

Chapter 6

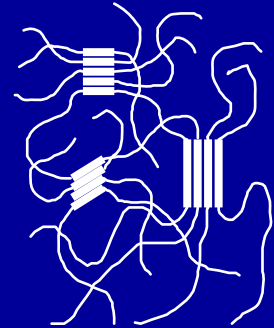
Crystalline State

Contents

- ❑ Melting
- ❑ Crystal structure
- ❑ Semicrystalline structure
 - Models
 - Lamella & Spherulite
- ❑ Crystallinity
- ❑ Crystallization
- ❑ Lamella re-entry
- ❑ Melting point
- ❑ Annealing

1. Melting

- semicrystalline = crystal + amorphous region
- melting = fusion of crystals
- Melting is a (true) **1st-order phase transition**



$$(dG/dP)_T = V$$

$$(dG/dT)_P = S$$

$$(d(G/T)/d(1/T)) = H$$

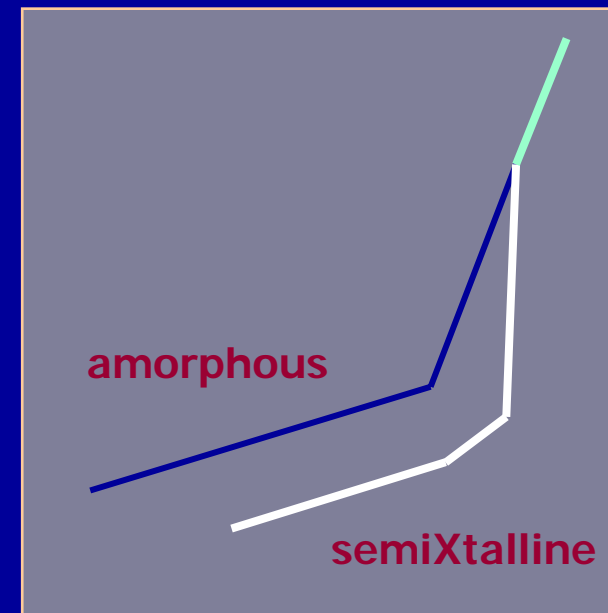
✓ V, S, H are discontinuous at T_m

- T_m measurement

- Dilatometry (V)  Fig 6.2


» on heating

Specific
volume



Temperature

Melting (cont'd)

- T_m measurement (cont'd)
 - DSC (H)
 - » on heating
 - » crystallinity  p244

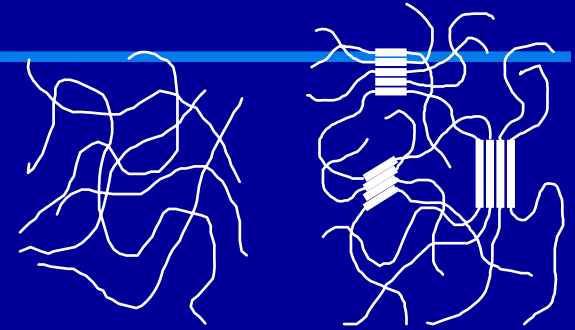
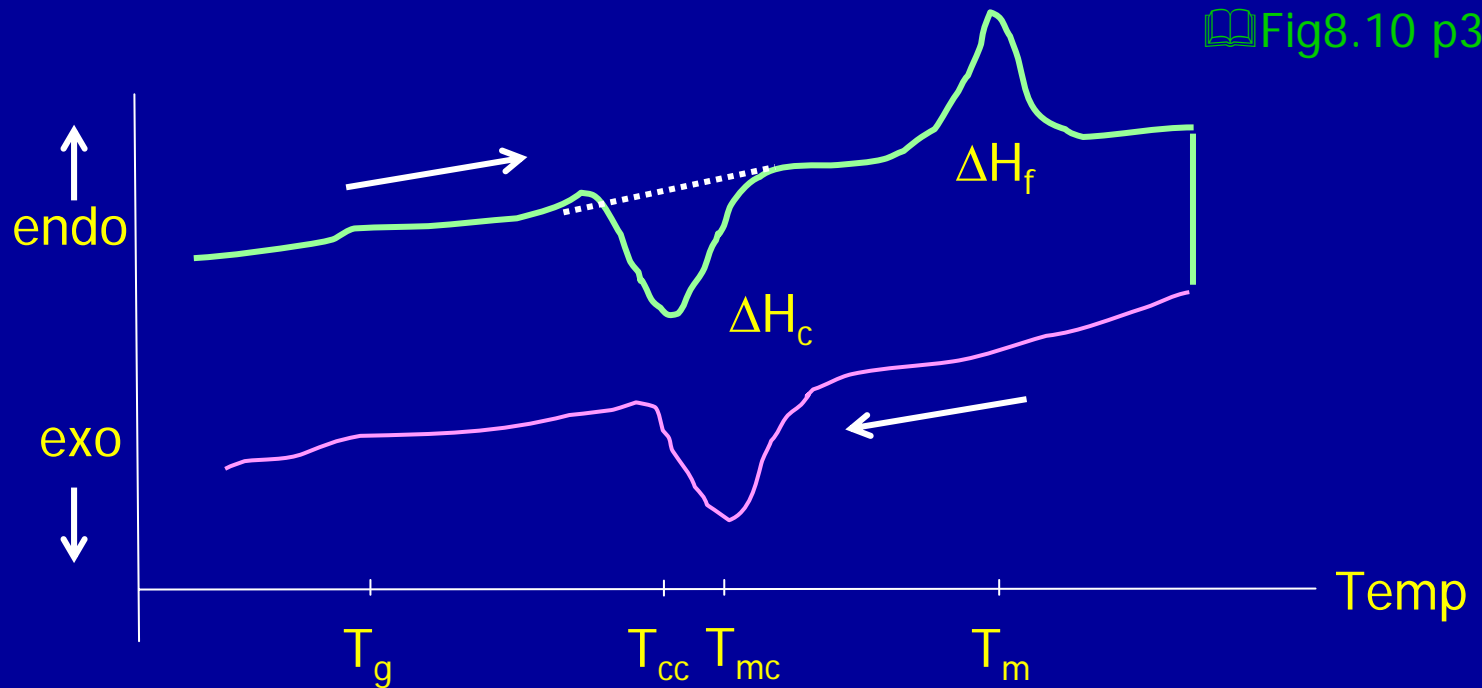


