2015 Fall

"Phase Equilibria in Materials"

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• How is the reaction in three phase region among liquid, α and β ?

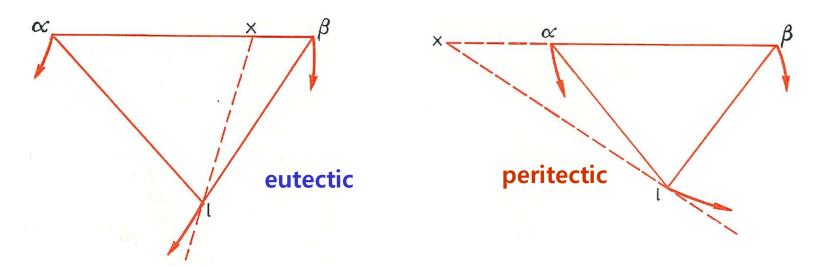


Fig. 149. Criteria for distinguishing eutectic and peritectic reactions in ternary three-phase equilibrium.

The tangent to the liquid curve at a particular temperature is extrapolated to meet the tie line connecting the α and β phases.

- 1) If the extrapolated line intersected the $\alpha\beta$ tie line, the equilibrium was considered to be eutectic
- 2) If it met the $\alpha\beta$ tie line only when the latter was extrapolated, the equilibrium was considered to be peritectic.

Similarly, a eutectoid reaction could be distinguished from a peritectoid and a monotectic from a syntectic.

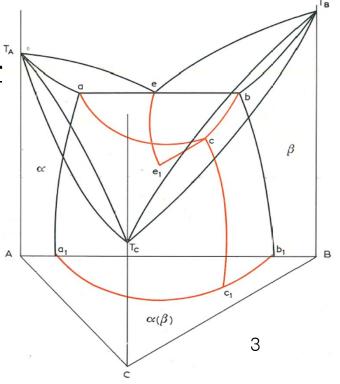
How is the reaction in three phase region?

<Hillert's criterion>

Basically, the reaction we can expect is eutectic reaction

 $(/ \rightarrow \alpha + \beta)$. But, in reality, we can have eutectic and peritectic reaction depending on the relative amount of three phase.

The <u>average composition of the alloy</u> then determines <u>for a particular temperature whether</u> the reaction will be eutectic or peritectic.



A small change in temperature, d7, causes a small change in the composition and amounts of each phase, but not of the alloy itself,

$$X_A$$
, X_B , X_C = constant, ΔX_A = 0, and $\Delta m_\alpha + \Delta m_\beta + \Delta m_l = 0$

$$\Delta X_A = m_\alpha \cdot \Delta X_A^\alpha + m_\beta \cdot \Delta X_A^\beta + m_l \cdot \Delta X_A^l + X_A^\alpha \cdot \Delta m_\alpha + X_A^\beta \cdot \Delta m_\beta + X_A^l \cdot \Delta m_l = 0$$

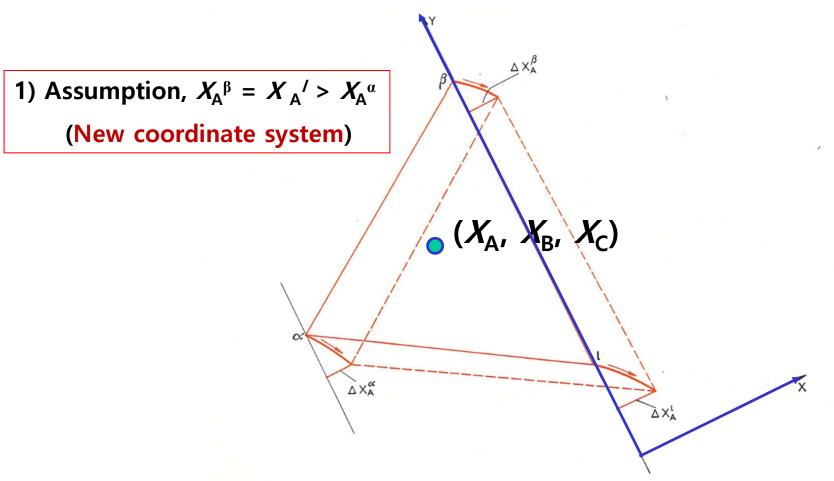


Fig. 150. Illustration of Hillert's criterion for distinguishing eutectic and peritectic reaction in ternary three-phase equilibrium; ————, equilibrium at T; —————, equilibrium at T-dT.

To simplify the calculation,

Assumption, $X_A^{\beta} = X_A^{\prime} > X_A^{\alpha}$ (New coordinate system) $\Delta m_{\beta} + \Delta m_{l} = -\Delta m_{\alpha}$

$$\Delta m_{\beta} + \Delta m_{l} = - \Delta m_{\alpha}$$

$$\Delta X_{A} = m_{\alpha} \cdot \Delta X_{A}^{\alpha} + m_{\beta} \cdot \Delta X_{A}^{\beta} + m_{l} \cdot \Delta X_{A}^{l} + X_{A}^{\alpha} \cdot \Delta m_{\alpha} + X_{A}^{\beta} \cdot \Delta m_{\beta} + X_{A}^{l} \cdot \Delta m_{l} = 0$$

$$\underline{-X_{A}^{\alpha} \Delta m_{\alpha} - X_{A}^{\beta} \Delta m_{\beta} - X_{A}^{l} \Delta m_{l}} = m_{\alpha} \Delta X_{A}^{\alpha} + m_{\beta} \Delta X_{A}^{\beta} + m_{l} \Delta X_{A}^{l}$$

$$\Delta m_{\alpha}(X_{A}^{\beta}-X_{A}^{\alpha})=m_{\alpha}\Delta X_{A}^{\alpha}+m_{\beta}\Delta X_{A}^{\beta}+m_{l}\Delta X_{A}^{l}$$

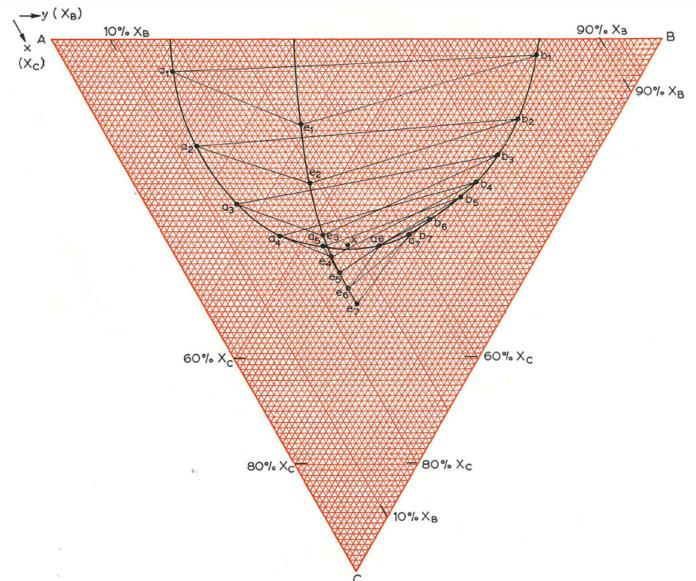
| Sign | Assumption | Sign | | |
|---------------------------------------------------------------------------|-------------------------------------------------------------------------------|----------------------------------------------------------------------------------------------------|--|--|
| $\Delta m_{\alpha}(X_{\mathbf{A}}{}^{\beta} - X_{\mathbf{A}}{}^{\alpha})$ | $X_{\mathbf{A}}{}^{\beta} = X_{\mathbf{A}}{}^{l} > X_{\mathbf{A}}{}^{\alpha}$ | $m_{\alpha} \Delta X_{A}{}^{\alpha} + m_{\beta} \Delta X_{A}{}^{\beta} + m_{l} \Delta X_{A}{}^{l}$ | | |
| $\Delta m_{\beta}(X_{A}{}^{\alpha}-X_{A}{}^{\beta})$ | $X_{\mathbf{A}}^{\alpha} = X_{\mathbf{A}}^{l} > X_{\mathbf{A}}^{\beta}$ | $m_{\alpha} \Delta X_{A}{}^{\alpha} + m_{\beta} \Delta X_{A}{}^{\beta} + m_{l} \Delta X_{A}{}^{l}$ | | |
| $\Delta m_l (X_{\mathbf{A}}{}^{\alpha} - X_{\mathbf{A}}{}^l)$ | $X_{\mathbf{A}}{}^{\alpha} = X_{\mathbf{A}}{}^{\beta} > X_{\mathbf{A}}{}^{l}$ | $m_{\alpha}\Delta X_{A}{}^{\alpha} + m_{\beta}\Delta X_{A}{}^{\beta} + m_{l}\Delta X_{A}{}^{l}$ | | |

here, Δm_{α} : change of α phase fraction with ΔT

| Δm_{lpha} | Δm_{eta} | Δm_l | | |
|-------------------|------------------|--------------|--------------------------------|------------|
| + | + | - | $l \rightarrow \alpha + \beta$ | eutectic |
| + | - | - | $l + \beta \rightarrow \alpha$ | peritectic |
| - | + | - | $l + \alpha \rightarrow \beta$ | peritectic |

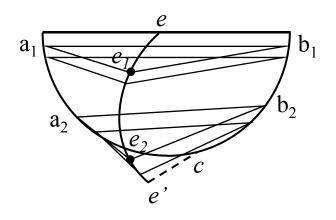
Hillert's criterion indicates that the relative amounts of the α , β and liquid phases (the average alloy composition) are of importance in determining the type of reaction.

