Chapter 9

The Behavior of Solutions

9.7 Nonideal Solutions





Figure 9.8 Activities in the system iron-nickel at 1600°C. (From G.R.Zellars, S.L.Payne, J.P.Morris, and R.L.Kipp, "The Activities of Iron and Nickel in Liquid Fe-Ni Alloys," *Trans.* AIME (1959), vol. 215, p. 181.)

9.7 Nonideal Solutions





Figure 9.10 Activity coefficients in the system iron-nickel at 1600°C.

Figure 9.9 Activities in the system iron-copper at 1550°C. (From J.P.Morris and G.R.Zellars, "Vapor Pressure of Liquid Copper and Activities in Liquid Fe-Cu Alloys," *Trans. AIME* (1956), vol. 206, p. 1086.)

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Figure 9.11 Activity coefficients in the system iron-copper at 1550°C.

(A) Method 1



Figure 9.12 A schematic representation of the variation of $\log a_B$ with X_B/X_A in a binary solution, and illustration of the application of the Gibbs-Duhem equation to calculation of the activity of component *A*.

(B) Method 2



(C) Method 3
<The α-Function>

ln.

$$\alpha_{i} = \frac{\ln \gamma_{i}}{(1 - X_{i})^{2}}$$
(9.56)

$$\alpha_{A} = \frac{\ln \gamma_{A}}{X_{B}^{2}} \text{ and } \alpha_{B} = \frac{\ln \gamma_{B}}{X_{A}^{2}}$$
(9.57)

$$\gamma_{A} = \alpha_{A} X_{B}^{2} \text{ and } \ln \gamma_{B} = \alpha_{B} X_{A}^{2}$$
(9.57)

$$d \ln \gamma_{B} = 2\alpha_{B} X_{A} dX_{A} + X_{A}^{2} d\alpha_{B}$$
(9.58)

$$d \ln \gamma_{A} = -\frac{X_{B}}{X_{A}} 2\alpha_{B} X_{A} dX_{A} - \frac{X_{B}}{X_{A}} X_{A}^{2} d\alpha_{B}$$
(9.59)

$$= -2X_{B} \alpha_{B} dX_{A} - X_{B} X_{A} d\alpha_{B}$$
(9.59)

<The α -Function> $\int d(x,y) = \int y dx + \int x dy$ $\int X_B X_A d\alpha_B = \int d(X_B X_A \alpha_B) - \int \alpha_B d(X_B X_A)$ $\ln \gamma_A = - \left[2X_B \alpha_B dX_A - \left[d(X_B X_A \alpha_B) + \right] \alpha_B d(X_B X_A) \right]$ $= -\int 2X_B \alpha_B dX_A - X_B X_A \alpha_B + \int \alpha_B X_B dX_A + \int \alpha_B X_A dX_B$ $= - \int 2X_B \alpha_B dX_A - X_B X_A \alpha_B + \int \alpha_B X_B dX_A - \int \alpha_B X_A dX_A$ $= -X_B X_A \alpha_B - \int (2X_B - X_B + X_A) \alpha_B dX_A$ $= -X_B X_A \alpha_B - \int_{a}^{x_A = x_A} \alpha_B dX_A$ (9.61)

<The α -Function>



Figure 9.14 Application of the Gibbs-Duhem equation to determination of the activity of iron in the system iron-nickel.

Figure 9.15 Application of the Gibbs-Duhem equation to determination of the activity of iron in the system iron-copper.

<The α -Function>



Figure 9.16 The variation of α_{Ni} with composition in the system iron-nickel.

<The Relationship between Henry's and Raoult's Laws>



Figure 9.17 The variation of α_{Cu} with composition in the system iron-copper.

(D) Method 4

<Direct calculation of the integral molar Gibbs free energy of mixing>

(9.62)

<Direct calculation of the integral molar Gibbs free energy of mixing>



Figure 9.18 Illustration of the direct calculation of the integral molar Gibbs freggnergies of mixing in the systems iron-copper at 1550°C and iron-nickel at 1600°C.

<Direct calculation of the integral molar Gibbs free energy of mixing>



Figure 9.19 The integral molar Gibbs free energies of mixing in the systems iron-copper at 1550°C and iron-nickel at 1600°C.

$$\alpha_A = \alpha_B = \alpha$$

$$\alpha = \frac{\alpha'}{RT} \qquad (9.69)$$

$$\Delta \overline{H}_i^M \neq 0$$
 and $\Delta \overline{S}_i^M = \Delta \overline{S}_i^{M,\text{id}} = -R \ln X_i$

<The properties of a regular solution>



2. $G^{XS} = \Delta H^M \neq f(T)$

It is thus seen that G^{XS} for a regular solution is independent of T

$$\left(\frac{\partial G^{\rm XS}}{\partial T}\right)_{P,\rm comp} = -S^{\rm XS}$$

 S^{XS} for a regular solution is zero, then G^{XS} , and hence ΔH^{M} , are independent of T

3. Application of RS

$$\overline{G}_A^{\rm XS} = RT_1 \ln \gamma_{A(T_1)} = RT_2 \ln \gamma_{A(T_2)} = \alpha' X_B^2$$

For a regular solution

$$\frac{\ln \gamma_A \text{ at the temperature } T_2}{\ln \gamma_A \text{ at the temperature } T_1} = \frac{T_1}{T_2}$$
(9.77)

Eq. (9.77) is of considerable practical use in converting activity data for a regular solution at one temperature to activity data at another temperature

9.9 Regular Solutions



Figure 9.20 Activities in the system thallium-tin. (From J.H.Hildebrand and J.N. Sharma, "The Activities of Molten Alloys of Thallium with Tin and Lead," J. Am. Chem. Soc. (1929), vol. 51, p. 462.)

