
Chapter 11 – 14

Determination of Molar Mass and MM Distribution

(Average) molar mass determination Ch 11-15 sl 2

□ M_n Chapter 11

- colligative property measurement
 - osmotic pressure, vapor pressure, bp, fp
- end-group analysis

$$\bar{M}_n = \frac{\sum N_i M_i}{\sum N_i}$$

□ M_w Chapter 12

- light scattering
- sedimentation (by ultracentrifugation)

$$\bar{M}_w = \frac{\sum N_i M_i^2}{\sum N_i M_i}$$

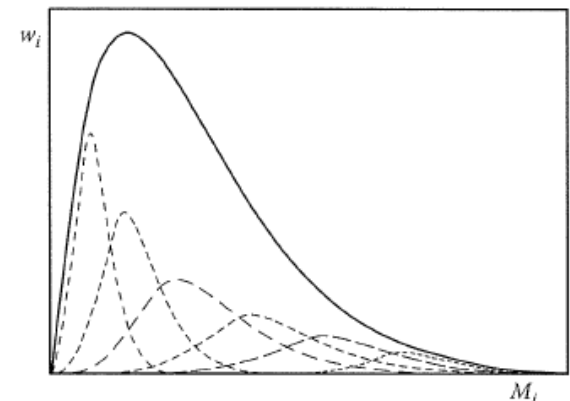
□ M_v Chapter 13

- (dilute solution) viscometry
 - volume \rightarrow mass $\sim M_v$ is a relative MM.

□ MMD [MWD] Chapter 14

- fractionation and GPC
- mass spectroscopy

Fig 14.4 p317



Colligative property

Ch 11-15 sl 3

= property that depends only on the # of solute
(not on the size or nature)

□ measuring gives M_n

□ M_n = total weight / total #

$$\bar{M}_n = \frac{\sum N_i M_i}{\sum N_i}$$

□ the 4 CP's

as MM ↑, signal ↓

TABLE 11.1

The Relative Magnitudes of Colligative Effects for Polymers Dissolved in Benzene at a Concentration of 10 g dm^{-3}

$\bar{M}_n / \text{g mol}^{-1}$	Osmotic Pressure ^a / mm Benzene	Lowering of Vapour Pressure / mm Hg	Elevation of Boiling Point / ° C	Depression of Freezing Point / ° C
10,000	288	799×10^{-5}	272×10^{-5}	577×10^{-5}
100,000	29	80×10^{-5}	27×10^{-5}	58×10^{-5}
500,000	6	16×10^{-5}	5×10^{-5}	12×10^{-5}
1,000,000	3	8×10^{-5}	3×10^{-5}	6×10^{-5}

Note: The values are given in practical units and have been calculated assuming ideal solution behaviour.

^a At 25 °C.

Membrane osmometry (MO)

Ch 11-15 sl 4

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□ osmotic pressure π

$$\mu_1 - \mu_1^0 = -\pi \bar{V}_1 \quad \Delta c \rightarrow \Delta \mu \rightarrow \pi$$

$$\pi = \rho g h$$

$$\mu_1 - \mu_1^0 = \frac{-RT\phi_2}{x} + RT\left(\chi - \frac{1}{2}\right)\phi_2^2 \quad \begin{array}{l} \text{Eqn (10.48)} \\ \text{F-H for dilute soln} \end{array}$$

$$\begin{aligned} \pi &= \frac{RT\phi_2}{x\bar{V}_1} + \frac{RT(1/2 - \chi)\phi_2^2}{\bar{V}_1} \\ &= RT\left(\frac{n_2}{V}\right) + RT\left(\frac{1}{2} - \chi\right)x^2V_1\left(\frac{n_2}{V}\right)^2 \end{aligned} \quad \begin{array}{l} \leftarrow \frac{\phi_2}{x\bar{V}_1} = \frac{n_2}{V} \\ \text{for dil soln p271} \end{array}$$

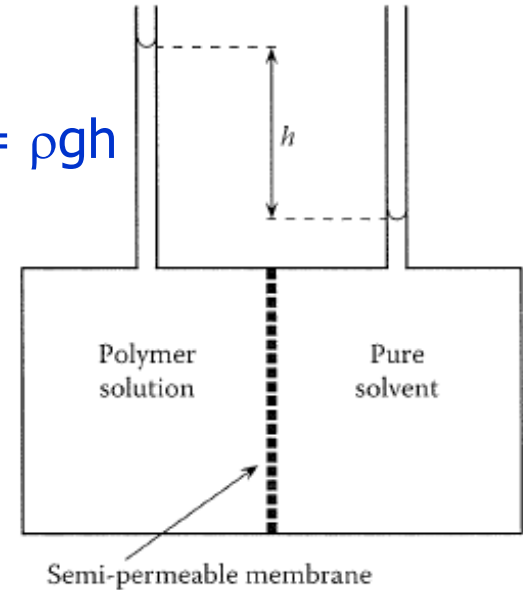


Fig 11.1 p270

$$\square \quad \bar{M}_n = \frac{\sum n_i M_i}{\sum n_i} = \frac{m}{n_2} \rightarrow \frac{n_2}{V} = \left(\frac{m}{V}\right)\left(\frac{n_2}{m}\right) = \frac{c}{\bar{M}_n}$$

□ reduced osmotic pressure π/c

$$\frac{\pi}{c} = \frac{RT}{\bar{M}_n} + \left(\frac{RT}{V_1}\right)\left(\frac{1}{2} - \chi\right)\left(\frac{xV_1}{\bar{M}_n}\right)^2 c$$

$$\square \quad \frac{\pi}{c} = \frac{RT}{\bar{M}_n} + \left(\frac{RT}{V_1} \right) \left(\frac{1}{2} - \chi \right) \left(\frac{xV_1}{\bar{M}_n} \right)^2 c$$

$$\square \quad \text{at } \theta \text{ condition, } \chi = 1/2, \text{ ideal} \quad \left(\frac{\pi}{c} \right)_{\theta} = \frac{RT}{\bar{M}_n}$$

$$\square \quad \text{as } c \rightarrow 0 \quad \left(\frac{\pi}{c} \right)_{c \rightarrow 0} = \frac{RT}{\bar{M}_n}$$

also ideal

□ a virial equation

$$\frac{\pi}{c} = RT \left[\frac{1}{\bar{M}_n} + A_2 c + A_3 c^2 + \dots \right]$$

