading to the field-displaced loop exhibiting **exchange anisotropy**, which results from an **exchange coupling** between the moments of Co and CoO.

- Definition of the coercivity \( H_c \) and exchange field \( H_{ex} \)
(2) Exchange Coupling

**Ferromagnetic-Antiferromagnetic (F/A) Exchange Coupling**

*M-H* loops of fine Co particles (*2r* ≈ 20 nm) (continued)

- A stronger negative field is required to demagnetize the sample than if it had been cooled in zero field. (Figs. in Cullity)

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(2) Exchange Coupling

**Interpretation**

*The actual microscopic moment arrangement of F/A exchanged coupled materials*

- Soft ferromagnetic layer exchange-coupled to an antiferromagnetic layer (see Fig. 12.4)
- The exchange coupling across the interface is such that the moments in the A material lie on an axis that is orthogonal to the F moment at the time of cooling through $T_N$, which is the lower energy configuration because it is easier to cant the A moments into the direction of an orthogonal field than one that is parallel to the preferred A axis.
- Why does the twist in magnetization occur mostly in the soft F layer?
  i) a small magnetic anisotropy in the F layer
  ii) no net torque on the A moments
The actual microscopic moment arrangement of F/A exchanged coupled materials (continued)

- In the case that the thickness of the F material is less than the thickness of a domain wall or exchange length (see Fig. 12.5 in O’Handley) : no magnetization twist in F

- If the A material were not sufficiently thick (see Fig. 12.6 in O’Handley)
In F/A exchanged coupled materials

- The exchange coupling \((i.e., \text{exchange anisotropy})\) disappear below a certain thickness of the A material.

The condition for no exchange anisotropy

\[ t_A \leq \frac{J}{K_A} = t_A^c \]

Coercivity

For a weak-A

- Coherent rotation of MF against its anisotropy KF

\[ H_c \leq 2 \frac{K_F}{M_F} \]

- Rotation of the MA against KA by the interface torque

\[ H_c \leq 2 \frac{K_A t_A}{M_F t_F} \]

For a strong-A

\[ H_c \leq 2 \frac{K_A \delta A}{M_F t_F} \approx \frac{\sigma_{dw}}{2M_F t_F} \]

Exchange field

\[ \frac{F}{Area} = -M_F H t_F \cos \theta + K_F t_F \cos^2 \theta - J \sin \theta \]

\[ \sin \theta \left[ \cos \theta + \frac{M_F H}{2 K_F} + \frac{J}{2 K_F t_F} \right] = 0 \]

\[ H_{ex} = \frac{J}{M_s t_F} \]

Anisotropy field

\[ \frac{F}{Area} = -M_F H t_F \cos \theta + K_F t_F \cos^2 \theta - J \sin \theta \]

\[ \cos \theta = m_x = \frac{M_F t_F}{2 K_F t_F + J} \]

\[ H = \frac{H}{H_a} \]

\[ H_a = \frac{2 K_F t_F + J}{M_F t_F} = H_{a0} + \frac{J}{M_F t_F} \]
In F/A exchanged coupled materials (continued)

- Exchange field and coercivity versus NiFe thickness (see Fig. 12.7 in O’Handley) \( H_c, H_{ex}, H_a \propto t_F^{-1} \)

- Anisotropy, exchange and coercivity fields versus permalloy thickness (see Fig. 12.8 in O’Handley)
A complementary microscopic model that addresses the strength of $J$ from a statistical perspective

Assuming Heisenberg exchange across an atomically smooth F/A interface

(See Fig. 12.11 and Fig. 12.12 in O’Handley)

\[
H_{ex} = \frac{j_{ex} S_F \cdot S_A}{a^2 M_s t_F}
\]
(2) Exchange Coupling

Examples of F-A exchange coupling across a nonmagnetic layer (see Fig. 12.13 in O’Handley)
**Ferromagnetic-Ferromagnetic Exchange Coupling**

**Origins: a direct Heisenberg-like exchange interaction or an indirect RKKY-like exchange interaction**

- When the magnetization at the interface has no perpendicular component:

  \[
  M_{\parallel} (\text{parallel to interface}) : \quad l_{\text{ex}\parallel(i)} = (A^{(i)}/K_{u}^{(i)})^{1/2} \quad i = \text{medium 1, 2}
  \]

  The spin orientation at the interface will be biased toward that of the stronger anisotropy material.

