

# X-ray for characterization of thin films

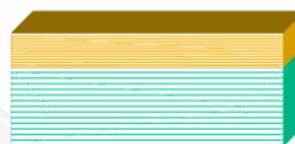
Many slides from the ppt made by Bruker, Panalytical, Rigaku and Dr. Scott A. Speakman ([prism.mit.edu/xray](http://prism.mit.edu/xray)) were used.

## Thin layer

- Dimension in one direction is much smaller than those in other two directions.



- Epitaxial layer

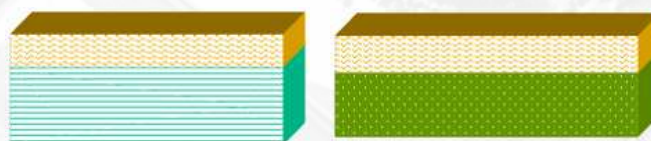


- Polycrystalline layer

- ✓ Non-single-crystal layer

- Amorphous layer

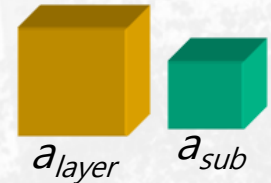
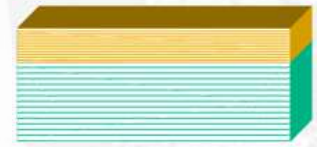
- ✓ lack of long-range ordering of atoms



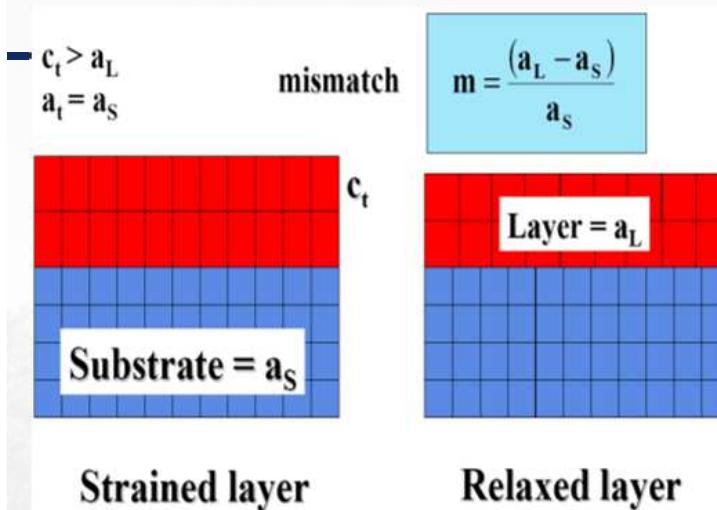
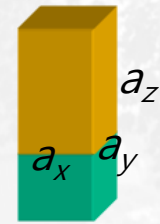
# Epitaxial layer

- Single crystal thin layer having crystal structure & orientation of substrate on which it was grown

- ✓ Homoepitaxial layer – layer and substrate are same materials (same lattice parameters)
- ✓ Heteroepitaxial layer – layer material is different from substrate (different lattice parameters)



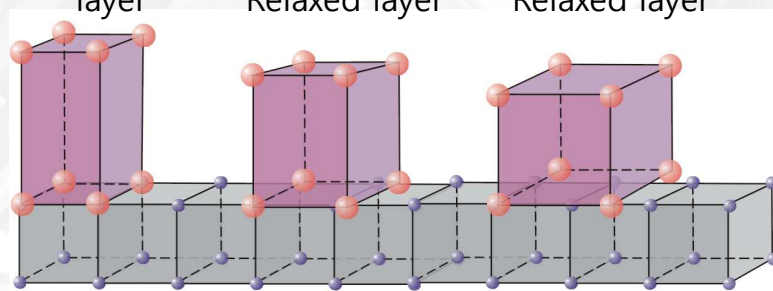
- Lattice mismatch  $f = (a_{layer} - a_{subs}) / a_{subs}$
- Critical thickness
  - ✓ thickness below which the layer grows pseudomorphically → the cubic unit cell of layer material is tetragonally distorted:  $a_z \neq a_x = a_y = a_s$  (layer is strained).
  - ✓ decreases when  $f$  increases.
- Layer relaxation;  $a_x, a_y \rightarrow a_z \rightarrow a_{relax} = a_{bulk}$



Epitaxial layer  
Mismatch  
Relaxation

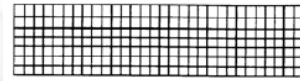


Pseudomorphic layer      Partially Relaxed layer      Completely Relaxed layer



# What we want to know

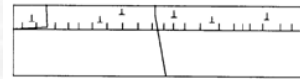
- crystalline state of layer/layers - epi?; polycrystalline?; amorphous?
- crystal quality                      chemical composition
- thickness                              surface and interface roughness
- superlattice period
- mismatch
- relaxation
- misorientation
- dislocation density
- mosaic spread
- curvature
- inhomogeneity



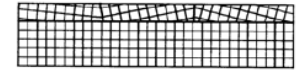
mismatch



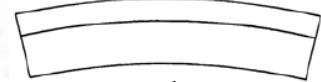
misorientation



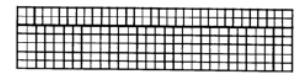
dislocation



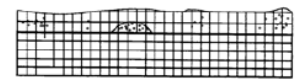
mosaic spread



curvature



relaxation

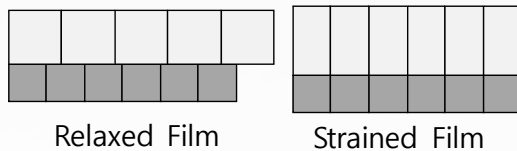


inhomogeneity

[www-ssrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering\\_epitaxiallayers\\_vailionis.pdf](http://www-ssrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering_epitaxiallayers_vailionis.pdf)

# Relaxation (Lattice Strain)

## Relaxation



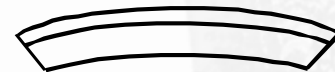
Relaxed Film

Strained Film

- If the film is mismatched to the substrate → the film can be strained. → lattice parameters in the lateral direction (within the plane of the film) are forced to match those of the substrate. → distorts unit cell of the film.
- Determine the degree of relaxation.
  - ✓ No relaxation (fully strained)- the lateral lattice parameters of the film are strained and identical to those of the substrate.
  - ✓ Fully relaxed- the lateral lattice parameters of the film are equal to the bulk values – they have not been distorted at all.

## Curvature

- The film and substrate may become slightly curved rather than perfectly flat. ← deposition process, thermal expansion mismatch between the film and substrate, etc

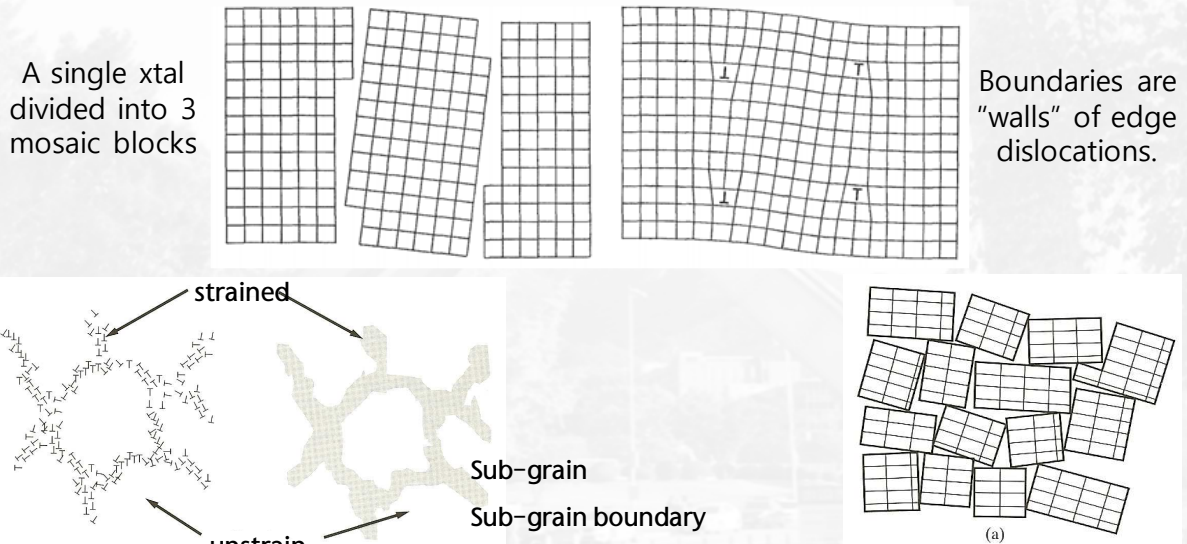


## Dislocations

- Interface dislocations may form to relieve lattice strain between film and substrate with a large amount of mismatch.
- Slip dislocations are created by plastic deformation due to thermal or mechanical strain in the layer.

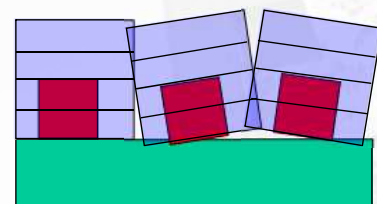
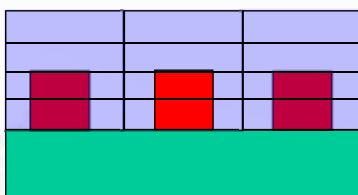
## Mosaic structure, mosaic blocks

- Angle of disorientation between the tiny blocks is  $\epsilon$ . → diffraction occurs at all angles between  $\theta_B$  and  $\theta_B \pm \epsilon$ .
- Increases the integrated intensity relative to that obtained (or calculated) for an ideally perfect crystal. ← strains & strain gradients associated with the groups of dislocations



## Mosaic spread

- Mosaicity is created by slight misorientations of different crystals as they nucleate and grow on the substrate. When the crystals join, they form boundaries.



In an ideal case, each nuclei (red) is perfectly oriented. When the crystals grow and meet, there is perfect bounding between the crystallites → no boundary.

If the nuclei (red) are slightly misaligned, then boundaries will be formed.



Perfect Epitaxy	Single crystal film in perfect registry with a substrate. There are no defects in the film or the substrate.
Nearly perfect Epitaxy	Single crystal film in nearly perfect registry with a substrate. Both film and substrate contain a low concentration of defects. Most defects are dislocations in the film.
Textured epitaxial	Film consists of mosaic domains in nearly perfect registry with the substrate. All domain boundaries are very low angle/low energy boundaries. There is nearly perfect bonding across domain boundaries.
Strongly textured polycrystalline	Film consists of grains with nearly perfect preferred orientation of all principle axes. This orientation is often strongly correlated to the substrate. Misorientation parameter for texture is small.
Textured polycrystalline	Film consists of grains with a preferred orientation for 3 principle axes or only along 1 axis out-of-plane.
Polycrystalline	Film consists of randomly oriented grains.
Amorphous	Film does not have long-range order.

