

재료상변태

Phase Transformation of Materials

2008.09.30.

박은수

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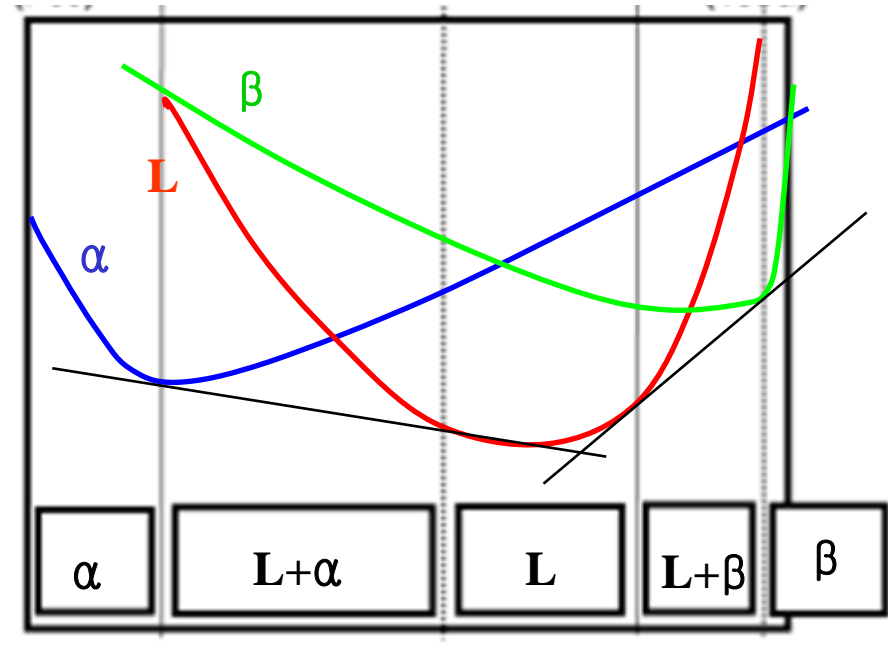
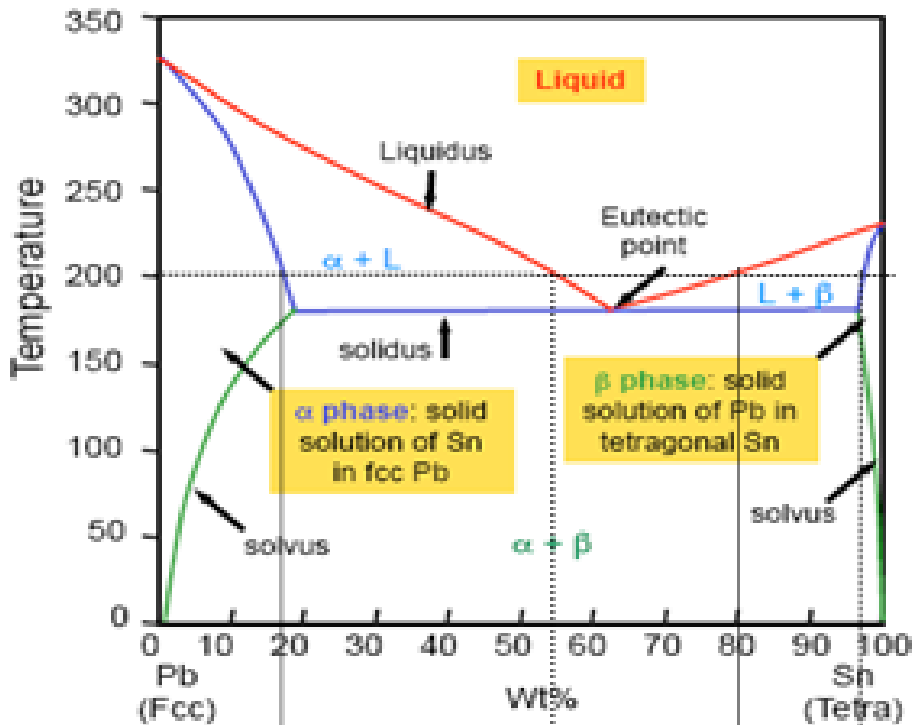
Contents for previous class

- Ternary Equilibrium: Ternary Phase Diagram

< Chapter 2 >

- Diffusion
- Interstitial Diffusion – Fick's First Law
- Effect of Temperature on Diffusivity