■ $A_2 = 2\text{nd virial coeff} \sim \text{non-ideality}$

□ MO experiment

□ static or dynamic method

□ for $5E4 < M_n < 1E6$

■ diffusion of solute; small signal (π)

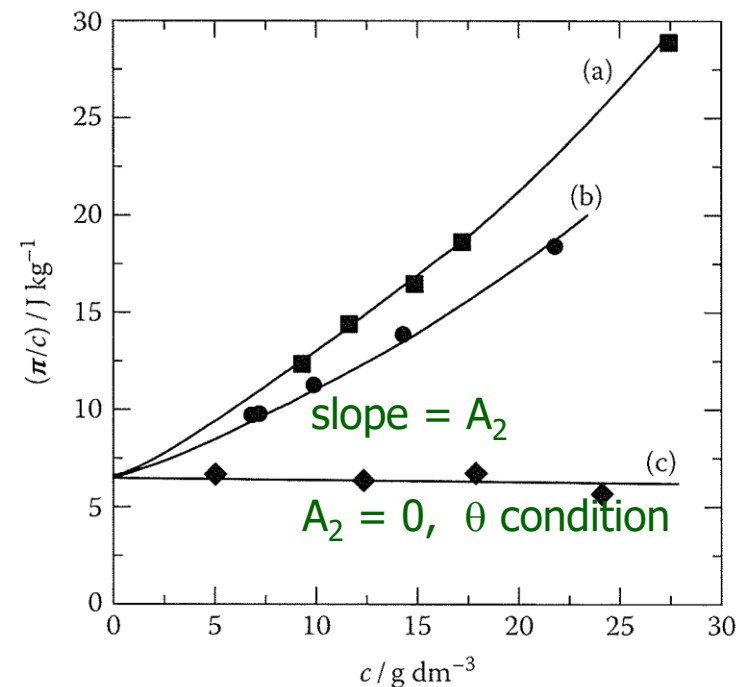


Fig 11.3

Vapor pressure osmometry (VPO)

Ch 11-15 sl 6

pp275-277

- lowering of solvent vapor pressure

$$\frac{1}{P} \left(\frac{dP}{dT} \right) = \frac{\Delta H_v}{RT^2}$$

$$\Delta T_e = - \left(\frac{RT_0^2}{\Delta H_v} \right) \ln a_{1,T_0}$$

$$\frac{\Delta T_{ss}}{c} = K_{ss} \left[\frac{1}{M_n} + A_2^{vpo} c + A_3^{vpo} c^2 + \dots \right]$$

$$\frac{\Delta R_{ss}}{c} = K_{Rss} \left[\frac{1}{M_n} + A_2^{vpo} c + A_3^{vpo} c^2 + \dots \right]$$

- for $M_n < 15000$
 - small signal (ΔR)
 - complementary to MO

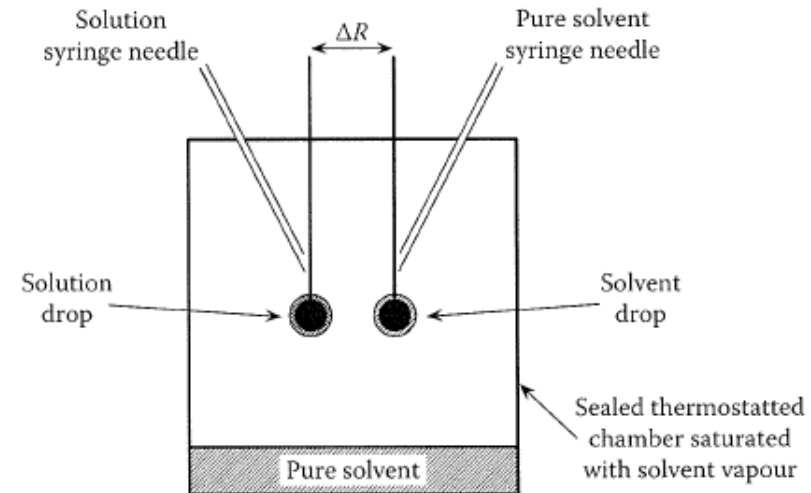


Fig 11.4

$$\Delta c \rightarrow \Delta \mu \rightarrow \Delta P \rightarrow \Delta H \rightarrow \Delta T \rightarrow \Delta R$$

Ebulliometry and cryoscopy

Ch 11-15 sl 7

pp277-278

- ❑ ebulliometry [bp elevation]
 - ❑ $\Delta T_b/c = K_e [(1/M_n) + A_2 c + A_3 c^2 + \text{----}]$
 - K_e calibrated with known MM
 - ❑ foaming
- ❑ cryoscopy [fp depression]
 - ❑ $\Delta T_f/c = K_c [(1/M_n) + A_2 c + A_3 c^2 + \text{----}]$
 - K_c calibrated with known MM
 - ❑ supercooling
- ❑ limited by precision of temperature measurement
 - ❑ useful only for $M_n < 5000$
 - ❑ not used these days

End-group analysis

Ch 11-15 sl 8

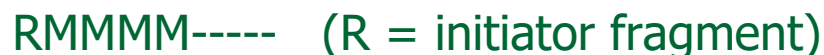
pp278-279

□ step polymers



- titration or spectroscopic methods

□ chain polymers



- spectroscopic methods

□ accurate but limited

- small conc'n of end-group
- for $M_n < 15000$

Light scattering (LS)

Ch 11-15 sl 9

pp281-291

□ light scattering

- static LS ~ time-averaged intensity of light measured
 - elastic [Rayleigh] scattering ~ same freq ~ M_w
 - inelastic [Raman] scattering ~ bond vibration energy
- dynamic LS ~ movement of particle ~ Doppler effect