## Techniques ↔ type of information of film

	Thick ness	Composition	Lattice Strain / Relaxation	Defects	Orien tation	Residual Stress	Crystallit e Size
Perfect Epitaxy	XRR HRXRD	HRXRD RC	Assume 100 %	Assume none	HRXRD	--	--
Nearly perfect Epitaxy	XRR HRXRD	HRXRD RC	HRXRD	RC	HRXRD	--	--
Textured epitaxial	XRR HRXRD	HRXRD	HRXRD IP-GIXD	RC	HRXRD	--	--
Strongly textured Polyxalline	XRR	XRPD IP-GIXD	IP-GIXD	XRPD, IP-G IXD	IP-GIXD PF	IP-GIXD	XRPD, IP -GIXD
Textured Polyxalline	XRR	XRPD, GIXD or IP-GIXD	--	XRPD, GIX D OR IP-GI XD	PF	Psi	XRPD GIXD
Polyxtalline	XRR	XRPD, GIXD	--	XRPD GIXD	PF	Psi	XRPD GIXD
Amorphous	XRR	--	--	--	--	--	--

HR- High Resolution XRD using coupled scan or RSM

IP-GIXD- in-plane grazing incidence XRD

RC- Rocking Curve

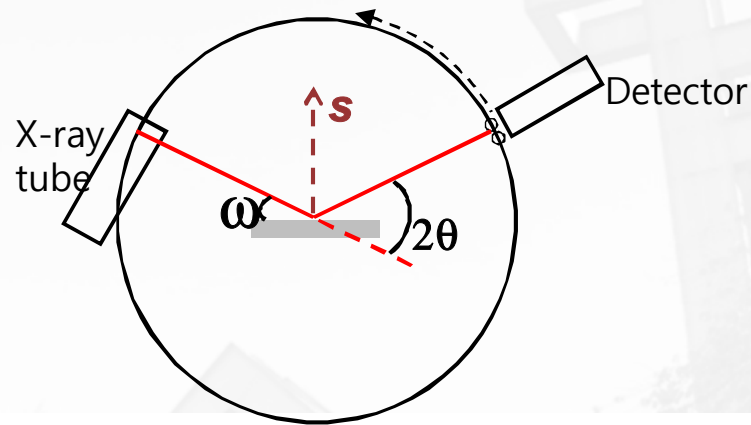
GIXD- grazing incidence XRD

PF- pole figure

XRR- X-Ray Reflectivity

XRPD- X-ray powder diffraction

Psi-  $\sin^2\psi$  using parallel beam

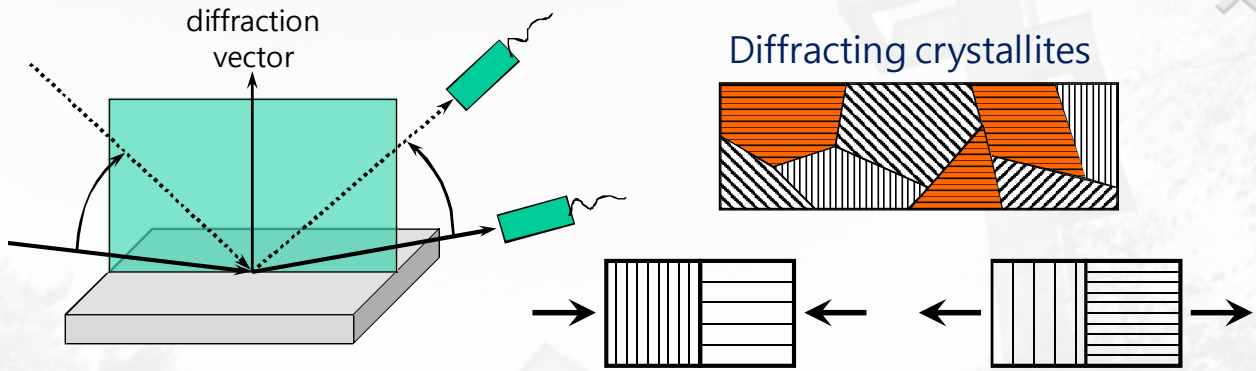


- The incident angle  $\omega$  (omega) (X-ray source – sample)
- The diffraction angle  $2\theta$  (incident beam – detector)
- **Rocking Curve;** X-ray intensity vs. Omega
- **Detector Scan;** X-ray intensity vs.  $2\theta$  without changing Omega
- **Coupled Scan;** X-ray intensity vs  $2\theta$ , but Omega also changes so that
$$\omega = \frac{1}{2} \times 2\theta + \text{offset.}$$

## X-ray technologies for thin films

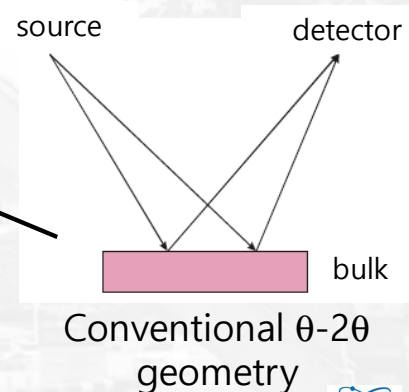
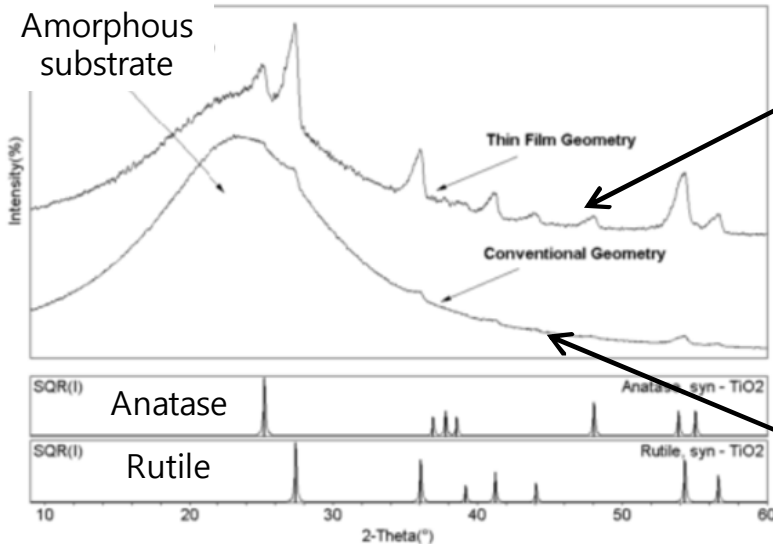
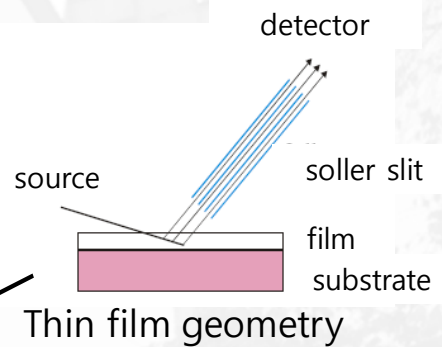
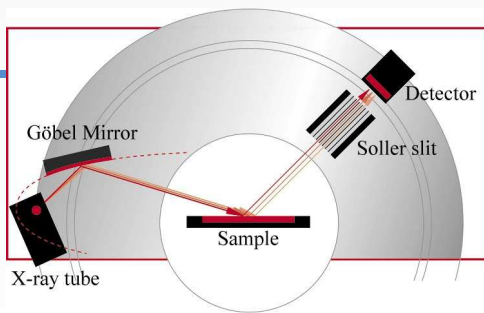
- Grazing incident X-Ray diffraction (GID)
  - ✓ poly-crystalline thin films; phase ID, crystallite size, crystallinity
- In-plane grazing incident diffraction (IP-GID)
  - ✓ Textured film and epitaxial film; in-plane orientation, in-plane lattice parameter, crystallinity, depth analysis
- Rocking curves → dislocation density, mosaic spread, curvature, misorientation, inhomogeneity, layer thickness, superlattice period, strain and composition profile, lattice mismatch, ternary composition, relaxation
- Coupled scans → lattice mismatch, ternary composition, relaxation, thickness, superlattice period
- Reciprocal space map (RSM) → composition, thickness (> 50 nm), mismatch, mosaicity, defects profile, etc. (most complete amount of information that are needed for the analysis of strained films)
- Reflectivity → composition, thickness (5-150 nm), interface/surface roughness; works with non-epitaxial and even non-crystalline thin films
- Grazing incident small-angle X-ray scattering (GISAXS)
  - ✓ pore structure (alignment), pore size distribution
- Pole figures → preferred orientation

# Conventional XRD

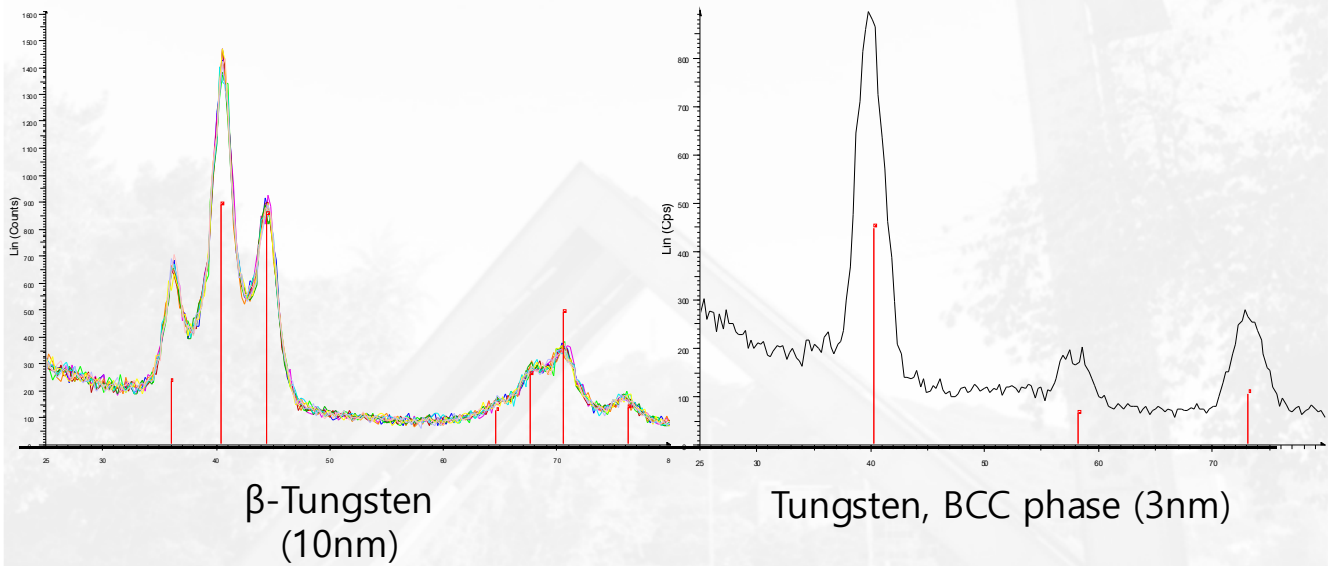


- Reliable information on
  - the preferred orientation of crystallites (out-of-plane orientation)
  - the crystallite size and lattice strain (in one direction)
- No information on the residual stress (← constant direction of the diffraction vector)
- Low scattering from the layer (← large penetration depth)

# GID



# GID > phase analysis on tungsten films

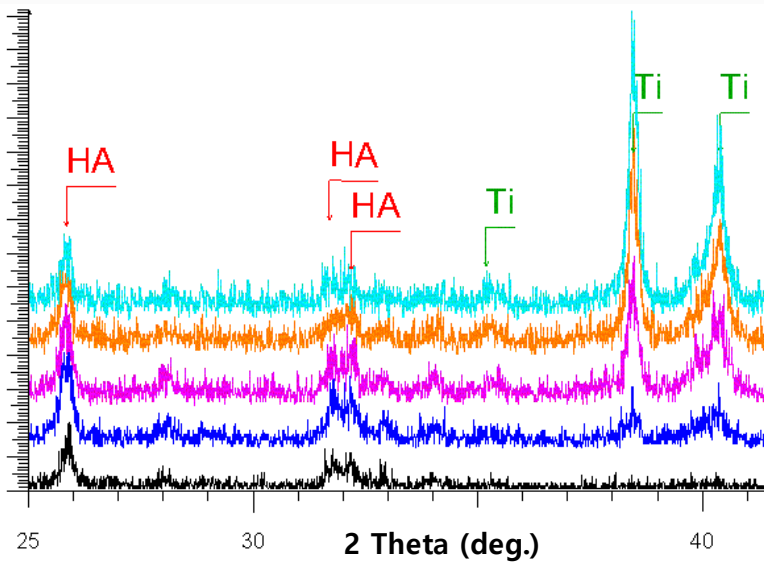


$\beta$ -Tungsten (10nm)

Tungsten, BCC phase (3nm)

# GID > depth profile

Ti coated with hydroxyapatite (HA)



$\omega$  (omega)

at progressively higher incidence angle, X-ray samples Ti substrate.

$\omega$

at shallow angles, X-ray beam samples HA coating.

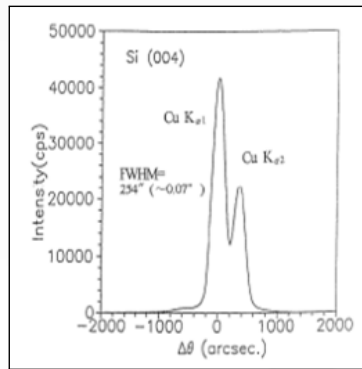
Parallel beam geometry must be used.



