Chapter 8. Polymers in Hyperstructures

8.1. Microstructure Based on Block Copolymers

- 8.1.1. Definition of Copolymers
- 8.1.2. Synthesis Methods for Block Copolymers
- 8.1.3. Block Copolymer Aggregate and Self-assembly
- 8.2. A Closer Look at Microstructure
 - 8.2.1. Block Copolymer Phase Behavior
 - 8.2.2. Amphiphilic Block Copolymers in Mixtures with Water and Oil

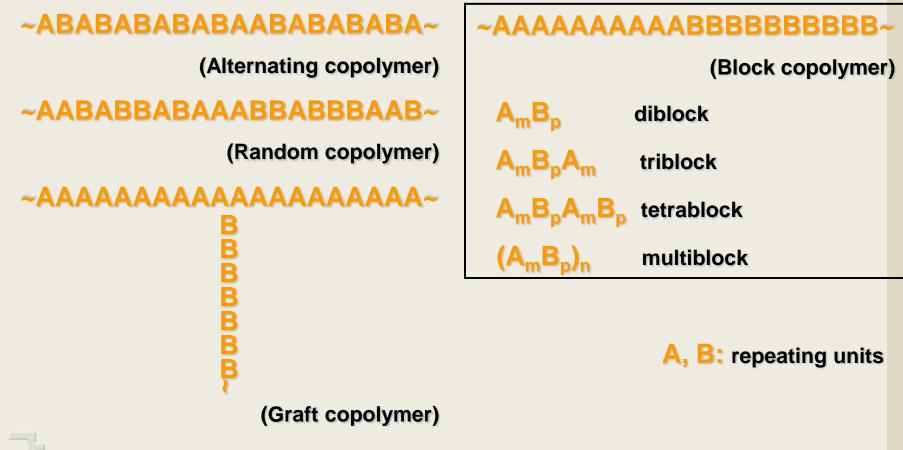
8.3. Applications of Copolymers

- 8.3.1. Commercialized Applications
- 8.3.2. Potential Applications



8.1.1 Definition of copolymers

: polymer having more than two different repeat units.

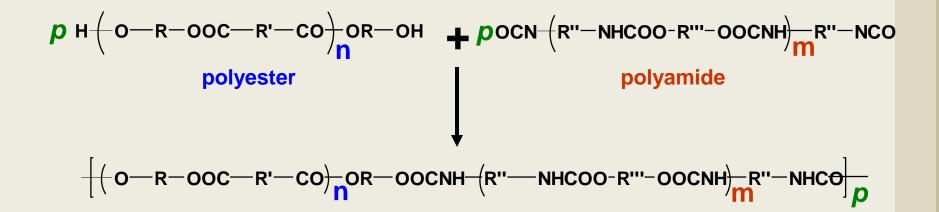




8.1.2 Synthesis Methods for Block Copolymers

Step copolymerization

→ The reaction between two different functional end groups of polymers

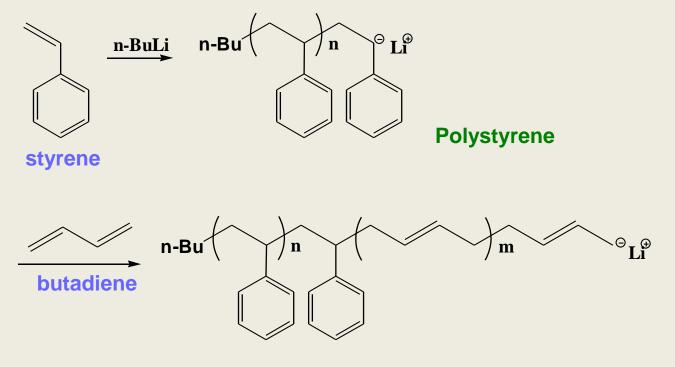






Sequential polymerization (Anionic polymerization)

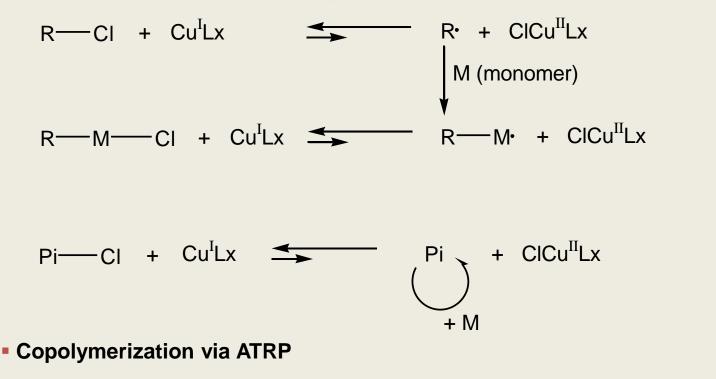
 \rightarrow A polymer is synthesized by anionic polymerization. Another monomer is then added to the living polymer carbanions.



Polystyrene-block-Poly(1,4-butadiene)



- Sequential polymerization (Living radical polymerization)
 - ATRP (Atom transfer radical polymerization)



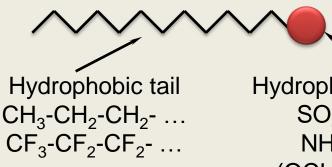
 $R \xrightarrow{A, CuX} R \xrightarrow{A_n} X \xrightarrow{B} R \xrightarrow{A_n} A_n$

A, B; monomer



8.1.3. Block Copolymer Aggregate and Self-assembly

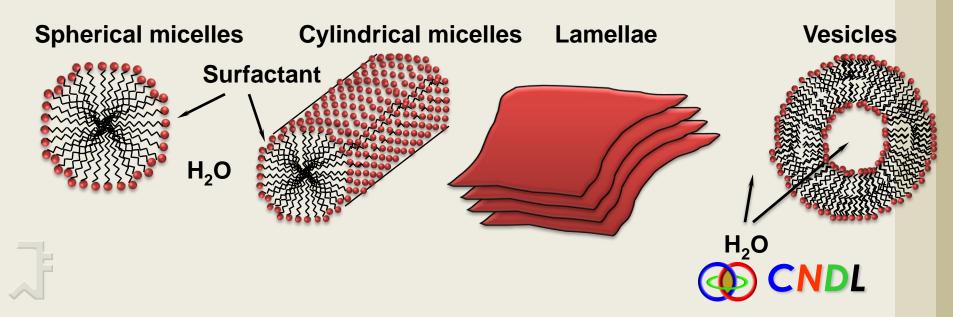
Low-molecular-weight surfactants



Hydrophilic head SO_3 -Na+ NH_3 +Cl--(OCH_2CH_2)_n-

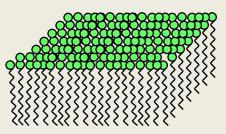
When surfactants are dissolved in water

- reduce the surface tension because they are adsorbed on the surfaces
- form variety of aggregates micelles, lamellae, vesicles, etc



***** Properties of the surfactant film

Surfactant film



Properties of the surfactant film change with:

- Molecular structure
- Additives
- Ionic strength
- Co-surfactant
- Temperature, pressure etc.