• three phase regions $a_1e_1b_1$, $a_2e_2b_2$, ..., $a_7e_7(b_7)$ projected on the concentration triangle.

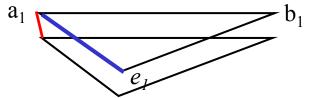


To determine whether the reaction is always a monovariant eutectic type, irrespective of alloy composition within the three-phase region, we apply Hillert's criterion to each pair of isotherms.

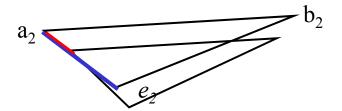
Relative position of vertex in tie triangle with ∆T

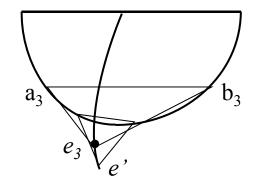


① Slope of tangent line at $a_1 > \text{slope of line } a_1e_1$

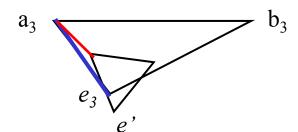


② Slope of tangent line at a_2 = slope of line a_2e_2





3 Slope of tangent line at a_3 < slope of line a_3e_3

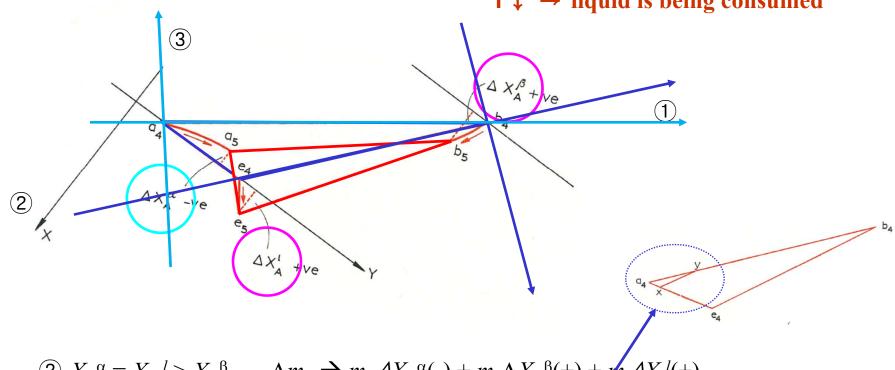


Consider tie triangle $a_4e_4b_{4,}$ and $a_5e_5b_5$

(1)
$$X_{A}^{\beta} = X_{A}^{l} > X_{A}^{\alpha}$$
 (3) $X_{A}^{\alpha} = X_{A}^{\beta} > X_{A}^{l}$

 $m_{\alpha} \Delta X_{\mathbf{A}}^{\alpha}(+) + m_{\beta} \Delta X_{\mathbf{A}}^{\beta}(+) + m_{l} \Delta X_{\mathbf{a}}^{l}(+) \rightarrow \Delta m_{\alpha}(+) \Delta m_{l}(-) \rightarrow m_{\alpha} \Delta X_{\mathbf{A}}^{\alpha}(-) + m_{\beta} \Delta X_{\mathbf{A}}^{\beta}(-) + m_{l} \Delta X_{\mathbf{a}}^{l}(-)$

 $T \downarrow \rightarrow liquid is being consumed$



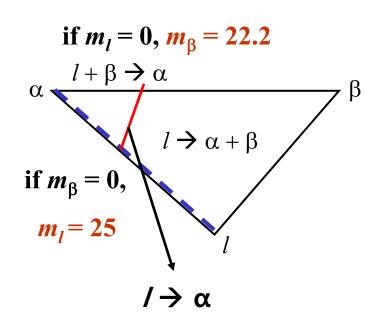
②
$$X_{\mathbf{A}}{}^{\alpha} = X_{\mathbf{A}}{}^{l} > X_{\mathbf{A}}{}^{\beta}$$
 $\Delta m_{\beta} \rightarrow m_{\alpha} \Delta X_{\mathbf{A}}{}^{\alpha}(-) + m_{\beta} \Delta X_{\mathbf{A}}{}^{\beta}(+) + m_{l} \Delta X_{\mathbf{a}}{}^{l}(+)$

 $\Delta m_{\alpha}(+)$, $\Delta m_{\beta}(-)$; if m_{α} is very larger than m_{β} and $m_{\beta} \Delta m_{\beta}(-) \rightarrow (/+\beta \rightarrow \alpha)$

if m_{α} is much smaller than m_{β} and $m_{\parallel} \rightarrow \Delta m_{\beta}$ (+) \rightarrow (/ $\rightarrow \alpha + \beta$) 8

How to decide the boundary btw eutectic & peritectic?

Reactions in the tie triangle $a_4e_4b_4$ along boundary, β plays no role $\rightarrow l = \alpha$



$$X_{\mathbf{A}}{}^{\alpha} = X_{\mathbf{A}}{}^{l} > X_{\mathbf{A}}{}^{\beta}$$

$$\Delta m_{\beta}(X_{A}^{\alpha}-X_{A}^{\beta})=m_{\alpha}\Delta X_{A}^{\alpha}+m_{\beta}\Delta X_{A}^{\beta}+m_{l}\Delta X_{A}^{l}$$

$$(\Delta X_A^{\alpha} = -1, \Delta X_A^{\beta} = 3.5, \Delta X_A^{l} = 3)$$
 (next page)

$$-m_{\alpha} + 3.5m_{\beta} + 3m_{l} = 0 \quad (m_{\alpha} + m_{\beta} + m_{l} = 100)$$

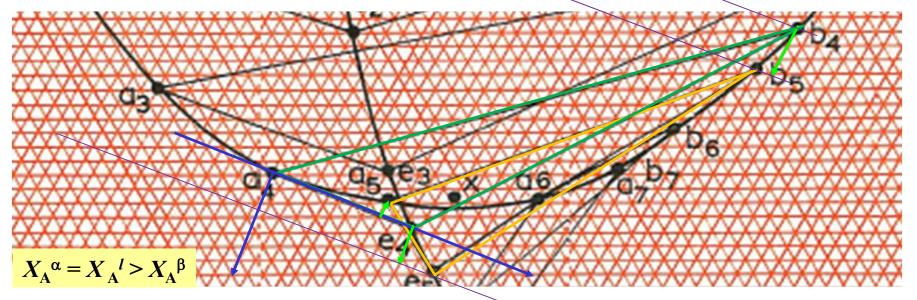
$$-100 + 4.5m_{\beta} + 4m_{l} = 0$$

if
$$m_{\rm B} = 0$$
, $m_l = 25$

if
$$m_l = 0$$
, $m_{\beta} = 22.2$

Initially, peritectic region confined the α corner.

Consideration of three-phase triangles at lower temperatures will indicate that the peritectic region sweeps round from the α corner towards the β and liquid corners. 9 • three phase regions $a_1e_1b_1$, $a_2e_2b_2$, ..., $a_7e_7(b_7)$ projected on the concentration triangle.



The boundary line can be determined by measuring $\Delta X_A{}^a$, $\Delta X_A{}^\beta$, and $\Delta X_A{}^l$.

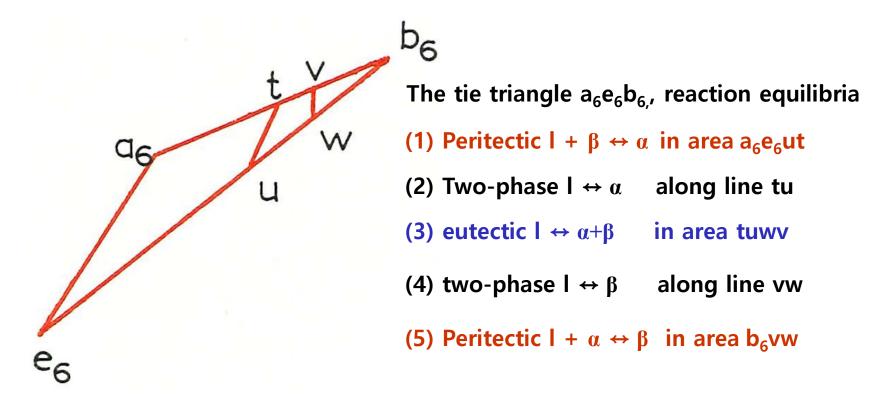
In Fig. 151,
$$\Delta X_A^{\alpha} = -1$$
, $\Delta X_A^{\beta} = -3.5$ and $\Delta X_A^{l} = 3$ units.