# High-resolution X-ray diffraction (HRXRD)

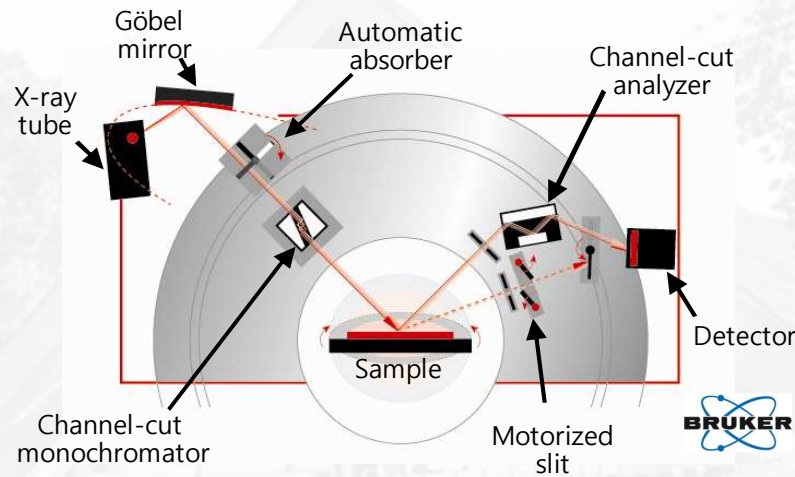
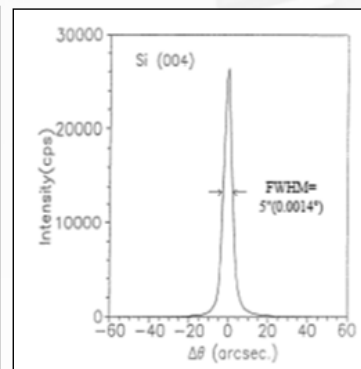
normal XRD

~ 0.065 deg  
(~ 230 arcsec)

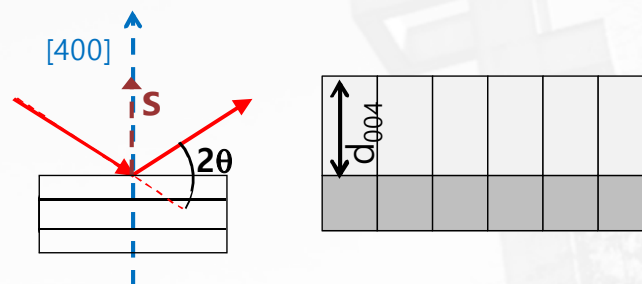
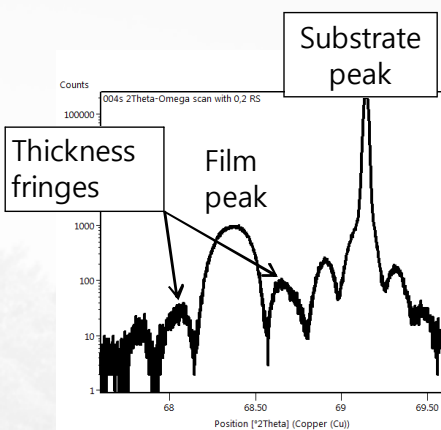


HR XRD

0.0015 ~ 0.008 deg  
(5 ~ 30 arcsec)

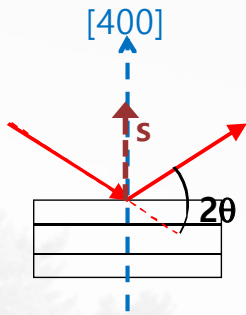


# Coupled diffraction scan → d-spacing and thickness

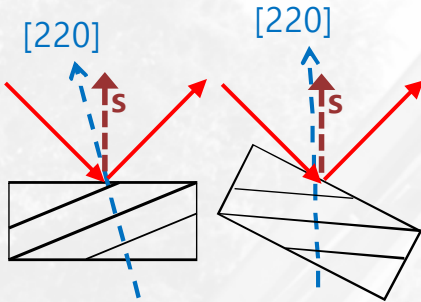


➤ Coupled scan:  $\omega$  &  $2\theta$  change in coupled manner so the direction being measured (scattering vector,  $s$ ) does not change.

- The peak position will give the d-spacing for the Bragg peak.
  - ✓ This will provide information on anything that changes the lattice parameter of the unit cell, such as *composition* or *strain/relaxation*.
  - ✓ This will only provide measurement of the **lattice parameter in one direction**.
- The width of film's Bragg peak can be used to quantify film thickness.
  - ✓ The thickness fringes can also be used to quantify the film thickness.

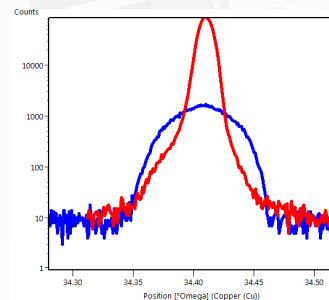
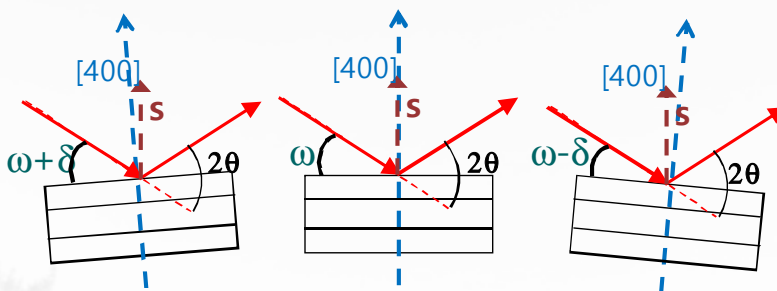


➤ A symmetric scan requires that  $\omega = \frac{1}{2} 2\theta$ , so that the scattering vector  $\mathbf{s}$  will be  $\perp$  to the sample surface.

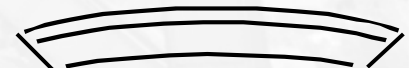
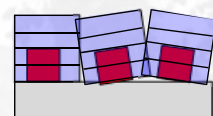


➤ In order to measure different crystallographic directions in the sample, sample can be tilted. → **asymmetric scan**

# Rocking curve ( $\omega$ scan)

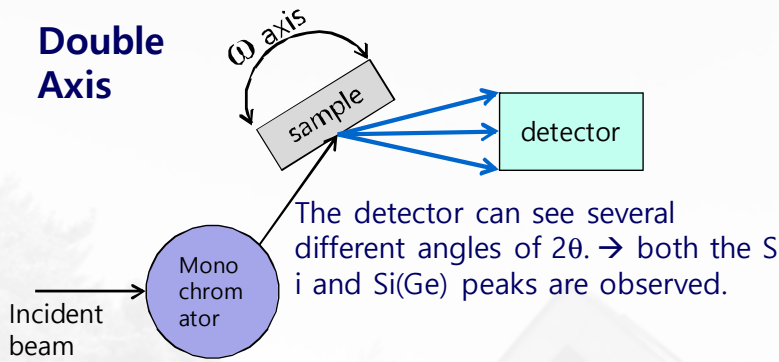


- The detector is set at a specific Bragg angle and the sample is tilted.
- A perfect crystal will produce a very sharp peak, observed only when the crystal is properly tilted so that the plane normal is // to the diffraction vector  $\mathbf{s}$ .
- Defects like mosaicity, dislocations, and curvature create disruptions in the perfect periodicity of atomic planes.
  - ✓ This is observed as broadening of the rocking curve.
  - ✓ The center of the rocking curve is determined by the d-spacing of the peaks.

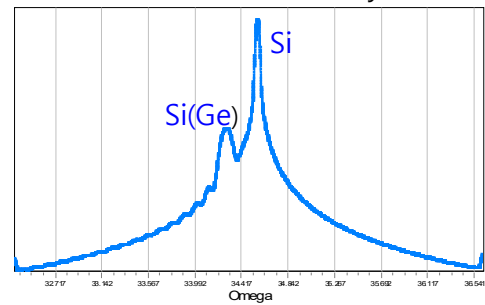


## Rocking curve > Double-axis vs. Triple-axis diffractometry

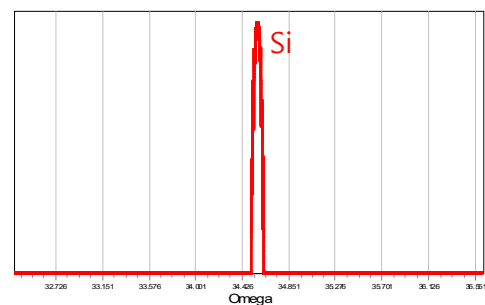
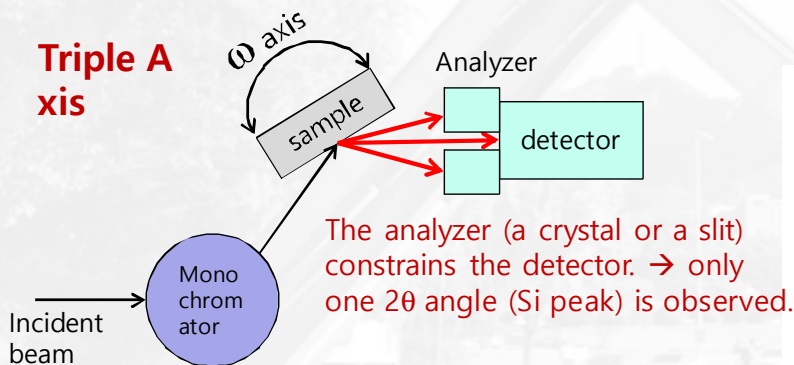
### Double Axis



Si(Ge) film on Si  
rotating omega while keeping the detector stationary



### Triple Axis



## Double-axis vs. Triple-axis diffractometry

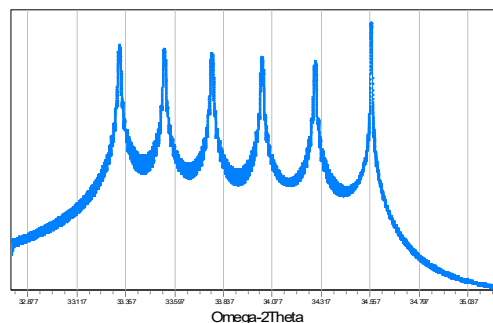
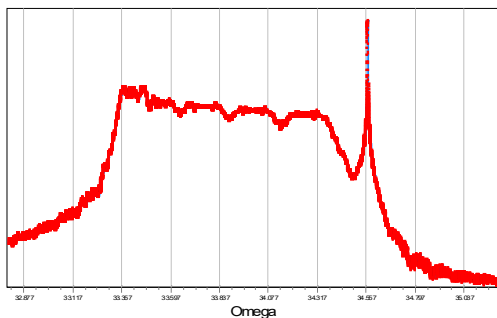
Two instrumental configurations for HRXRD

### ➤ Double-axis (double-crystal)

- ✓ The detector does not discriminate between different diffraction angles  $2\theta$ .
- ✓ All Bragg angles are measured simultaneously (over a limited range).
- ✓ The sample is rotated about its  $\omega$  axis (changing the incident angle) to produce a rocking curve (intensity vs  $\omega$ ).

### ➤ Triple-axis (triple-crystal)

- ✓ A slit or analyzer crystal determines the angular acceptance of the detector.
- ✓ The analyser crystal enables to distinguish between mosaic spread and strain contributions in the diffracted intensity distribution.
- ✓ While a rocking curve (intensity vs  $\omega$ ) can be measured, it is more common to collect data by using a coupled scan.
- ✓ As the sample is rotated about  $\omega$ , the detector is rotated at twice the rate so that  $2\theta = 2\omega$ , producing a coupled  $\omega - 2\theta$  scan.
- ✓ Reciprocal space maps are collected by collecting coupled scans at different  $\omega$  offsets, where  $2\theta = 2\omega - \text{offset}$ .



- The double-axis rocking curve
- A Si wafer coated with 5 slightly relaxed Si(Ge) layers of varying Ge concentration
- The Ge concentrations were 10, 20, 30, 40, and 50%.
- Each Ge layer was 500nm thick.