Properties of the surfactant film:

- Interfacial tension
- Lateral elasticity

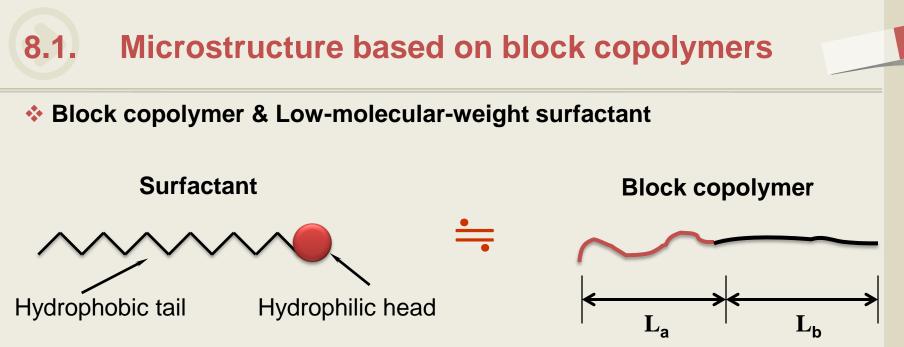


- Spontaneous curvature
- Bending elasticity
- Saddle splay elasticity









More than two polymeric units (different solubility)

HLB value (hydorphilic-liphophilic balance) Geometrical factor (size) → Appendix 8.1 Length and ratio of polymer segments

Like surfactant, block copolymers can form various aggregates as varied polymer segments.



***** Appendix 8.1 : HLB Concept

 In an attempt to extend it into some kind of quantitative relationship between surfactant hydrophilicity and function in solution, Griffin introduced the concept of the hydrophilic-lipophilic balance (HLB) of a surfactant.
 Griffin's HLB numbers were restricted to non-ionic surfactants.

An arbitrary scale of 0 to 20 :

- : 0 ~ completely hydrophobic molecule
- 20 ~ completely hydrophilic molecule

HLB number range	Appearance of aqueous solution
1-4 3-6 6-8 8-10 10-13	No dispersibility Poor dispersibility Milky dispersion after agitation Stable milky dispersion From translucent to clear
13–20	Clear solution
HLB number range	Application
3-6 7-9 8-14 9-13 10-13 12-17	w/o Emulsifier Wetting agent o/w Emulsifier Detergent Solubilizer Dispersant

Ref.) W.C.Griffin, Journal of the Society of Cosmetic Chemists, 1949, 1, 311 W.C.Griffin, Journal of the Society of Cosmetic Chemists, 1954, 5, 259

Table. Use of Griffin's HLB number concept

***** Appendix 8.1 : HLB Concept

- Griffin's HLB number concept was later extended by Davies, who introduced a scheme to assign HLB numbers to chemical groups which compose as surfactant.

HLB =

7 + \sum (hydrophilic group numbers) + \sum (lipophilic group numbers)

Ref.) K.Holmberg, Surfactants and polymers in aqueous solution, 2nd Ed., John Wiley & Sons Ltd, England, 2003, p.460

J.T.Davies, Proceedings of the International Congress of Surface Activity, 1957, 426-438

Group	HLB number
Hydrophilic	
-SO ₄ Na	35.7
$-CO_2K$	21.1
$-CO_2Na$	19.1
-N (tertiary amine)	9.4
Ester (sorbitan ring)	6.3
Ester (free)	2.4
$-CO_2H$	2.1
-OH (free)	1.9
-0-	1.3
–OH (sorbitan ring)	0.5
Lipophilic	
$-CF_3$	-0.870
$-CF_2-$	-0.870
$-CH_3$	-0.475
$-CH_2-$	-0.475
-CH-	-0.475
-CH-	

Determination of HLB numbers according to Davies

"The HLB method of selecting an emulsifier is crude but simple."

- Some general guidelines for the selection of surfactants as emulsifier
- (1) The surfactant should have a strong tendency to migrate to the oil-water interface.
- (2) Oil-soluble surfactants preferably form w/o emulsions, and vice versa.
- (3) Good emulsions are often formed by using a mixture of one hydrophilic and one hydrophobic surfactant.
- (4) The more polar the oil phase, then the more hydrophilic the emulsifier should be, and vice versa.

Table. The use of the HLB method in selecting an emulsifier.

Emulsification of a mixture of 20% paraffin oil (HLB=10) and 80% aromatic mineral oil (HLB=13) in water

HLB number of oil: $10 \times 0.20 + 13 \times 0.80 = 12.4$

A mixture of $C_{12}E_{24}$ with HLB = 17.0 and $C_{16}E_2$ with HLB = 5.3 is used. A 60:40 mixture of the two gives a surfactant HLB number as follows:

 $17.0 \times 0.60 + 5.3 \times 0.40 = 12.3$

This surfactant combination is found to give excellent emulsion stability.

Ref. M.J.Rosen, Surfactants and interfacial phenomena, 2nd Ed., Wiley, New York, New

***** Appendix 8.1 : Geometric Factors

- The structures of micelle can be simply determined by the geometric factors of the surfactant at the interface.
- Critical packing parameter (CPP)



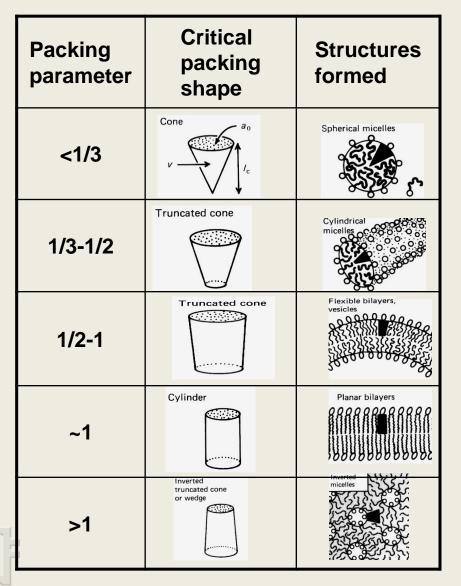
- v : partial molecular volume of surfactant
- a_{o} : area of head group of surfactant
- $I_{\rm c}$: maximum chain length

(1) Spherical micelles : CPP < 1/3
(2) Non-spherical micelles : 1/3 < CPP < 1/2
(3) Vesicles or bilayers : 1/2 < CPP < 1
(4) Inverted micelles : 1 < CPP

- The change of environment (pH, temperature, ionic strength) will affect these parameters.



*** Appendix 8.1 : Geometric Factors**



-The relationship between aggregate type and geometry on the packing requirements of surfactant head group and chains

Ref. Intermolecular and Surface Forces, Israelachvili, Jacob N.

(1) Spherical micelles

- Usually formed by anionic surfactants
- For an o/w micelles, this can be done by adjusting the repulsion between head groups, resulting in large values for a_0 .

(2) Cylindrical micelles

- It is a quite common phenomenon that micelles grow as the preferred surface curvature decreases. Any change that reduces the effective head group area will lead to the growth of micelles.
- Basic three ways to form cylindrical micelles

1 addition of a cosurfactant with a very compact head group (i.e. n-alkanol)

2 changing the counterion

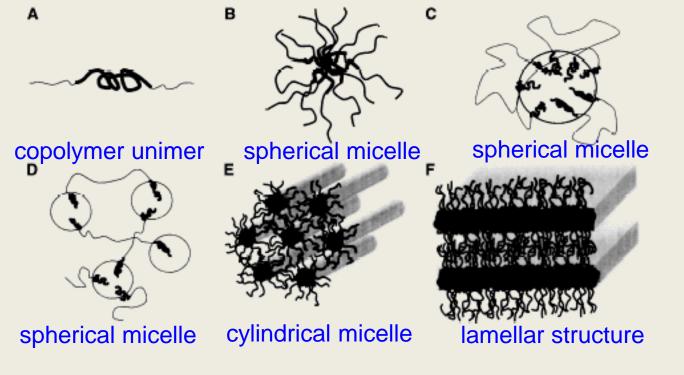
(i.e. changing Na⁺ to Mg²⁺ reduce the effective volume of head groups.)