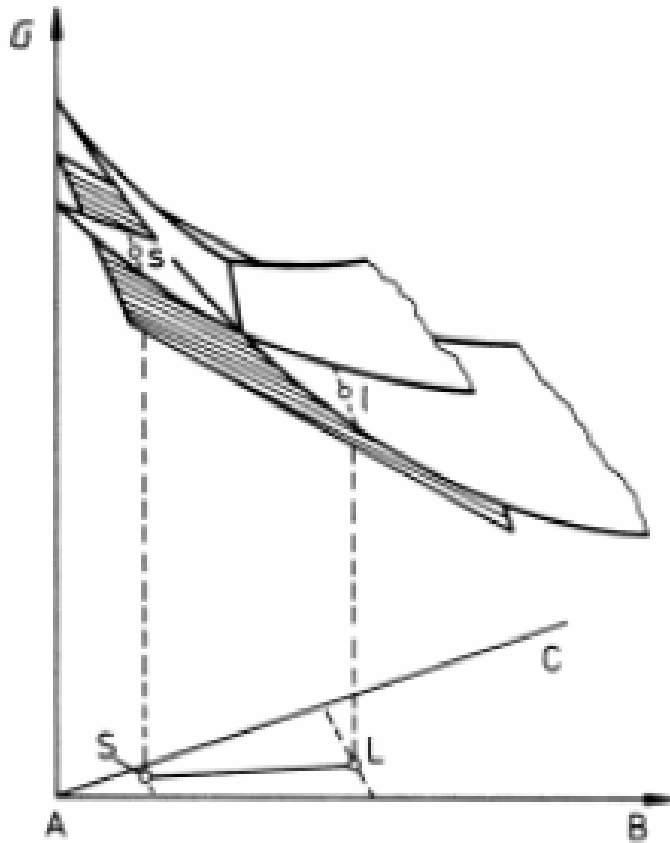
Tie line: binary system



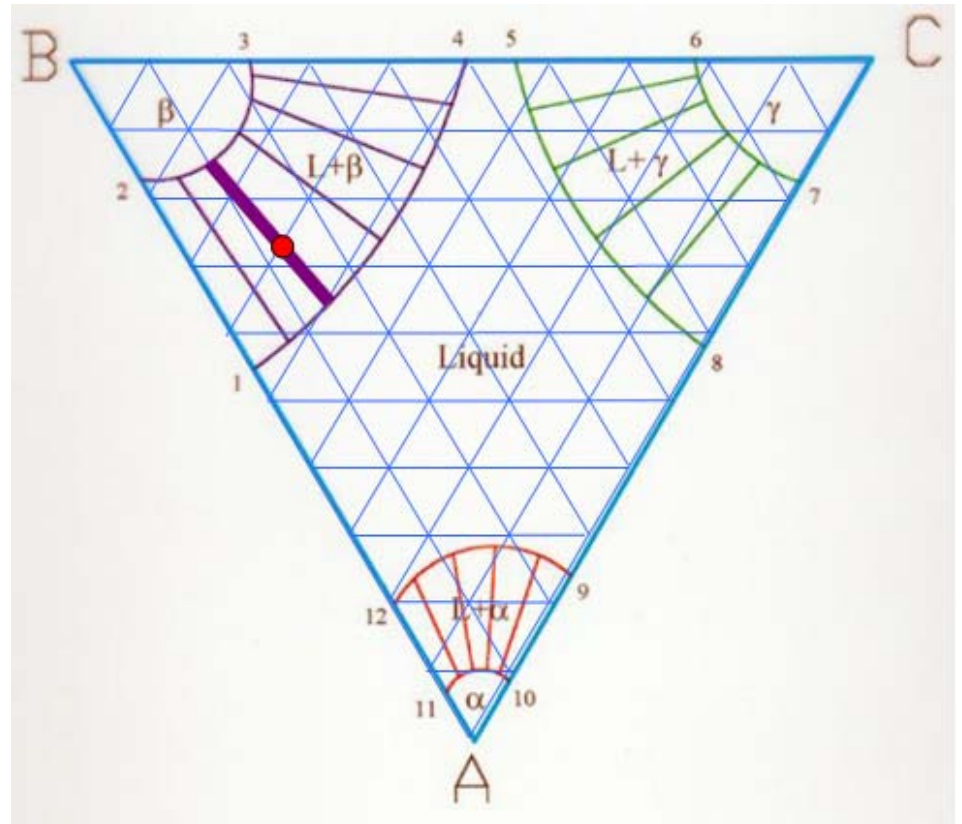
* Simple Eutectic Systems

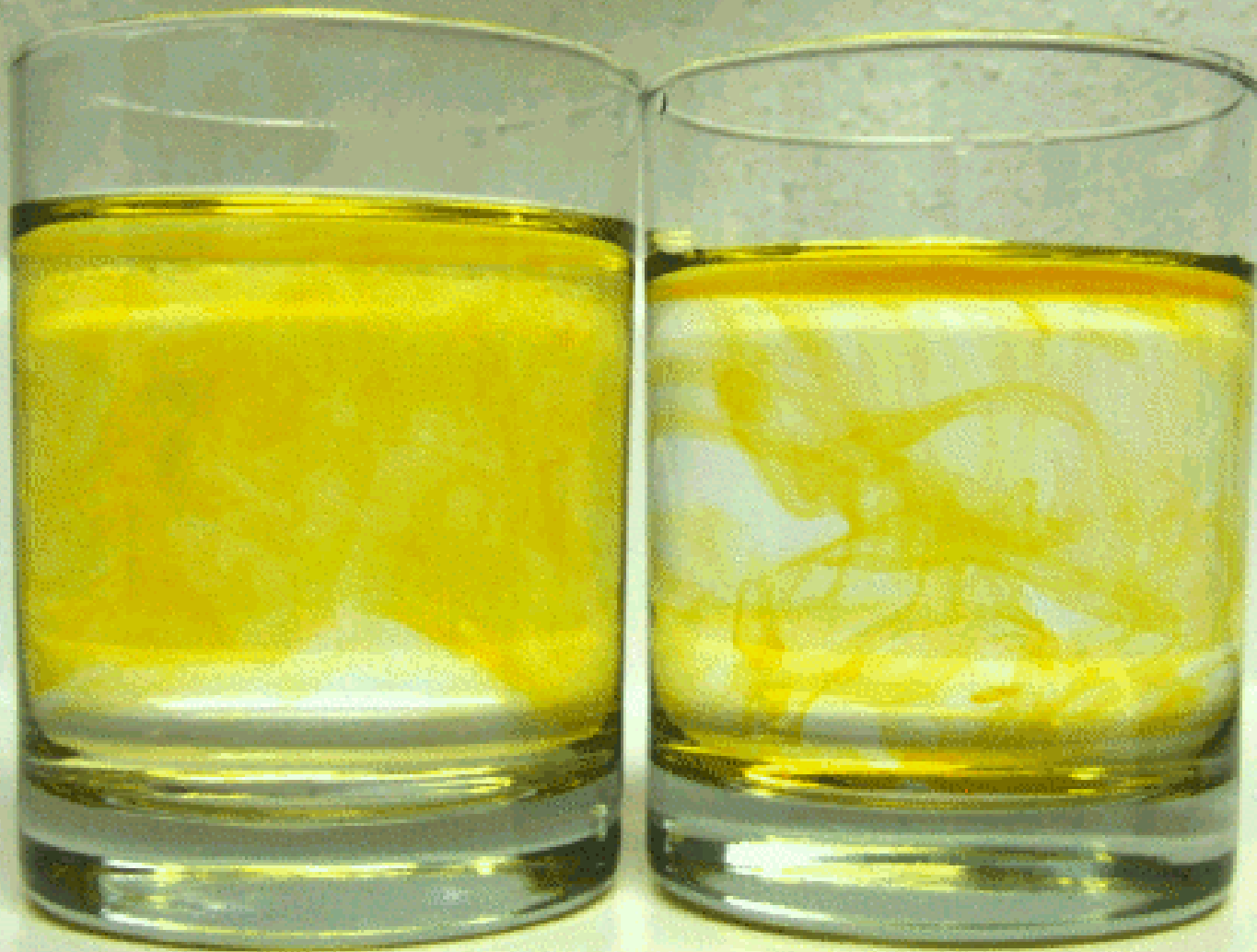
$$\Delta H_{mix}^L = 0 \quad \Delta H_{mix}^S \gg 0$$

Tie line: ternary system



(b)





Diffusion

Diffusion

