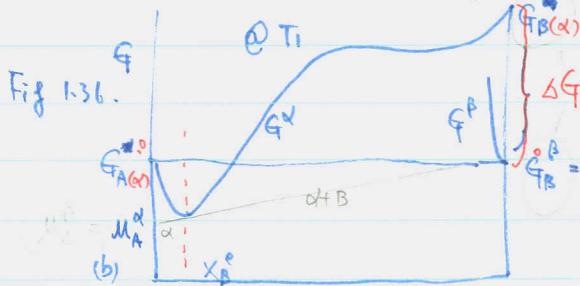


7. The Effect of Temp on Solid Solubility.

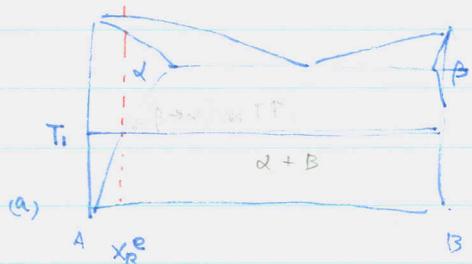


Phase diagram B soluble in A

A insoluble in B.

for x_B^e (max. solubility of B in A)

$$\mu_B^\alpha = \mu_B^\beta \approx G_B^\circ(\beta)$$



$$\text{For a regular soln. } \frac{G_B^\circ(\beta)}{G_B^\circ(\alpha)} \mu_B^\alpha = G_B^\circ(\beta) + RT \ln \alpha_B$$

$$= G_B^\circ(\beta) + RT \ln \alpha_B \rightarrow z_1.$$

$$\left[G_B^\circ - \frac{1}{2}(E_A + E_B) \right] \downarrow \frac{G_B^\circ}{G_B^\circ(\alpha)} \mu_B^\alpha = G_B^\circ(\beta) + RT \ln \alpha_B + RT \ln X_B.$$

$$\Delta H = N_A Z \epsilon$$

$$G_B^\circ(\alpha) = G_B^\circ(\beta)$$

From fig 1.36 b. $\approx G_B^\circ(\alpha) - \mu_B^\alpha = \Delta G_B(\beta \rightarrow \alpha)$ the diff in f.E.
bet'n pure B (in ~~the~~ the stable β -form) and
unstable α -form.

$$\text{for } x_B = x_B^e$$

$$(1.51) \quad \therefore \Delta G_B(\beta \rightarrow \alpha) = -RT \ln x_B^e - \Delta H (1-x_B^e)^2$$

$$(1.52) \quad \text{If } x_B^e \ll 1 \quad (1.51) \rightarrow x_B^e = \exp\left(-\frac{\Delta G_B + \Delta H}{RT}\right)$$

$$\text{Since } \Delta G_B = \Delta H_B - T \Delta S_B$$

$$(1.53) \quad x_B^e = A \exp\left(-\frac{Q}{RT}\right)$$

$$(1.54) \quad \text{where } A = \exp\left(\frac{-\Delta S_B}{R}\right), \quad Q = \Delta H_B + \Delta H \quad \begin{array}{l} \text{NaZ} \\ \text{change in E.} \end{array}$$

$$\Delta S_B = \Delta S_B(\beta \rightarrow \alpha) \neq f(T)$$

~~방법~~. $x_B^e \uparrow$ as $T \uparrow$, rate determined by Q .

$\therefore Q = \text{Heat absorbed}$
When 1 mole of B in β
dissolves in A(α) to make a dilute
soln.



8. Equil^m Vac. Conc.

- Always there are vacancies in the lattice. $\Delta S_{\text{Harrow}} = \frac{\text{small}}{q}$
Vacancy (removal of atoms) cause U (int. E) and $\Delta S_{\text{conf.}}$ to \uparrow .
 $\therefore G$ of the alloy $\stackrel{\text{is}}{=} f(\text{vac. conc.})$ and X_v^e will give G_{\min} .
↓ due to vibrat.
- Assume that calculation of X_v^e is identical to that of ΔG° for A + B when $\Delta H^{\circ} > 0$ (endo)
- Since X_v^e normally small, ignore vac-vac interactions
 $\therefore \Delta H \approx \Delta H_{X_v}$
↪ $\Delta H/\text{mole vac. added.}$
- $\Delta S_{\text{conf}} \gg \Delta S_{\text{thermal by vibration of atoms due to vac. existence}}$
↪ $\Delta S_{\text{conf}} \text{ due to mixing}$

$$\Delta S = X_v \Delta S_v - R [X_v \ln X_v + (1-X_v) \ln (1-X_v)]$$

Thermal energy
↓ Vacant site ΔG

$$(1.55) \quad G = G_A^{\circ} + \Delta G_v = G_A^{\circ} + X_v (\Delta H_v - T \Delta S_v) + RT [X_v \ln X_v + (1-X_v) \ln (1-X_v)]$$

