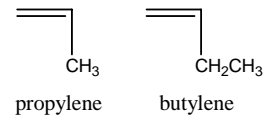


Total of 100 points. Each question is worth of 5 points, unless otherwise noted.

1. For a polymer sample with  $M_w$  of 500000, the plateau shear modulus,  $G_N^0$ , was determined as 1.0 MPa at 27 °C.
  - (a) Estimate the Young's modulus of this polymer sample at the glassy state.
  - (b) What would be the plateau shear modulus, when the molecular weight is doubled up?
  - (c) What would be the slope of  $\log \eta - \log M_w$  plot of this polymer sample when it is melted? Explain your answer with a calculation.  
[You may need this:  $R = 8.3 \times 10^7$  dyn cm/mol K]

2. Choose one in the bracket, and explain why using "free volume."

(a) The glass transition temperature of polybutylene is [lower, higher] than that of polypropylene.



(b) As the molecular weight of a polymer increases, glass transition temperature [increases, decreases] up to a certain molecular weight.

(c) As the crosslinking density of a network polymer increases, glass transition temperature [increases, decreases].

3. You observed that zero-shear viscosity of polystyrene ( $T_g = 100$  °C) decreases by 5% by raising the temperature from 200 °C to 210 °C.
  - (a) What would be the % change in zero-shear viscosity, when the temperature is raised from 210 °C to 230 °C?
  - (b) What would be the % change in zero-shear viscosity, when the temperature is raised from 150 °C to 160 °C?

4. Answer the following questions briefly in your own words.

- (a) Why is rubber called an 'entropy spring'?
- (b) What is the 'Kauzmann paradox,' and how is the paradox resolved?
- (c) What is the 'two-thirds rule,' and why is it observed for polymers?

5. Fill the blanks. No need to explain. [Each right answer counts 3 points.]

The reason why fracture stress of polymers is much lower than (a) \_\_\_\_\_, which is about one-tenth of its modulus, is the existence of flaw like crack, inclusion, or notch.

In front of a crack, there are two factors that lead to brittle fracture, (b) \_\_\_\_\_ and (c) \_\_\_\_\_. In front of a sharp crack (b) makes the stress very high, but the stress is cut-off by (d) \_\_\_\_\_, which let the material in front of a crack undergo plastic deformation. This is the reason why measured (e) \_\_\_\_\_ is much higher than that expected based on the (f) \_\_\_\_\_ created by crack propagation.

However, the plastic deformation in front of a crack is restricted by (c), which is the result of triaxial stress state in front of a crack and let the material yield at a higher stress than its inherent (d). If the effective (d) is higher than its (g) \_\_\_\_\_, the polymer

fractures in a brittle manner. At surface or edge of a fracture test specimen, the stress state is in (h) \_\_\_\_\_ condition, and a larger volume of material yields. At inside of a specimen, (c) cut the (i) \_\_\_\_\_ to about one-third of that at surface. This is the reason for (j) \_\_\_\_\_ transition when specimen thickness varies.

6. The drawing below shows yield and crazing criteria of polymers.
- (a) Is this polymer ductile or brittle upon uniaxial tension? Explain your answer using a drawing.
  - (b) Is this polymer ductile or brittle upon biaxial compression? Explain your answer using a drawing.
  - (c) Show how the drawing changes when the strain rate increases.