Diffusion : Mechanism by which matter transported through matter

What is the driving force for diffusion?

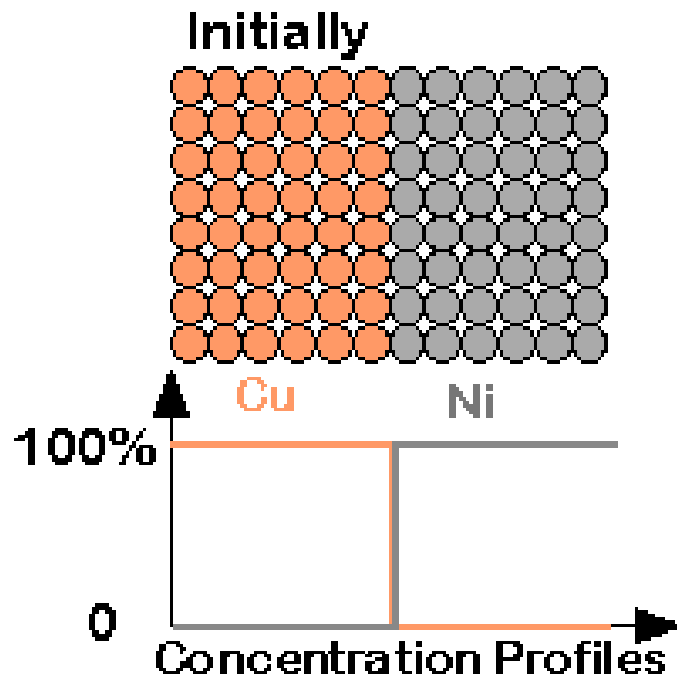
⇒ a concentration gradient (x)

⇒ a chemical potential (o)

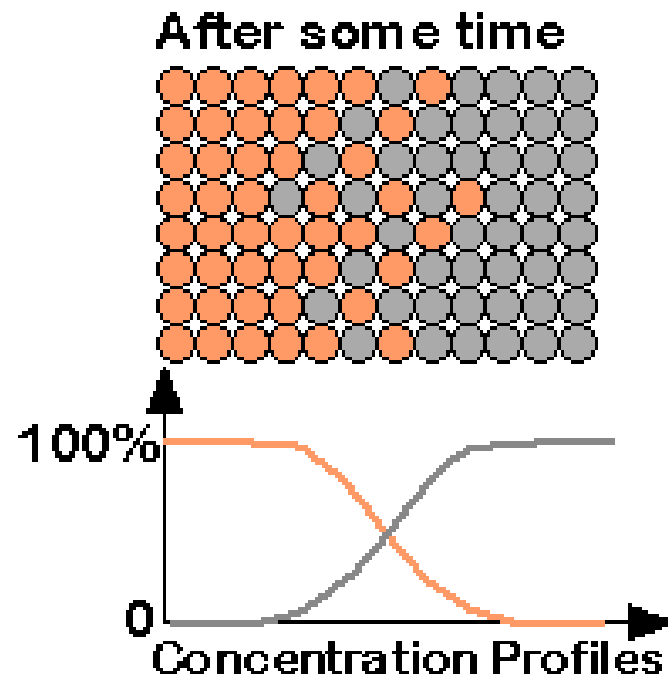
But this chapter will explain with concentration gradients for a convenience.