 Fig6.3 p244

 Fig8.10 p369



Melting (cont'd)

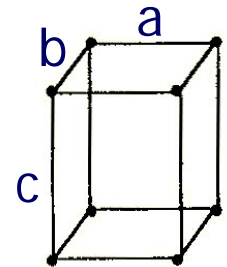
- T_m measurement (cont'd)
 - IR, microscopy, X-ray
 - » conventional ~ snapshot ~ indirect and not popular
 - » real-time ~ measurement possible

2. Crystal Chemistry

- (semi)crystalline = crystal + amorphous
- crystal = regular array of atoms
- unit cell = smallest volume of repeating structure
 - crystal system ~ shape of unit cell
 - **Orthorhombic** is popular for polymers

Table 7.1 Crystal systems

Systems	Axes	Axial angles	Minimum symmetry
Triclinic	$a \neq b \neq c$	$\alpha \neq \beta \neq \gamma \neq 90^\circ$	None
Monoclinic	$a \neq b \neq c$	$\alpha = \gamma = 90^\circ; \beta \neq 90^\circ$	One two-fold rotation axis
Orthorhombic	$a \neq b \neq c$	$\alpha = \beta = \gamma = 90^\circ$	Three perpendicular two-fold rotation axes
Tetragonal	$a = b \neq c$	$\alpha = \beta = \gamma = 90^\circ$	One four-fold rotation axis
Hexagonal	$a = b \neq c$	$\alpha = \gamma = 90^\circ; \beta = 120^\circ$	One six-fold rotation axis
Rhombohedral	$a = b = c$	$\alpha = \beta = \gamma \neq 90^\circ$	One three-fold rotation axis
Cubic	$a = b = c$	$\alpha = \beta = \gamma = 90^\circ$	Three four-fold rotation axes




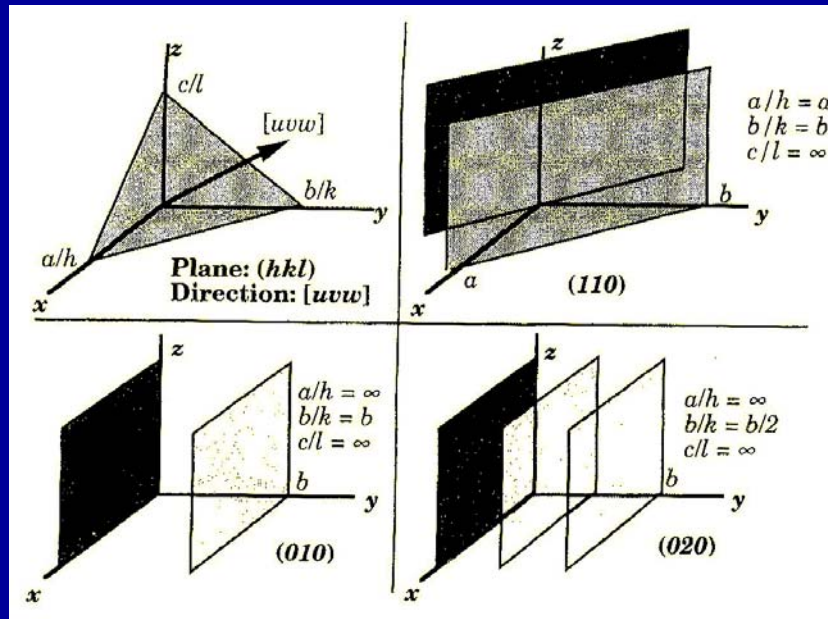
$$\alpha = \angle bc$$

$$\beta = \angle ac$$


$$\gamma = \angle ab$$

Crystal Chemistry (cont'd)

- observing crystal structure
 - X-Ray (WAXD)  Fig 6.4 & 6.5(b)
 - ED, IR, Raman
- Miller index See the supplement
 - The plane passing $(a/h, b/k, c/l)$ is $(h k l)$ plane.

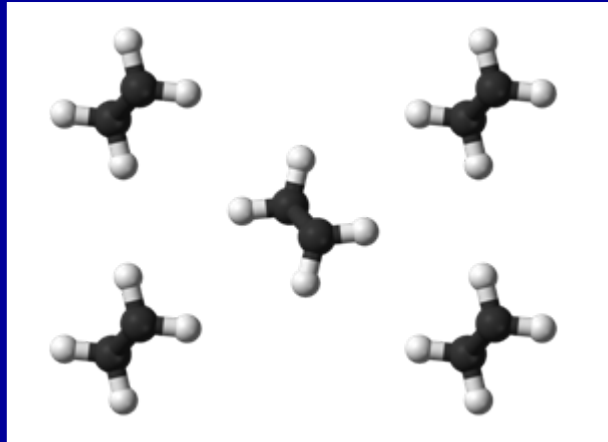


3. Crystallographic Structure of Polymers

- crystal(lographic) structure = unit cell structure
- Chains are of their preferred conformation given by RIS model. ← minimum energy  p256 #2

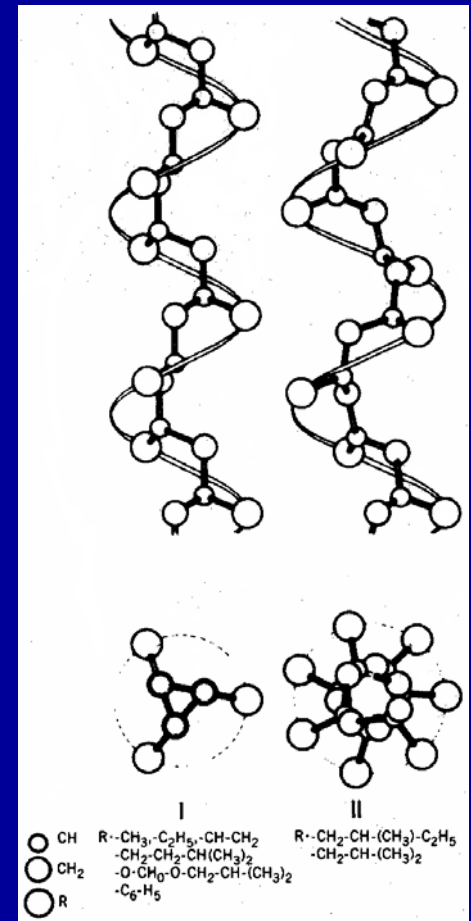
- Polyethylene; *ttttt...*; planar zigzag