Figure 151

Table 2

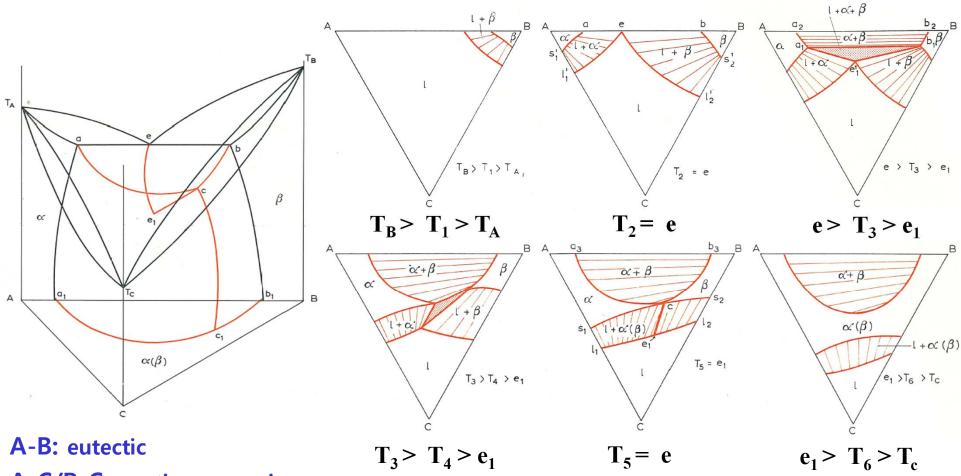
| | X_{B} , | $X_{\mathbf{C}}$ | | $X_{\mathbf{B}}$, | $X_{\mathbf{C}}$ | | $X_{\mathbf{B}}$, | $X_{\mathbf{C}}$ |
|-------|--------------------|------------------|------------|--------------------|------------------|-------|--------------------|------------------|
| · e, | 33, | 16 | a_1 | 17, | 6 | b_1 | 78, | 3 |
| e_2 | 29, | 27 | a_2 | 14, | 20 | b_2 | 69, | 15 |
| e_3 | 26, | 37 | a_3 | 15, | 31 | b_3 | 62, | 22 |
| e_4 | 25.3, | 41 | a_4 | 19, | 37 | b_4 | 56, | 27 |
| e_5 | 25, | 44 | a_5 | 25, | 39 | b_5 | 52, | 30 |
| e_6 | 25, | 47 | a_6 | 34, | 39 | b_6 | 45, | 34 |
| e_7 | 25, | 50 | $a_7(b_7)$ | 40, | 37 | | | |

Monovariant β curve coincides with the α tie line between isotherms $a_5e_5b_5$ and $a_6e_6b_6$ Second peritectic reaction area appears at the β corner of the three-phase triangle



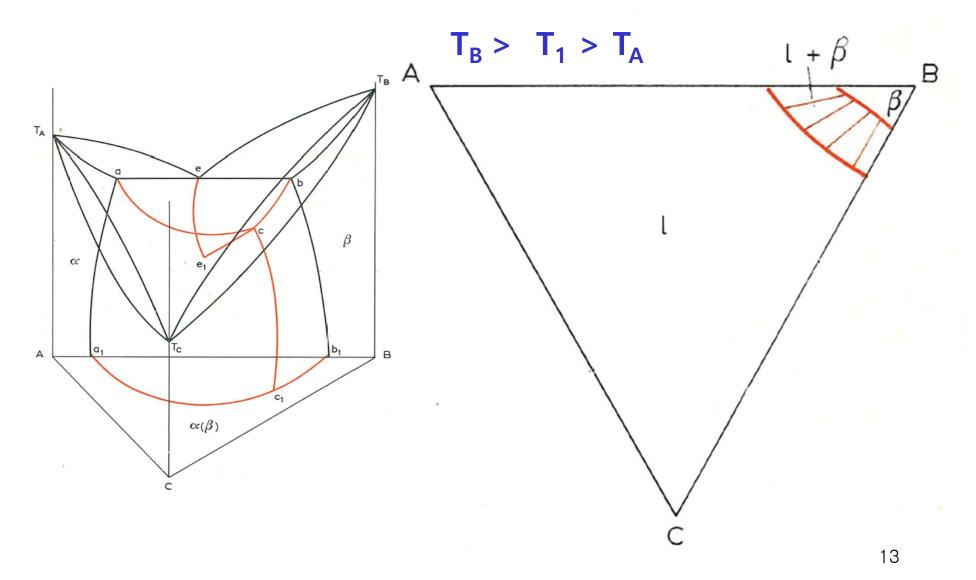
To summarise, the three-phase reaction is initially eutectic for all alloys untill the temperature of the three phase triangle $a_4e_4b_{4,}$ is reached. From that temperature until the end of the three-phase reaction at the tie line $e_7a_7(b_7)$, the reaction type is dependent on the alloy composition within the sequence of the three-phase triangles.

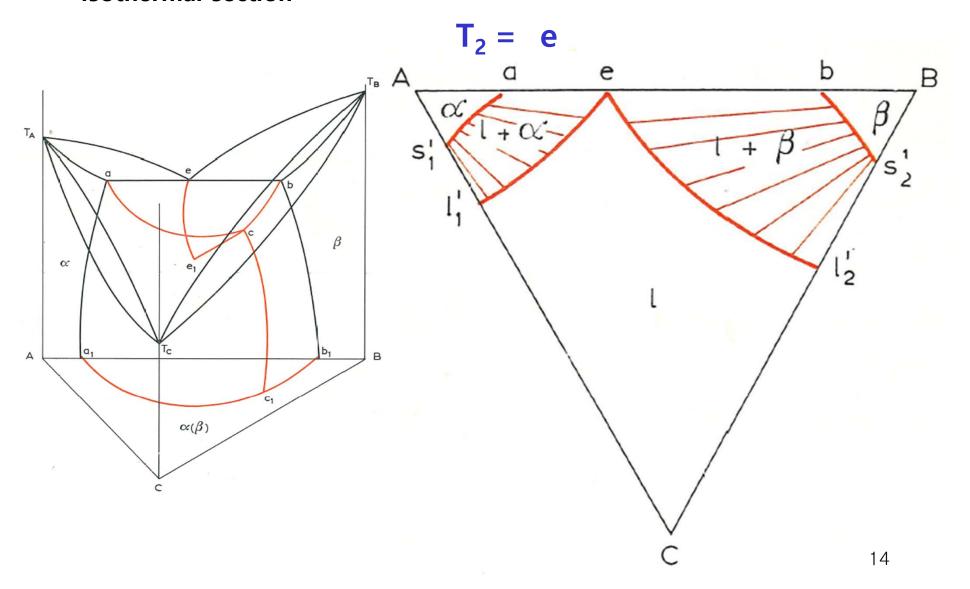
Isothermal section

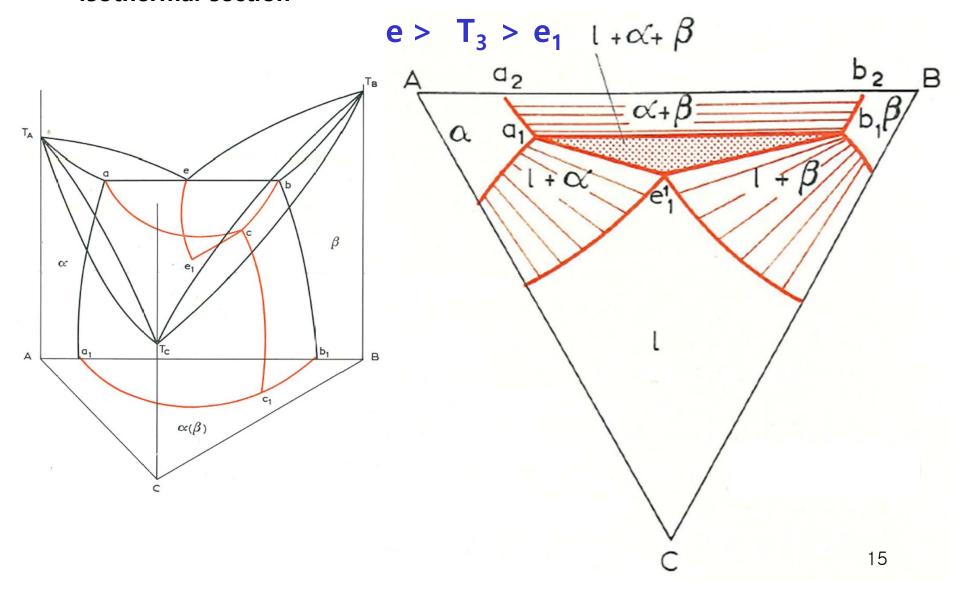


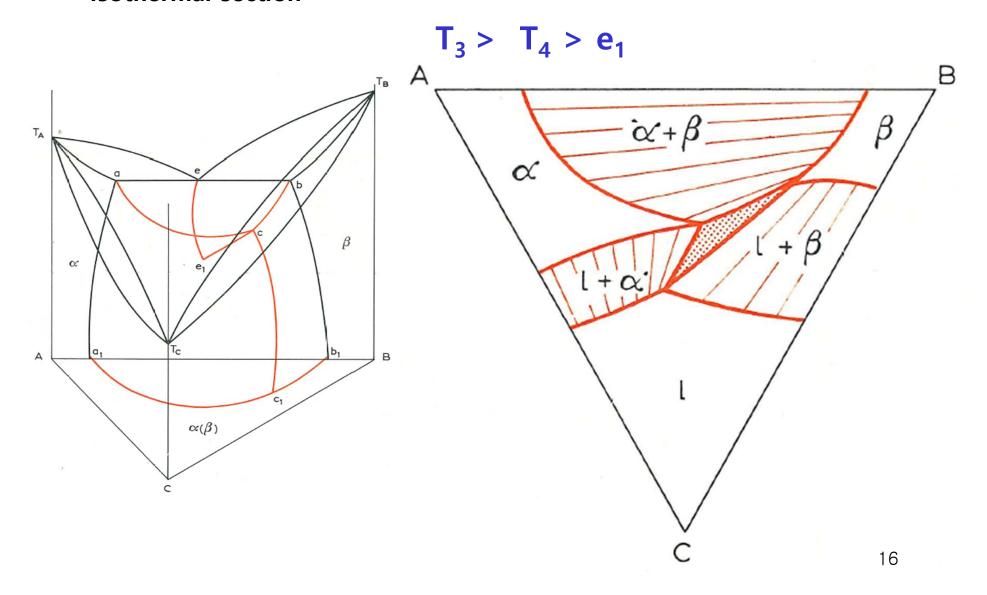
A-C/B-C: continuous series of solid soln