$X_A + X_V$

$$= G_A^{\circ} + (\Delta H - T \Delta S)$$

$$= G_A^{\circ} + \Delta H_v X_v - T \Delta S_v X_v + RT [$$

For X_v^e

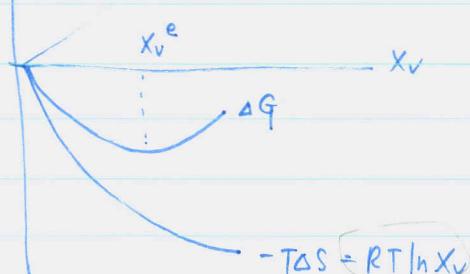
$$\left. \frac{dG}{dX_v} \right|_{X_v=X_v^e} = 0$$

$$\Delta H = \Delta H_v X_v$$

$$\therefore \Delta H_v - T \Delta S_v + RT \ln X_v^e = 0$$

$$(1.56) \quad \therefore X_v^e = \exp \left(\frac{\Delta S_v}{R} \right) \cdot \exp \left(-\frac{\Delta H_v}{RT} \right)$$

$$(1.57) \quad \text{or} \quad X_v^e = \exp \left(-\frac{\Delta G_v}{RT} \right)$$



In practice $\Delta H_v \sim 1 \text{ eV/atom}$

$X_v^e \sim 10^{-4} \sim 10^{-3}$ at T_m of solid.

1.6. The Influence of Interfaces on Equil^m

So far assumed a perfect single Xtal.

In reality, various defects exist. (disl. q.b., interphase interface)

- 1 In the early stages of phase transf.

fine β in α matrix

$$\text{due to the curvature } \Delta P = \frac{2\gamma}{r} \text{ exerted (5!!)} \\ \rightarrow U + PV. \quad \text{on the } \beta.$$

$$G = H - TS$$

$$(dG = -SdT + VdP)$$

$$\therefore \Delta G = \Delta P \cdot V \quad \text{molar vol. of } \beta.$$

$$(1.58) \quad \Delta G_f = \frac{2\gamma V_m}{r} : \text{Gibbs-Thompson effect} \\ \text{or capillary effect.}$$

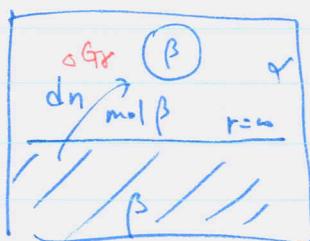
the concept of a $\Delta P \rightarrow$ good for spherical liq. particles

less important convenient in solids. \rightarrow usually nonspherical

- 2 two β particles : [one w/ radius r

in α matrix. [one w/ " " (flat). in α .]

if the molal f.e. difference bet'n two particles is ΔG_f



the transfer of d_n mole of β from β_{ss} to β_r .

$$dG = \Delta G_f d_n.$$

if the surface Area of β_{ss} unchanged.

then dG is due to the increase in the interfacial area of the β_r (dA).

$$dG = \gamma dA.$$

$$(1.59) \quad \therefore \Delta G_f = \gamma \frac{dA}{dn}$$

$$n = \frac{4\pi r^3}{V_m}, A = 4\pi r^2 \quad \therefore \frac{\partial A}{\partial n} = \frac{8\pi r V_m}{4\pi r^2} = \frac{2V_m}{r}$$

$$(1.58) \quad \therefore \Delta G_f = \frac{2\gamma V_m}{r} \quad \text{Gibbs-Thompson Effect.}$$

- G-T Effect \rightarrow see AT solubility of β in α \rightarrow sensitive to the size of β
 $x_{\infty} \rightarrow x_r$. Assuming that α is a regular sol'n
 in ~~the~~ Figure β is almost pure B, $x_{\beta} \sim 1$.

$$(1.52) \rightarrow x_B^e = \exp \left[- \frac{\Delta G_B + \gamma_2}{RT} \right] = x_{\infty}$$

Similarly $x_r = \exp \left[- \frac{(\Delta G_B - \frac{2\gamma V_m}{r}) + \gamma_2}{RT} \right]$

$$(1.60) \quad \therefore x_r = x_{\infty} \exp \left(\frac{2\gamma V_m}{RT r} \right) \quad \text{if this is small.}$$