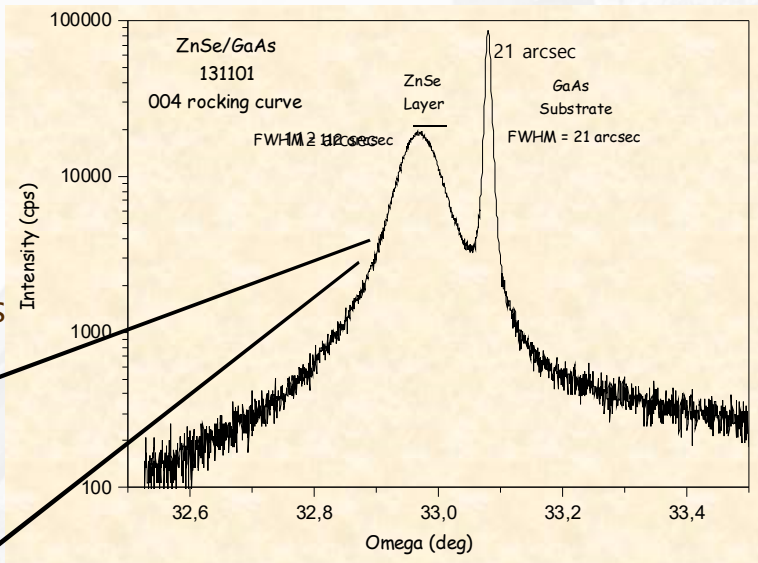
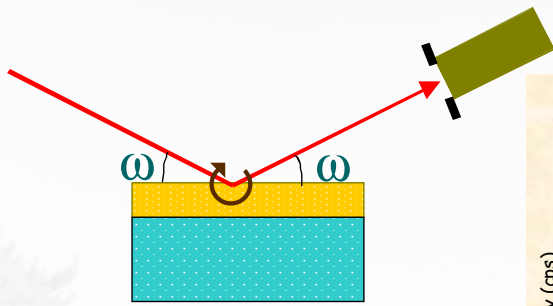
- The triple-axis coupled omega-2theta scan of the same Si wafer
- A rocking curve in triple-axis mode can be collected for each individual peak to determine the tilt variation of each individual Si(Ge) layer.

## Triple-axis > coupled scans vs Reciprocal Space Map

- Coupled scan collects data as  $\omega$ - $2\theta$ 
  - ✓  $2\theta$  is moved at twice the rate as the sample rotation about  $\omega$ .
  - ✓  $2\theta = 2\omega + \text{tilt}$
  - ✓ This will observe peaks with different Bragg angles, but only for one specific tilt.
  - ✓ If the epilayers are tilted w.r.t. the substrate, then a single coupled scan cannot observe both substrate and film peaks. → must collect coupled scans for a range of tilts: this is the **Reciprocal Space Map (RSM)**.
- The **RSM** collects several  $\omega$ - $2\theta$  coupled scans, but each coupled scan is collected with a slightly different tilt (offset) in the  $\omega$  direction.
  - ✓ When the scan is collected,  $2\theta$  still moves at twice the rate as the sample rotation so that  $2\theta = 2\omega + \text{tilt}$ .
  - ✓ The tilt value is slightly different for each coupled scan that is collected.
  - ✓ Complete map of  $\omega$ - $2\theta$  vs tilt ( $\omega$ ).



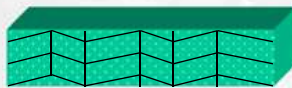
# Rocking curve > crystal quality



Lattice parameter fluctuations

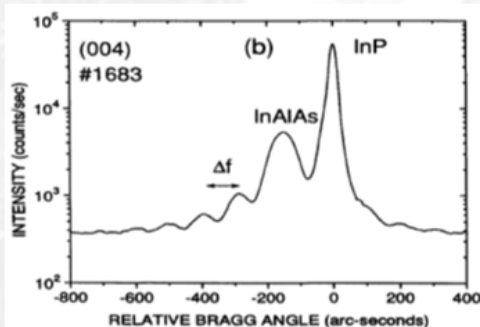
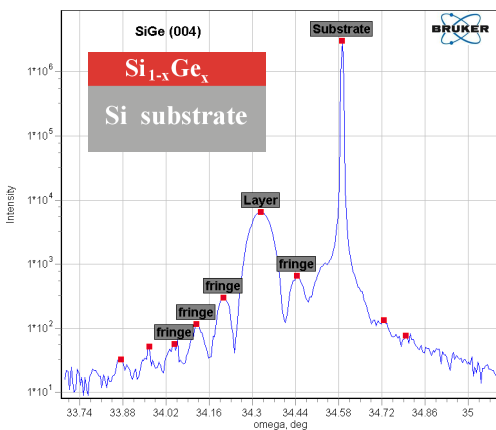


Mosaic structure



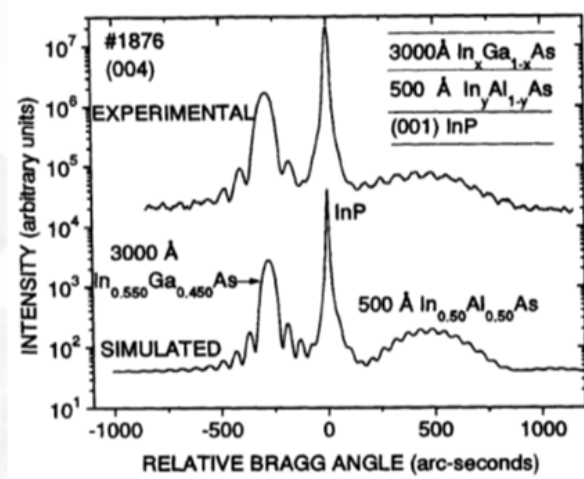
?

# Rocking curve > thickness and composition



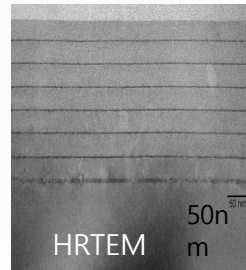
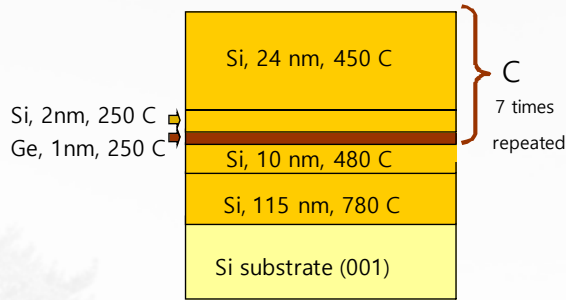
HRXRD (004) Rocking curve for  $In_{0.53}Al_{0.47}As$  on InP substrate

The best way to determine layer compositions & thicknesses is to compare the experimental rocking curve with simulated curves.



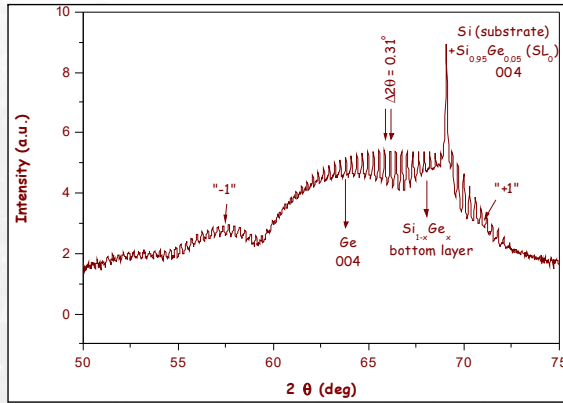
HRXRD (004) experimental and simulated rocking curves

# Rocking curve - superlattice of self-assembled ultra-small Ge quantum dots

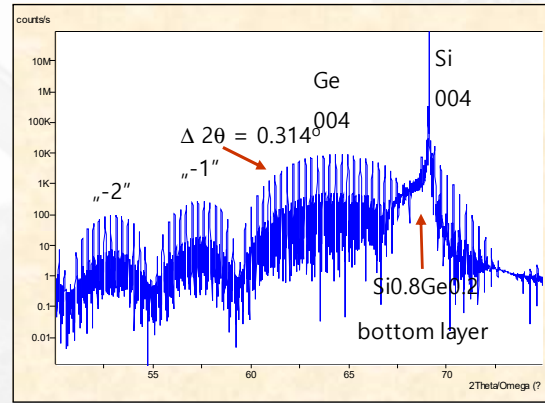


**Results:**

	HREM	XR
superlattice period C.....	≈ 33.5 nm	33 nm
thickness of Ge.....	≈ 1.8 nm	2.0 nm
thickness of SiGe <sub>x</sub>		
bottom layer.....	≈ 6.7 nm	6.7 nm
Compositon.....	----	x ≈ 0.2

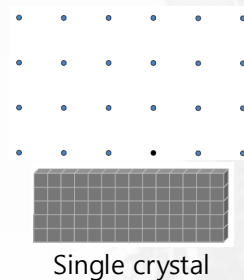
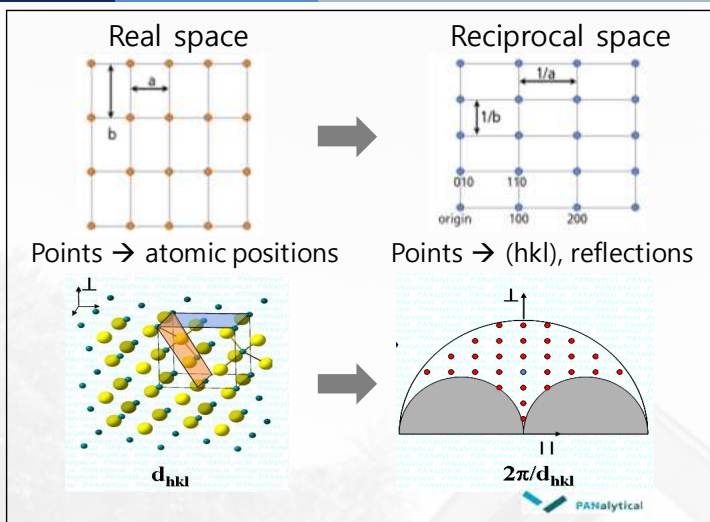


Experimental pattern, HasyLab (Hamburg)

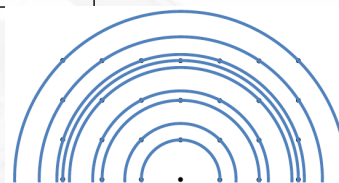


Simulated pattern, X'Pert Epitaxy and Smoothfit software

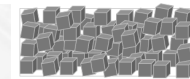
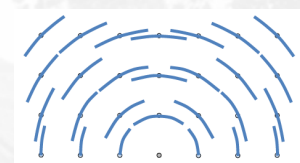
## Real vs. Reciprocal space