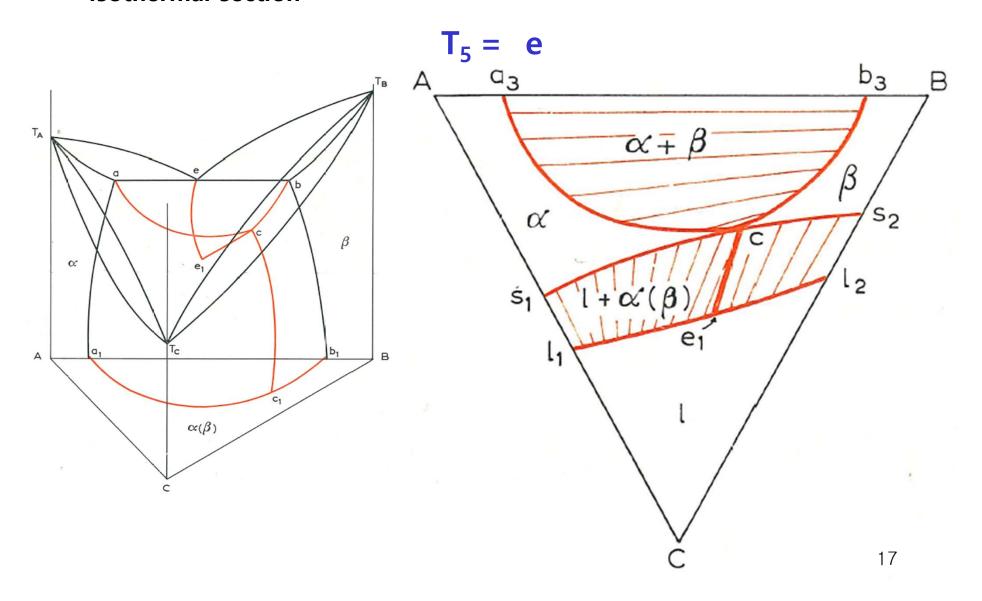
cf) Movie

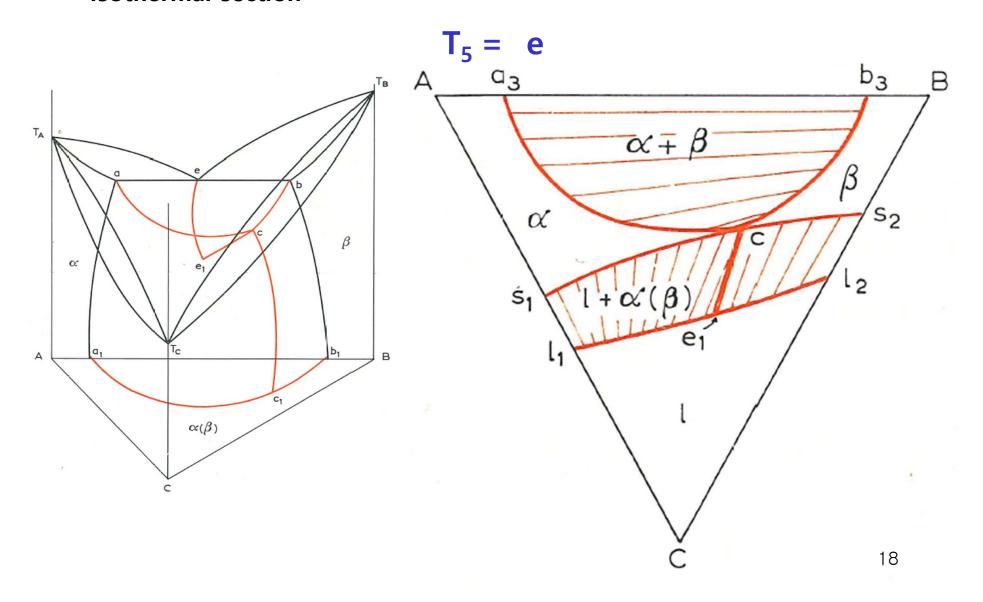












9.2. THREE-PHASE EQUILIBRIUM

- 1 Coalescence of miscibility gap and two phase region
 - How we can have 3 phase equil.?

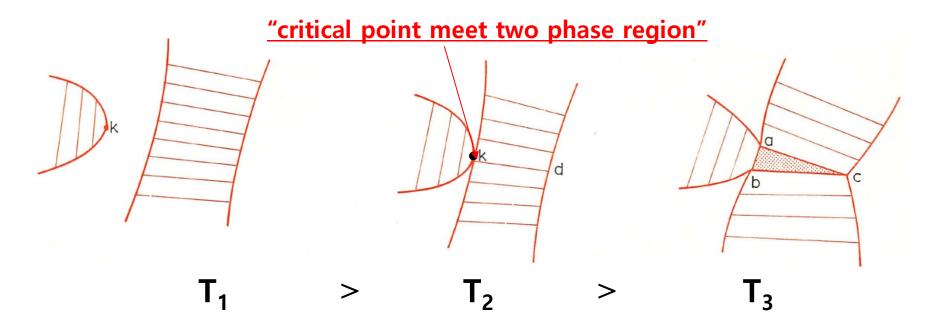


Fig. 136. Production of a ternary three-phase equilibrium by the coalescence of two two-phase regions

9.2. THREE-PHASE EQUILIBRIUM

② Coalescence of two two-phase region

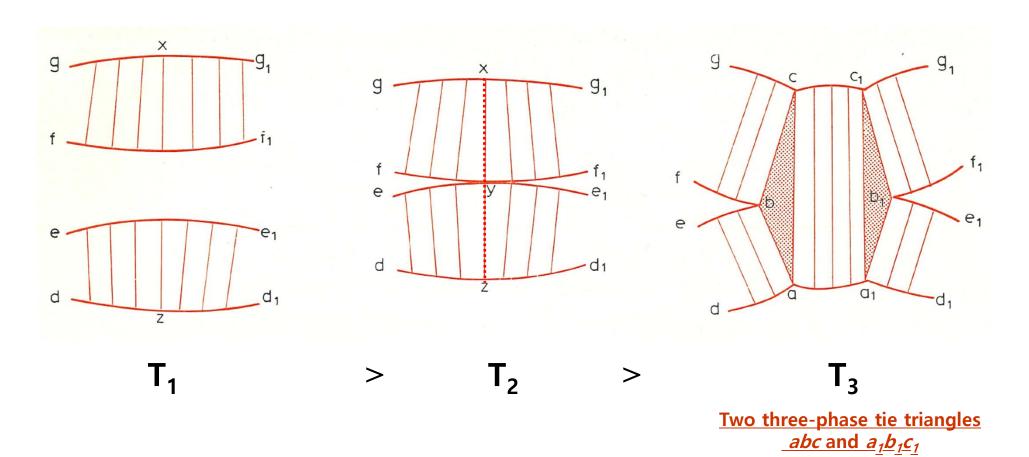
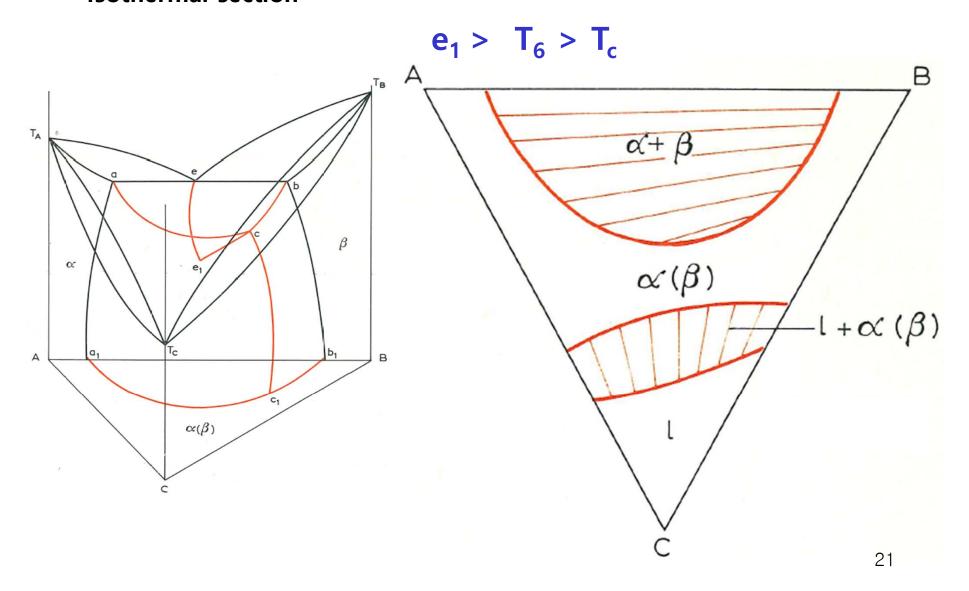
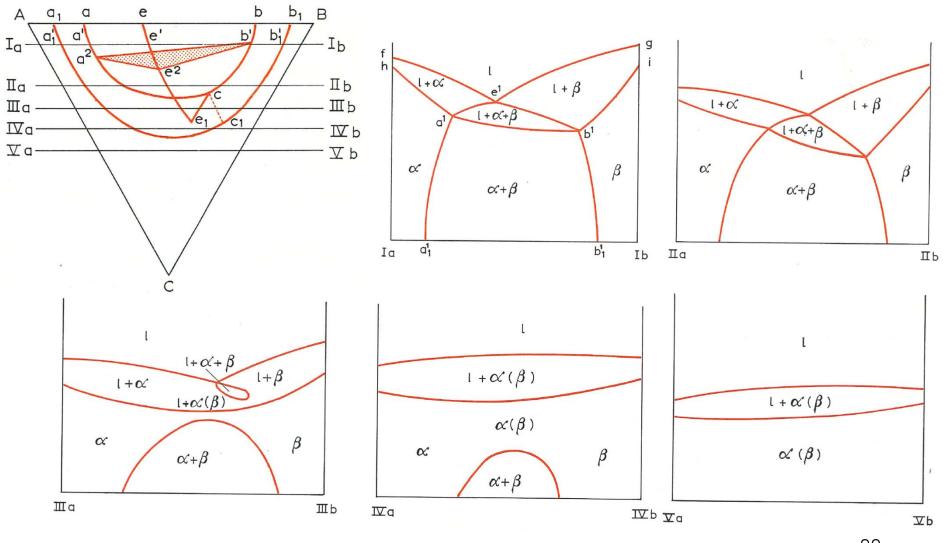
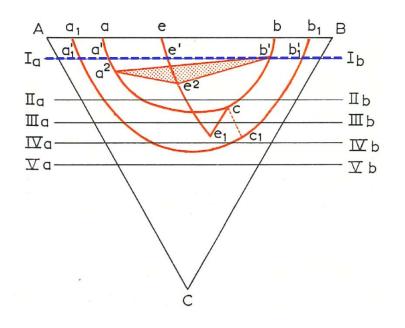
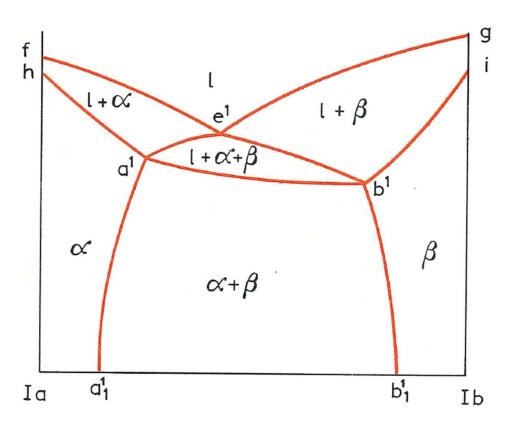