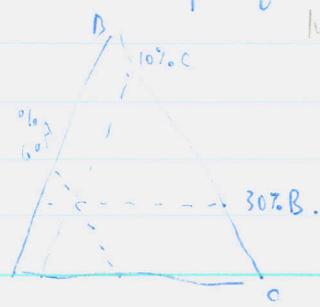
$$(1.61) \quad x_r \approx x_{\infty} \left(1 + \frac{2\gamma V_m}{RT r} \right)$$

- ex. $\gamma = 200 \text{ mJ/m}^2, V_m = 10^{-5} \text{ m}^3, R = 8.3 \text{ J/mol}^\circ\text{K}, T = 500 \text{ K}$.

$$\frac{x_r}{x_{\infty}} \approx 1 + \frac{1}{r(\text{nm})} \quad \leftarrow \begin{array}{l} \text{prov} \\ \text{effective for } r=1-100 \text{ nm.} \end{array}$$

for $r=10 \text{ nm}$ $\frac{x_r}{x_{\infty}} = 1.1$.

7. Ternary Equil. m.



A.B.C axes by
 Intercept on a tangential plane to the f.e. surface as in Fig
 → gives the chem. pot. of A, B, C

Fig 1.41 → simple Eutectic.

Fig. 1.40 The Gibbs Triangle.

- Construction of phase diagram at T from G-curve curve.

1. Points \times by common tangential plane (\therefore equal chem. pot.) connected

2. Mark these points on an isothermal section of the diagram.

3. get the series of tie line by rolling the tangential plane.

- Relative amount (solid & liquid) determined by a lever rule.

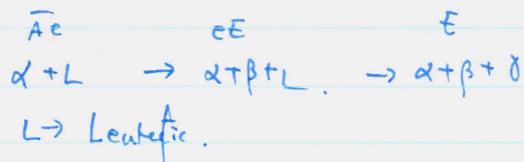
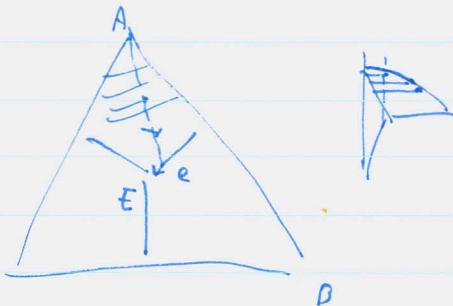
- At lower temp.

tie line bundles.

tie-triangles

ternary eutectic : $L + \alpha + \beta + \gamma$

- The course of solidification. assyng equl^m maintained.

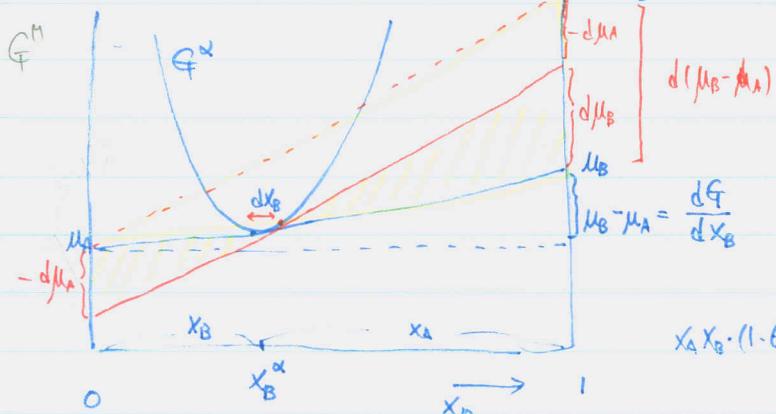


1.8. Add'l Thermo Relationships for Binary Soln.

Interested in the calculation of $d\mu$ (Δ in chem. pot.). due to dx .

$$G = H - TS \quad \text{or} \quad \frac{\partial G}{\partial T} = \mu_A + x_B \frac{\partial \mu_B}{\partial x}$$

$$\text{Fig. 1.46.} \quad (1.63) \quad -\frac{d\mu_A}{dx} = \frac{d\mu_B}{x_A} = \frac{d(\mu_B - \mu_A)}{x_A} \quad \text{in } \frac{d(\mu_B - \mu_A)}{dx}$$



The slope of G-comp. curve

$$\frac{dG}{dx_B} = \frac{\mu_B - \mu_A}{1} \quad (1.64)$$

$$x_A x_B \cdot (1.63) \leftarrow (1.64)$$

$$= x_A x_B d(\mu_B - \mu_A)$$

$$(1.65) \quad -x_A d\mu_A = d\left(\frac{dG}{dx_B}\right)$$

$$-x_A d\mu_A = x_B d\mu_B = x_A x_B \frac{d^2G}{dx_B^2} \cdot dx_B$$

G-D eq.