□ LS by small gas molecule

$$\frac{I_\theta}{I_0} = \frac{8\pi^4 \alpha^2 (1 + \cos^2 \theta)}{\lambda^4 r^2}$$

□ Rayleigh Eqn

- α ~ polarizability of molecule
- α^2 gives M_w
 - ❖ intensity \propto [amplitude] $^2 \propto$ [mass] 2

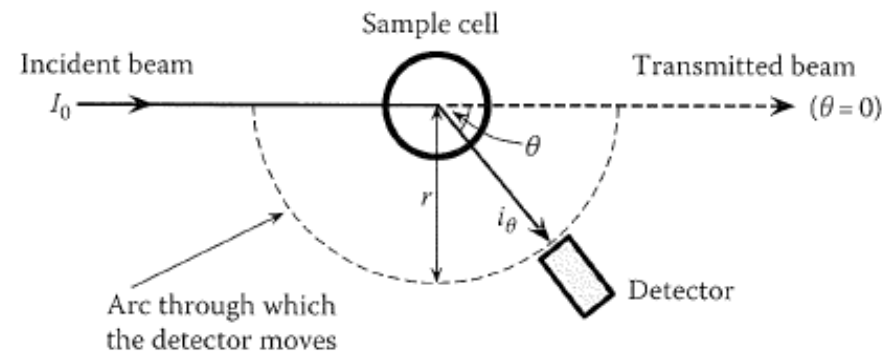


Fig 12.1

$$\bar{M}_w = \frac{\sum N_i M_i^2}{\sum N_i M_i}$$

□ LS by gas (cont'd)

$$\frac{i_\theta}{I_0} = \frac{2\pi^2 (dn/dc)^2 M c (1 + \cos^2 \theta)}{\lambda^4 r^2 N_A}$$

molar mass
↓

$$R = \frac{2\pi^2 (dn/dc)^2 M c}{\lambda^4 N_A}$$

Rayleigh ratio

$$R = \frac{i_\theta r^2}{I_0 (1 + \cos^2 \theta)} \sim \text{measuring}$$

□ LS by (dilute) solution of small molecules

$$\Delta R = R_{\text{solution}} - R_{\text{solvent}}$$

$$\Delta R = \frac{2\pi^2 n_0^2 (dn/dc)^2 c}{N_A \lambda^4 [(1/M) + 2A_2 c + 3A_3 c^2 + \dots]}$$

$$\frac{Kc}{\Delta R} = \frac{1}{M} + 2A_2 c + 3A_3 c^2 + \dots$$

← $K = \frac{2\pi^2 n_0^2 (dn/dc)^2}{N_A \lambda^4} = \text{optical constant (exp'tal condition)}$

then, as $c \rightarrow 0$, $\Delta R/Kc = M$?

- LS by sol'n of large molecules ($D > \lambda/20 \approx 20 \text{ nm}$)
 - asymmetry + interference \rightarrow angular dependence of i

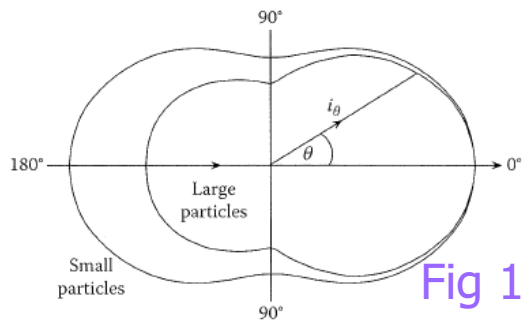


Fig 12.2

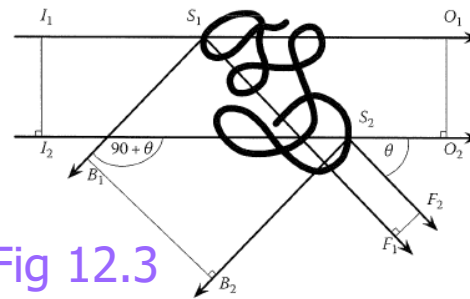
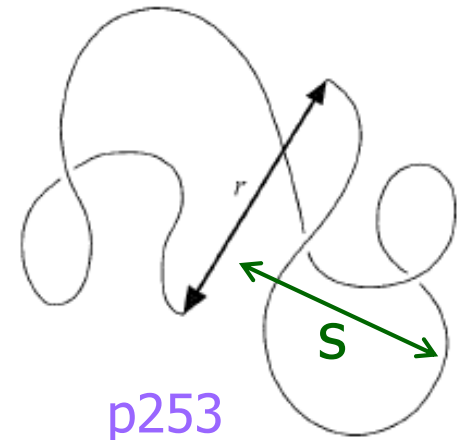


Fig 12.3

$$\frac{Kc}{\Delta R_\theta} = \frac{1}{P(\theta)} \left[\frac{1}{M} + 2A_2c + 3A_3c^2 + \dots \right]$$

$$P(\theta) = \frac{\Delta R_\theta}{\Delta R_{\theta=0}} = \text{scattering factor} \\ [\text{angular dependency of } \Delta R]$$

$$\frac{1}{P(\theta)} \approx 1 + \left(\frac{q^2 \langle s^2 \rangle}{3} \right) \quad \begin{array}{l} q = \text{scattering vector} \propto \sin(\theta/2) \\ \langle s^2 \rangle = \text{MS radius of gyration} \end{array}$$



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LS by polymer sol'n (cont'd)