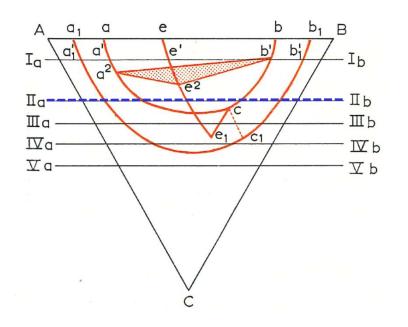
Fig. 138. Alternative method to Fig. 136 for the production of a ternary three-phase equilibrium by the coalescence of two two-phase regions

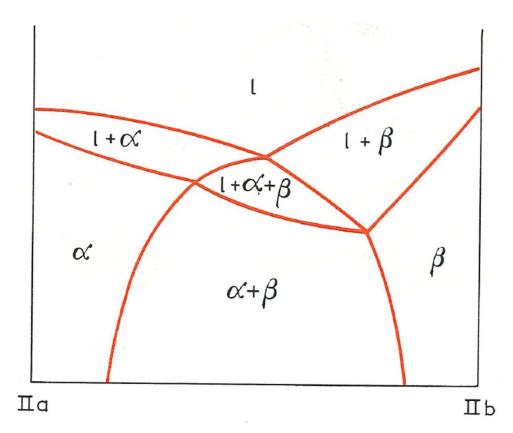


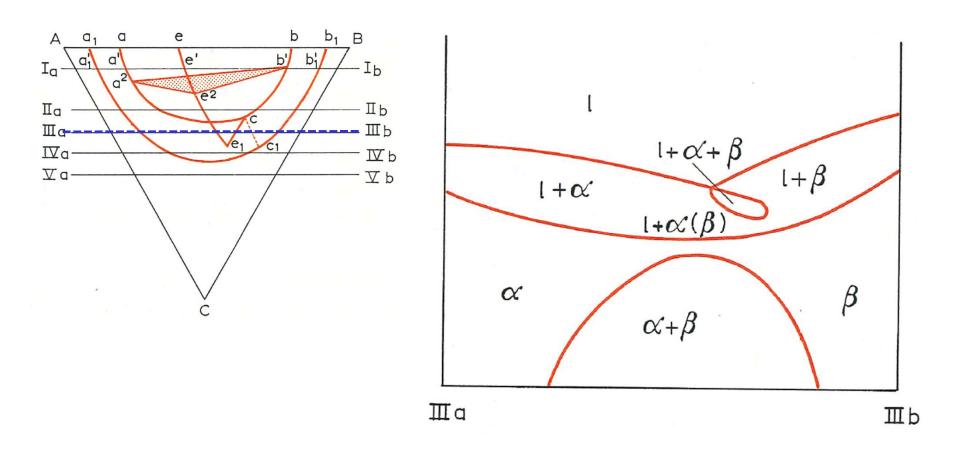


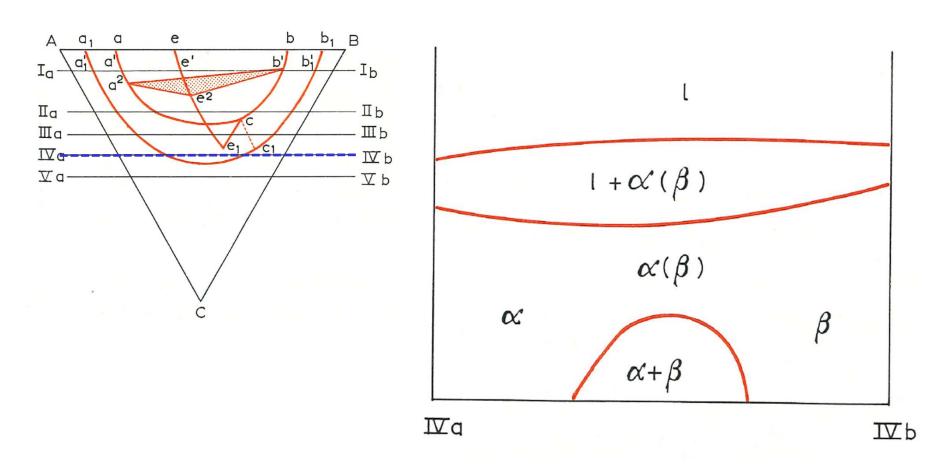


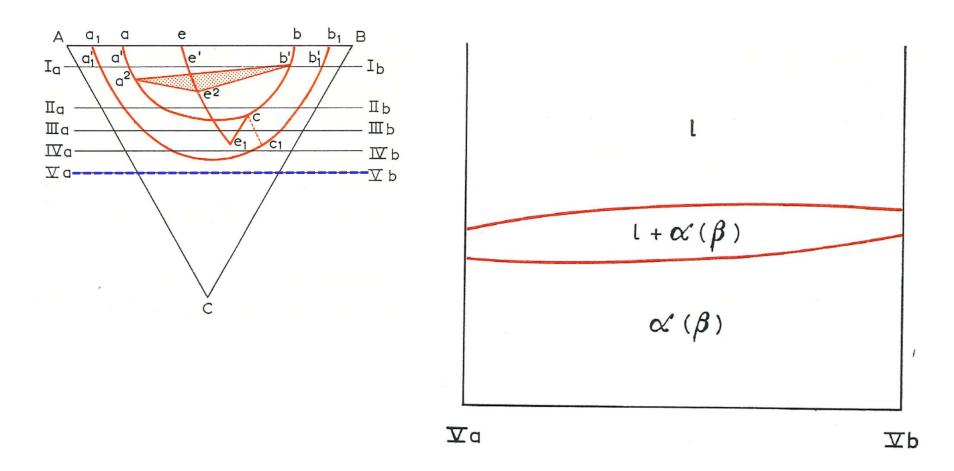








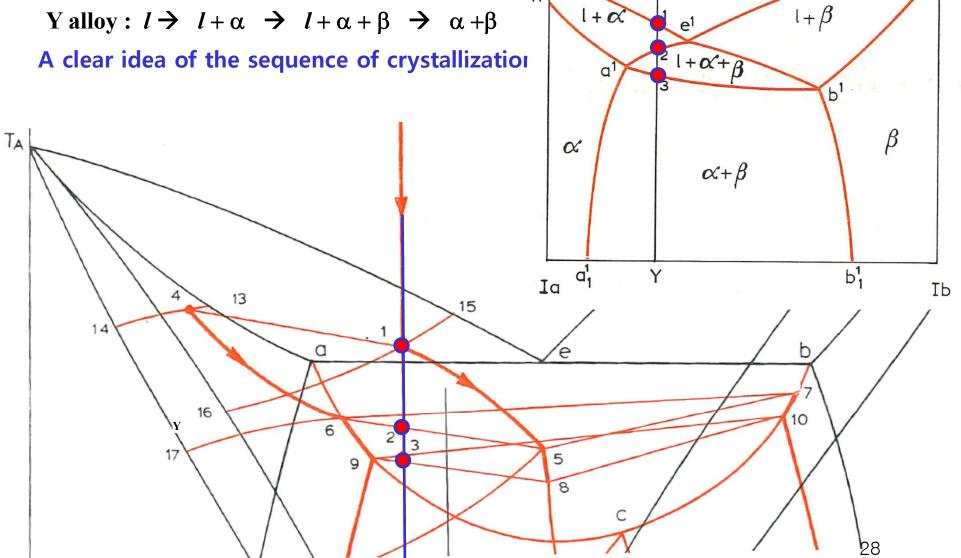




1+00

• Transformation on cooling

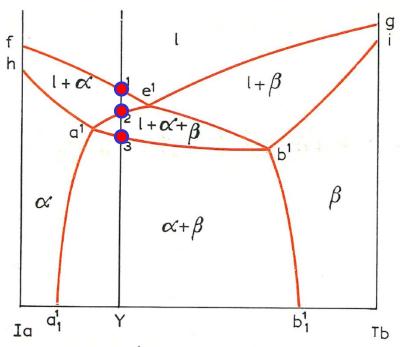
Yalloy: $l \rightarrow l + \alpha \rightarrow l + \alpha + \beta \rightarrow \alpha + \beta$