Single crystal

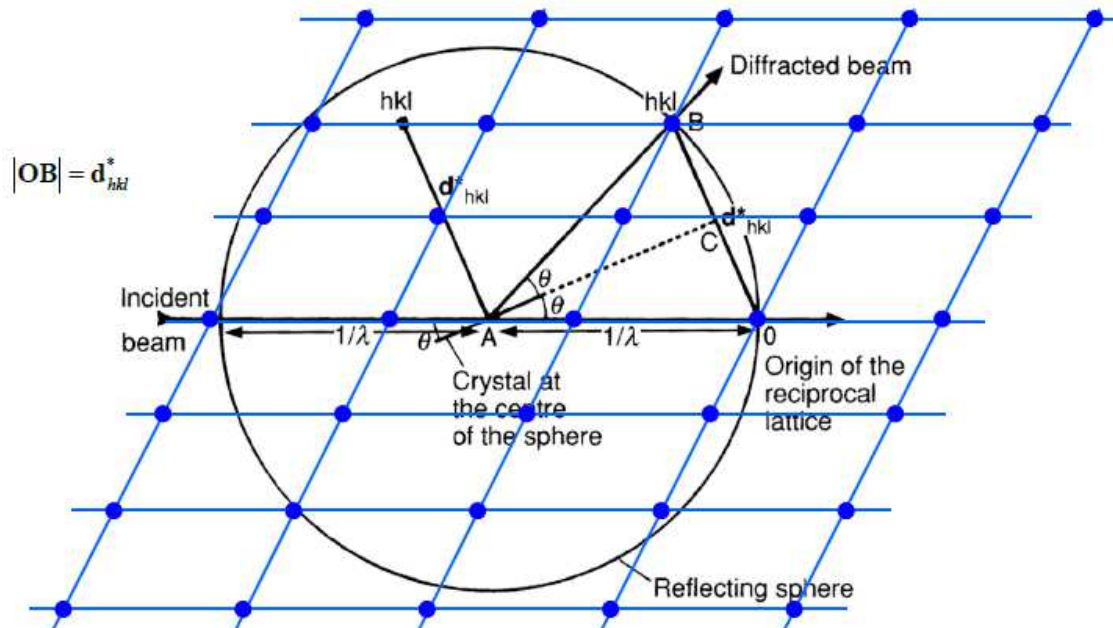


Polycrystalline random



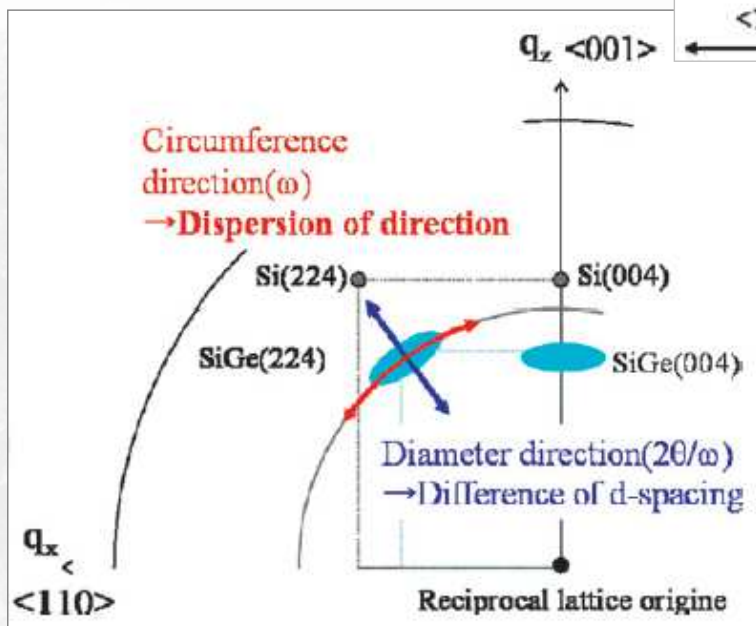
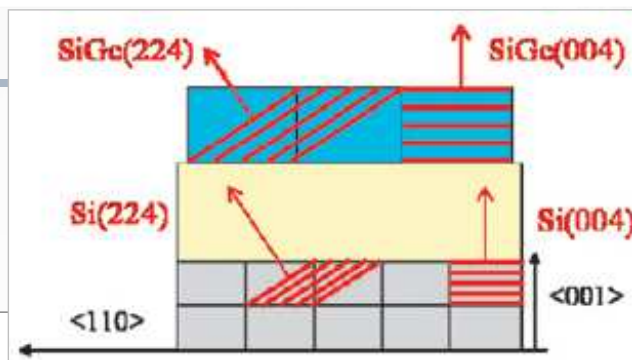
Polycrystalline textured

$$|OC| = \frac{1}{\lambda} \sin \theta = \frac{1}{2} |d_{hkl}^*| = \frac{1}{2d_{hkl}} \rightarrow \lambda = 2d_{hkl} \sin \theta$$



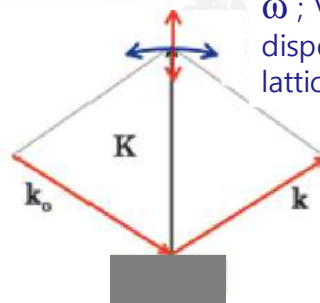
[www-sslrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering\\_epitaxiallayers\\_vailionis.pdf](http://www-sslrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering_epitaxiallayers_vailionis.pdf)

Reciprocal space map (RSM)



$2\theta/\omega$ ; Change or spread in lattice constant (d-spacing)

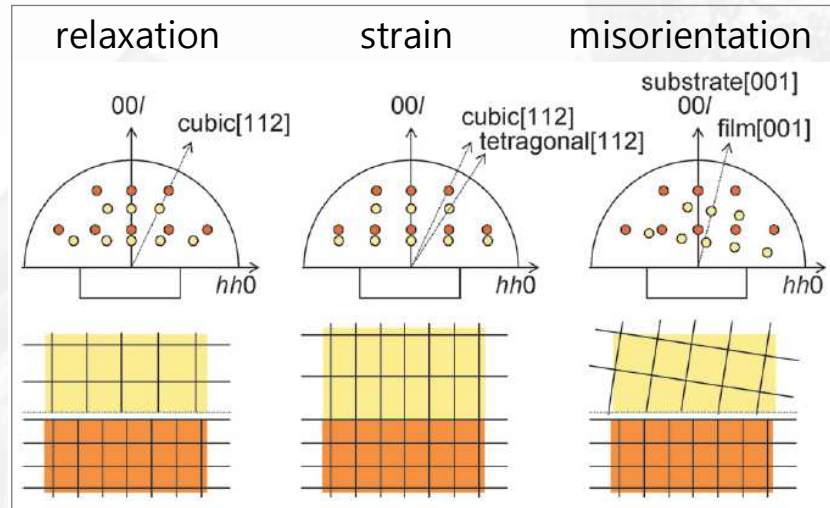
$\omega$ ; Variation and dispersion of lattice direction





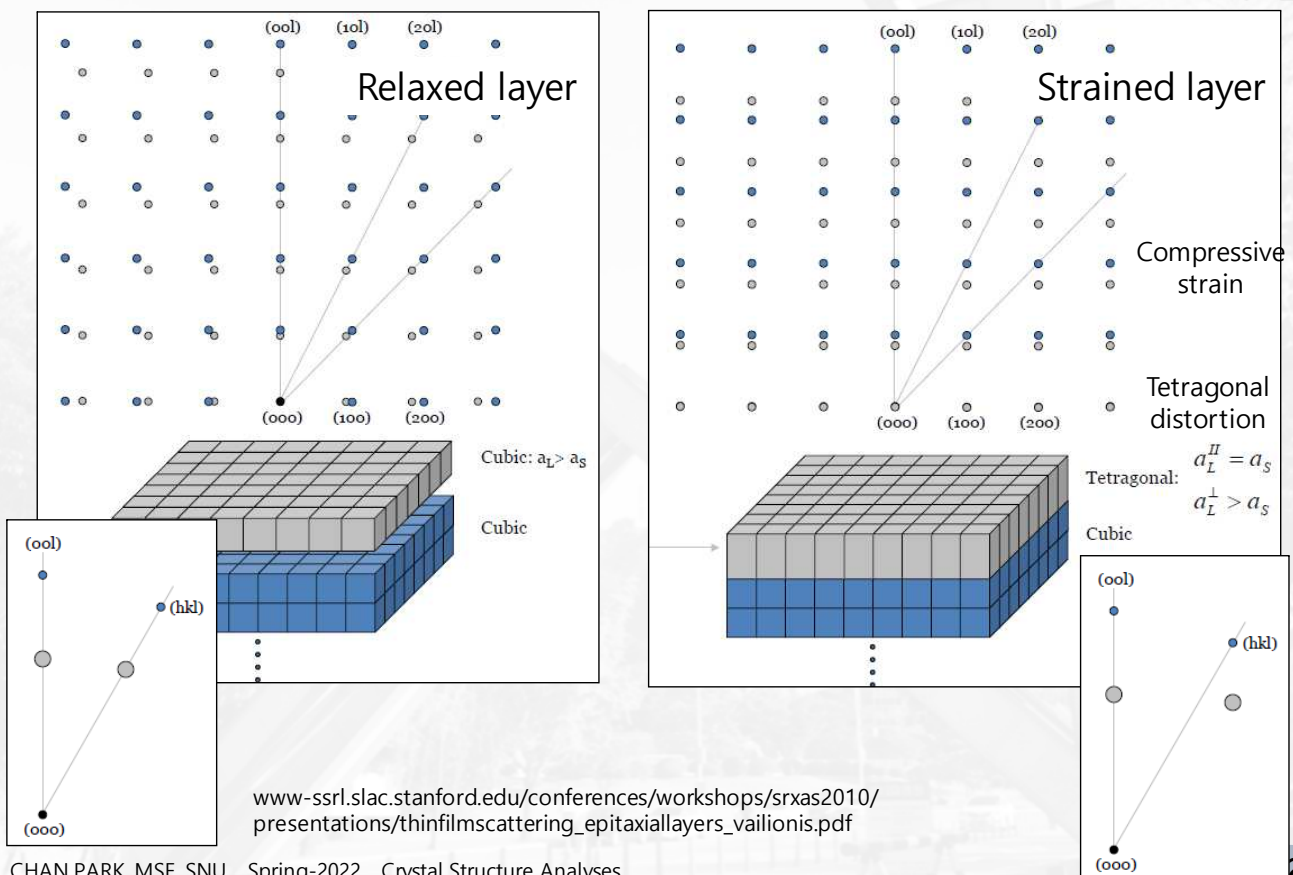
- RSM records diffraction intensity distributions by scanning both diffraction angle and sample rotation axes, and plots the result in the reciprocal space.
- RSM can provide info on orientation relationships, composition, thickness, mismatch, relaxation, layer tilt, mosaicity, defects profile, xtallinity, preferred orientation, etc.