»  Fig 6.5









- Isotactic polypropylene; *tgtg...*; 3_1 helix


»  Fig 6.6, 6.7

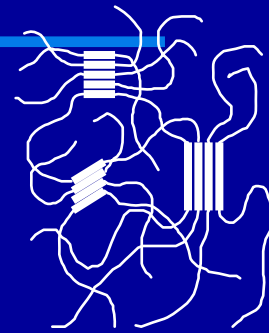




Crystal Structure (cont'd)

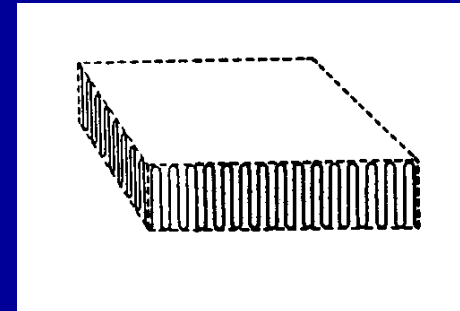
- Polyoxymethylene; $g+g+g+g+\dots$ or $g-g-g-g-\dots$; 2_1 helix
 - Nylon; hydrogen bonding; planar zigzag  Fig 6.8
-
- orthogonal popular  packing  p256 #3
 - anisotropic  packing  p256 #3
 - polymorphic  Table 6.2

4. Semicrystalline State


- Fringed micelle model  Fig 6.9
 - intuitive
 - A chain passes through crystallites.
 - 2-phase morphology (crystallites + amorphous)

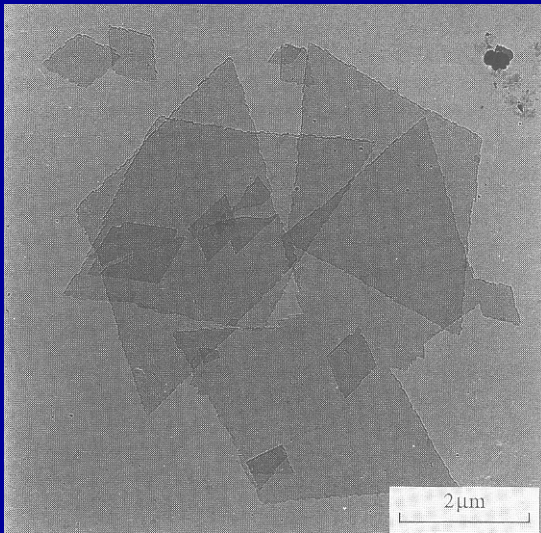


- Folded chain model  Fig 6.10
 - Single crystal **lamella** from dilute solution  Fig 6.5(a)
 - Chains are perpendicular to **lamella** by ED
 - Chains got to be *folded*
 - 3-phase morphology (lamellae + interface + amorphous)
 - Lamella re-entry – adjacent or switchboard?

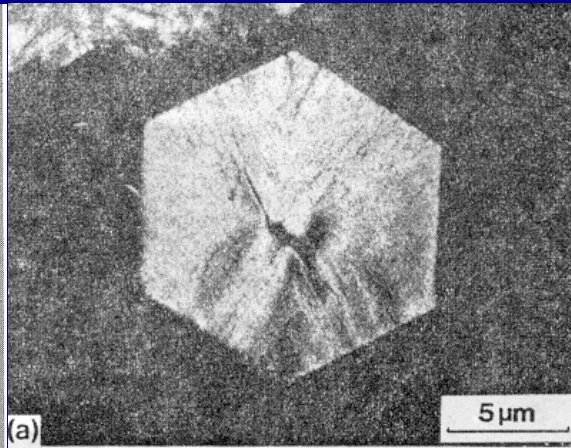


Lamella 1

- Single crystal from solution
- Part of **spherulite** from melt
- Shape depends on crystal structure  Fig 6.11 vs 6.12



Polyethylene
(orthorhombic)



Polyoxymethylene
(hexagonal)