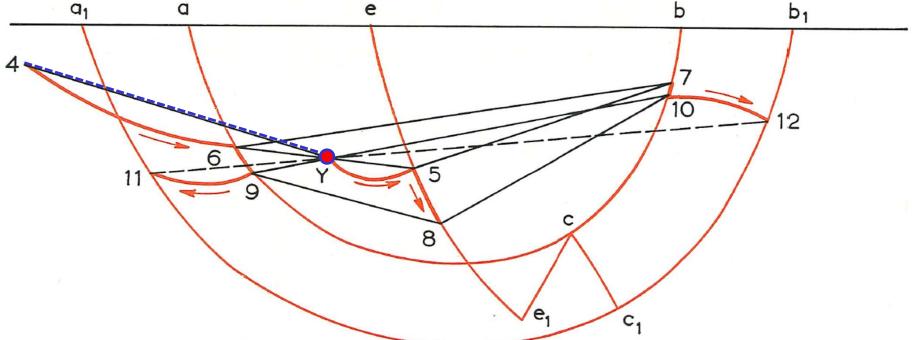


- > Point 1: 4 on the α solidus surface
- > Point 1- Point 2
 - * $4\rightarrow 6$ on the α solidus surface
 - * $1\rightarrow 5$ on the α liquidus surface

Three phase equilibrium 15, α 6, β 7

- * α : $6 \rightarrow 9$, β : $7 \rightarrow 10$, β : $5 \rightarrow 8$
- > Point 3: on the tie line 9-10
- > Point 3-Y: α : $9 \rightarrow 11$, β : $10 \rightarrow 12$

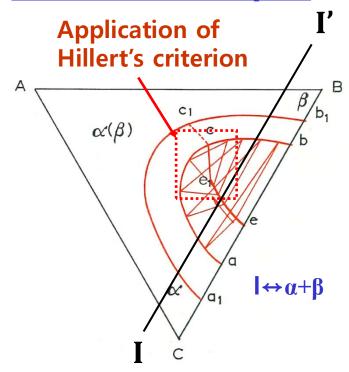


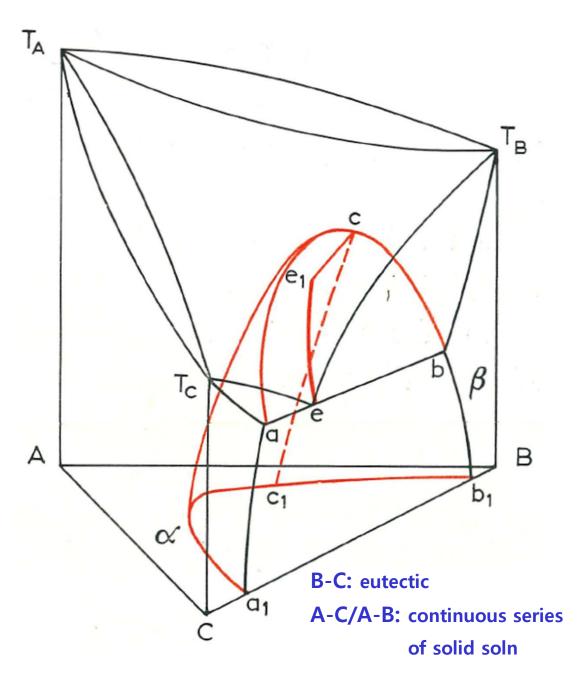


Projection of the solidification sequence for alloy Y on the concentration triangle

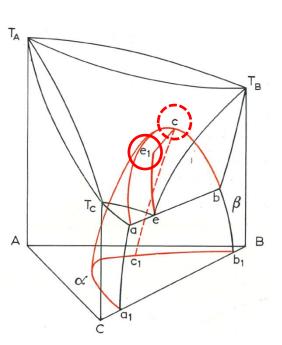
A maximum critical point

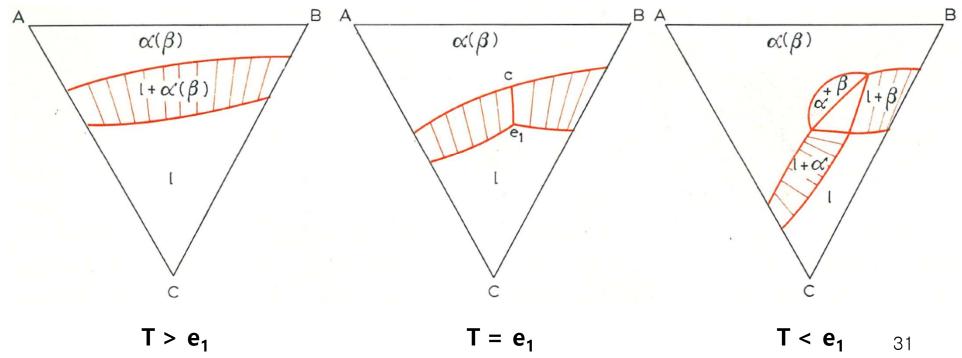
Basically, the reaction we can expect is eutectic reaction ($l \rightarrow \alpha + \beta$). But, in reality, we can have eutectic and peritectic reaction depending on the relative amount of three phase.