- ③ by electrolyte addition or temperature change
- (i.e. reduce the area of the head groups)



Micelle formation of block copolymer

- ; A solvent that is good for one of the units but poor for the other(s) leads to an amphiphilic behavior. (similar to low-molecular-weight surfactant solution)
- ; Block copolymer micelles are aggregates that resemble many properties of micelles formed by low-molecular weight surfactants.



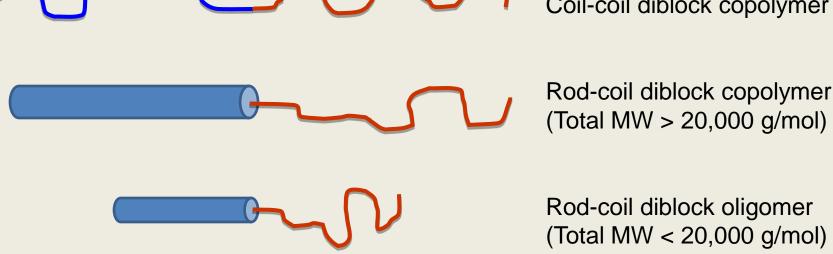


Self-assembly of Block copolymer based on type of polymer segments

- ; The types of polymer segments can affect block copolymer aggregation.
- ; Three different classes of diblock copolymer type will be discussed in this section.

Coil-coil diblock copolymer

(self-assembly)





Coil-coil diblock copolymers

- ; Block copolymers comprised of two flexible, chemcally incompatible and dissimilar blocks can microphase separated into a variety of morphologies. eg) Polystyrene-*b*-Polyisoprene
- ; The degree of microphase separation is determined by $\chi N.$
 - $\rightarrow \chi$: Flory interaction parameter,

a measure for the incompatibility between the two block, be written in terms of solubility parameter(δ)

$$\chi = \frac{V_0}{kT} (\delta_A - \delta_B)^2 \quad \delta_A = \sqrt{\frac{\Delta E_A}{V_A}}$$

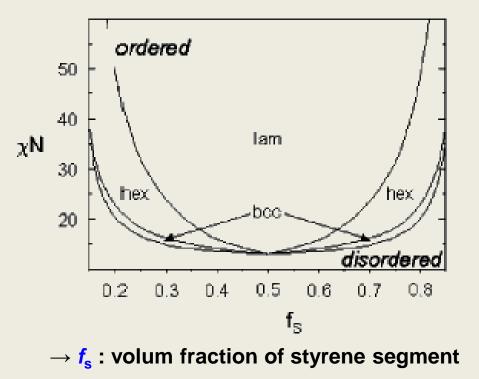
V_o: the lattice site volume

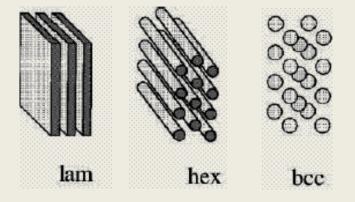
- k : Boltzmann constant
- T : temperature
- ΔE_A : the evergy of vapourization of A molecule
- V_A : molecular volume of A units
- ※ Solubility parameter ↔ cohesive energy density, interaction energy per unit volume between the molecules
- → N : Total degree of polymerization, N=N_A+N_B





- ; Separated morphology can be varied from spheres via lamellae to inverse spheres by changing the volume fractions of the blocks (*f*).
 - ***** Experimental phase diagram for polystyrene-b-polyisoprene





 χ N≤10, weak-segregation limit 10 < χ N≤50, intermediate segregation region χ N→∞, strong segregation limit



Rod-coil diblock copolymers

- ; Replacing one of the blocks of a coil-coil diblock copolymer by a stiff, rigid segment results in a rod-coil type diblock copolymer. eg) poly(*p*-phenylene)-*b*-poly(styrene), poly(phenylquinoline)-*b*-poly(styrene), poly(heyl isocyanate)-*b*-poly(styrene)
- ; The self-assembly is no longer solely determined by phase-separation, but is also affect by several other process.
 - \rightarrow aggregation of the rigid segments into (liquid-)crystalline domains
 - \rightarrow increase in the Flory interaction parameter due to stiffness asymmetry

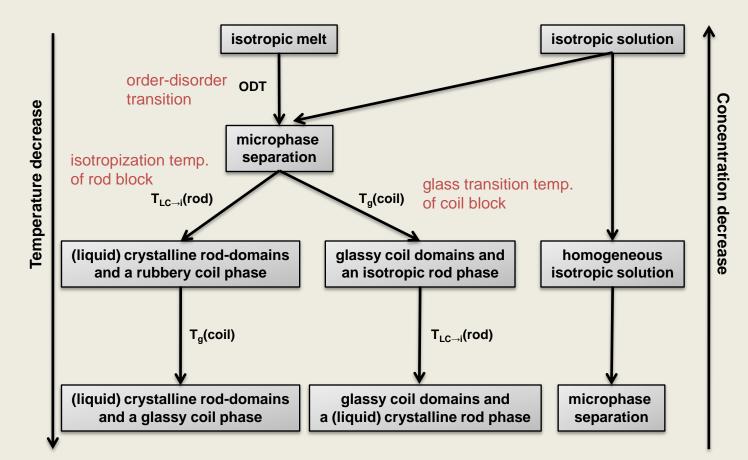


 $(\delta_A - \delta_B)^2 > (\delta_A - \delta_B)^2$ $\therefore \chi$ is increased in rod-coil diblock copolymers



reference 6

***** Possible routes for the self-assembly of rod-coil type diblock copolymer





- ; Cooling from an isotropic melt
 - 1) The system passes ODT and undergo microphase separation
 - 2) If $T_{LC \rightarrow i}$ of rod > T_g of coil,
 - further cooling 1 formation of (liquid-)crystalline domains surrounded by a layer of rubbery coils
 - further cooling 2 (liquid-)crystalline domains of rod segments surrounded by glassy phase of coil segments

If T_{LC→i} of rod < T_g of coil, further cooling 1 - formation of glassy matrix further cooling 2 - crystallization of rod blocks can only take place in confined domains imposed by the glassy matrix formed by the coils



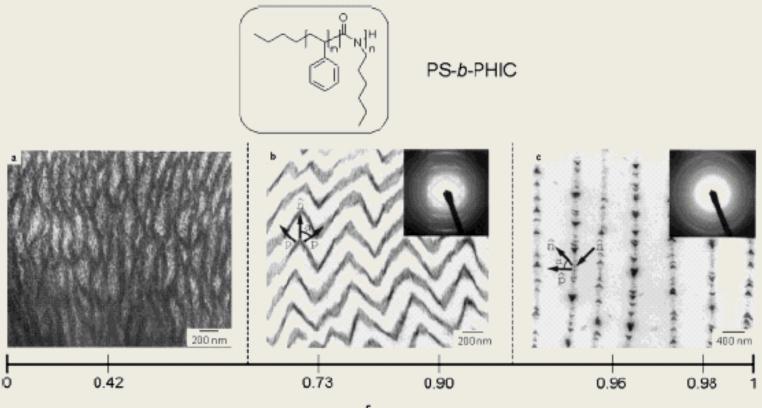
; Evaporation of the solvent from isotropic solution

- 1) If a good solvent for both blocks is used, microphase separation will continuously complete with crystallization of the rod segments during the evaporation of solvent until a stable structure is obtained.
- If a good solvent for only one block is used, the rod-coil block copolymers can already be pre-assembled into a particular supramolecular structure prior to film casting and evaporation of the solvent.