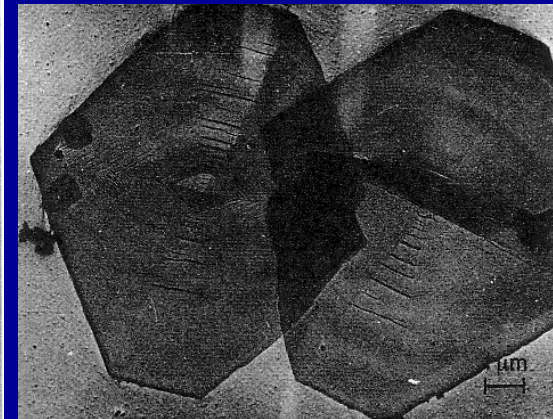
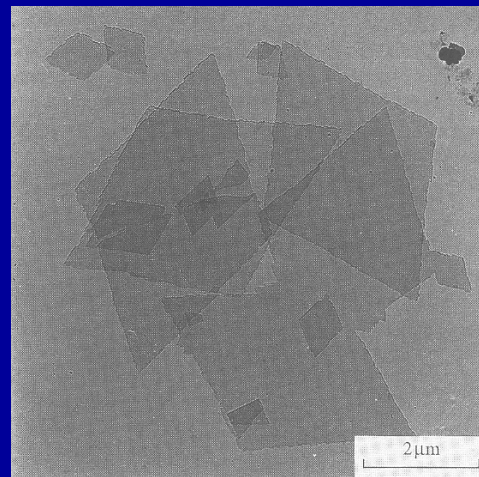
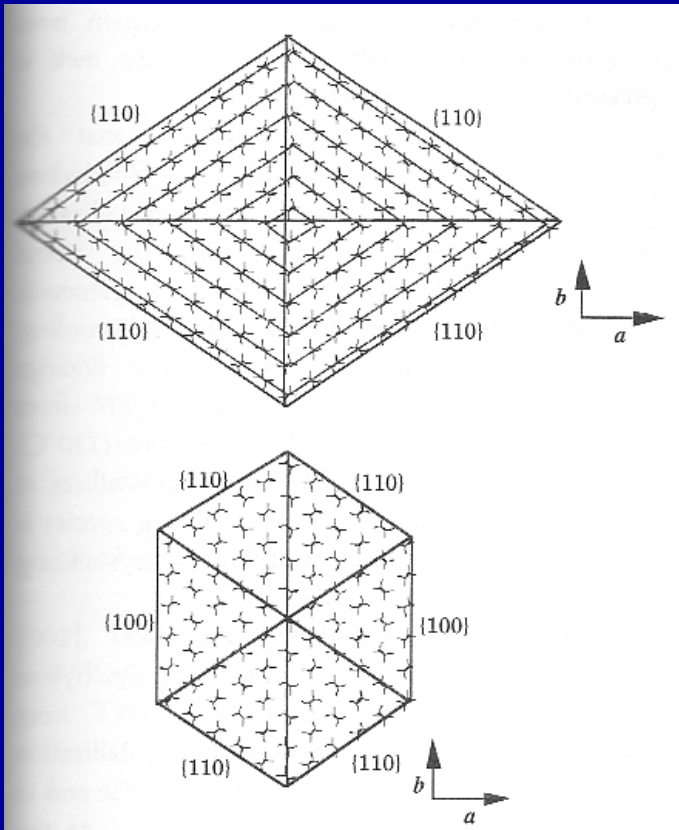
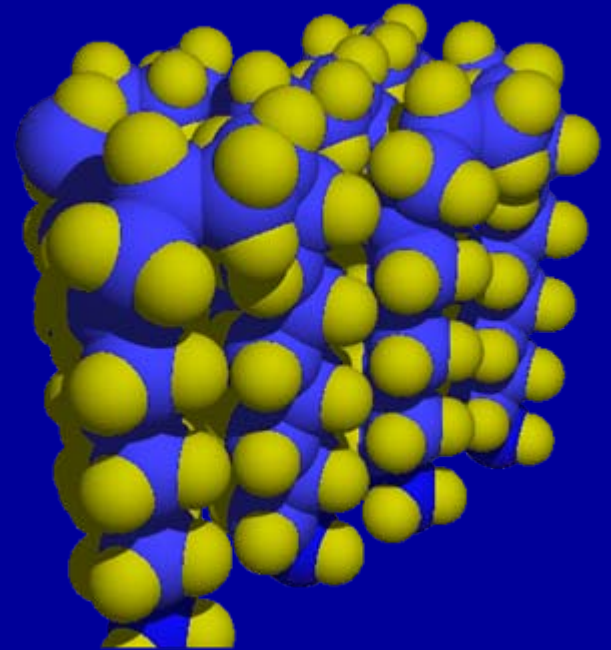


Poly-4-methyl-1-pentene
(tetragonal)


Lamella 2

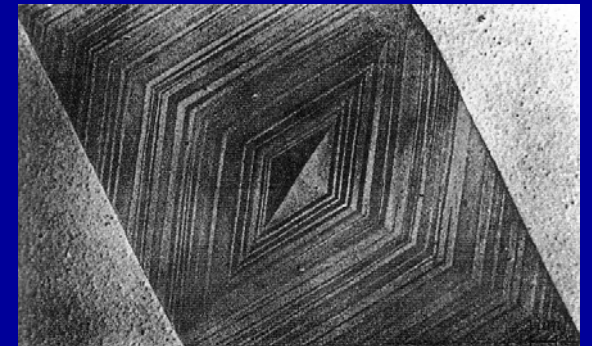
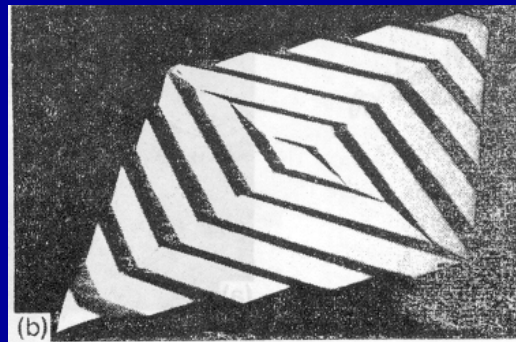
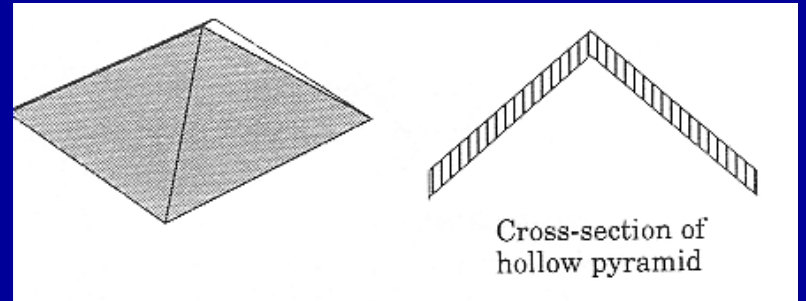
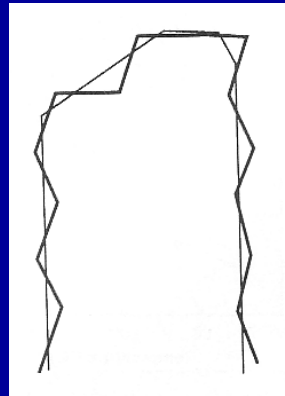
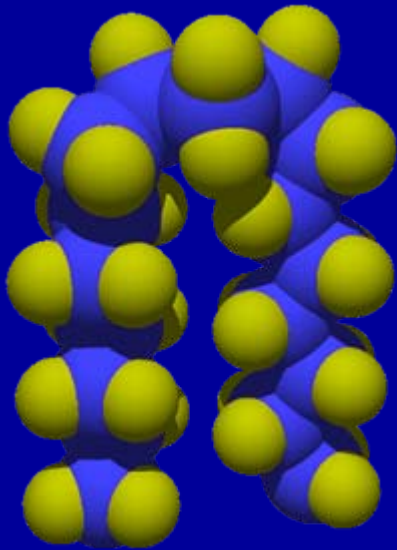
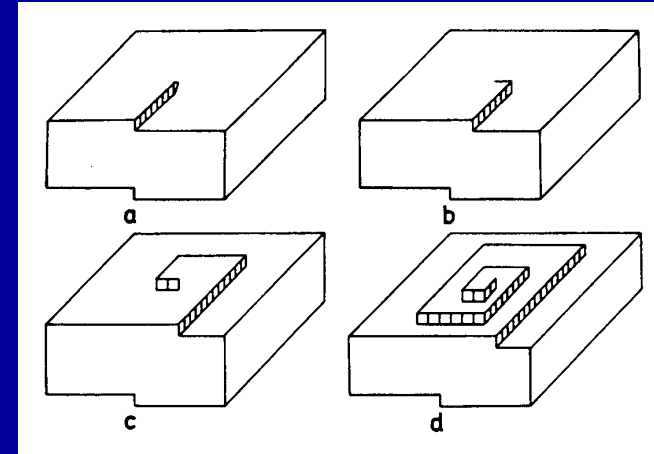
□ Growth