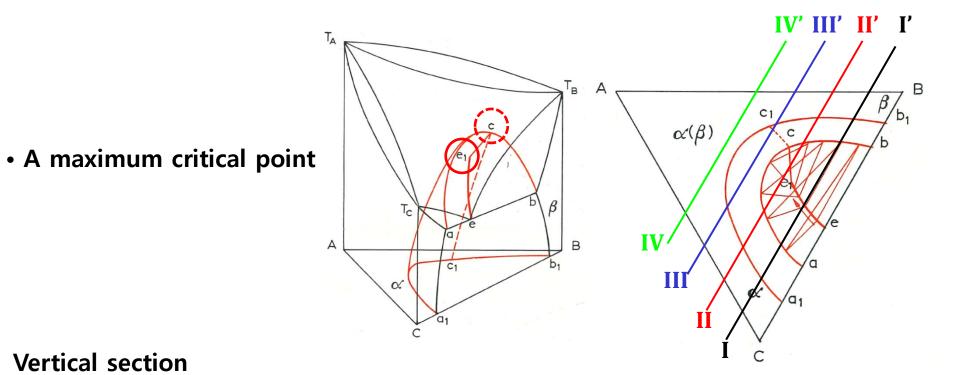


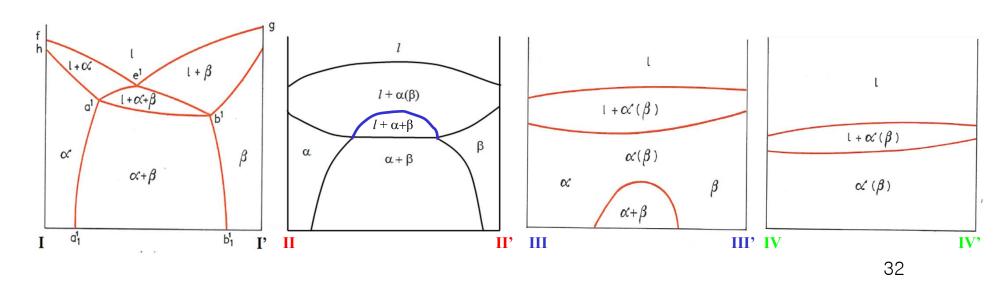


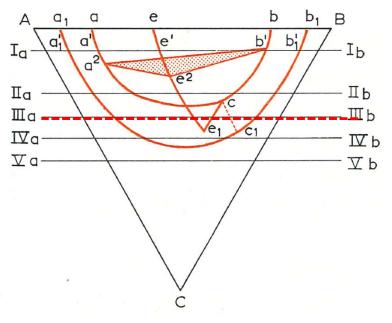
• A maximum critical point

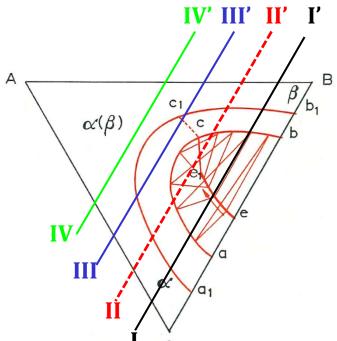


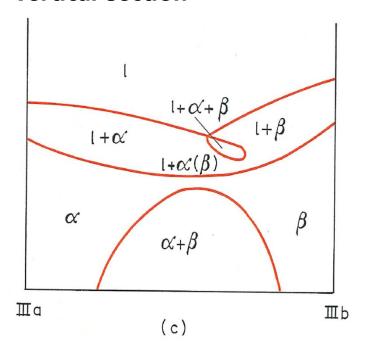


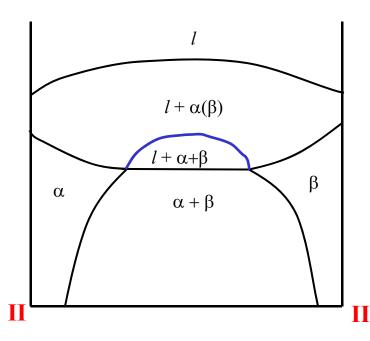




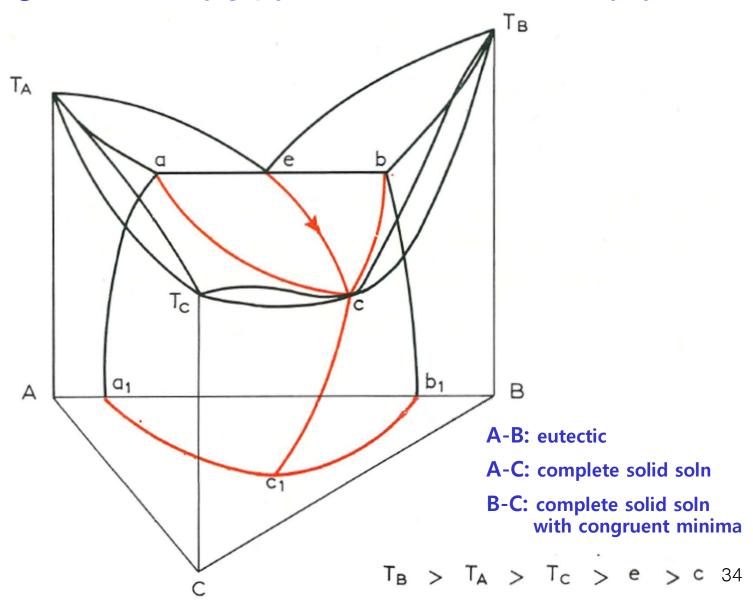




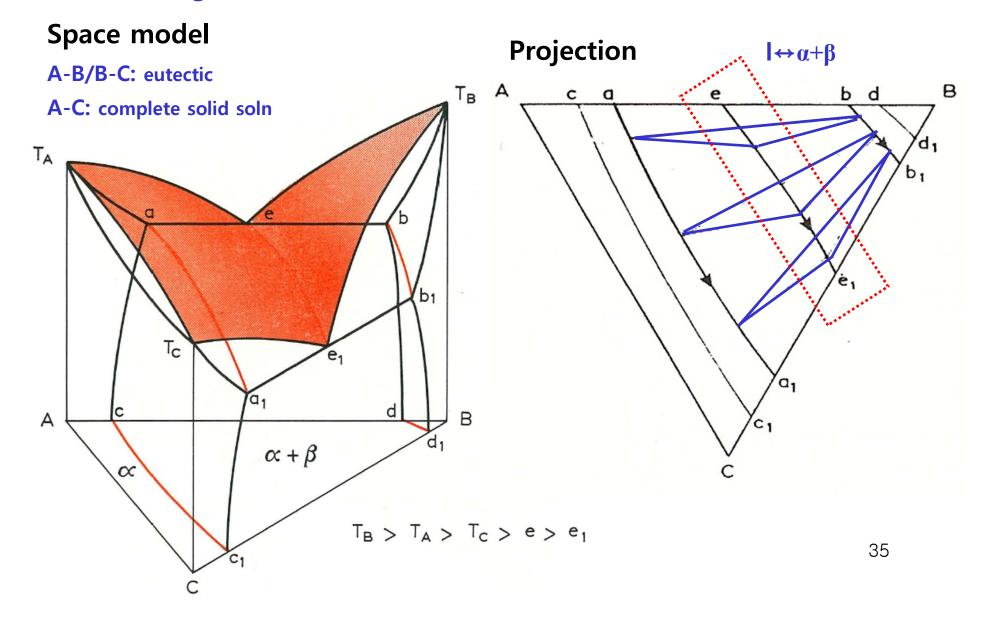


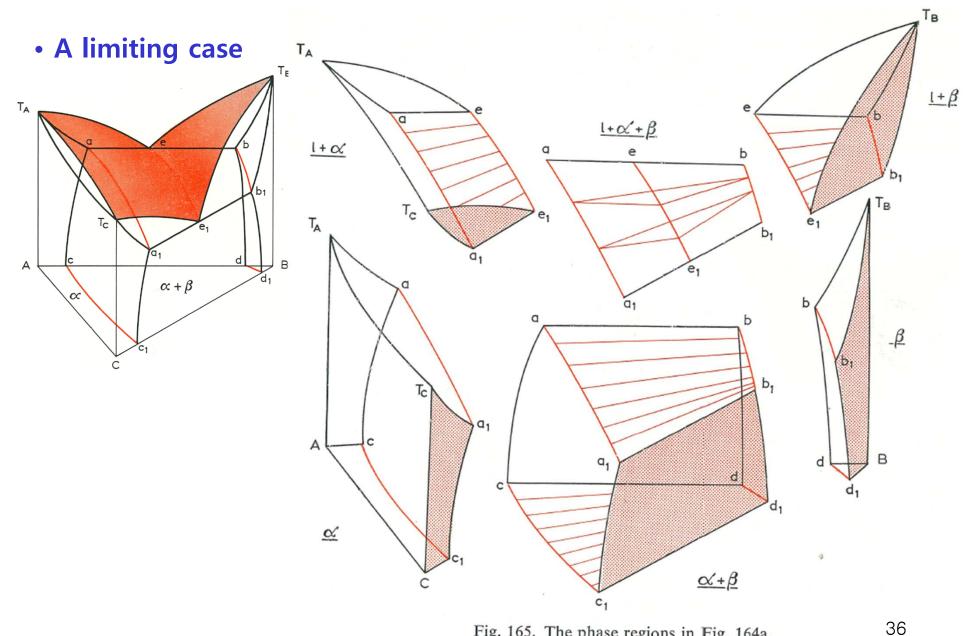


• A limiting case: solubility gap just reaches one of the binary systems.

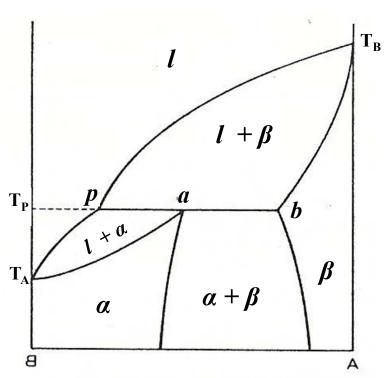


• A limiting case





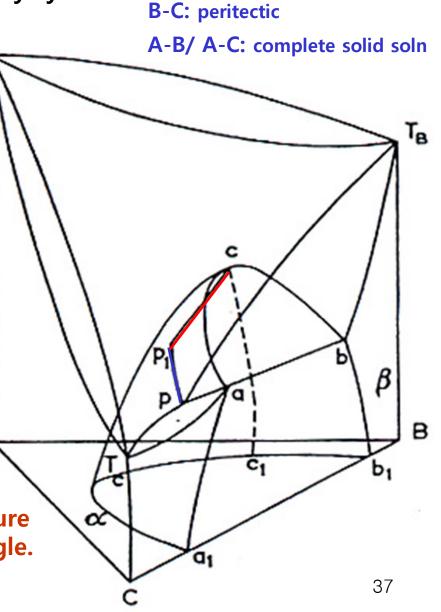
• A peritectic solubility gap in one binary system



: A minimum or a maximum may appear in the monovariant liquid curve.

PP₁: monovariant curve for liquid

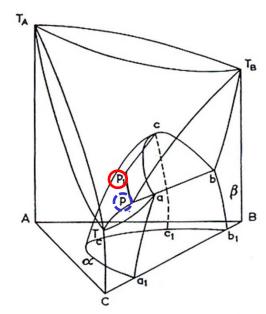
Points P_1 and c lie at the same temperature and the line P_1 c is a degenerate tie triangle.