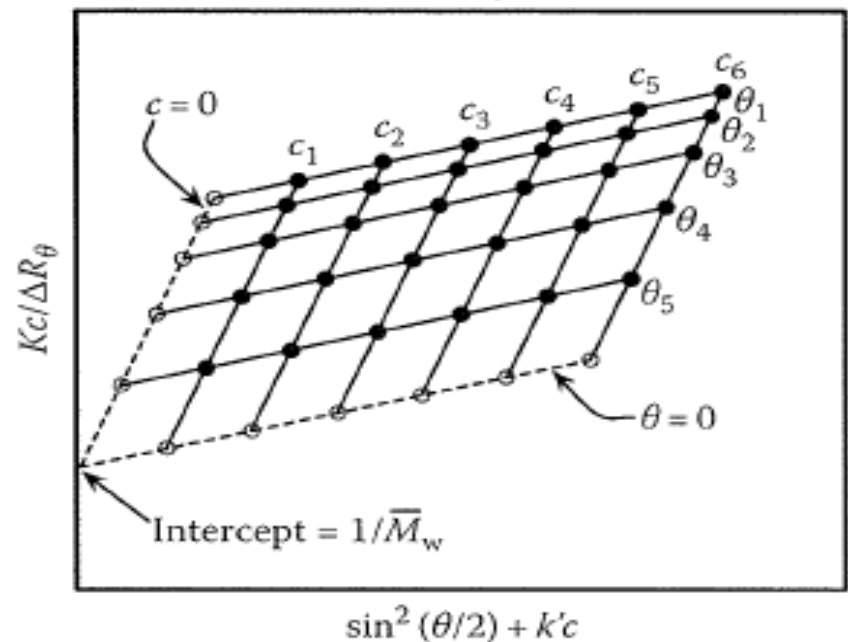
$$\frac{Kc}{\Delta R_\theta} = \left[\frac{1}{M} + 2A_2c + 3A_3c^2 + \dots \right] \left[1 + \left(\frac{16\pi^2 n_0^2 \sin^2(\theta/2)}{3\lambda^2} \right) \langle s^2 \rangle \right]$$

$$\left(\frac{Kc}{\Delta R_\theta} \right)_{\substack{c \rightarrow 0 \\ \theta \rightarrow 0}} = \frac{1}{\bar{M}_w}$$

for $2E4 < M_w < 5E6$

- small intensity (i_θ)
- cancelling if too large
- sample must be dust-free

Fig 12.4 'Zimm plot'



SAXS and SANS

pp294-296

Ch 11-15 sl 13

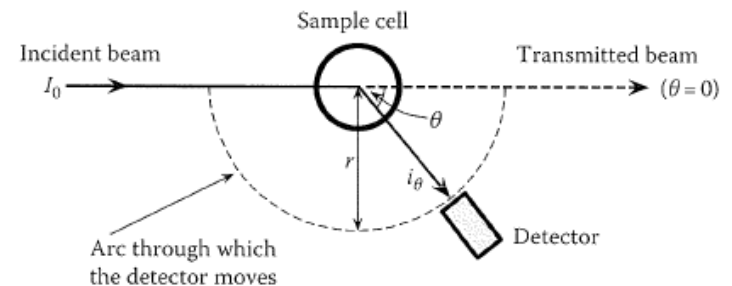
- ❑ LS uses visible light
- ❑ XS and NS
 - ❑ uses smaller λ
 - smaller scale structure
 - ❑ at small angle $< 2^\circ$
 - to avoid angular dependence $P(\theta)$
 - ❑ detector far from sample
 - to separate from $\theta = 0^\circ$
 - ❑ high penetration depth
 - can be used for solid polymers
 - ❑ SAXS for phase structure
 - morphology of blends, copolymers
 - ❑ SANS for dimension of chain in bulk
 - with deuterated sample

TABLE 12.1

Values of the Reciprocal Scattering Vector q^{-1} Calculated from Equation 12.23 for Some Typical λ, θ Combinations^a

Scattering Method	λ / nm	θ	q^{-1} / nm
SAXS	0.154	1.0°	1.4
	0.154	0.5°	2.8
	0.154	0.2°	7.0
SANS	1.0	1.0°	9.1
	1.0	0.5°	18.2
	1.0	0.2°	45.6
Light scattering	632.8	60.0°	67.1
	632.8	30.0°	129.7
	632.8	10.0°	385.2

^a The refractive index n_0 of the medium is assumed arbitrarily to be 1.0 for SAXS and SANS, and 1.5 for light scattering.

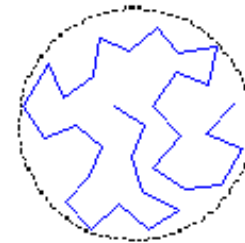


Dilute solution viscometry [DSV]

Ch 11-15 sl 14

- polymer/solvent friction is
 - non-draining = moves together
 - (most) flexible chains
 - free-draining = solvent pass-through
 - only when very short or rod-like
- volume V_h and viscosity η

§10.4 pp262-264



$$\square \quad \eta = \eta_0 \left[1 + \left(\frac{5}{2} \right) \phi_2 \right] \quad \text{Einstein eqn}$$
$$\phi_2 = \left(\frac{c}{M} \right) N_A V_h \quad \frac{[\text{g/L}]}{[\text{g/mol}]} \text{ [mol] [L]}$$

$$\square \quad \eta_{sp} = (\eta - \eta_0) / \eta_0 = \left(\frac{5}{2} \right) \left(\frac{c}{M} \right) N_A V_h$$

η_0 = solvent viscosity
 ϕ_2 = vol fraction of solute
 c = conc'n of polymer [g/L]
 M = mol wt of polymer
 V_h = hydrodynamic volume
= vol of equiv. sphere
 η_{sp} = specific viscosity

□ intrinsic viscosity $[\eta]$ and MM

$$\square [\eta] = \lim_{c \rightarrow 0} \left(\frac{\eta_{sp}}{c} \right) = \left(\frac{5}{2} \right) \frac{N_A V_h}{M}$$

$$= \Phi_0^s \alpha_\eta^3 \left(\frac{\langle s^2 \rangle_0^{3/2}}{M} \right) = K_\theta \alpha_\eta^3 M^{1/2} \quad \text{Flory-Fox eqn}$$

$$K_\theta = \Phi_0^s \left(\frac{\langle s^2 \rangle_0^{3/2}}{M} \right)$$

$$\begin{aligned} V_h &= (4/3) \pi R_h^3 \\ R_h &\propto \alpha_\eta \langle s^2 \rangle_0^{1/2} \\ \alpha_\eta &= \text{expansion parameter} \\ \alpha_\eta &\propto x^\Delta \propto M^\Delta \end{aligned}$$

eqn (10.79) – (10.89)
pp260-262

$$\square [\eta] = KM^a \quad \text{Mark-Houwink-Sakurada [MHS] eqn} \quad a = 0.5 + 3\Delta$$