- When there is a perpendicular component of magnetization at the interface

  \[
  M_{\perp} (\text{perpendicular to interface}) : \quad l_{\text{ex}\perp(i)} = \left\{ A^{(i)}/(K_{u}^{(i)} + 2\pi\Delta M_{\perp}^{2}) \right\}^{1/2}
  \]

  The exchange interaction through the parameter \( l_{\text{ex}} \), has the effect of communicating the magnetization direction in one region over distances of several nanometers into another, exchange-coupled region.
Domain Walls near Interfaces: The Exchange Length, $l_{ex}$

The exchange length $l_{ex}$ (or pinned wall thickness) is the thickness of the magnetization orientation transition when the magnetization is pinned in a direction different from the easy axis in the interior of the material.

- Two cases

  (i) No perpendicular magnetization component near the interface

  $M$ rotation is driven toward the easy axis by the interior anisotropy energy.

  $$M_{//}(\text{parallel to interface}) : l_{ex} = (A/K_u)^{1/2} = \frac{\delta_{dw}}{\pi}$$

  (ii) Non-zero perpendicular magnetization component near the interface $M$ rotation is driven toward the easy axis by both the interior anisotropy energy and the magnetostatic energy associated with the charged interface.

  $$M_{\perp}(\text{perpendicular to interface}) : l_{ex} = \left\{(A/K_u + 2\pi \Delta M_{\perp}^2)\right\}^{1/2}$$

**Figure 8.15** Illustration of the two cases important for determining the range of the twist in magnetization on moving from an interface at which the spins are pinned in a direction different from the interior of a ferromagnetic material. At left, the surface pinning holds the magnetization in the plane of the interface so magnetostatic energy is not an issue. At right, the surface spin pinning is such that a perpendicular component of magnetization exists near the interface. The magnetic charge at the interface gives rise to a local magnetostatic field that tends to shorten the exchange length.
(2) Exchange Coupling

- **Oscillatory Exchange Coupling**: A long-range oscillation exchange coupling

  Physical mechanism: Long-range oscillation in exchange coupling in two-dimensional magnetic nanostructures is believed to be related to the spin polarization of conduction electrons as described by the RKKY model.

  ![Graph showing oscillatory exchange coupling](image)

  **Key points:**
  - Large saturation fields at 10, 24, 45 Å of Cr: strong antiferromagnetic coupling between the Fe (or Co) layers
  - Small saturation fields: ferromagnetic coupling between the Fe layers
  - \([\text{Co/Cu}]_N\) multilayers: an exchange coupling of the Co layers that oscillates from antiferromagnetic to ferromagnetic as \(t_{\text{Cu}}\) increases
  - Typical oscillation period \(\sim 1\) nm, spacer thickness \(\sim 5-6\) nm

Saturation field of \([\text{Fe/Cr}]_N\) multilayers along a hard \(<110>\) in-plane direction at 4.2 K vs Cr thickness for Si(111)/100Å Cr/[20Å Fe/\(t_{\text{Cr}}\)Cr]/50Å Cr, open circle and square: \(N=30\), deposited at 40°C, closed circle and square: \(N=20\)°C, deposited at 125°C
(2) Exchange Coupling

Oscillatory Exchange Coupling (continued)

Key points:
- The Fe film is exchange coupled through the Cr layer to the magnetization in the whisker below.
- As $t_{Cr}$ increases, the exchange coupling is directly observed to oscillate to antiferromagnetic and back to ferromagnetic with a period corresponding to Cr atomic layers.
- No oscillatory exchange coupling was observed in F/A exchange coupling
(2) Exchange Coupling

- **Random Anisotropy in Nanostructured Materials**
  - Randomly oriented local magnetic anisotropy in amorphous alloys and nanostructured materials
  - Origin: Local crystal field giving rise to a local magnetocrystalline anisotropy, $K_{loc}$
  - In amorphous alloys, the orientation and strength of this local anisotropy varies with position.
    (see Fig. 11.11 in O’Handley)

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Figure 11.11 Schematic representation of the variation of local anisotropy easy axis with position and the variation of magnetization direction in response to the local anisotropy and exchange coupling. $M(x)$ closely follows a strong local anisotropy and is a smoother function of $x$ for weak local anisotropy.
(2) Exchange Coupling

