



NUMERICAL SIMULATION OF MICROPOROSITY FORMATION IN A356 ALUMINUM ALLOY

Ahmad Bahmani Current Status of Structural Materials 2014-04-08

A. Bahmani, N. Hatami, N. Varahram, P. Davami, "A Mathematical Model for Prediction of Microporosity Formation in Aluminium Alloy A356", International Journal of Advanced Manufacturing Technology, Vol. 64, pp 1313-1321, 2013,.



CONTENTS

- Introduction
- Governing Equations
- Experimental Procedures
- **Results and Discussions**
- Model Verification
- Industrial Application
- Conclusions





INTRODUCTION

- Definitions and Theories
 - Casting, Solidification, Feeding
 - Porosity and its importance
- Numerical Model for Prediction of Microporosities
 - Criteria (Niyama (G/V), Hansen Sahm (G/ $t_f^{3/2}$), ...)
 - Conservation Equations



- Solution of Heat, Solidification, Continuity and Gas Conservation Equations
- Presentation of a 3D Model to:
 - Solve Pressure Distribution, and Porosity Contents in Interdendritic Region
 - Considering Variant Properties (K, d, Cp, HTC and so on)
- Verification of Model with Experiments
- Coupling with SUTCAST, Commercial Software



Porosities Classification

- Based on Size (Micro and Macro)
 - Microporosities and Microshrinkages
- Based on the Formation Mechanism (Gas and Shrinkage)



NUCLEATION AND GROWTH

Pores Formation

- Solidification Shrinkage (Density Change via the Transformation)
- Gas Segregation (Solubility Change via the Transformation and Temperature Decrease)

Level of Growth

- Heterogenous Nucleation (on Oxides or Inclusions)
- Spherical Growth by Gas Diffusion
- Growth Limitation by Solidification Front (Dendrites)
- Deformation of Pore within Solidification Growth





GOVERNING EQUATIONS

- Permeability
- Dendrite Arm Spacing
- Darcy's Equation
- Viscosity
- Hydrogen Solubility
- Gas, Liquid and Surface Tensions Relationship
- Pressure Applied Due to Surface Tension
- Pore Radius

$$K = \frac{1 - g_s^3}{g_s^2} \times \frac{\lambda_2^2}{180}$$
$$\lambda_2 = \operatorname{at_f^b}$$
$$u_l = -\frac{\kappa}{\mu g_l} (\operatorname{grad} P_L - \rho_L g_l)$$
$$\mu = \mu_0 \exp (E_{\mu} / RT)$$
$$[H]_l = K_l \sqrt{P_i}$$
$$P_g = P_1 + P_{\sigma}$$
$$P_{\sigma} = \frac{2\sigma_{Pl}}{r}$$
$$r = \frac{\lambda_2}{4}$$



NUMERICAL METHOD

Gas Conservation

Discretized Eq.

• Continuity

$$\rho_{L}[H]_{o} = \rho_{s}g_{s}[H]_{s} + \rho_{L}g_{L}[H]_{L} + \frac{\alpha g_{p}P_{g}}{T}$$

$$\frac{\partial \rho}{\partial t} + div(\rho u) = 0$$

$$\frac{\partial \overline{\rho}}{\partial t} - \rho_{l}\frac{\partial g_{p}}{\partial t} + \rho_{1}div\left(-\frac{k}{\mu g_{l}}(\operatorname{grad}P_{l} - \rho_{l}g)\right) = 0$$
Where $\overline{\rho} = \rho_{l}g_{l} + \rho_{s}g_{s}$

$$g_{p}^{n} = g_{p}^{o} + \frac{\overline{\rho}^{n} - \overline{\rho}^{o}}{\rho_{l}} - \rho_{l}\frac{K\Delta t}{\mu}\left[\frac{P^{o}_{i+1jk} + P^{o}_{i-1jk} - 2P^{o}_{ijk}}{\Delta Z^{2}}\right]$$
Fig. 2. Flowchart of program for computation of pressure and porosity in each time setp.

7

- Experimental Procedure
- Aluminum A356
- Steel Chill
- Hydrogen Degassing by Rotary Degassing
- Gas Measurement by High Scan Measurement Device
- Microporosity Measuring by Densitometer (Archimedes Equation)





SECONDARY DENDRITE ARM SPACING (SDAD)

✓ Good Agreement between the Results

✓ Increase in SDAS by getting Distance from Chill (Decreasing of Cooling Rate)





/łæ°Łį#ĭßæ ˘°,ĿĿÊ

- Suddenly Drop of Permeability by Initial formed Dendrites
- Change in the Curvature at Coherency Point
- Lower Permeability for Finer Structures





MICROPOROSITY DISTRIBUTION

✓ More porosities in Regions with Less Cooling Rates
✓ Good Agreement between Results in Low and High Contents of Gas





COUPLING WITH SUTCAST SIMULATION SOFTWARE (LOW PRESSURE DIE CASTING)

- Counter Gravity Feeling
- High Quality Casting
- Expensive Process





COUPLING WITH SUTCAST SIMULATION SOFTWARE (LOW PRESSURE DIE CASTING)

 Microporosity and Feeding Behavior Prediction for Industrial Castings







CONCLUSIONS

- Proportionally Good Agreements between Results
- Solidification development makes the metal more complex and more difficult to feed
- Increasing of cooling rate make the structure finer and resistance of solid domain increases but anyway there is less time in this regions for pore's growth.
- In lower values of cooling rate more porosities observed which is related to the diffusion of gas and shrinkage growth.
- The Model combined with SUTCAST simulation software and examined successfully in an industrial casting.



THANKS FOR your attention