- on $[110]$ fold plane  Fig 6.10
- on $[110]$ and $[100]$ plane




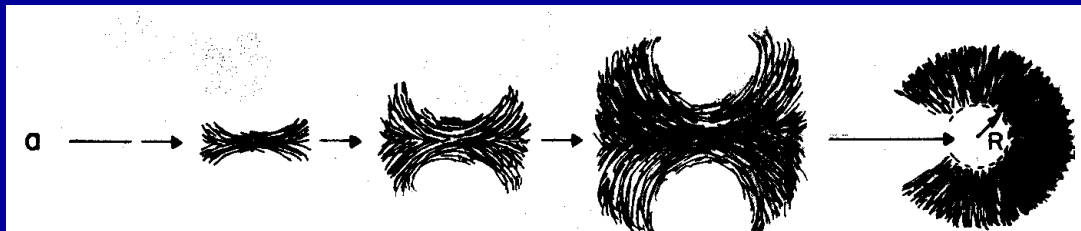
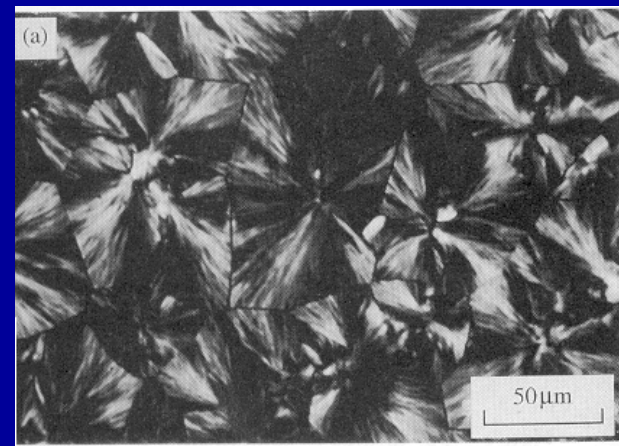
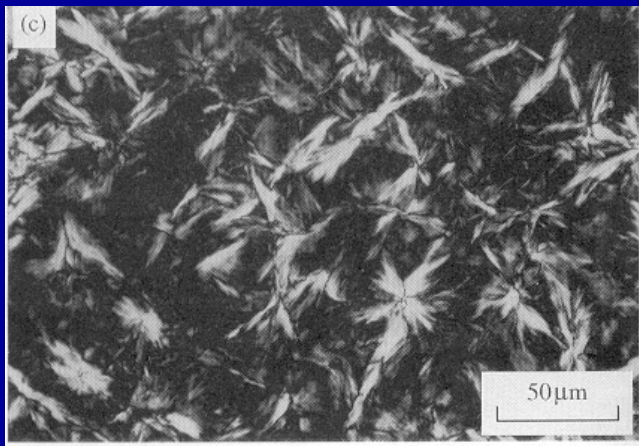
Lamella 3

- Multi-layer lamella  Fig 6.11 & 6.12
 - Screw dislocation
- Hollow Pyramid
 - lattice mismatch



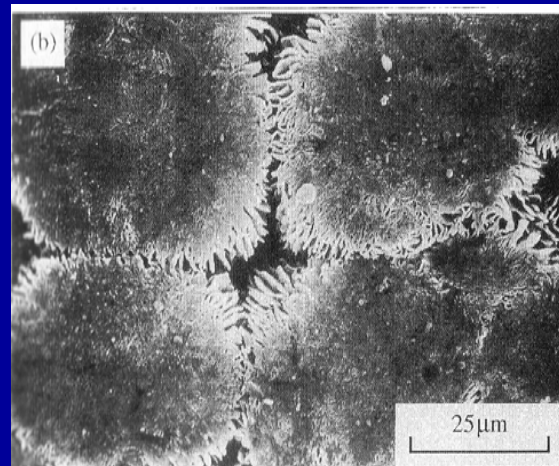
5. Spherulite

- from melt
- Melt crystallized
 - at high temperature – axialite (sheaf-like)
 - at low temperature – spherulite (dendritic growth)
- Axialite to spherulite  Fig 6.19



Spherulite (cont'd)

- Growing of lamellae from the center  Fig 6.17



- under POM ~ Maltese cross pattern  Fig 6.13

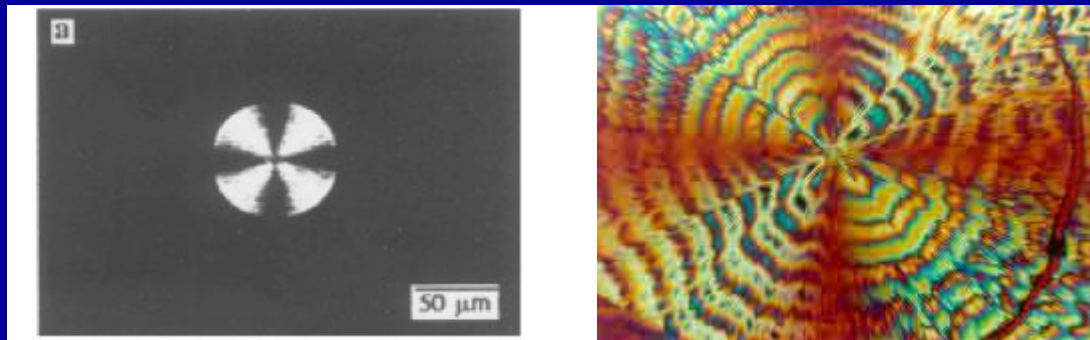


Figure 3. Micrographs of the same spherulite grown in two stages at different T_c : (a) after 16 hours at 140°C, (b) followed by 3 hours at 180°C.