$$= \frac{d^2G}{dx_B^2} \cdot dx_B$$

$$\therefore \frac{d^2G}{dx_B^2} = \frac{d^2G}{dx_A^2} \rightarrow (1.68) \quad \checkmark$$

For a regular soln.

$$= x_B (1 - x_B)$$

$$(1.39) \quad G = x_A f_A^\circ + x_B f_B^\circ + \Delta x_A x_B + RT (x_A \ln x_A + x_B \ln x_B)$$

$$(1.66) \quad \frac{d^2G}{dx^2} = \frac{RT}{x_A x_B} - 2\Delta \quad \frac{df}{dx_B} = -f_A^\circ + f_B^\circ + \Delta (1 - 2x_B)$$

$$\therefore x_A x_B \frac{d^2G}{dx^2} = RT \left[1 - \frac{2\Delta x_A x_B}{RT} \right] = RT \left[1 - \frac{2\Delta H_M}{RT} \right] + RT (-\ln(1-x_B) + 1 \ln x_B + 1)$$

for ideal soln. $\Delta = 0$.

$$= (1.71)$$

$$(1.67) \quad \frac{d^2G}{dx^2} = \frac{RT}{x_A x_B}$$

$$\frac{d^2G}{dx_B^2} = -2\Delta + RT \left(\frac{1}{1-x_B} + \frac{1}{x_B} \right)$$

$$= -2\Delta + RT \left(\frac{x_A + x_B}{x_A x_B} \right)$$

$$= -2\Delta + \frac{RT}{x_A x_B}$$

$$(1.68) \quad (1.4) + (1.43) \Rightarrow (1) \quad \mu_B = f_B^\circ + RT \ln x_B$$

Differentiate

$$(1.67) \quad d\mu_B = RT [\ln \gamma_B X_B]' = RT [\ln \gamma_B + \ln X_B]'$$

(22)

$$\textcircled{2} \quad = RT \left[\frac{d\gamma_B}{\gamma_B} + \frac{dX_B}{X_B} \right]$$

$$\therefore \frac{d\mu_B}{dX_B} = \frac{RT}{X_B} \left[\frac{X_B}{\gamma_B} \frac{d\gamma_B}{dX_B} + 1 \right]$$

$$(1.65) \rightarrow X_B d\mu_B = X_A X_B \frac{d^2G}{dX^2} dX_B = -X_A d\mu_A$$

$$+ (1.68) \rightarrow \frac{d\mu_B}{dX_B} = \frac{RT}{X_B} \left[1 + \frac{X_B}{\gamma_B} \frac{d\ln \gamma_B}{dX_B} \right] = \frac{RT}{X_B} \left[1 + \frac{d\ln \gamma_B}{d\ln X_B} \right]$$

$$(1.70) \quad (1.65) \quad -X_A d\mu_A = X_B d\mu_B = RT \left[1 + \frac{d\ln \gamma_B}{d\ln X_B} \right] dX_B$$

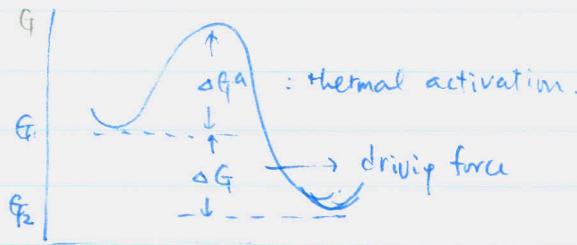
$$(1.71) \quad \therefore X_A X_B \frac{d^2G}{dX^2} = RT \left[1 + \frac{d\ln \gamma_A}{d\ln X_A} \right] = RT \left[1 + \frac{d\ln \gamma_B}{d\ln X_B} \right]$$

For a regular sol.

for an ideal sol

9. The kinetic of Phase Transf.

- the Thermo func. : be used to calculate the driving force for a transf.
how fast the transf will occur \rightarrow kinetics.



the probability of an atom reaching
the activated state
 $= \exp(-\Delta G^\ddagger/kT)$

the rate \propto # of atoms which reached
the activate state.

\therefore the rate $\propto \exp(-\frac{\Delta G^\ddagger}{kT})$ $\Delta G^\ddagger = \Delta H^\ddagger - T\Delta S^\ddagger$.

$$= \underbrace{\exp\left(\frac{\Delta S^\ddagger}{R}\right)}_{\text{const}} \exp\left(-\frac{\Delta H^\ddagger}{kT}\right) \quad \text{Arrhenius rate eq.}$$