Random Anisotropy in Nanostructured Materials (continued)

- When interparticle exchange and random anisotropy combine to affect the magnetic properties of discrete magnetic systems such as nanocrystalline alloys,

  Two important effects
  (i) The magnetization experiences an anisotropy reduced from its local value $K_{loc}$ by exchange-averaging over the random local anisotropy: $<K> = K_{loc}(l/L)^{3/2}$ (see Fig. 11.11 in O’Handley)
  (ii) The magnetization shows an orientational coherence over a length, $L = l_{ex} : L = l_{ex} = 16A^2/9K_{loc}^2l^3$ (see Fig. 12.17 in O’Handley)

  The appropriate length scale $l$ for the random anisotropy magnetization direction is given by the thickness of the interparticle layer.

  Suggestion by Herzer (1993)
  From $l_{ex}^{(i)} = [A^{(i)}/K_u^{(i)}]^{1/2}$ and $l_{ex} = 16A^2/9K_{loc}^2l^3$
  $<K> = 0.32(K_u^4/A^3)l^6$: explainable for a steep rise in $H_c$ in nanostructured system (see Fig. 9.19 in O’Handley)
(3) Nanostructured Magnetic Materials

- **Processing**
  - Bulk specimens:
    - Quenching certain alloy compositions from the melt
    - Heat-treatment of an amorphous alloy precursor
    - Addition of nucleation centers or grain growth inhibitors
  - Films:
    - Fabrication of two-dimensional films and multilayers:
      - Sputtering, Electrodeposition, MOCVD, E-beam evaporation/condensation

- **Examples**
  - The Co-Nb-B system (Fig. 9.19 in O’Handley)
    - Annealing of amorphous Co-Nb-B
    - Hc depends on the size of nanocrystallites
    - AC loss greatly decreases with decreasing the size

Variation of coercivity in amorphous and crystallized CoNbB as a function of mean particle size. Average D of nanocrystallites was determined by TEM.
(3) Nanostructured Magnetic Materials

- The Fe-B-Si-Nb-Cu system (Figs. 12.18-19 in O’Handley)

- SD $\alpha$-Fe$_3$Si particles: exchange coupled for $T < T_c$ (amorphous)
  decoupled for $T > T_c$ (amorphous)

- Origin of the peak in $H_c$ at the decoupling temperature:
  No longer switch in unison for the entire magnetization cycle as the exchange coupling weakens (See Fig. 12.6 in O’Handley)

- $H_c \rightarrow 0$ for $T_c$ (amorphous) $< T < T_c$ $\alpha$-Fe$_3$Si):
  Superparamagnetic behavior of SD $\alpha$-Fe$_3$Si particles
(3) Nanostructured Magnetic Materials

- $(\text{Fe,Co})_{81}M_9\text{C}_{10}$ (M = Ta, Hf, Zr, Nb,...) (Figs. 12.20-24 in O’Handley)

  Inclusion of refractory metals: Increase in the strength, stability, and corrosion resistance of alloys
  Annealing of metastable amorphous transition metal carbon alloys
    - Grain size of $\alpha$-FeCo: 5-10 nm
    - Grain size of M carbide nanocrystals (stable grain boundary phase): 1-4 nm
(3) Nanostructured Magnetic Materials

- **CoCrTa-(Pt) Recording Media** (Figs. 12.25 in O’Handley)
  CoCr alloys:
  - HCP phase below 18% Cr exhibiting strong magnetic anisotropy with high $H_c$
  - Important in magnetic recording media
  - Ta or Pt addition retard grain growth and promote segregation of nonmagnetic refractory metals and oxides to the grain boundaries.

![Co-Cr-Ta Nanostructured Film](image.png)
(3) Nanostructured Magnetic Materials

- Nanostructured Permanent Magnets (Figs. 12.26-28 in O’Handley)

Improvement in Br by exchange-coupling the high-anisotropy Nd$_2$Fe$_{14}$B$_1$ grains (20-40 nm) to a larger magnetization $\alpha$-Fe (5-10 nm) intergranular phase.

Pure 2-14-1 : Br =16 kG

iHc > 20 kOe

- Nanotubes and Nanowires (Fig. 12.1 in O’Handley)