RSM of hetero-epitaxial layer

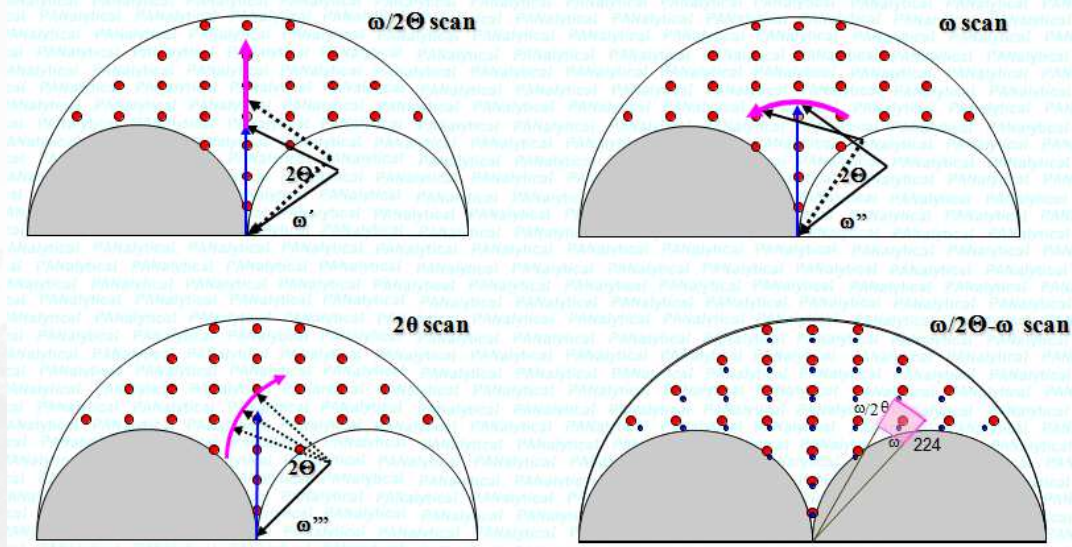


Rigaku Journal, 25(2), 2009, X-ray thin film measurement techniques

## Relaxed layer vs. Strained layer

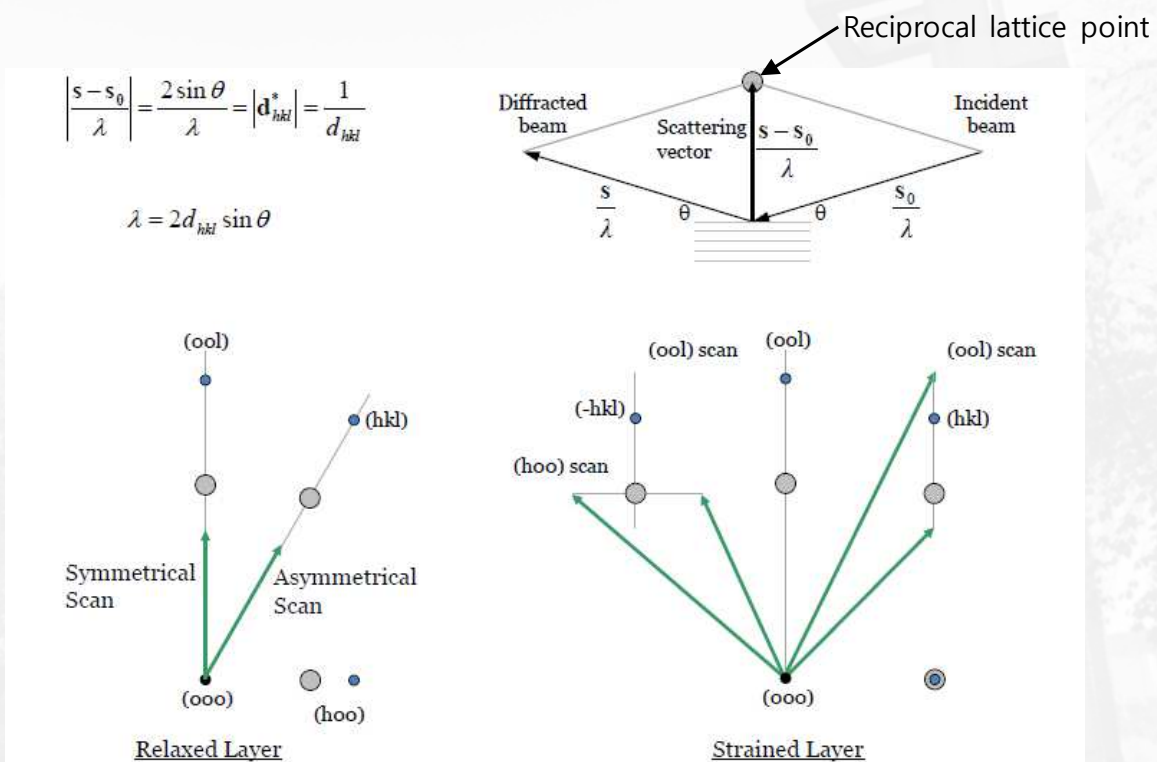






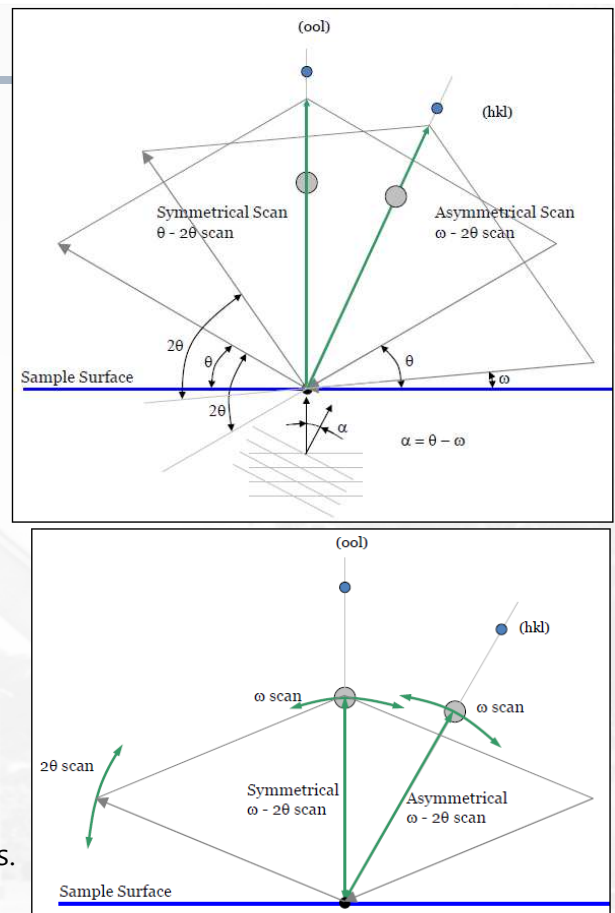
- Rocking curve ( $\omega$  scan) - arc centered on the origin
- Detector scan ( $2\theta$  scan) - arc along the Ewald sphere circumference
- Coupled scan ( $2\theta$ - $\omega$  scan) - straight line pointing away from the origin

# Reciprocal lattice – scattering vector



## Scan directions

- One family of planes is // or nearly // to the surface of the sample.
  - ✓ These are the only planes examined in a symmetric scan.
  - ✓ The sample is not tilted.  $\rightarrow 2\theta = 2\omega$
- Other planes can only be observed by tilting the sample.
  - ✓ Asymmetric scans are used to collect peaks from these other planes by tilting the sample about  $\omega$ .  $\rightarrow 2\theta = 2\omega + \text{tilt}$
- Several properties can only be determined by collecting both symmetric and asymmetric scans.

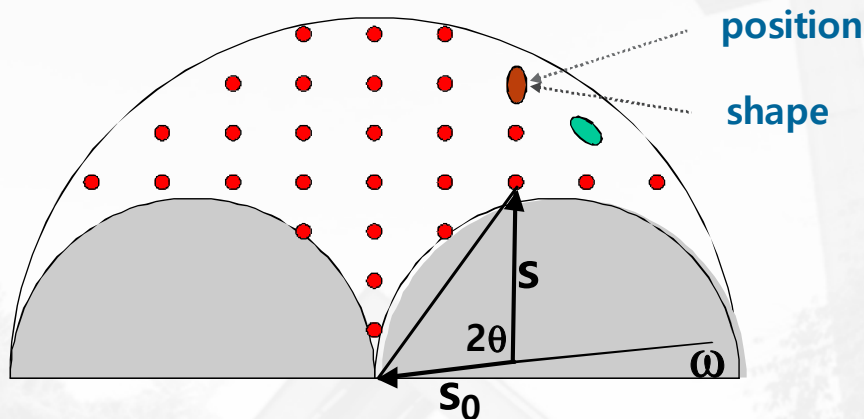


www-ssrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering\_epitaxiallayers\_vailionis.pdf  
 CHAN PARK, MSE, SNU Spring-2022 Crystal Structure Analyses

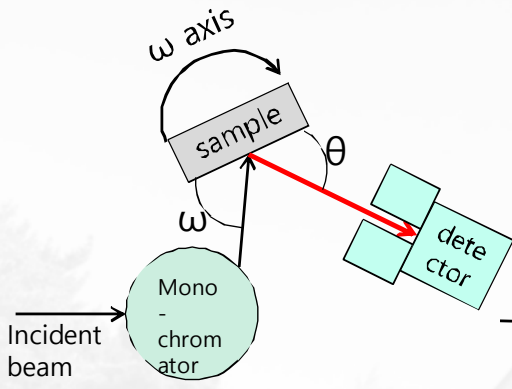
Scott A. Speakman

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## Symmetric vs Asymmetric

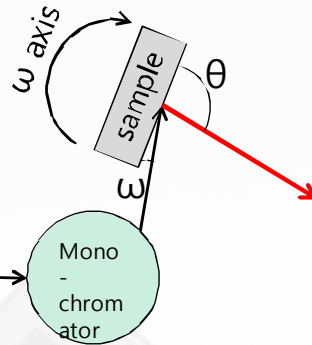
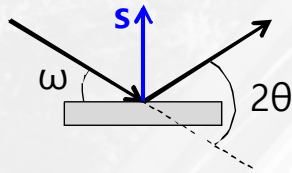


- Symmetrical reflection = planes parallel to surface
- **Asymmetrical** reflection = planes inclined to wafer surface
  - ✓ High angle of incidence or glancing exit =  $\omega > 2\theta/2$
  - ✓ Glancing incidence =  $\omega < 2\theta/2$



Symmetric Scan

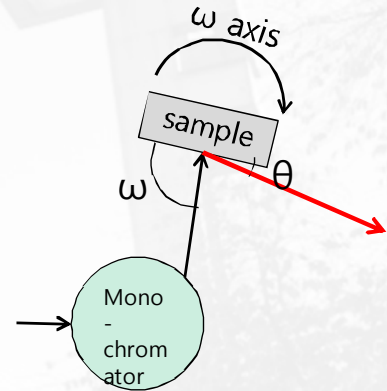
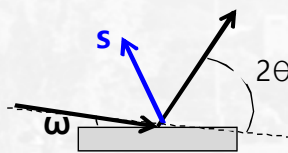
$\omega = \theta$



Asymmetric Scan

Grazing Incidence

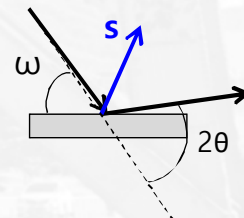
$\omega = \theta - \text{Tilt}$



Asymmetric Scan

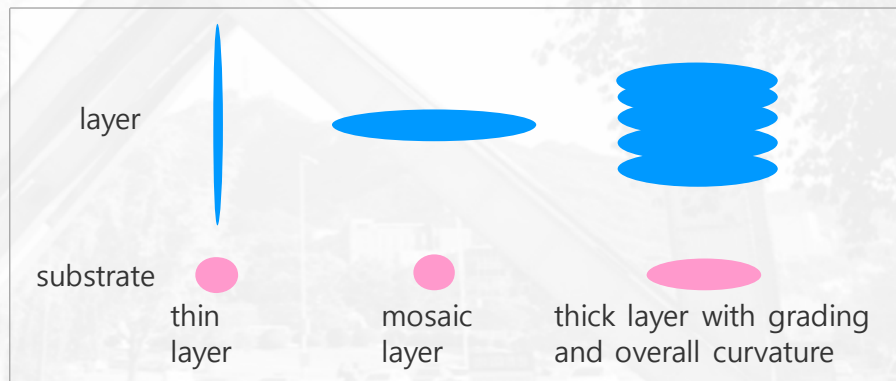
Grazing Exit

$\omega = \theta + \text{Tilt}$

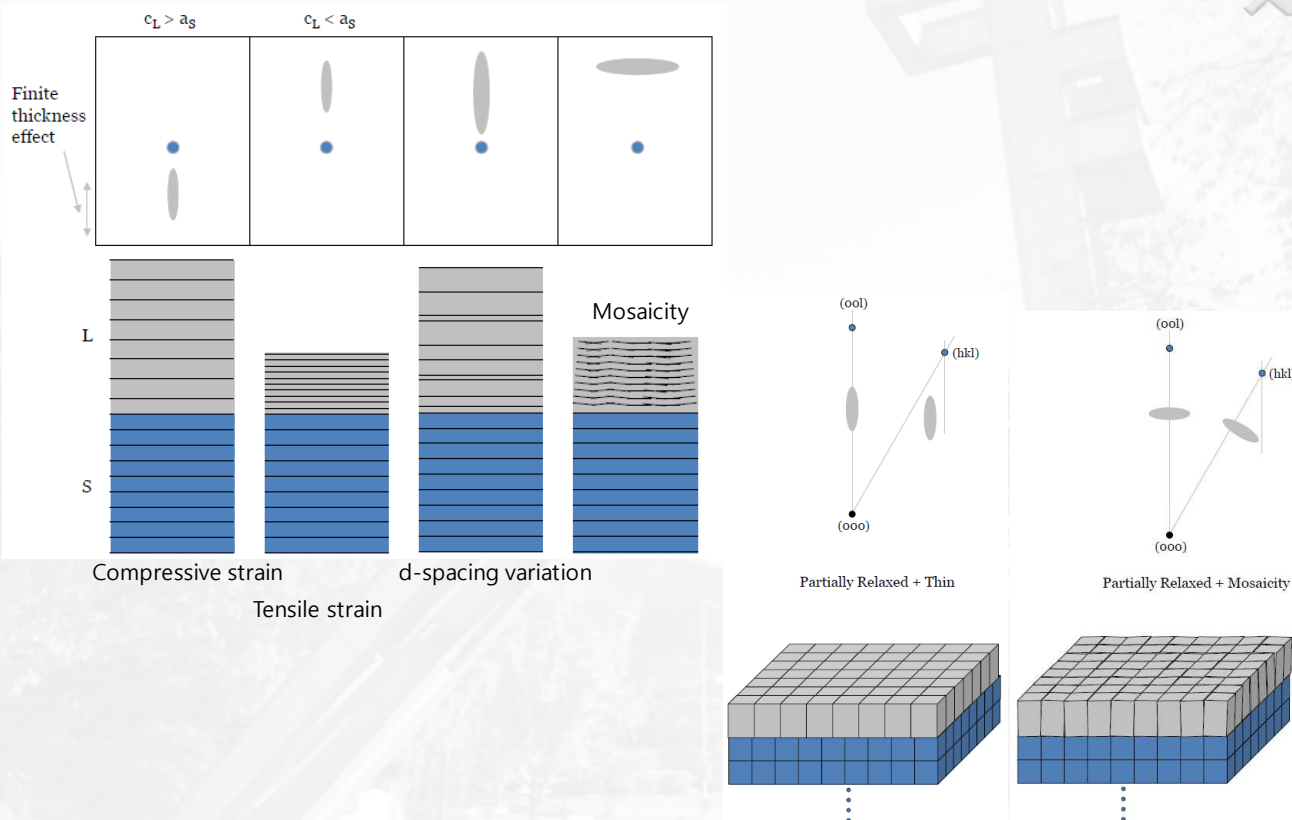


## Shape of reciprocal lattice point

- high quality substrate - sharp peak
- broadening normal to sample surface
  - ✓ thin layers
  - ✓ d spacing variation with depth
- broadening parallel to surface
  - ✓ mosaic structure
  - ✓ variable tilts (curvature or dislocations)