Diffusion: THE PHENOMENON

- **Interdiffusion:** in a solid with more than one type of element (an alloy), atoms tend to migrate from regions of large concentration.



Substitutional diffusion



Interstitial diffusion

Interstitial diffusion

Random jump of solute B atoms in a dilute solid solution of a simple cubic lattice

Assume that there is no lattice distortion and also that there are always six vacant sites around the diffusion atom.

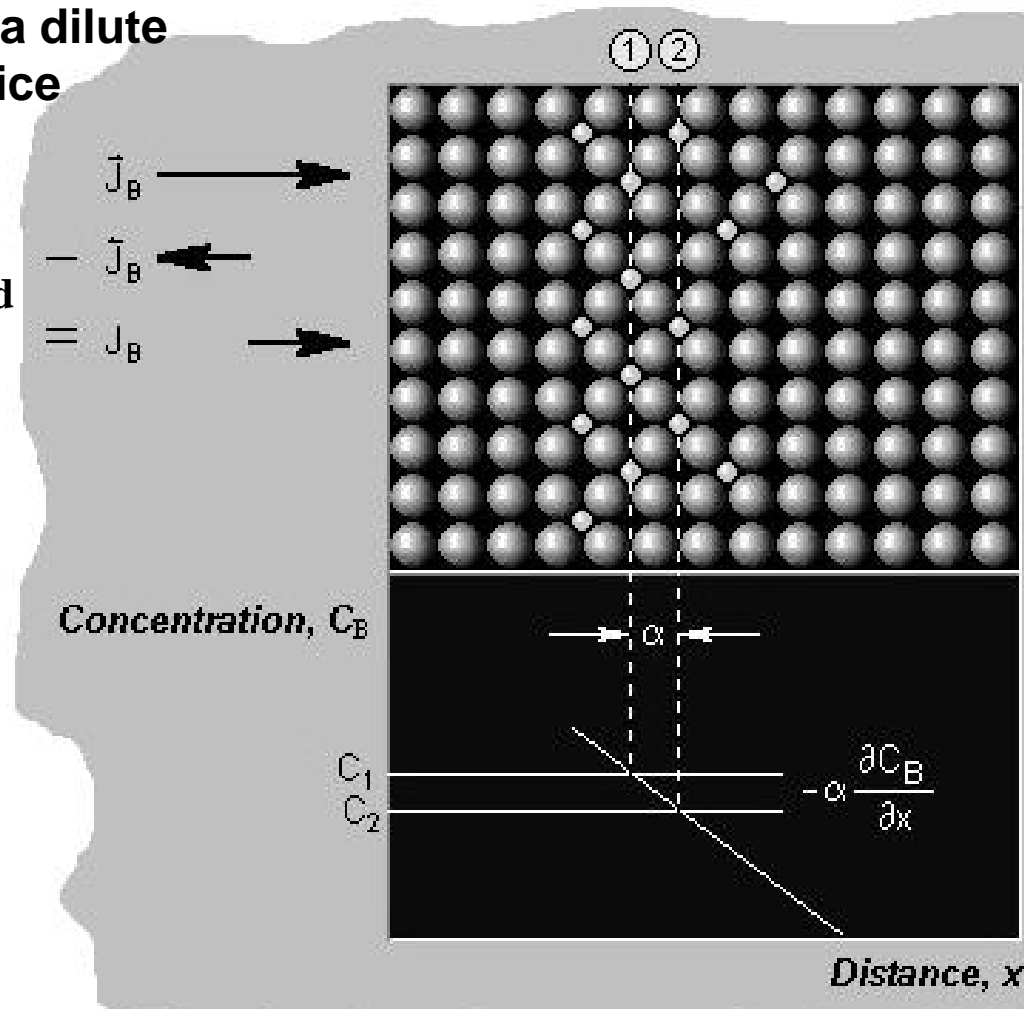
$$J_B = \frac{1}{6} \Gamma_B n_1 - \frac{1}{6} \Gamma_B n_2$$