Self-assembly of Poly(styrene)-b-poly(hexyl isocynate) with the volum fraction of hexyl isocynate segment



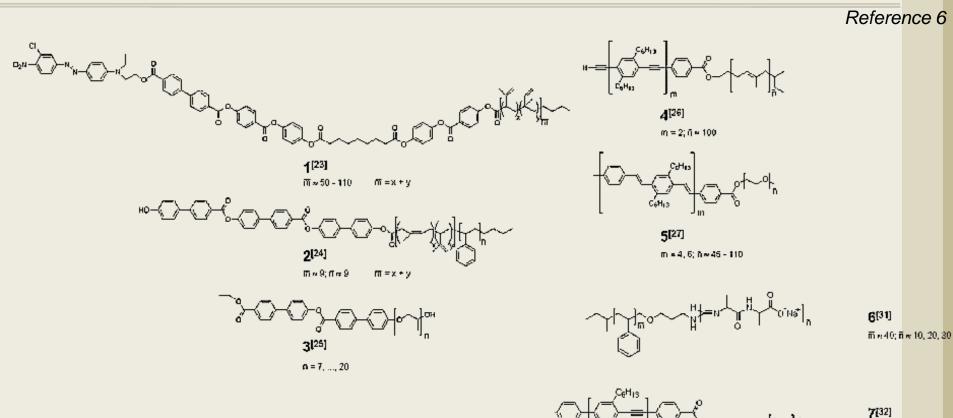




Rod-coil diblock oligomers

- ; Like rod-coil diblock copolymer, the oligomers also indicates the competition between microphase separation of the blocks and aggregation of the rigid rods.
 - \rightarrow increase in the Flory interaction parameter due to their stiffness-asymmetry
 - → Rod-coil diblock oligomers can allow access to phase-separated morphologies with domain sizes that can not be attained with traditional coil-coil diblock copolymers.
- ; Rod-coil diblock copolymers can be divided into two major classes.
 - \rightarrow i) perfectly monodisperse rod-segments, ii) polydisperse rigid rod
 - → Diblock oligomers can be simply synthesized by organic reactions, so monodisperse rod-segments can be obtained. cf) Polymer's MW is polydisperse
 - ※ Polydispersity (MW) is known to have a strong influence on the liquid crystalline properties of rod-like macromolecules.





***** Example of rod-coil oilgomers

8.1.

8[34] m = 8; n = 115

m = 12; n = 15



8.2.1 Block Copolymer Phase Behavior (in the "melt" state)

(1) One-component block copolymer system

- The different blocks of a block copolymer are able to segregate (microphase separation) and form domains with different morphologies.
- The stability of different morphologies

 \rightarrow From an interplay between enthalpic (contact between chemically different blocks) and entropic (chain stretching, confinement, frustration) contribution to the system free energy.

 \rightarrow Described in terms of a χN vs *f* phase (Fig 8.1)

χ : Flory-Huggins interaction parameter between the different blocks

 \rightarrow Enthalpic contribution

N : degree of polymerization (copolymer molecular weight)

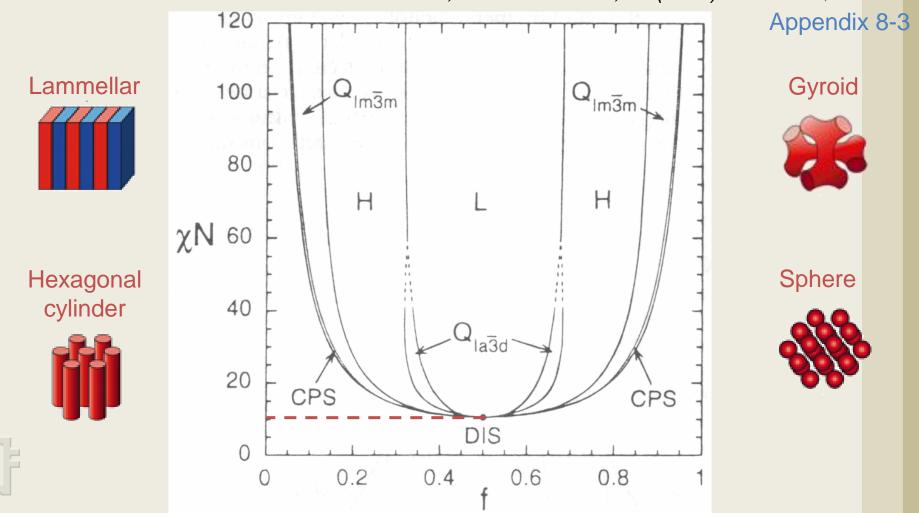
 \rightarrow Tendency for block segregation \rightarrow Entropic contribution

f: volume fraction of one block (composition variable)

- The higher χN → higher degree of segregation between the different blocks
 → higher tendency for forming ordered microstructures
- Order-disorder transition : at xN≈10.5
- The morphologies (sphere, cylinder, lamellar) depend on the composition f



Fig 8.1 χN vs f phase diagram for symmetric AB block copolymer, predicted from <u>mean-field</u> <u>theory</u>. L (lamellar), H (hexagonal-packed cylinders), Q_{la3d} (gyroid), Q_{lm3m} (bcc spheres), CPS (close-packed FCC spheres), and DIS (disorderd). Ref. M. W. Matsen and F. S. Bates, Macromolecules, 29 (1996) 478



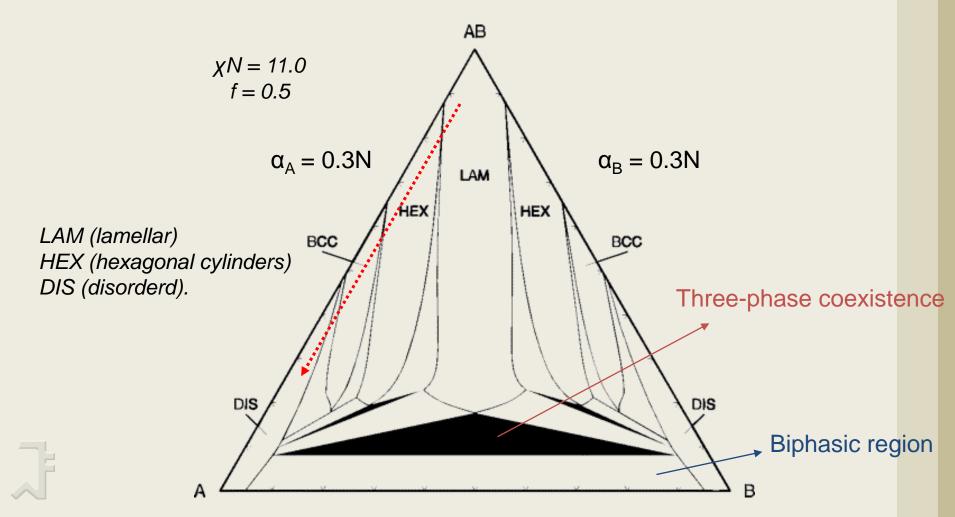
(2) Two- and three-component block copolymer system