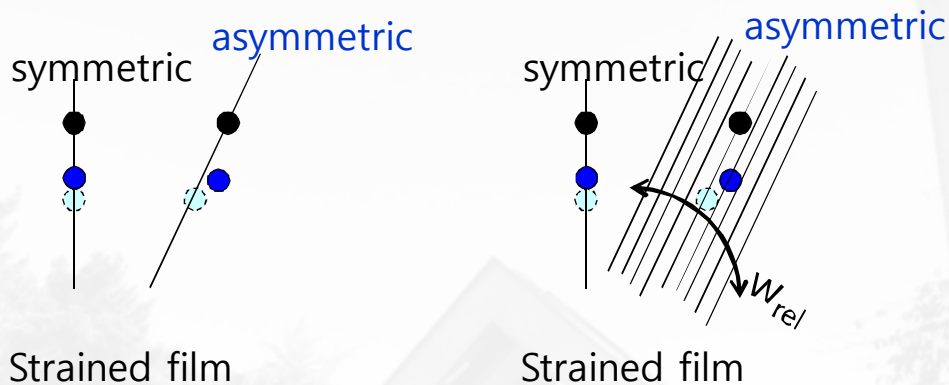


# Shape of reciprocal lattice point



[www-ssrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering\\_epitaxiallayers\\_vailionis.pdf](http://www-ssrl.slac.stanford.edu/conferences/workshops/srxas2010/presentations/thinfilmscattering_epitaxiallayers_vailionis.pdf)  
 CHAN PARK, MSE, SNU Spring-2022 Crystal Structure Analyses

# Highly strained film

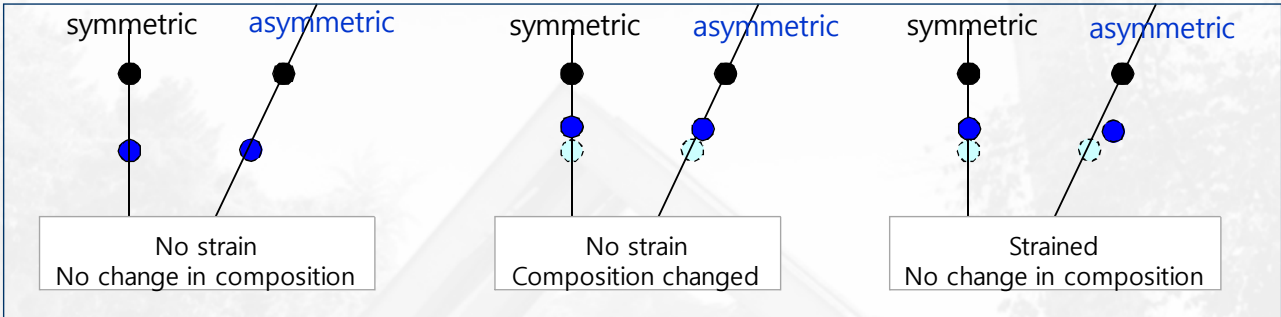


- The typical way to collect reciprocal space maps is to vary relative  $\omega$  and collect multiple  $2\theta/\omega$  coupled scans.



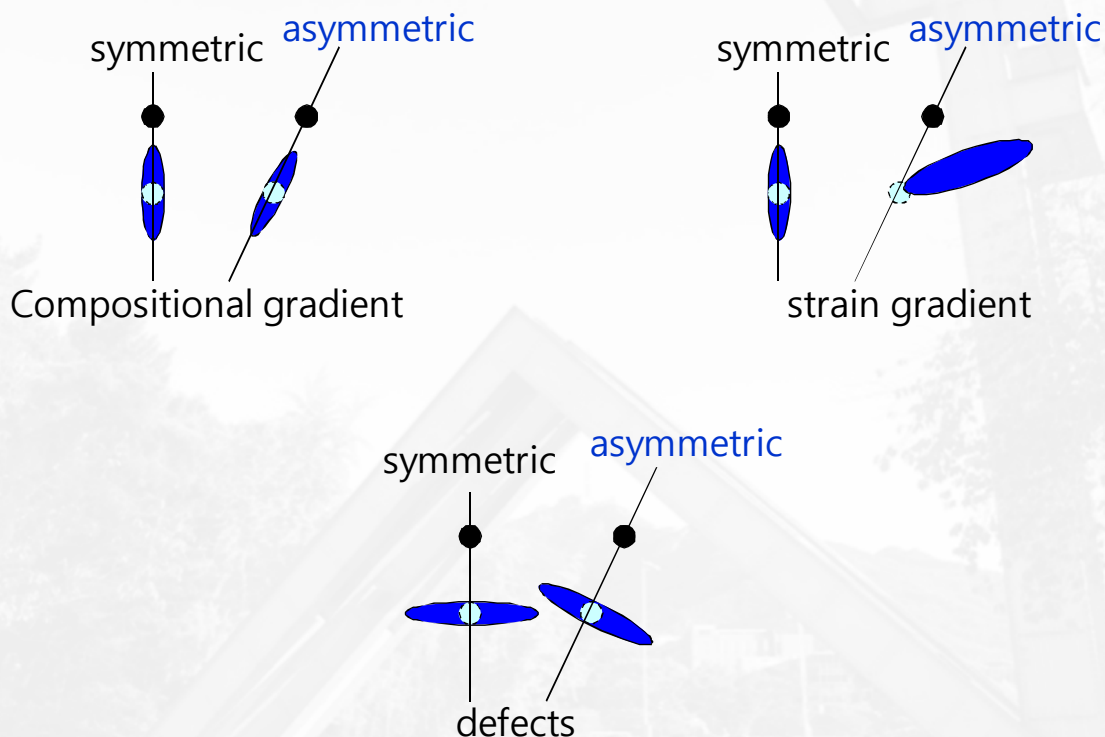
# Change of Strain and Composition

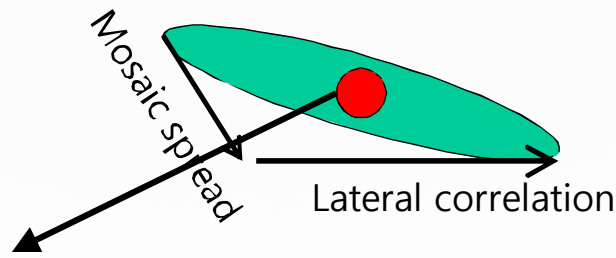
- In substitutional solid solutions, the composition can vary.
- Changes in the composition will change the lattice parameters, which will change  $d_{hkl}$  and therefore the Bragg peak positions.
  - ✓ Unlike relaxation, changes in composition will not change lattice tilts.



- Symmetric scans cannot distinguish between strain & compositional changes.
- In the symmetric scan, strain and compositional changes produce similar peak shifts.
- In order to quantify both strain and composition, must combine a symmetric scan with an asymmetric scan.

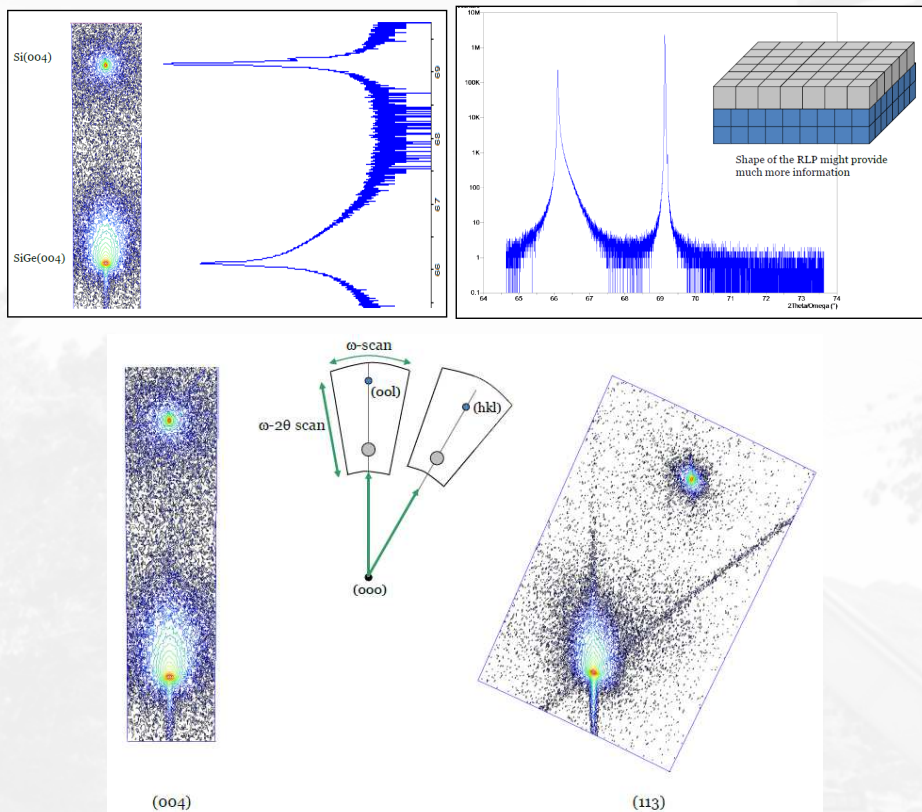
# Defects & gradients can produce spreading of RLP



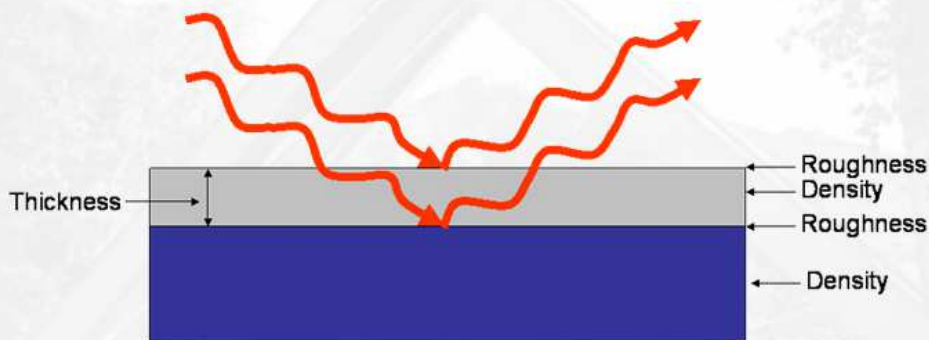


- The amount of broadening of the reciprocal lattice point that is  $\perp$  to the reflecting plane normal can be attributed to mosaic spread.
- The peak broadening  $//$  to the interface can be attributed to lateral correlation length.

Relaxed SiGe on Si(001)

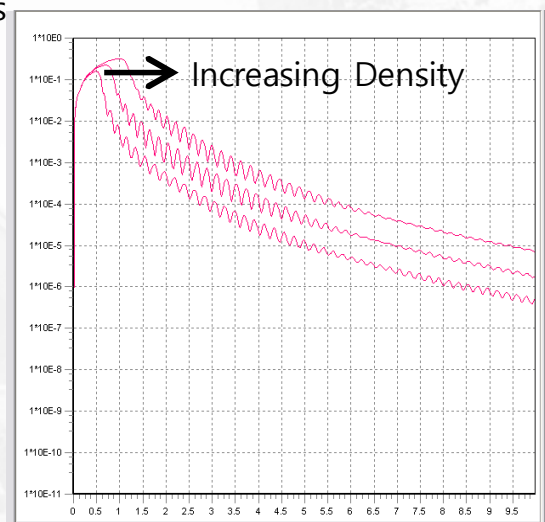


- The same equipment that is optimized for HRXRD can also be used for XRR analysis of thin films.
- X-ray waves reflecting from each different surfaces in a multilayer thin film.
  - ✓ The multiple reflected waves interfere with each other. → reflectivity curve
  - ✓ The XRR scan can be used to determine the density, thickness, and roughness of each layer in a multilayer thin film.