J_B : Flux of B atom

Γ_B : Average jump rate of B atoms

n_1 : # of atoms per unit area of plane 1

n_2 : # of atoms per unit area of plane 2

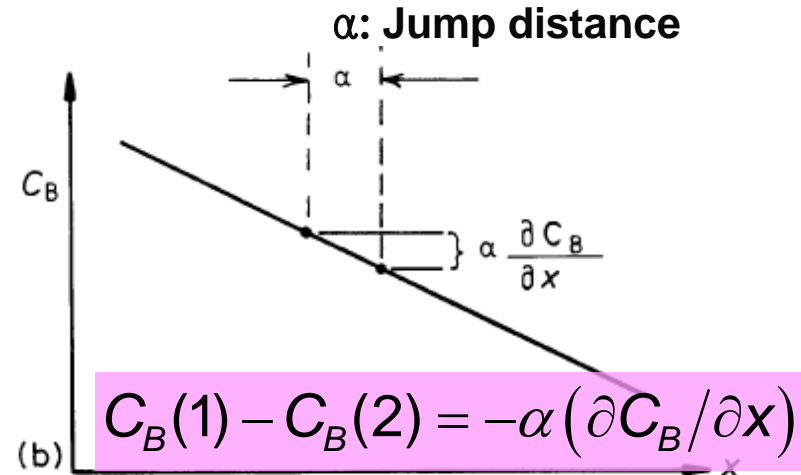


Fick's First Law of Diffusion

$$J_B = \frac{1}{6} \Gamma_B n_1 - \frac{1}{6} \Gamma_B n_2 = \frac{1}{6} \Gamma_B (n_1 - n_2)$$

$$C_B(1) = n_1/\alpha, \quad C_B(2) = n_2/\alpha$$

$$(n_1 - n_2) = \alpha (C_B(1) - C_B(2))$$



$$J_B = \frac{1}{6} \Gamma_B (n_1 - n_2) = - \left(\frac{1}{6} \Gamma_B \alpha^2 \right) \frac{\partial C_B}{\partial x} = -D_B \frac{\partial C_B}{\partial x}$$

(atoms $m^{-2} s^{-1}$)

➔ Concentration varies with position.

**D_B : Intrinsic diffusivity or
Diffusion coefficient of B
⇒ depends on
microstructure of materials**

Magnitude of D in various media

Gas : $D \approx 10^{-1} \text{ cm}^2/\text{s}$

Liquid : $D \approx 10^{-4} \sim 10^{-5} \text{ cm}^2/\text{s}$

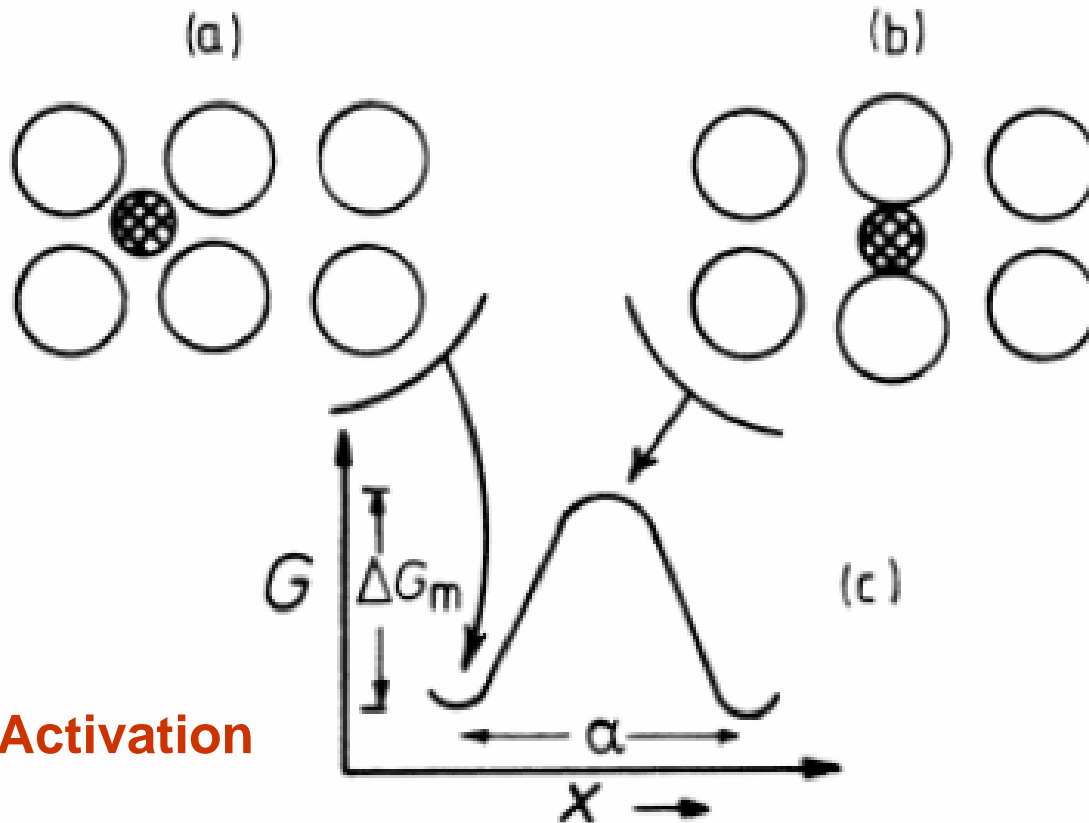
Solid : Materials near melting temp. $D \approx 10^{-8} \text{ cm}^2/\text{s}$

Elemental semiconductor (Si, Ge) $D \approx 10^{-12} \text{ cm}^2/\text{s}$

EFFECT OF TEMPERATURE on Diffusivity

Temperature \rightarrow Γ \rightarrow D, Diffusivity

(Interstitial atom jumps times per second)



Thermal Activation

Thermally activated process

jump frequency Γ_B ?