Banded spherulite

- lamellar twisting

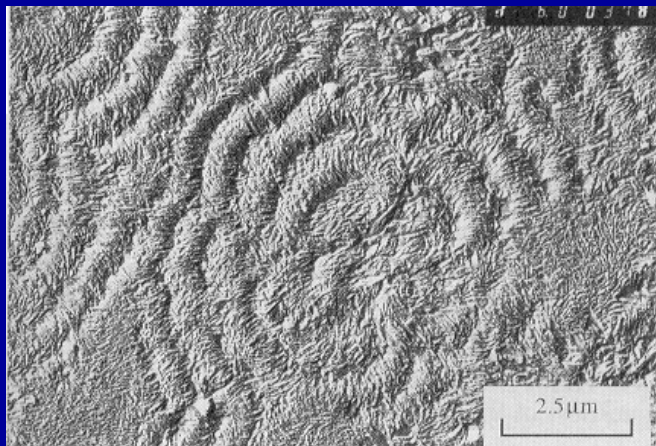
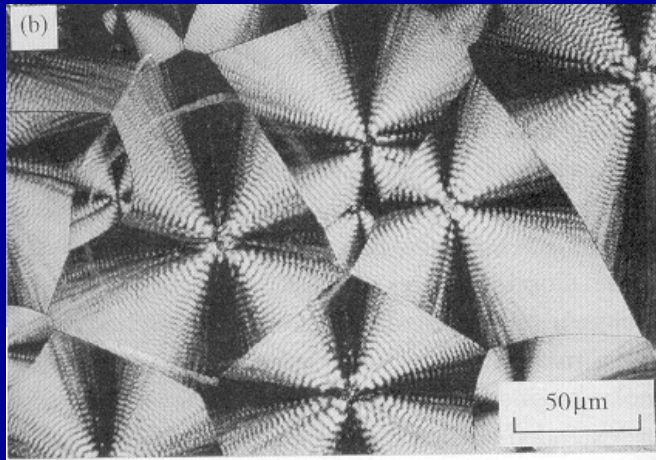
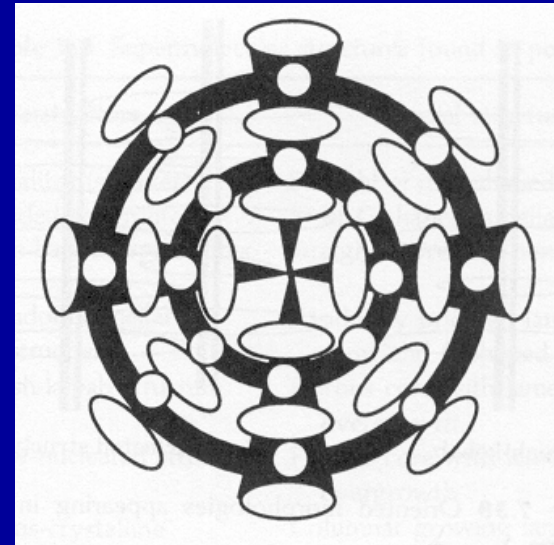


Fig 6.22



Crystallinity

- Crystallinity = % crystallinity = degree of crystallinity
 - X_c = volume of Xtal / total volume = 30 ~ 70% for polymers
- X_c depends on
 - repeat unit structure $X_c(\text{PE}) > X_c(\text{PEster})$
 - cooling rate
- Measuring X_c
 - Volumetric ~ density gradient column
 - Crystallographic ~ WAX
 - Thermal ~ DSC
 - Spectroscopic ~ IR
 - 100% crystal data ← unit cell structure, T_m depression
 - 100% amorphous data ← quenched sample, extrapolation from melt

6. Crystallization

□ Nucleation

- at $T < T_m$
- thermodynamic control
- faster at large $\Delta T (= T_m - T)$



□ Growth

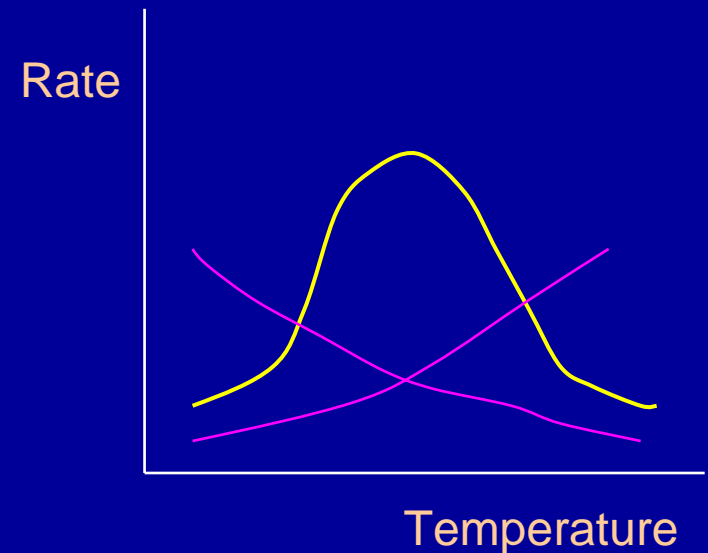
- at $T > T_g$
- diffusion (kinetic) control
- faster at high T

higher chain mobility and lower viscosity at large $T - T_g$

□ Fig 6.27 (T or $T - T_g$)