Common Name	Name Proposed by IUPAC ^a	Symbol and Definition ^b
Relative viscosity	Viscosity ratio	$\eta_r = \eta / \eta_0$
Specific viscosity	—	$\eta_{sp} = \eta_r - 1$
Reduced viscosity	Viscosity number	$\eta_{red} = \eta_{sp} / c$
Inherent viscosity	Logarithmic viscosity number	$\eta_{inh} = \{ \ln(\eta_r) \} / c$
Intrinsic viscosity	Limiting viscosity number	$[\eta] = \lim_{c \rightarrow 0} (\eta_{red})$

viscosity-average mol wt, M_v

pp300-301

$$[\eta] = KM^a \quad [\eta] = K\bar{M}_v^a$$

$$\bar{M}_v = \left(\frac{\sum n_i M_i^{1+a}}{\sum n_i M_i} \right)^{1/a}$$

$0.5 \leq a \leq 0.8$

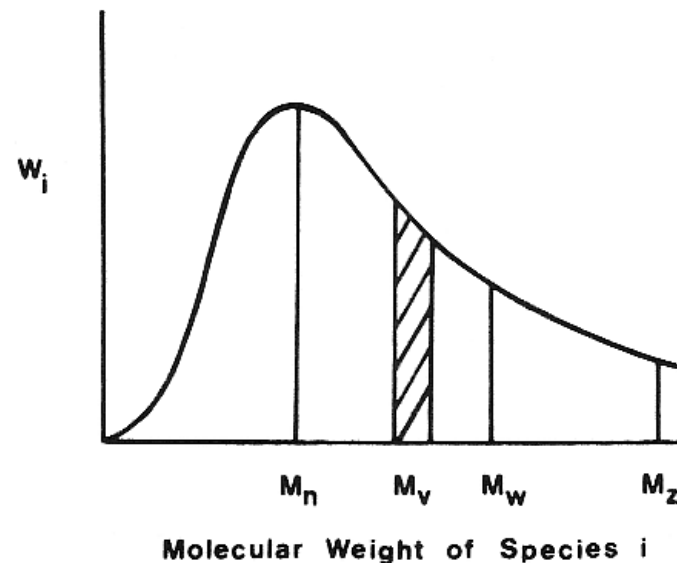
0.5 at θ -condition; often > 0.7

$M_n < M_v < M_w$

M_v close to M_w

$a = 1 \rightarrow M_v = M_w$

$a = -1 \rightarrow M_v = M_n$



DSV experiment pp299-304

$$\eta_{sp} = k_0[\eta]c + k_1[\eta]^2c^2 + k_2[\eta]^3c^3 + \dots$$

$$\frac{\eta_{sp}}{c} = [\eta] + k_H[\eta]^2c \quad \text{Huggins eqn}$$

$$\frac{\ln(\eta_r)}{c} = [\eta] + k_K[\eta]^2c \quad \text{Kraemer eqn}$$

$$[\eta] = K\bar{M}_v^a$$

- K, a from handbook Table 13.2
- at the same solvent and temp

cautions

- Temp control < 0.01 °C
- $t_0 > 100$ s (laminar)
- $c < 1$ g/dL (Newtonian)

$$\frac{V}{t} = \frac{\pi r^4 P}{8\eta l}$$

$$\eta_r = \frac{\eta_s}{\eta_0} = \frac{\rho_s t_s}{\rho_0 t_0}$$

$$\eta_{sp} = \eta_r - 1$$

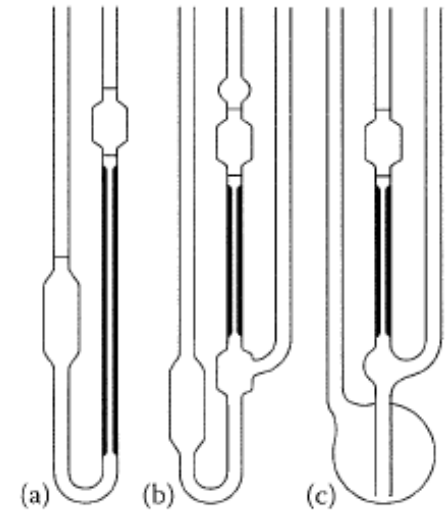


Fig 13.2

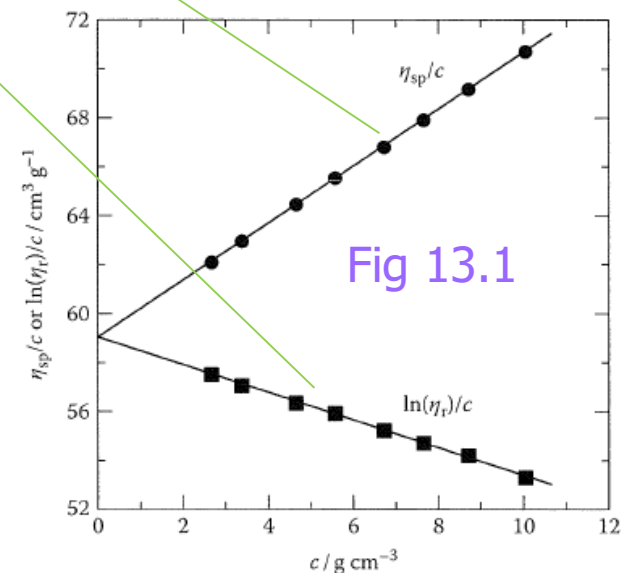


Fig 13.1

Sedimentation (by ultracentrifugation) Ch 0-1 sl 18

- ❑ molecule with higher MM sediments faster
 - ❑ ultracentrifuge for sedimentation > diffusion
- ❑ sedimentation equilibrium exp't
 - ❑ 5000 – 10000 rpm, days for equili
 - ❑ refractive index [conc'n] profile
 - ❑ M_w and M_z
- ❑ sedimentation velocity exp't
 - ❑ 60000 – 70000 rpm, dynamic
 - ❑ movement of boundary layer
 - ❑ M_w
- ❑ MMD?
 - ❑ width of boundary layer