$$\Gamma_B = Z \nu \exp(-\Delta G_m / RT)$$

Z : nearest neighbor sites

ν : vibration frequency

ΔG_m : activation energy for moving

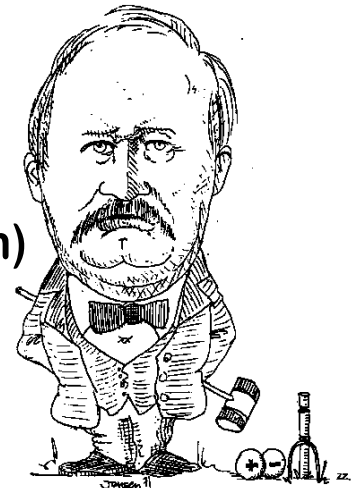
$$\left\{ \Delta G_m = \Delta H_m - T \Delta S_m, D_B = \frac{1}{6} \Gamma_B \alpha^2 \right\}$$

$$D_B = \left[\frac{1}{6} \alpha^2 Z \nu \exp(\Delta S_m / R) \right] \exp(-\Delta H_m / RT)$$

$$\Delta H_m \equiv Q_{ID}$$

$$D_B = D_{B0} \exp \frac{-Q_{ID}}{RT}$$

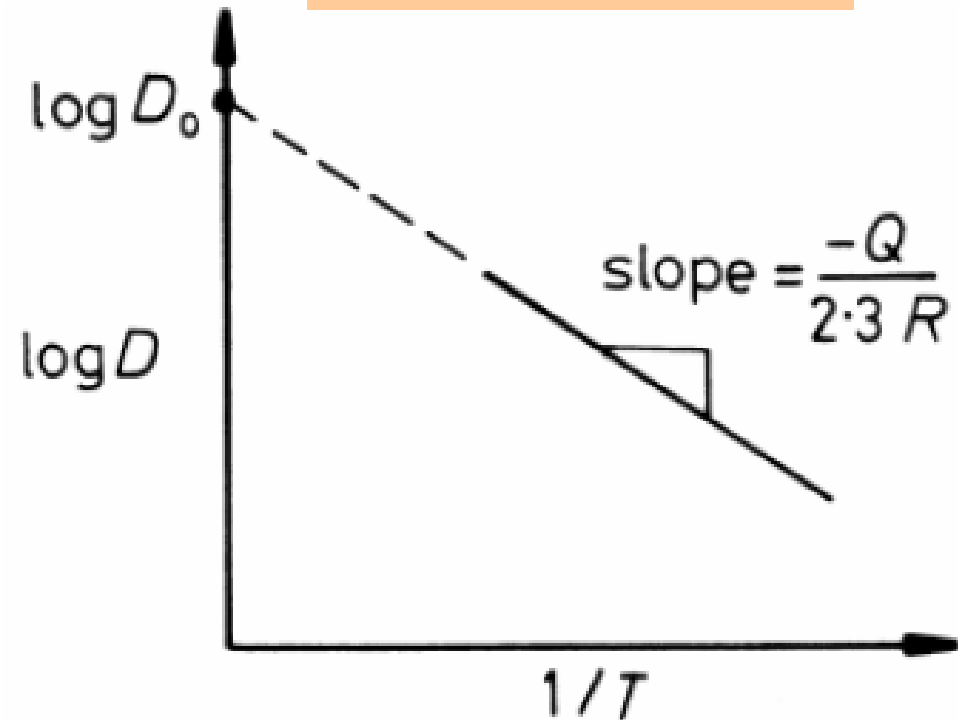
(Arrhenius-type equation)



Temperature Dependence of Diffusion

$$D_B = D_{B0} \exp \frac{-Q_{ID}}{RT}$$

How to determine Q_{ID} experimentally?



$$\log D = \log D_0 - \frac{Q}{2.3R} \left(\frac{1}{T} \right)$$

→ Therefore, from the slope of the D -curve in an $\log D$ vs $1/T$ coordinate, the activation energy may be found.

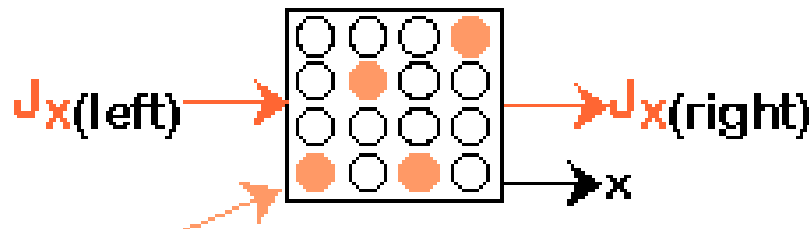
Fig. 2.7 The slope of $\log D$ v. $1/T$ gives the activation energy for diffusion Q .

Contents for today's class

- **Steady-state diffusion – Fick's First Law**
- **Nonsteady-state diffusion – Fick's Second Law**
- **Solutions to the diffusion Equations**

Steady-state diffusion

The simplest type of diffusion to deal with is when the concentration at every point does not change with time.