9.9 Regular Solutions





Figure 9.21 Activities coefficients in the system thallium-tin. (From J.H.Hilde-brand and J.N.Sharma, "The Activities of Molten Alloys of Thallium with Tin and Lead," J. Am. Chem Soc. (1929), vol. 51, p. 462.)

Figure 9.22 Log γ₁₁ vs. X²_{Sn} in the system thallium-tin. (From J.H.Hildebrand and J.N.Sharma, "The Activities of Molten Alloys of Thallium with Tin and Lead," J. Am. Chem. Soc (1929), vol. 51, p. 462.)

4. What can we use R.S?

 $\Delta H^{M} = bX_{A}X_{B} \text{ or } G^{XS} = b'X_{A}X_{B}$ where b and b' are unequal $\Delta S^{M} \neq \Delta S^{M,\text{id}}$

$$G^{\rm XS} = RTX_B \int_0^{X_A} \frac{\ln \gamma_A}{X_B^2} \, dX_A$$

$$\begin{array}{c} 300 \\ 280 \\ 260 \\ 260 \\ 240 \\ 220 \\ 200 \\ 500 \\ \hline \\ 600 \\ \hline \\ \\ 7,K \end{array}$$

Figure 9.23 The variation of the product αT with T in the system T1–Sn.

$$\gamma_A/X_B^2 = \alpha, G^{XS} = RT\alpha X_A X_B$$



Figure 9.24 The molar enthalpy, entropy, and Gibbs free energy of mixing of thallium and tin at 414°C.

- Application of the statistical mixing model to two components which have equal molar volumes
- In solution the interatomic forces exist only between neighboring atoms
- The energy of the solution is the sum of the interatomic bond energies
- Consider 1 mole of a mixed crystal containing N_A atoms of A and N_B atoms of B

 $X_A = \frac{N_A}{N_A + N_B} = \frac{N_A}{N_O}$ and $X_B = \frac{N_B}{N_O}$ N_o is Avogadro's number

- 1. A-A bonds the energy of each of which is E_{AA}
- 2. B-B bonds the energy of each of which is E_{BB}
- 3. A-B bonds the energy of each of which is E_{AB}

<Quasi-Chemical Model>

• N_A atoms in pure A

 $2 \times$ the number of A-A bonds

= the number of atoms × the number of bonds per atom

$$P_{AA} = \frac{1}{2} N_A z$$
$$P_{BB} = \frac{1}{2} N_B z$$

• If $\Delta H^{M} = 0$, then the mixing of the N_A atoms with the N_B atoms of B is random \leftarrow random mixing

 $\Delta S^{M} = \Delta S^{M,\text{id}} = -R(X_A \ln X_A + X_B \ln X_B)$

- |ΔH^M|≤RT The mixing of the atoms is also approximately random
- Consider two neighboring lattice sites in the crystal The probability that site 1 is occupied by an A atom is

 $\frac{\text{the number of } A \text{ atoms in the crystal}}{\text{the number of lattice sites in the crystal}} = \frac{N_A}{N_O} = X_A$

- The probability that site 2 is occupied by a B atom is X_B
- Thus the probability of A-B pair is $2X_A X_B$,A-A pair is X_A^2 , B-B pair is X_B^2

- Henry's law requires that γ_A , and hence $ln\gamma_A$, approach a constant value as X_B approaches unity Thus,

$$\gamma_A \rightarrow \ln \gamma_A^\circ = \Omega/RT$$

The applicability of the statistical model to real solutions decreases as the magnitude of Ω increases, i.e., if the magnitude of EAB is significantly greater or less than the average of EAA and EBB then random mixing of the A and B atoms cannot be assumed



Figure 9.25 Illustration of the origins of deviation from regular solution behavior

9.11 Subregular Solutions

$$\Omega = a + bX_B + cX_B^2 + dX_B^3 + \cdots$$
 (9.92)

$$G^{\rm XS} = (a + bX_B)X_A X_B \tag{9.93}$$

$$\overline{G}_{A}^{XS} = aX_{B}^{2} + bX_{B}^{2}(X_{B} - X_{A})$$
(9.94a)

$$\overline{G}_B^{\rm XS} = aX_A^2 + 2bX_A^2 X_B \tag{9.94b}$$

$$\frac{dG^{\rm XS}}{dX_B} = 0$$

$$G^{XS} = aX_B + 2(b - a)X_B^2 - bX_B^3$$

$$\frac{dG^{\rm XS}}{dX_B} = a + 2(b - a)X_B - 3bX_B^2 = 0$$

$$X_B = \frac{2(b-a) \pm 2\sqrt{b^2 + ab + a^2}}{6b}$$

9.10 Subregular Solutions

$$G^{\rm XS} = (a_0 + b_0 X_B) X_A X_B \left(1 - \frac{T}{\tau}\right)$$
(9.95)

$$S^{XS} = -\frac{\partial G^{XS}}{\partial T}$$

$$= \frac{(a_0 + b_0)X_A X_B}{\tau}$$
(9.96)

$$\Delta H^{M} = G^{XS} + TS^{XS} = (a_{0} + b_{0}X_{B})X_{A}X_{B}\left(2 - \frac{T}{\tau}\right)$$
(9.97)



Figure 9.26 Excess molar Gibbs free energy curves generated by the subregular solution model.