MMD and fractionation

Ch 11-15 sl 19

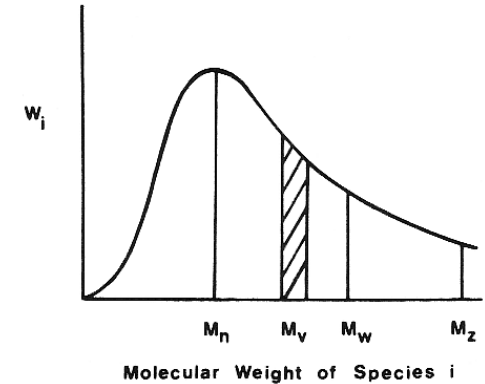
MM of polymers has distribution

- MM averages

- MMD [MWD]

- $PDI = M_w/M_n$

- MMD affects property? some on rheology

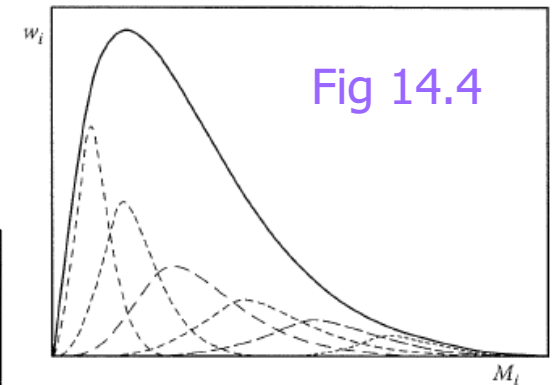
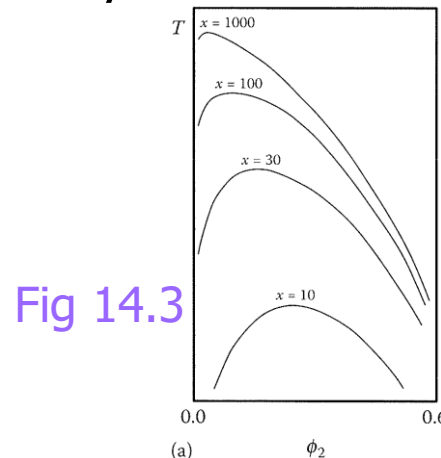
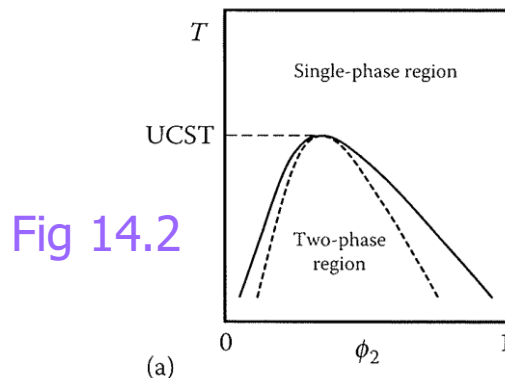


fractionation p317

- separate fractions of narrow(er) distribution

- by lowering Temp or adding non-solvent

- manually or (semi-)automatically



Gel permeation chromatography (GPC)

Ch 11-15 sl 20

pp318-325

■ a size-exclusion chromatography (SEC)

- separation by size using porous gel
- Larger molecules elute earlier.

■ elution volume V_e and size

$$V_e = V_0 + V_i \exp\left(\frac{-A_s \bar{L}}{2}\right) \quad \text{smaller } V_e \text{ for larger } L$$

$L \sim$ diameter of molecule in sol'n

■ V_e to mol wt

$$[\eta] = \left(\frac{5}{2}\right) \frac{N_A V_h}{M} \rightarrow V_h = \left(\frac{2}{5}\right) \frac{[\eta] M}{N_A}$$

$$[\eta] = KM^a$$

$$M(V_e) = \left[\frac{([\eta]M)_{V_e}}{K} \right]^{1/(1+a)}$$

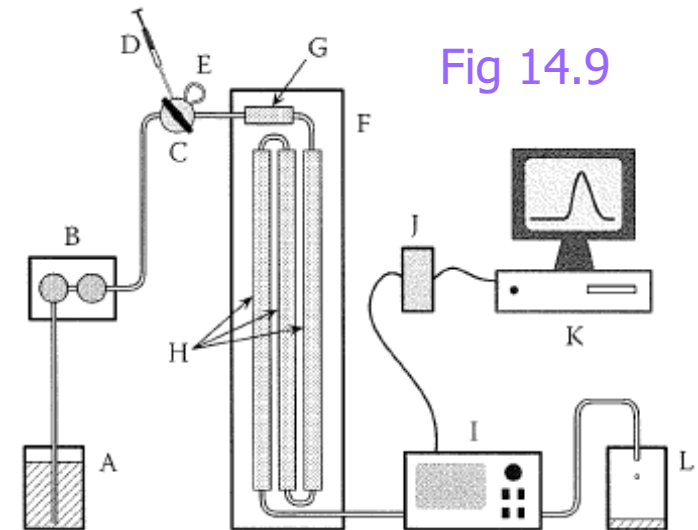


Fig 14.9

Fig 14.5

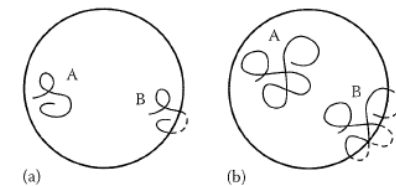
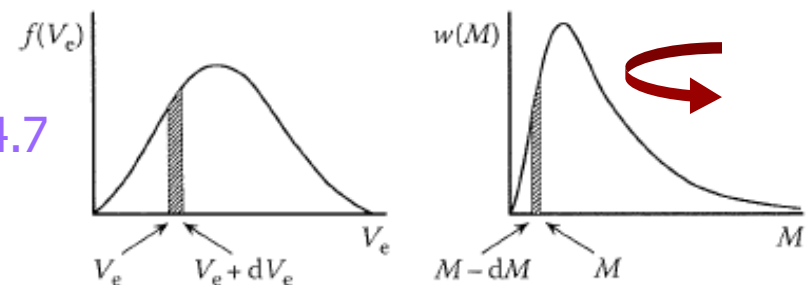