- The interfacial curvature in solvent-free block copolymers \rightarrow by the composition f
 - \rightarrow At given block composition can result in only one type of microstructure
 - \rightarrow For different structures, different composition are required (synthesis of other polymers)
- Multicomponent system: homopolymer or copolymer added to a block copolymer

 -> can result into a wide variety of morphologies
- For example, addition of an A-homopolymer to an AB-block copolymer
 - \rightarrow modifying the ratio of A and B: increase of A \rightarrow shift the phase stability
 - \rightarrow predicting by self-consistent field theory (Appendix 8-4)
- Fig 8.2 : isothermal (constant χN) phase diagram
 - : AB-block copolymer + A-homopolymer + B-homopolymer
 - \rightarrow Morphologies change: lamellar \rightarrow cylinder \rightarrow sphere (red dotted line)
- → Reducing the need to synthesize a new block copolymer for a specific morphology



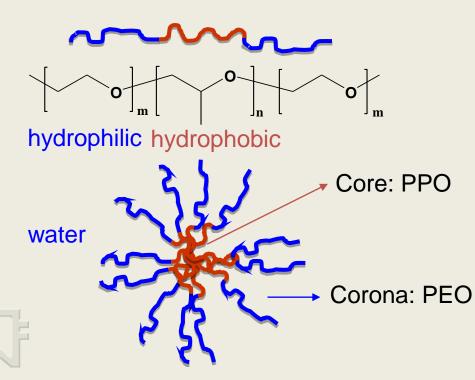
Fig 8.2 Phase diagram for a symmetric AB-diblock copolymer (N) with an A-homopolymer (0.3N) and a B-homopolymer (0.3N), predicted from mean-field theory. Three-phase coexistence; shaded and biphasic region; unlabeled Ref. P. K. Janert and M. Schick, Macromolecules, 30 (1997) 137



8.2.2 Amphiphilic Block Copolymers in Mixtures with Water and Oil

- : The main features of the self-assembly of block copolymers in binary and ternary systems with solvents and effects on self-assembly of the block copolymer MW and composition
- (1) Binary amphiphilic block copolymer water system

PEO-PPO-PEO

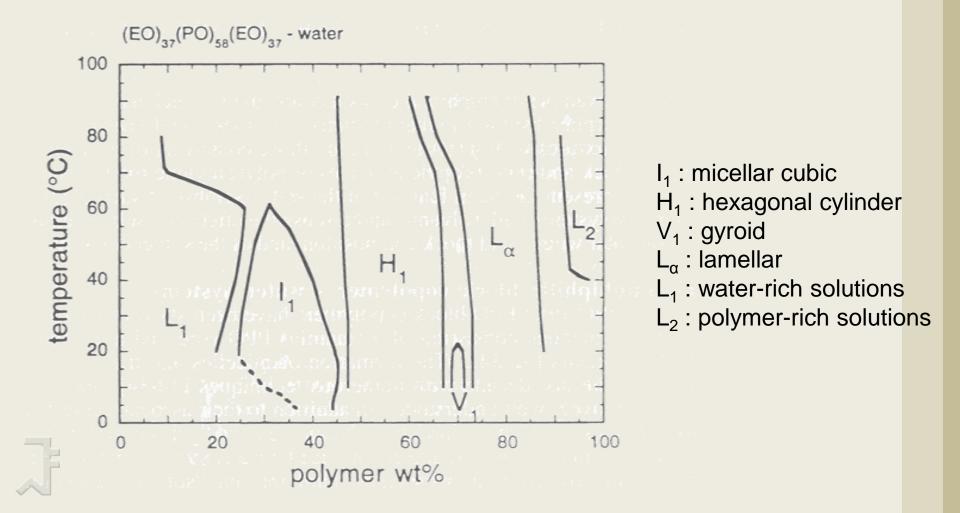


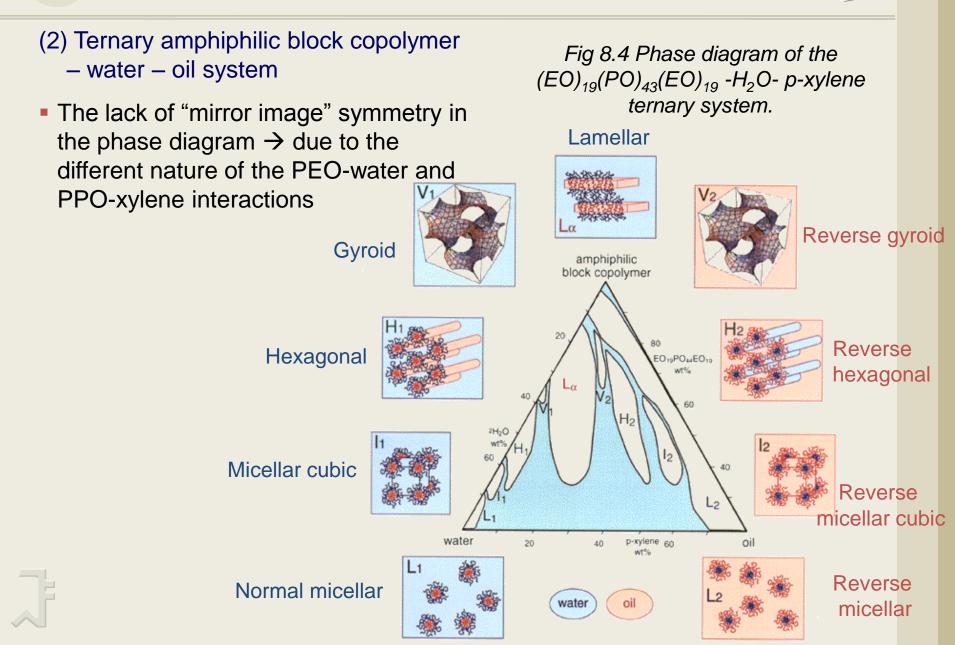
- The temperature effects on the phase behavior of copolymers → understood by invoking the <u>reverse solubility</u> (temperature↑ → solubility↓)
- The temp.↑
 → PEO-water and PPO-water interaction parameter χ ↑
 → PEO-PPO interaction parameter χ↓
 → hydrophobicity of the polymer ↑
 - ightarrow solubility \downarrow