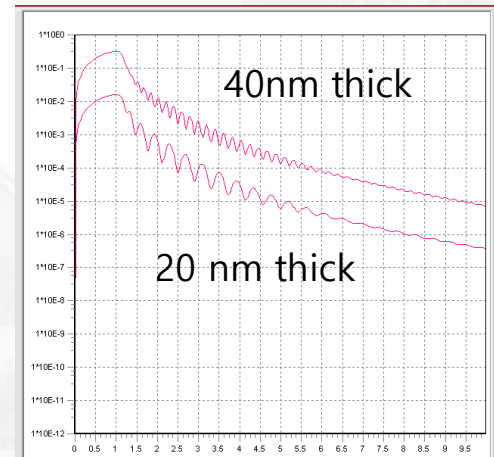
## XRR > critical angle

- The critical angle is a function of the density & composition of the layer.
- Below the critical angle,  $\theta_C$ , the X-ray beam is completely reflected (total external reflection).
- The critical angle for a layer is a function of its electron density.
  - ✓ This is a convolution of density and composition.
  - ✓ If one is known, the other can be determined using XRR.
  - ✓ For example, for a given composition, as the density of the film  $\uparrow$ , the critical angle  $\theta_C$  often  $\uparrow$ .



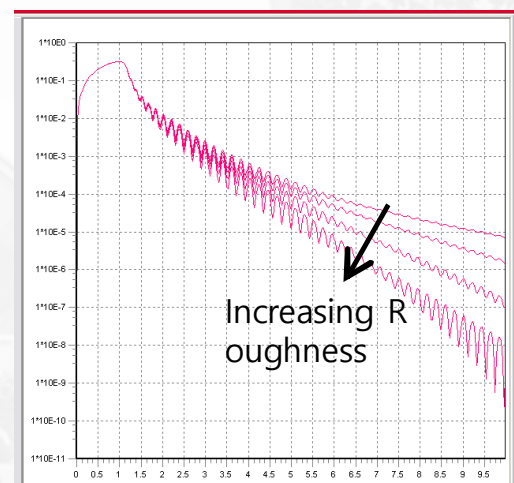
## XRR > distance between interference fringes

- The distance between interference fringes is a function of the thickness of the layers.
- Interference fringes are created by the phase difference between X-rays reflected from different surfaces.
- The distance between the fringes is inversely proportional to the thickness of the layer.
  - ✓ Because of this, thicker films need better resolution (use a monochromator) and thinner films need more intensity (use only the mirror).

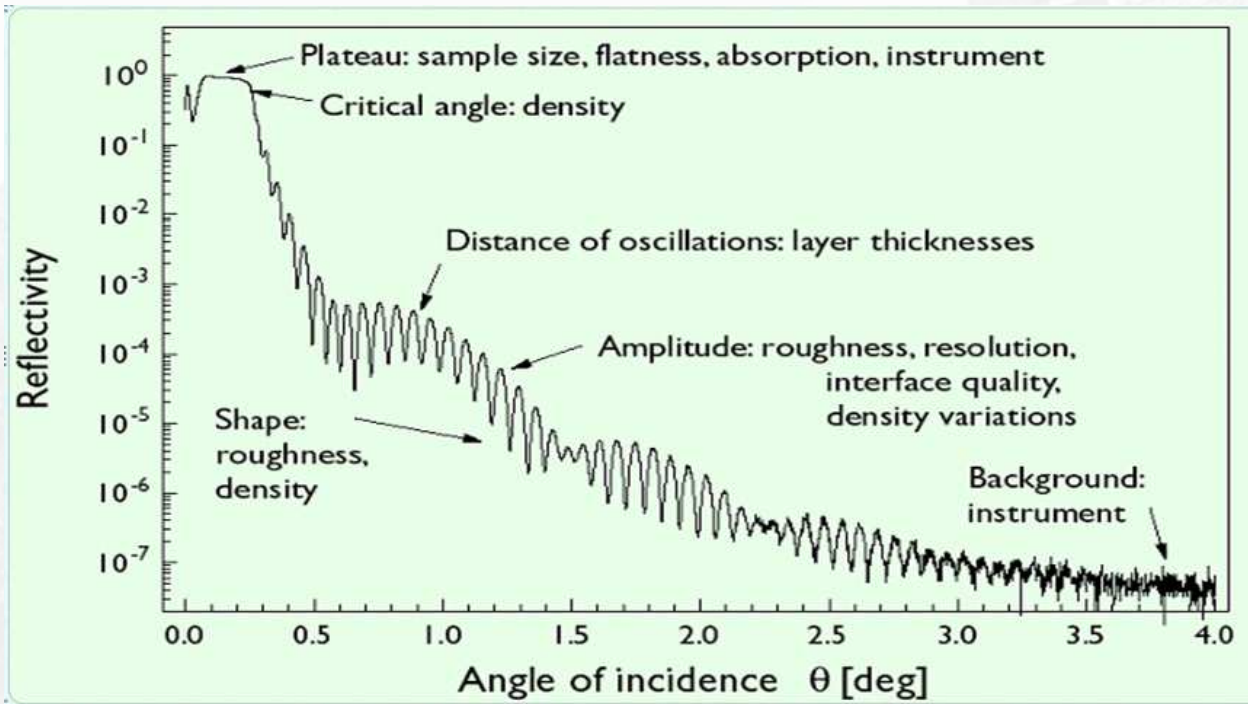


## XRR > how quickly the reflected signal decays

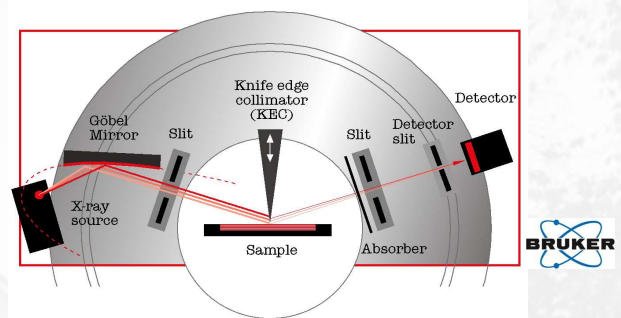
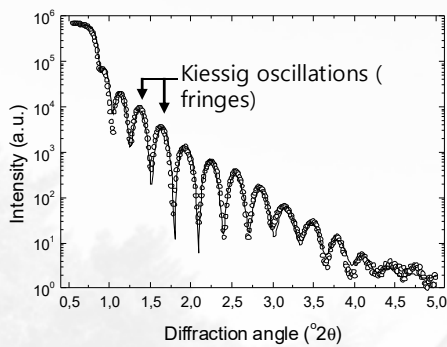
- Roughness determines how quickly the reflected signal decays.
- Roughness causes X-rays to be scattered rather than reflected.
  - ✓ This produces a decay in the reflected beam intensity.
  - ✓ The loss of beam intensity  $\uparrow$  with  $\theta$ .
- A rougher surface produces more diffuse scatter, causing the reflected beam intensity to decay more with  $\theta$ .
  - ✓ The diffuse scatter can be measured to look for order in the roughness of the film.



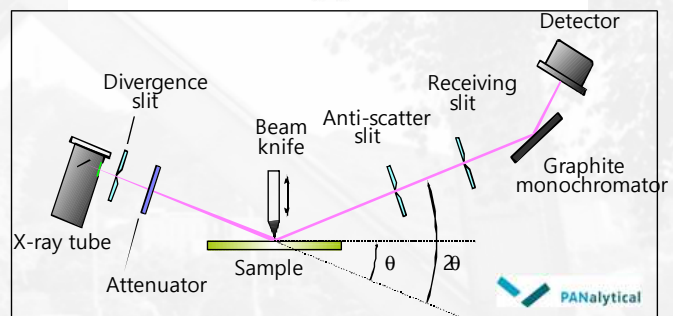




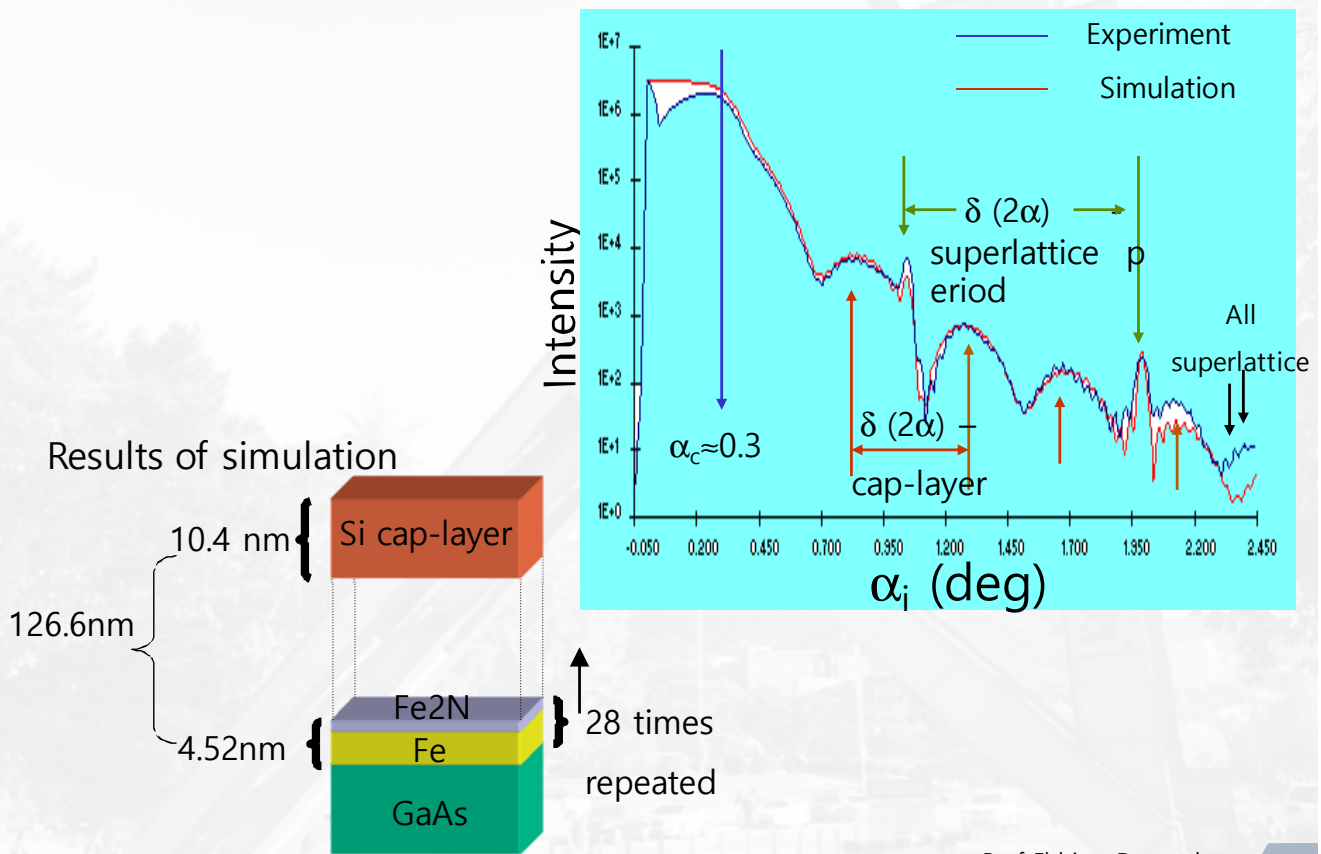
➤ density, thickness and interface roughness for each layer ← multi-layer thin film



	$\rho$	$t$ [Å]	$\sigma$ [Å]
Mo	0.68	19.6	5.8
Mo	0.93	236.5	34.0
Mo	1.09	14.1	2.7
W	1.00	5.0	2.7
Si	1.00		2.8



The surface must be smooth (mirror-like).



GMR Heterostructure

N	R	Material	Thickness	Roughness	Density T
1	1	Ta <sub>2</sub> O <sub>5</sub>	3.2159	0.84140	7.43604
2	1	Ta	3.3487	1.04430	16.99877
3	1	NiFe amorph	8.7575	0.91430	8.07869
4	1	Ru	0.7350	0.32740	12.19874
5	1	Co	1.5981	0.46450	8.60127
6	1	PtMn crystalline	18.1875	0.39210	13.67539
7	1	NiFe amorph	2.9749	0.24740	8.64525
8	1	Al <sub>2</sub> O <sub>3</sub>	4.6321	0.27800	3.00021
SUB	1	SiO <sub>2</sub>	0.0000	0.30580	2.20000