Fig 14.7



□ V_e to mol wt (cont'd)

□ 'universal calibration'

$$M(V_e) = \left[\frac{([\eta]M)_{V_e}}{K} \right]^{1/(1+a)}$$

- Many polymers fall on the same curve on $\log[\eta]M - V_e$ plot.

'universal calibration curve'

□ experiment

- Run GPC with sample and (anionic PS) standards.
- For each **fraction** (V_e), get M .
 - $[\eta]_{PS}M_{PS} = [\eta]_{sample}M_{sample}$
 - $K_{PS}M_{PS}^{a(PS)+1} = K_{sample}M_{sample}^{a(sample)+1}$
- Calculate M_n , M_w , and MMD.

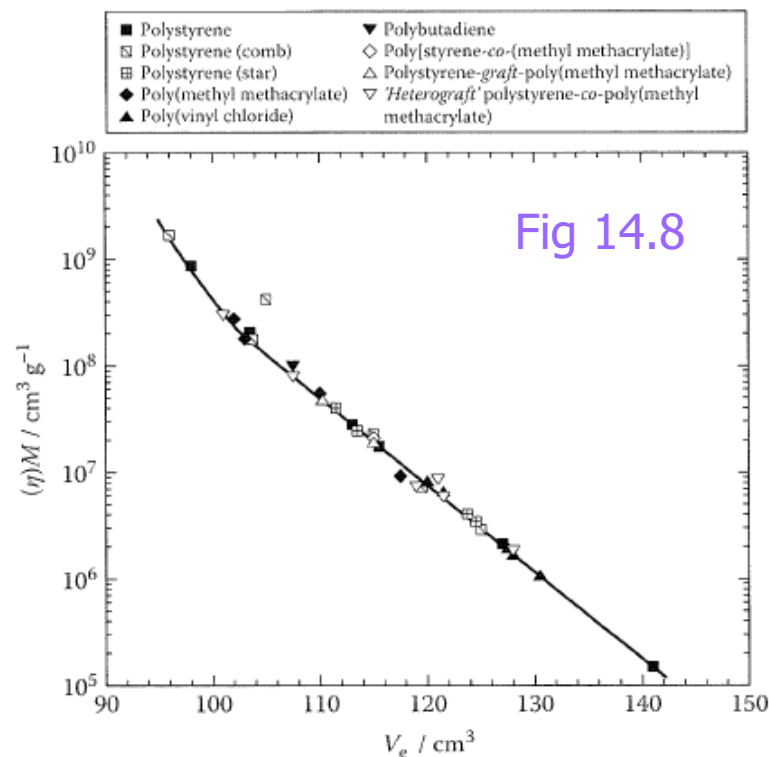


Fig 14.8

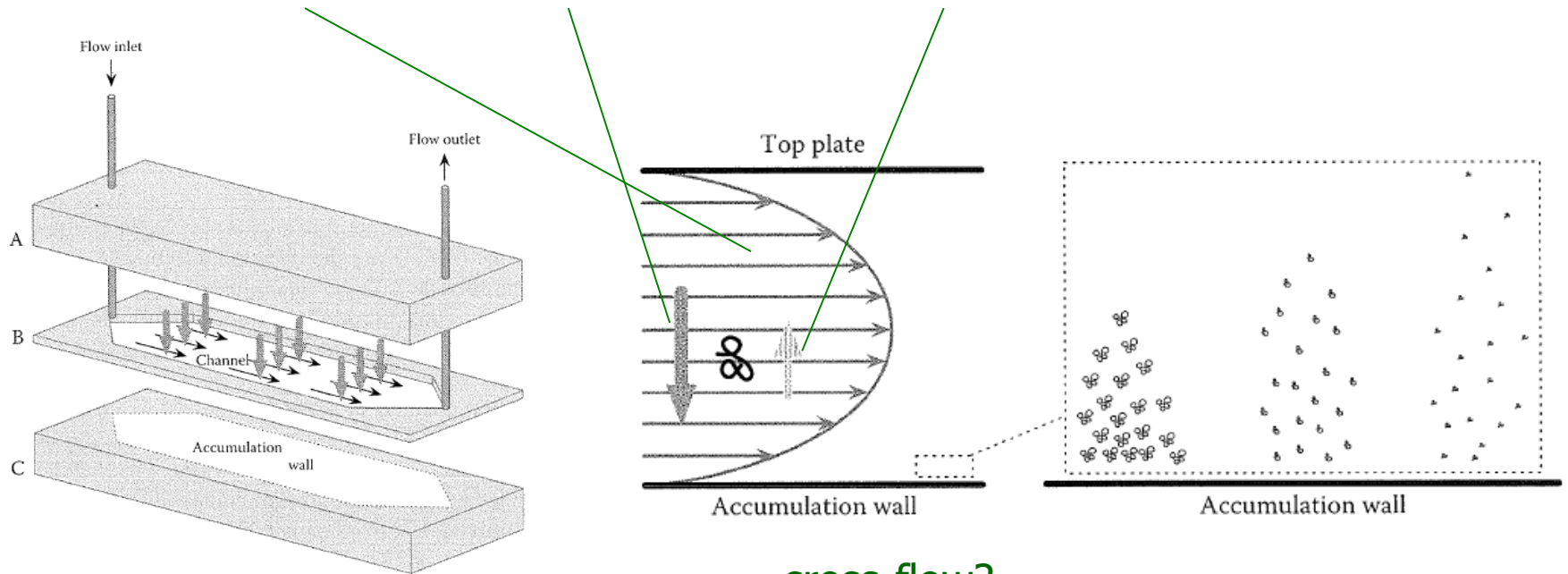
for each 'fraction'
 $M \approx M_n \approx M_v \approx M_w$

$$[\eta]M = KM^{1+a}$$

Field-flow fractionation

Ch 11-15 sl 22

□ forward-flow vs cross-flow and diffusion



cross-flow?
non-solvent, centrifuge, etc

□ useful for

- higher MM (than for GPC)
- biopolymers
- sometimes preparatory

Mass spectroscopy [--metry; MS]