Steady State:
 $J_x(\text{left}) = J_x(\text{right})$

Concentration, C , in the box

- Apply Fick's First Law:

$$J_x = -D \frac{dC}{dx}$$

If $J_x(\text{left}) = J_x(\text{right})$, then

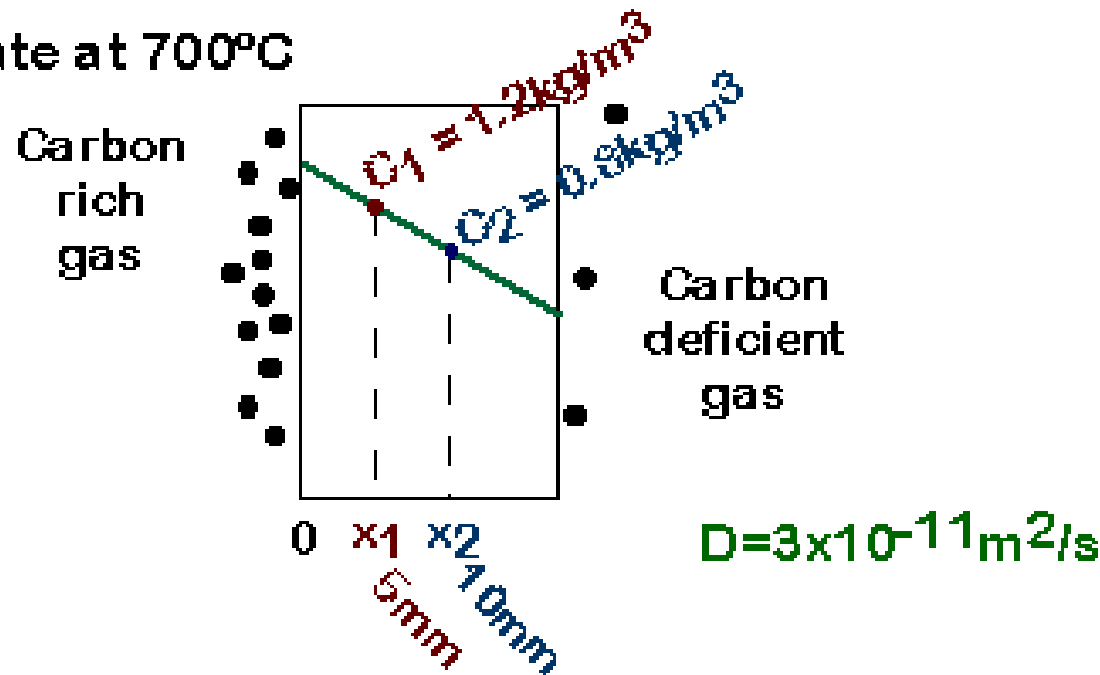
$$\frac{dC}{dx}(\text{left}) = \frac{dC}{dx}(\text{right})$$

\Rightarrow the slope, dC/dx , is constant
(does not vary with position)!

Steady-state diffusion

The simplest type of diffusion to deal with is when the concentration at every point does not change with time.

- Steel plate at 700°C



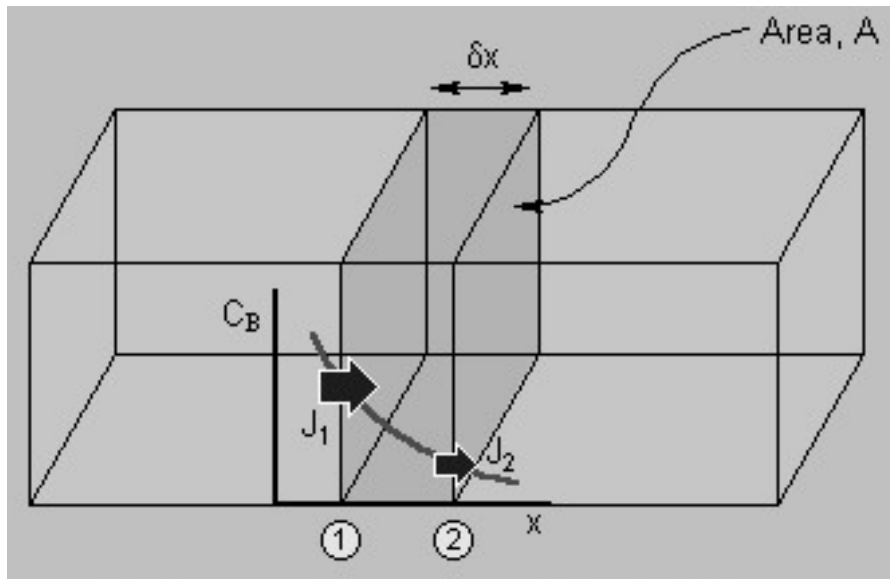
Q: How much carbon is transferring from the rich to deficient side?

$$J = -D \frac{C_2 - C_1}{x_2 - x_1} = 2.4 \times 10^{-9} \frac{\text{kg}}{\text{m}^2 \text{ s}}$$

Nonsteady-state diffusion

In most practical situations steady-state conditions are not established, i.e. concentration varies with both distance and time, and Fick's 1st law can no longer be used.