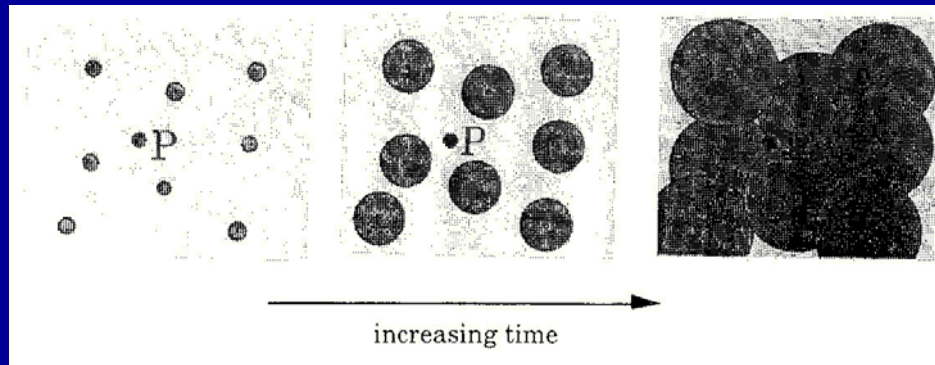
□ Measurement of Crystallization rate

- dilatometry  Fig 6.25
- microscopy  Fig 6.26



Theories of Crystallization Kinetics

- Avrami equation ~ time-dependency





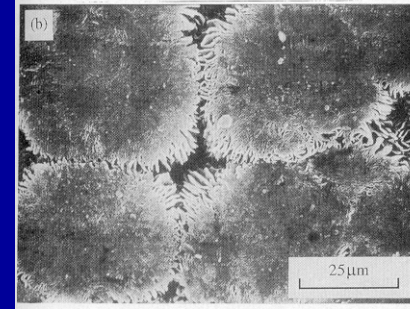
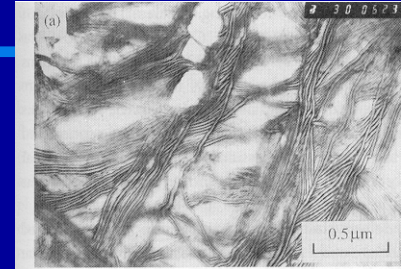
- $p_x = \exp(-E) E^x / x!$
 - » probability of P crossed by x fronts out of E fronts
- $p_0 = \exp(-E) = 1 - X_t = \exp(-Z t^n)$  Eqn (6.16)
- athermal (predetermined) N & G; $n = 3$
- thermal (sporadic) N & G; $n = 4$
- diffusion (of impurity) controlled; $n = 2.5$



 Table 6.4, 6.5

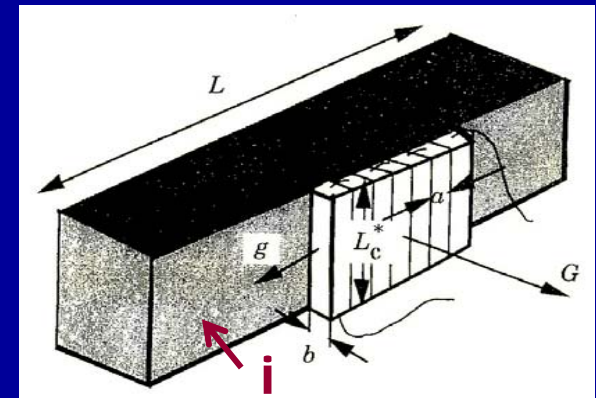
Theories of Crystallization Kinetics

- Keith-Padden theory ~ structure  Fig 6.28
 - $\delta = D / G$ (Diffusion of impurity / radial Growth)
 - $\approx \delta \sim$ lateral dimension of lamellae
 - » small $\delta \sim$ coarse spherulites



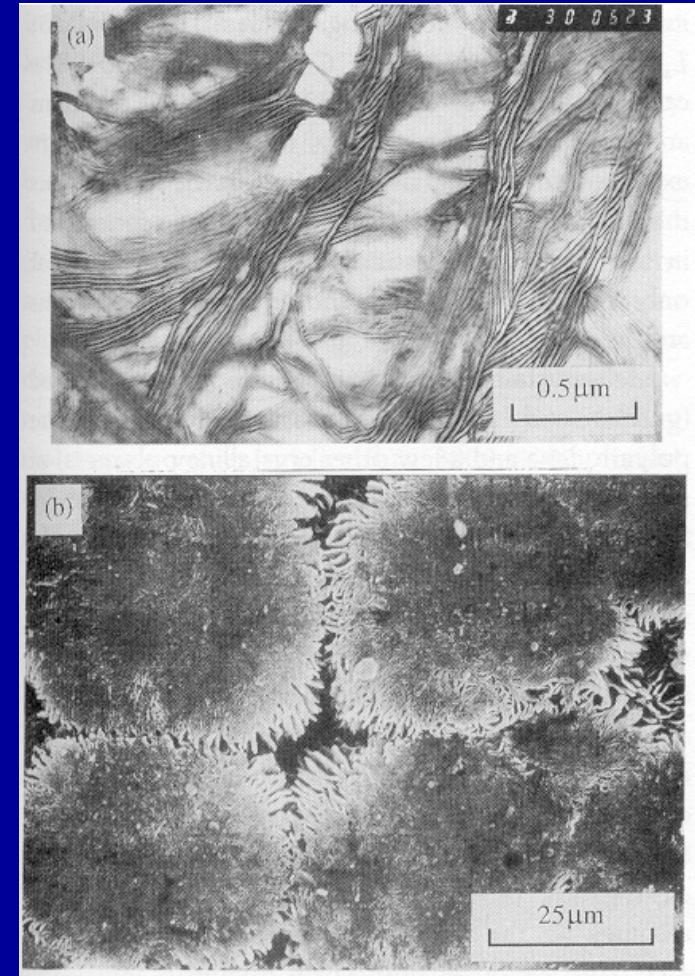
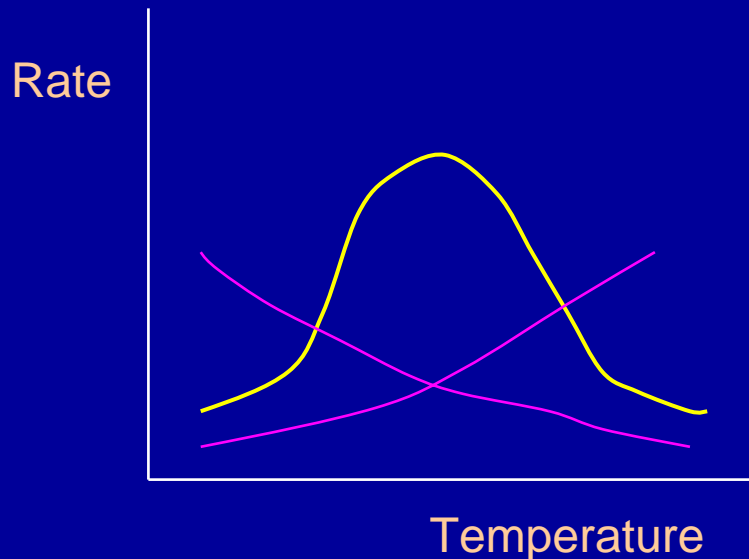
Hoffman