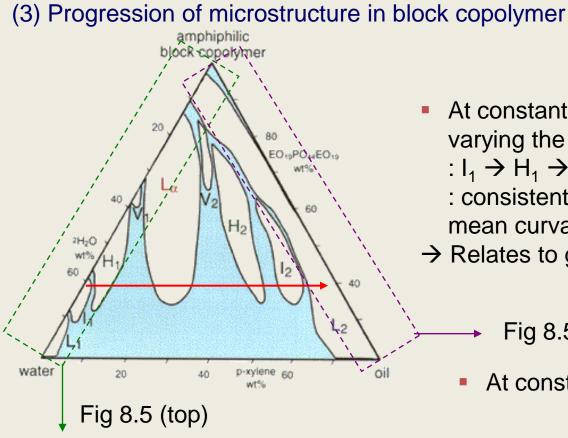
 At Fig 8.3, Lα regions shift to lower polymer concentration as the temp ↑



Fig 8.3 The concentration-temperature phase diagram of the EO₃₇PO₅₈EO₃₇ (Pluronic P105) block copolymer-H₂O binary system. Ref. P. Alexandridis, D. Zhou, A. Khan, Langmuir, 12 (1996) 2690







At constant copolymer/oil ratio

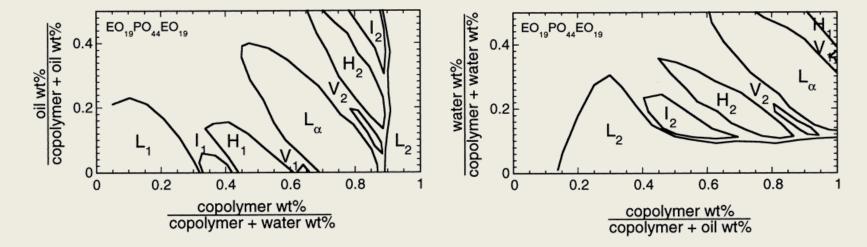
 $: L_1 \rightarrow I_1 \rightarrow H_1 \rightarrow V_1 \rightarrow L_\alpha \rightarrow L_2$

At constant copolymer content (40 wt%), varying the water/oil ratio → phase changes : I₁ → H₁ → L_α → H₂ → I₂ → L₂ (→→) : consistent with a decreasing interfacial mean curvature (H); H=0 in lamellar phase → Relates to geometric factor (Appendix 8-1)

Fig 8.5 (bottom)

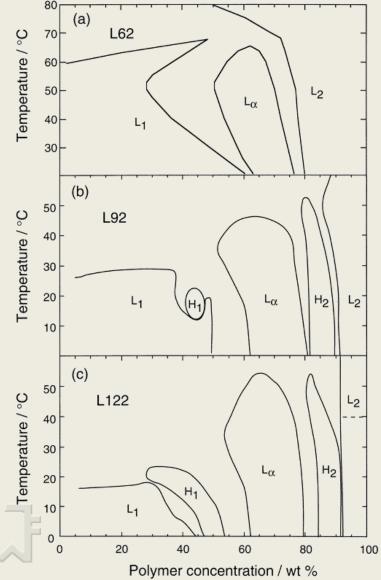
• At constant copolymer/water ratio

Fig 8.5 (top) Boundaries of the various phases along the copolymer-water side of the ternary phase diagram (x-axis: copolymer weight fraction; y-axis: oil weight fraction).
(bottom) Boundaries of the various phases along the copolymer-oil side of the ternary phase diagram (x-axis: copolymer weight fraction; y-axis: water weight fraction).



- For a PEO/PPO block copolymer of a given block composition and molecular weight, the types of structures appear to be a function of the volume fraction of the polar (water-like) / apolar (oil-like) components.
- The microstructure in such systems is not tied up to a specific block copolymer molecular weight and block composition, which define a point in the χN vs f phase diagram

(4) Effect of block copolymer molecular weight



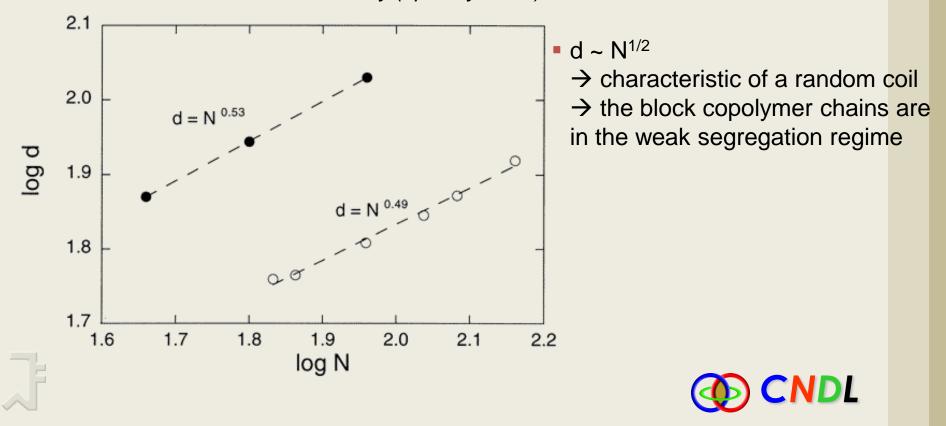
- An increase in copolymer MW → increase the block segregation and the tendency for organization
- Fig 8.6 confirms that a certain minimum MW is required for PEO and PPO blocks to segregate (a) only one mesophase L_α

(b), (c) three mesophase and the increase of the range of the hexagonal regions when L92 \rightarrow L122

Fig 8.6 Concentration-temperature phase diagrams for the (a) Pluronic L62/water, (b) Pluronic L92/water, and (c) Pluronic L122/water systems. L62, L92, and L122 have the same EO/PO ratio but different molecular weight (increasing in the order L62<L92<L122).

 An important result from the study of MW affects concerns the dependence of the lamellar characteristic spacing on the polymer MW

Fig 8.8 Logarithm of the lamellar domain spacing d, vs the logarithm of number of polymer monomers N, for aqueous solution of Pluronic L62, L92, and L122 at 75 wt% (filled symbols) and calculated values from mean field theory (open symbols)



8.2. A Closer Look at Microstructure

(5) Effect of block copolymer composition

 The block composition is the main determinant of the microstructure observed in solvent-free block copolymers, and the chemical composition of typical surfactants affects their hydrophile/lipophile ratio and self-assembly properties.

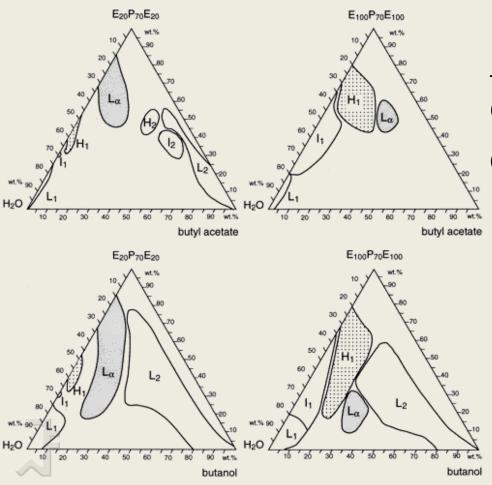
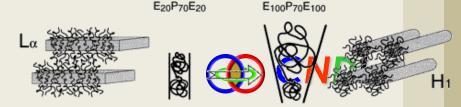