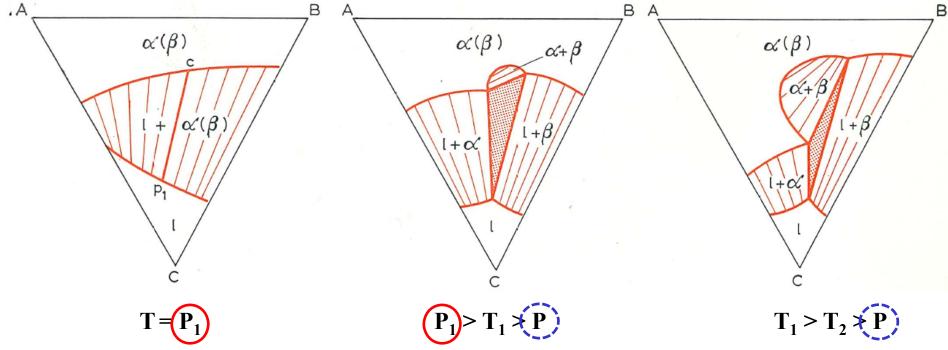


• A peritectic solubility gap in one binary system

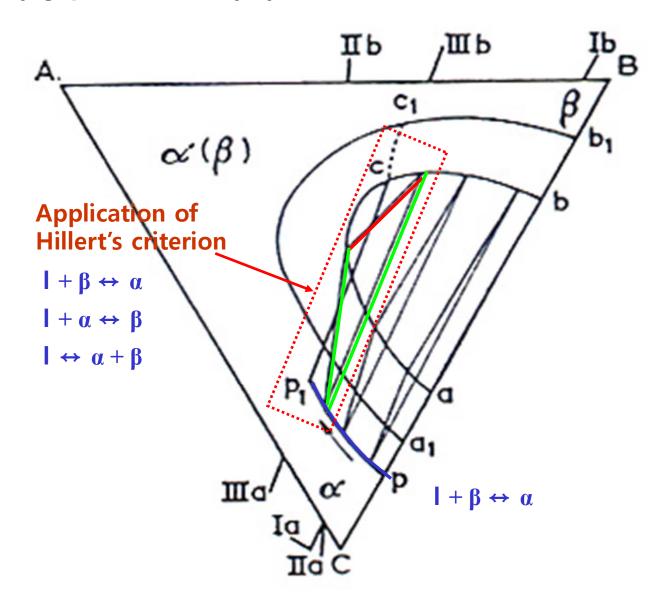
PP₁: monovariant curve for liquid

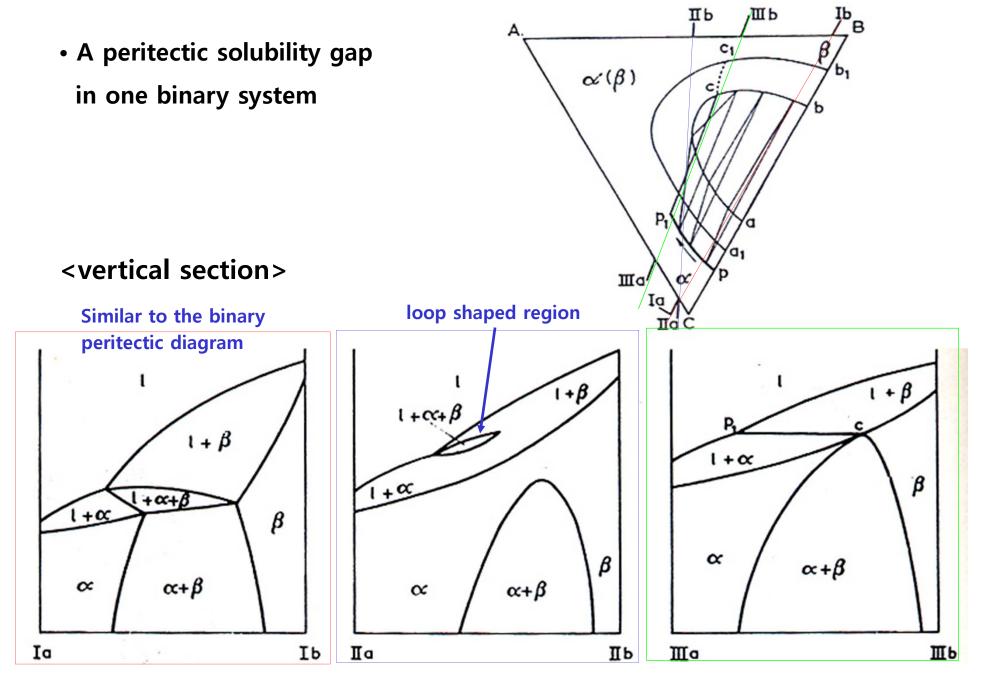
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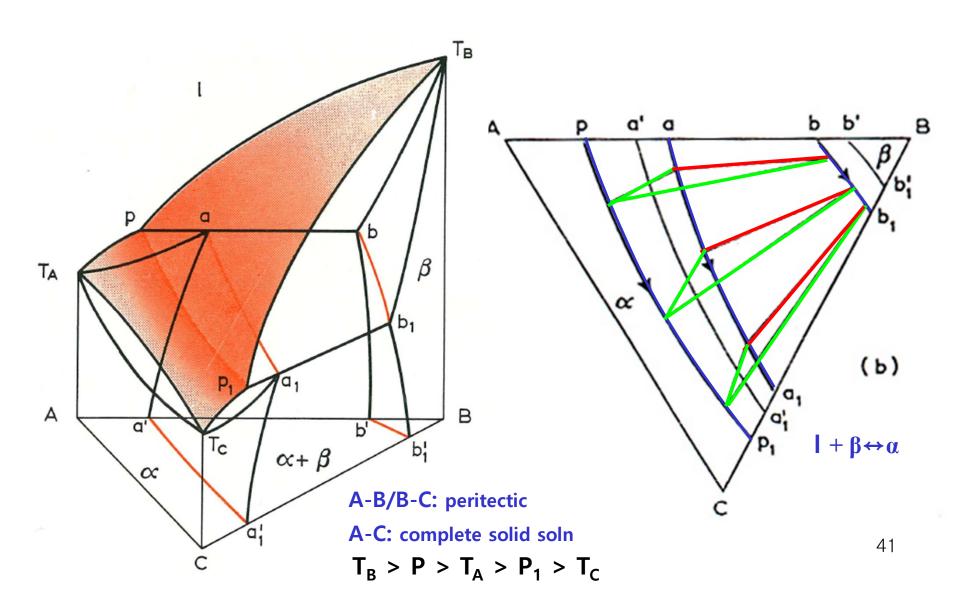


• A peritectic solubility gap in one binary system

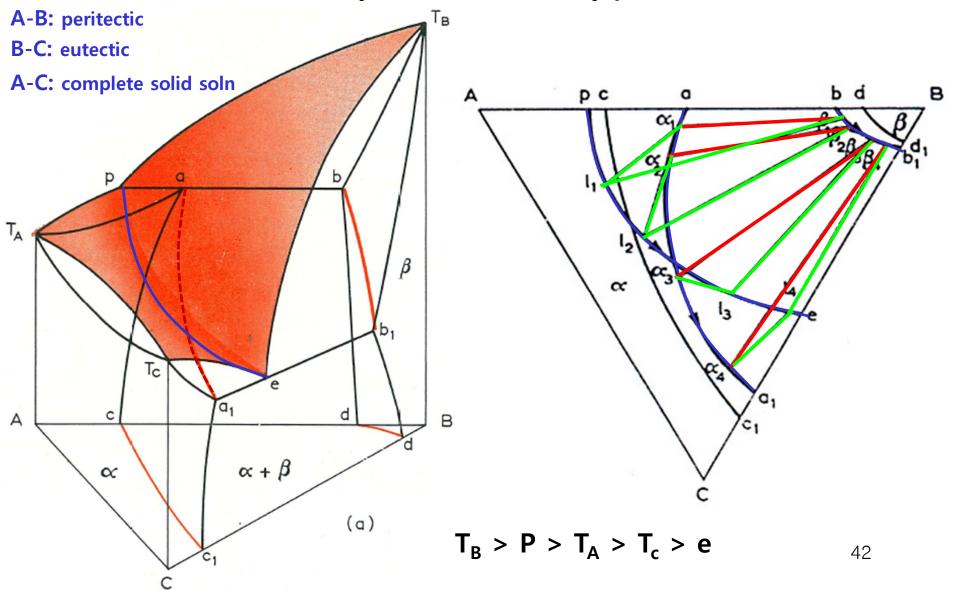




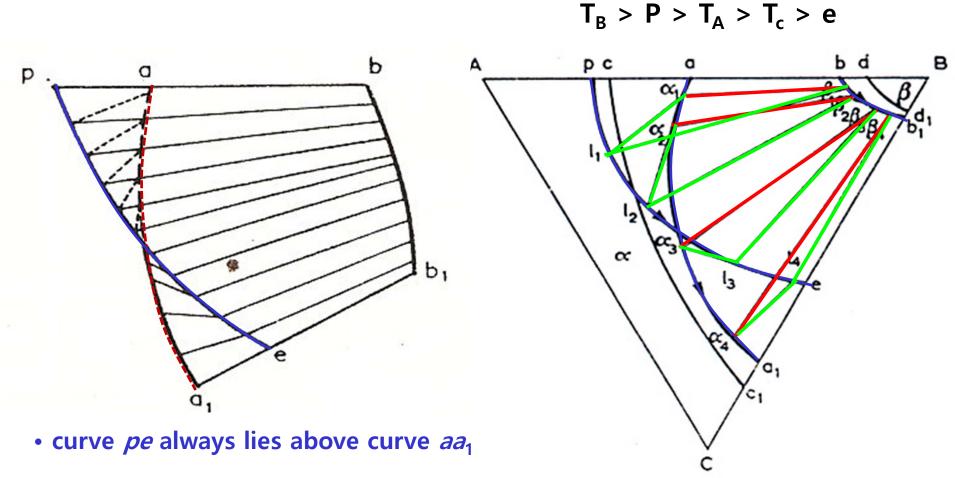
• A peritectic solubility gap in two binary system



• A transition from a binary eutectic to a binary peritectic reaction



• A transition from a binary eutectic to a binary peritectic reaction



- Tie lines are drawn on the $I\beta$ and $I\alpha$ surfaces only.
- By Hillert to show that the transition form a peritectic to a eutectic reaction does not occur at a unique temperature.

• Binary Monotectic, syntectic and metatectic reactions in combination with each other as well as with binary eutectic and peritectic reactions.