- Lauritzen-~~Hopffmann~~ theory ~ mechanism  Fig 6.30-31
 - three regimes of Xtallization  Fig 6.32
 - » Regime I ~ at high T , $g \gg i$
 - ◆ axialite, adjacent reentry
 - » Regime II ~ at lower T , $g < i$, spherulite
 - » Regime III ~ at even lower T , $g \sim i$
 - ◆ random, switchboard



Crystallization and Property

- Degree of crystallinity
 - modulus, yield strength
- Spherulite size
 - clarity, toughness
- Processing condition







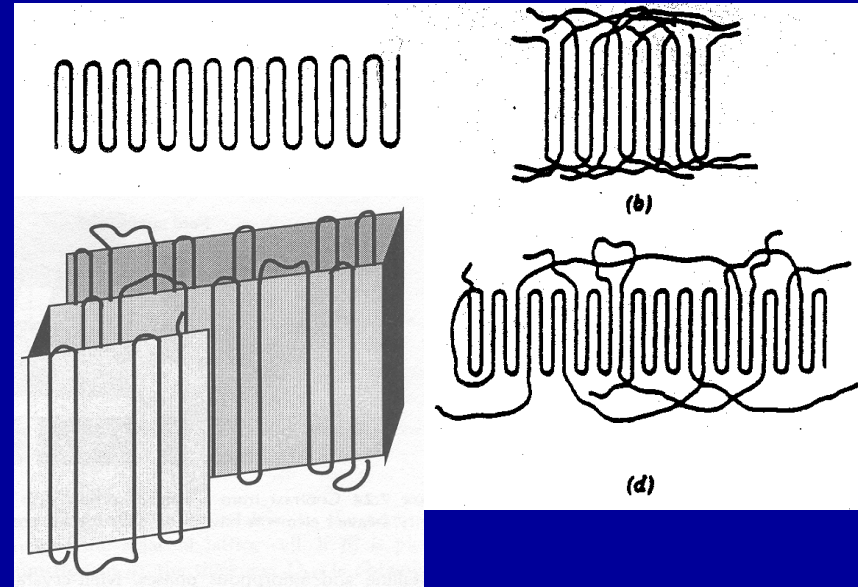
7. Lamella Re-entry

□ 2 models

- Adjacent → folding
 - » IR of mixed crystal
- Switchboard → fringed micelle
 - » etching experiment

□ 4 models

- Adjacent with superfolding  Fig 6.34(a)
- Adjacent without superfolding  Fig 6.34(c)
- Mixed or partial adjacent  Fig 6.34(b)
- Switchboard  Fig 6.34(d)



Lamella Re-entry 2

- (sol'n grown) single crystal lamellae
 - super-folding for intermediate MW
 - » $\langle r^2 \rangle^{1/2} \propto M^{0.1}$ by SANS
 - » 75% adjacent
 - Mixed or partial non-adjacent for high MW
- Melt crystallized lamellae
 - Switchboard + some folding
 - $\langle r^2 \rangle^{1/2} \propto M^{0.5}$ by SANS
 - 3 phases
 - » crystal + interface + amorphous

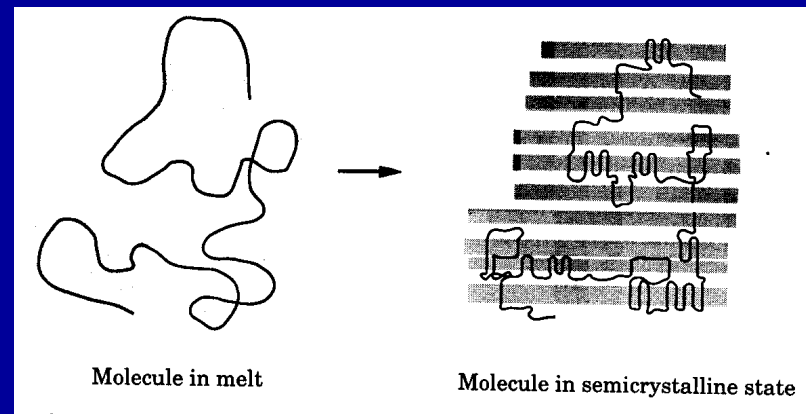




Fig 6.36

8. Melting Temperature

- $\Delta G_f = \Delta H_f - T\Delta S_f$
- At pure state, $T_m^0 = \Delta H_f^0 / \Delta S_f^0$
- Equilibrium melting temperature, T_m^0
 - for infinitely thick crystal (lamella)
 - for infinitely high molecular weight
- Melting point depression by impurity
 - Copolymer
 - Molecular weight
 - 2nd component (low MW or other polymer)
- T_m^0 measured by Hoffman-Weeks plot  Fig 6.40
 - isothermal crystallization at various T_c

9. Structure and T_m

- $T_m^0 = \Delta H_f^0 / \Delta S_f^0$
- $\Delta H_f \sim$ interchain interaction
 $\Delta S_f \sim$ chain flexibility
- Higher for more regular, rigid, closely-packed, and stronger interaction

 (6.54-56) p305-306

- T_g and T_m are correlated.
$$T_g = (0.5 \sim 0.8) T_m \text{ (in K)}$$

Annealing

- At $T_g < T < T_m$
- Due to
 - activated internal mobility
 - promoted higher stability
- Results
 - Growth of crystalline region
→ increase in crystallinity
 - Increase in crystal perfection
→ increase in T_m
 - More stable structure
→ lamella thickening