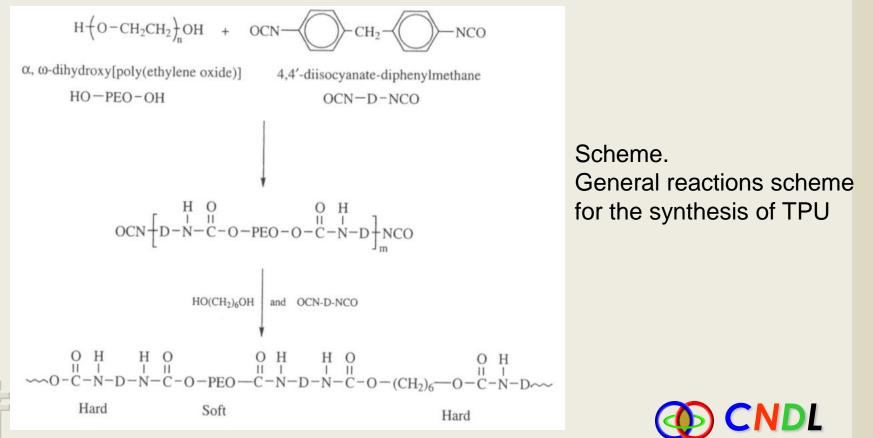
Fig 8.8 Phase diagrams of the copolymer-oilwater ternary systems.

The change in phase behavior when
(i) the size of the copolymer E block changes from E₂₀ to E₁₀₀
(ii) the "oil" changes from butyl acetate to butanol (top to bottom). The schematic represents the change of the self-organization of the amphiphilic block copolymers from the lamellar (L) to the normal hexagonal (H₁) structures upon an increase of the E block size and the corresponding increase in the interfacial curvature. The shaded areas represent the apolar domains.



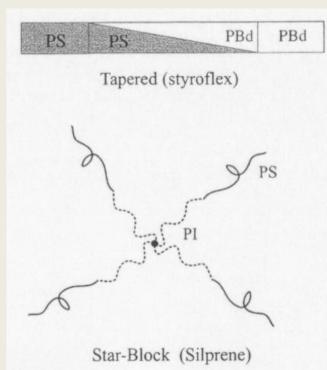
8.3.1 Commercialized applications

- TPU
 - \rightarrow linear multiblock copolymer
 - \rightarrow the first commercially available TPEs which based on polyurethanes
 - \rightarrow automotive bumpers, snowmobile treads, etc.



- Kraton shell introduced the styrenic TPEs
- \rightarrow linear triblock copolymer, made by anionic polymerization
- \rightarrow well-defined materials with low molecular weight and compositional heterogeneity

 \rightarrow include footwear, bitumen modification, thermoplastic blending, adhesive, and cable insulation and gaskets.



Scheme. Tapered and star block copolymers



• TPEs based on polyesters, TEPS (du Pont), and polyamides, TPA (Huls and Ato Chimie)

→Linear or polyamide multiblock copolymers

 \rightarrow Linked together by ester or amide linkage instead of urethane

 \rightarrow hose tubing, sport goods, automotive components, etc.

•All TPEs exhibit properties characteristic of chemically crosslinked <u>elastomers</u> at room temperature

 \rightarrow At elevated temperature, behave as <u>thermoplastics</u>.

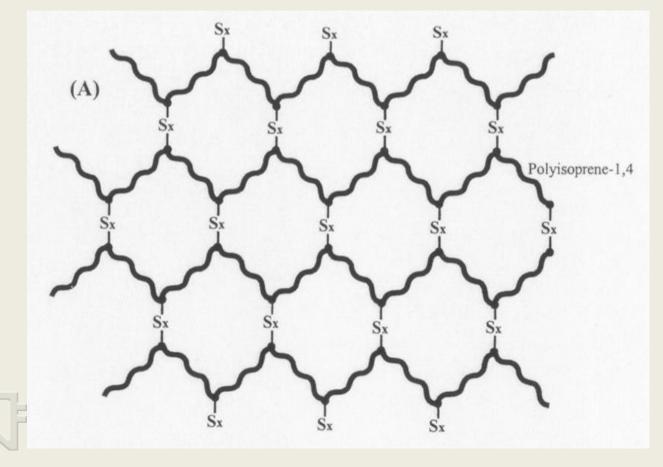
•They can be processed with high speed, efficiency and economy on conventional thermoplastic equipment.

•TPEs were considered to be one of the breakthroughs in rubber technology.

On the molecular level,
 Thermoplasticity – consequence of noncrosslinked chains
 Thermoelasticity – consequence of crosslinked chains

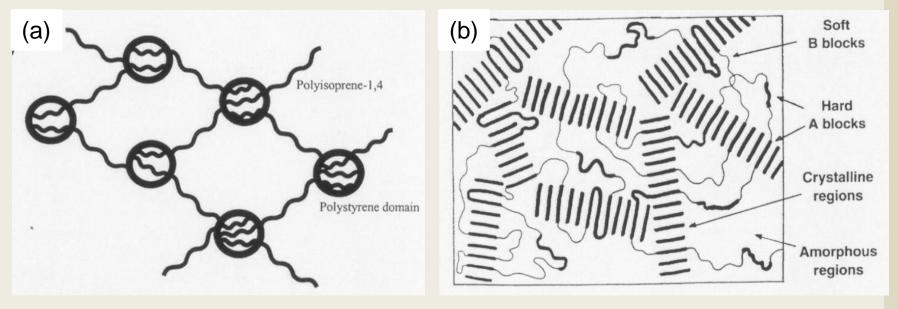


- In TPEs the crosslinks are physical rather than chemical, which is the case of vulcanized rubber.
- Thus, the hard domains lose their strength when the material is heated (thermoreversible) or dissolved in a solvent (solvoreversible).





• Thermoplastic elasticity is illustrated schematically for (a) styrenic TPEs and (b) multiblock TPEs.



• Crystallinity (TPES, TPA), hydrogen bonding (TPU, TPA), and van der Waals interactions (styrenic) all have been shown to cause microphase separation in these systems.



• Triblock copolymers with polydienes end blocks and diblock copolymers are not TPEs because the flexible chains are not immobilized at both ends by the glassy hard domains of PS and, consequently, cannot be elastic.

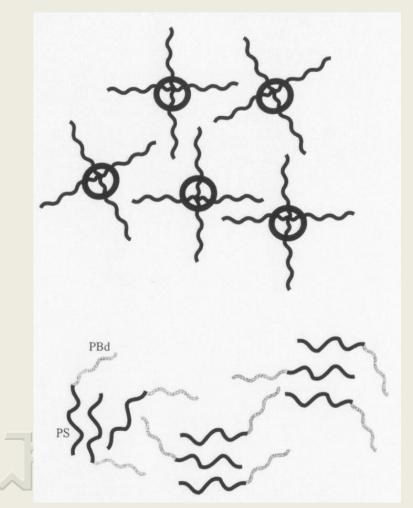


Figure 8.9 Styrenic triblock copolymer with polydiene external block or diblock copolymers (styrene or PU) are not elastic.



8.3.2 Potential Applications

(1) Drug release in target cells

 A block copolymer for use in drug delivery as a microcontainer device must consist of a water-soluble block(hydrophilic), in order to impart blood solubility of the microcontainer, and a water-insoluble block (hydrophobic) compatible with the drug to be carried.