Ch 11-15 sl 23

pp331-340

❑ MS determines MM

- ❑ by detecting molecular ion, M^+
- ❑ in vapor phase
- ❑ For polymers;
 - non-volatile
 - fragmentation

❑ MALDI-ToF technique

- ❑ a soft ionization
 - choice of the **matrix** critical
- ❑ still for not-too-high MM
 - incomplete ionization, fragmentation, etc

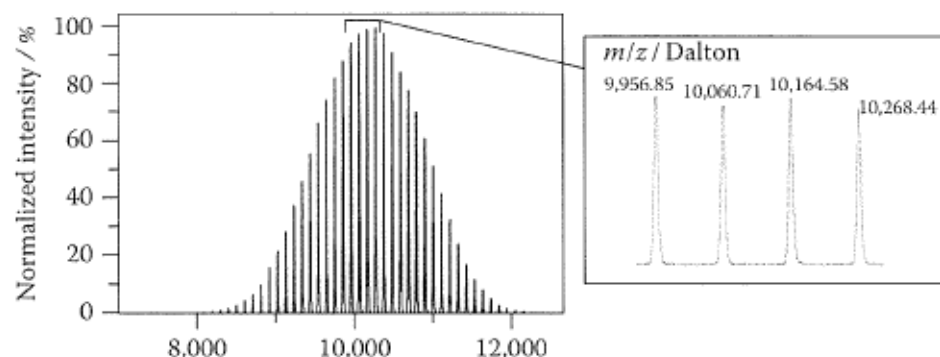


Fig 14.13

Chapter 15

Structure Determination

Analysis of
chemical composition and microstructure

□ UV/Vis

- double bonds and conjugations
- change in structures, especially outdoor
- additives (with =)

□ IR

- characteristic groups ~ identification
- copolymer composition, degree of branching, crystallinity, etc

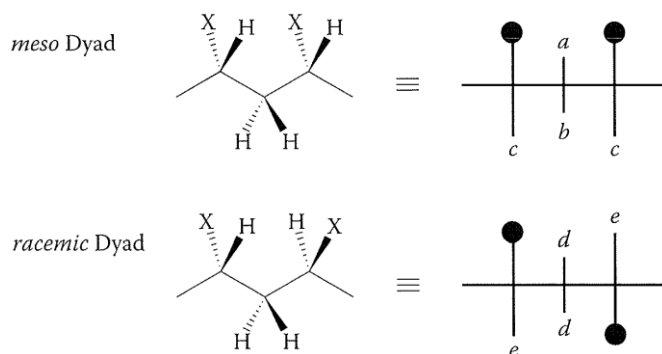
□ Raman

- vibrational spectroscopy detecting change in polarizability
 - IR detects change in dipole moment.
 - mutual exclusion and complementary → application similar to IR
- sensitive to local environment and orientation of bonds
 - some more applications; LAM, micromechanics, etc

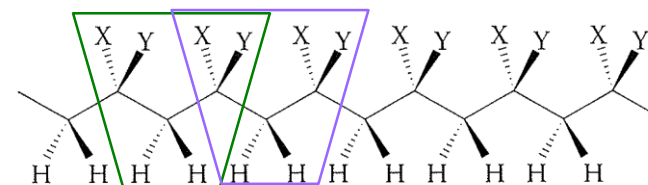
NMR

Ch 11-15 sl 26

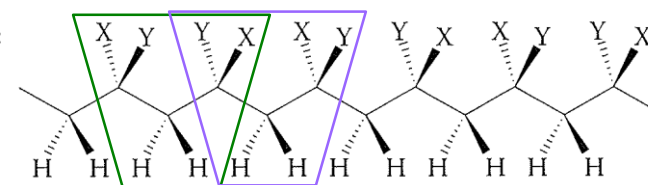
- ❑ chemical structure
- ❑ chain dynamics and motion
 - ❑ using relaxation time and peak width
- ❑ sequence distribution in copolymers
- ❑ **tacticity**
 - ❑ affects crystallinity (and hence properties)



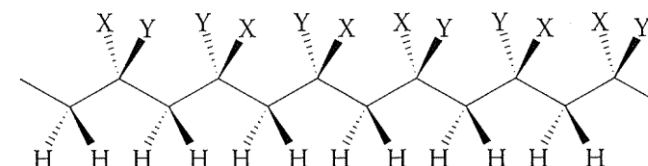
Isotactic



Syndiotactic



Atactic



~mm~

TABLE 15.3
Sequences in Vinyl Homopolymers and Their Mole Fractions according to Bernoullian Statistics

Sequence Type	Designation ^a	Projection	Mole Fraction of Sequence in Terms of P_m ^a
Dyad	m		P_m
	r		$1 - P_m$
Triad	mm		P_m^2
	mr (and rm)		$2P_m(1 - P_m)$
	rr		$(1 - P_m)^2$
Tetrad	$mmmm$		P_m^3
	mmr (and rmr)		$2P_m^2(1 - P_m)$
	rmr		$P_m(1 - P_m)^2$
	$mrmm$		$P_m^2(1 - P_m)$
	rrm (and mrr)		$2P_m(1 - P_m)^2$
	rrr		$(1 - P_m)^3$

isotactic

syndiotactic

atactic