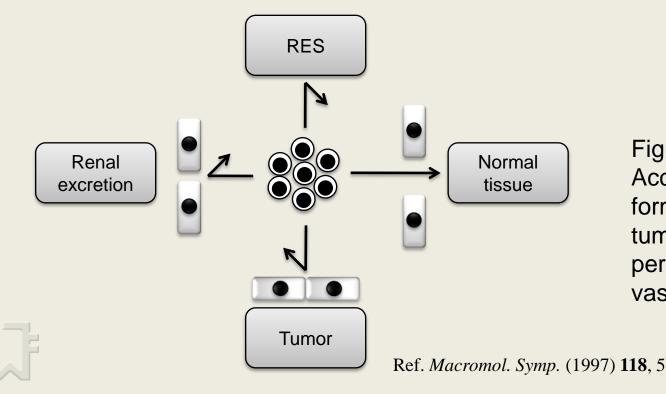
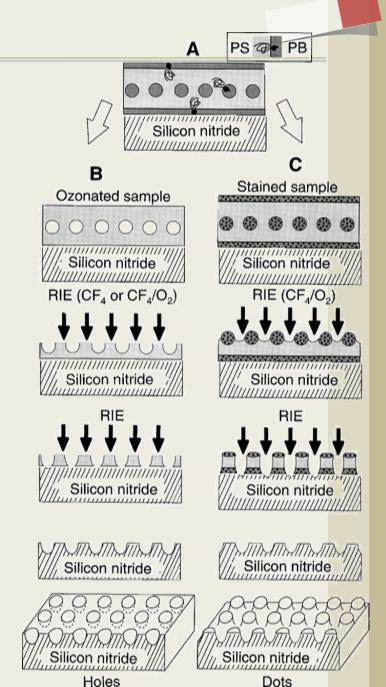


Figure 8.10 Accumulation of micelleforming microcapsules in a tumor utilizing enhanced permeability of tumor vasculature

(2) Nanopatterning

- Sizes less than 30nm are not easily obtained standard lithography (photolithography, electron beam lithography).
- One way to overcome this problem is by using block copolymer

Fig. 8.11 (A) Schematic cross-sectional view of a nanolithography template consisting of a uniform monolayer of PB spherical microdomains on silicon nitride. PB wets the air and substrate interfaces. (B) Schematic of the processing flow when an ozonated copolymer film is used, which produces holes in silicon nitride. (C) Schematic of the processing flow when an osmium-stained copolymer film is used, which produces dots in silicon nitride. Ref. Science (1997) 276, 1401



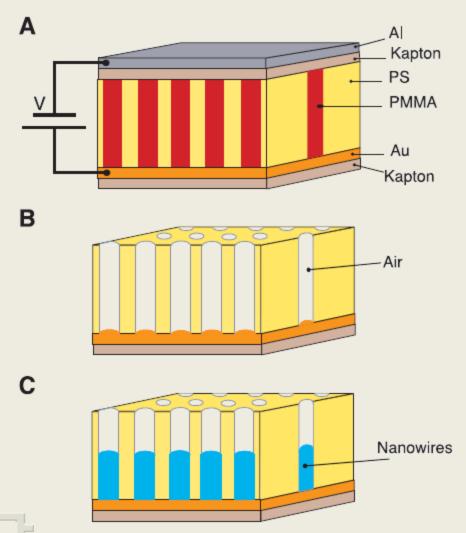


Fig. 8.12. A schematic representation of high density nanowire fabrication in a polymer matrix.

(A) An asymmetric diblock copolymer annealed above the glass transition temperature of the copolymer between two electrodes under an applied electric field, forming a hexagonal array of cylinders oriented normal to the film surface. (B) After removal of the minor component, a nanoporous film is formed. (C) By electrode position, nanowires can be grown in the porous template, forming an array of nanowires in a polymer matrix.

Ref. Science (2000) 290, 2126.



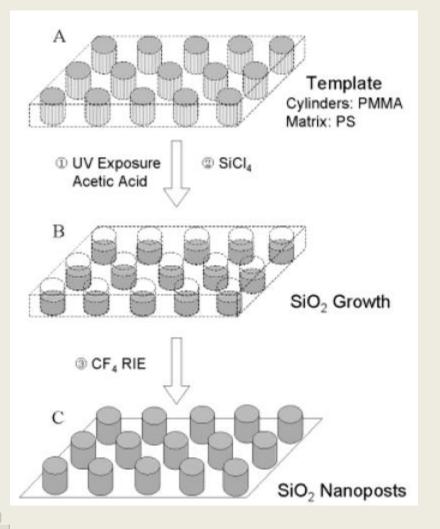


Fig. 8.13 Schematic diagram of the steps required to generate SiO₂ nanoposts.

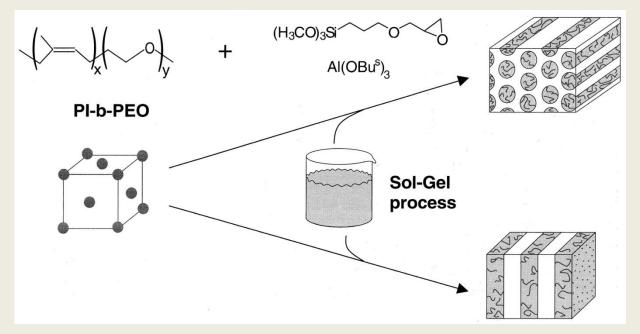
A) Block copolymer films having cylindrical microdomains oriented normal to the surface. B) Growth of SiO₂ within the nanopores generated by selective elimination of PMMA cylinders. C) Array of SiO₂ nanoposts after removing PS matrix with CF_4 RIE.



Ref. Adv. Mater. (2001) 13, 795.

(3) Organic-inorganic hybrid mesostructures

 By using an amphiphilic block copolymer as a structure-directing agent, organicinorganic hybrid materials with nanoscale structures can be prepared.



Figu 8.14 Schematic drawing of our approach for synthesizing organically modified silica mesostructures. Ref. Science (1997) 278, 1795.





•By thermal treatment, single ceramic nanoobjects of different shapes and sizes can be prepared.

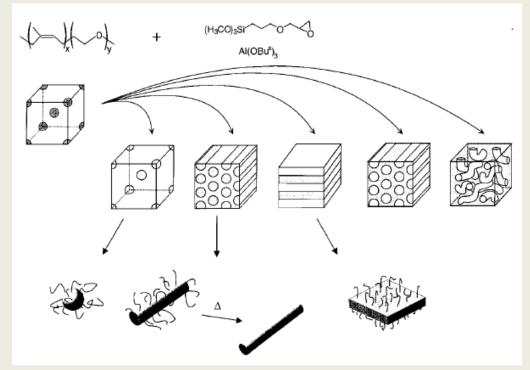


Fig. 8.15 Schematic drawing of our approach for synthesis of single nano-objects with controlled shape, size, and composition. In the upper part all morphologies obtained from PI-b-PEO and different amounts of metal alkoxides are shown. As displayed in the lower part of the figure, the single "hairy" hybrid nano-objects of different shape are isolated by dissolution. Ref. Adv. Mater. (1999) **11**, 141.



Keywords in Chapter 9

- Five regions of viscoelastic behavior
- DSC (Differential Scanning Calorimetry)
- The free volume theory for glass transition
- The Gibbs and DiMarzio theory for glass transition

-Factors Influencing on the glass transition temperature