How do we know the variation of C_B with time? → Fick's 2nd law



The number of interstitial B atoms that diffuse into the slice across plane (1) in a small time interval dt :

$$\rightarrow J_1 A dt$$

$$\text{Likewise : } J_2 A dt$$

Sine $J_2 < J_1$, the concentration of B within the slice will have increased by

Due to mass conservation

$$(J_1 - J_2) A \delta t = \delta C_B A \delta x \quad \delta C_B = \frac{(J_1 - J_2) A \delta t}{A \delta x}$$

Nonsteady-state diffusion

$$J_2 = J_1 + \frac{\partial J}{\partial x} \delta x \quad (\delta x \text{ is small})$$

as $\delta t \Rightarrow 0$

$$\frac{\partial C_B}{\partial t} = - \frac{\partial J_B}{\partial x}$$

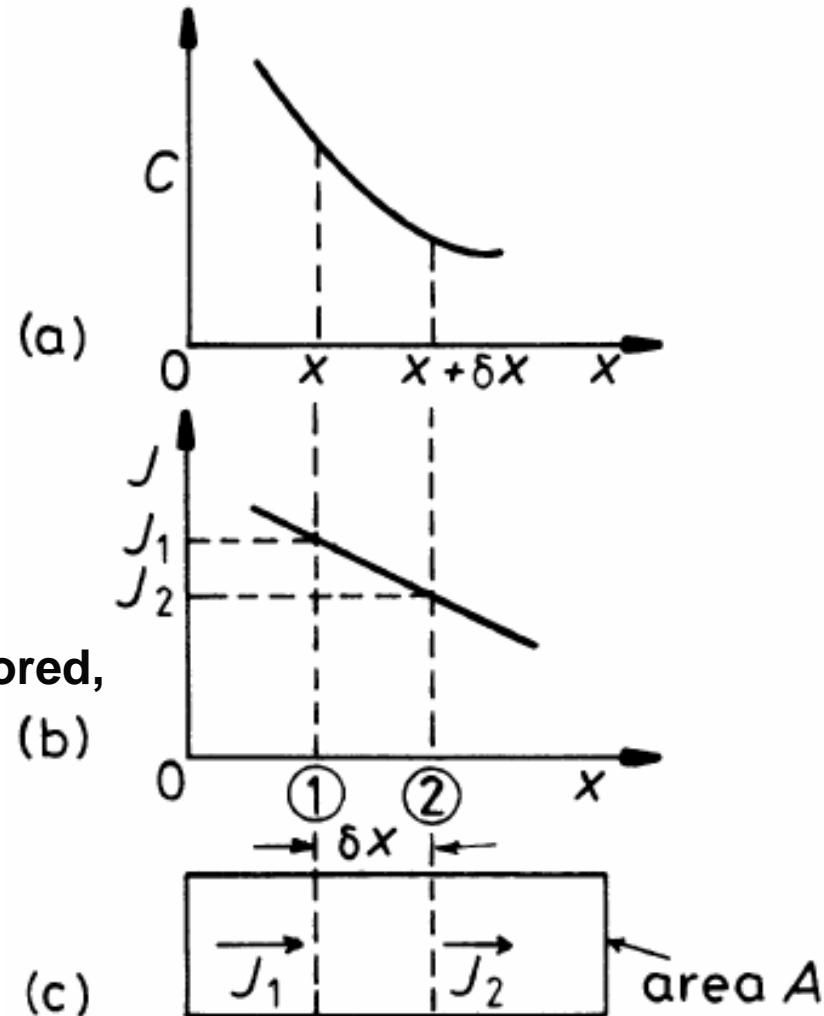
substituting Fick's 1st law gives

$$J_B = -D \frac{\partial C_B}{\partial x} \quad \frac{\partial C_B}{\partial t} = \frac{\partial}{\partial x} \left(D_B \frac{\partial C_B}{\partial x} \right)$$

If variation of D with concentration can be ignored,

$$\frac{\partial C_B}{\partial t} = D_B \frac{\partial^2 C_B}{\partial x^2}$$

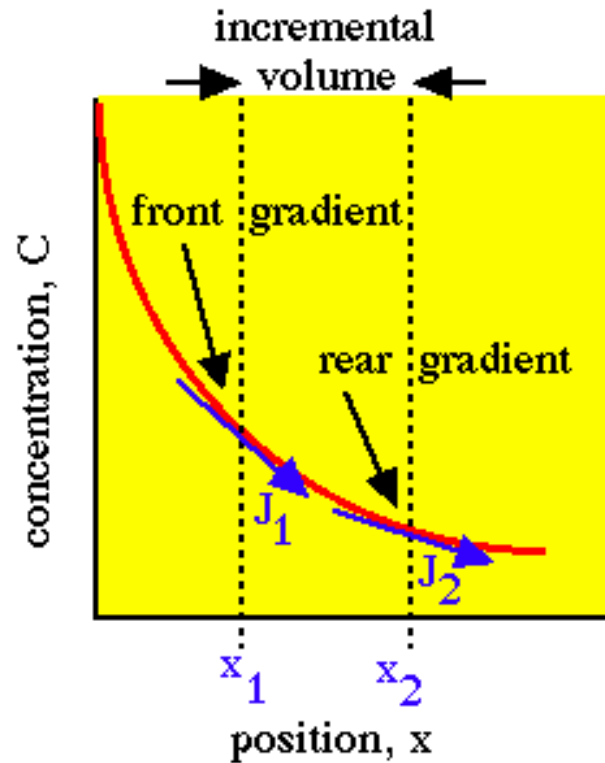
Fick's Second Law



Fick's Second Law

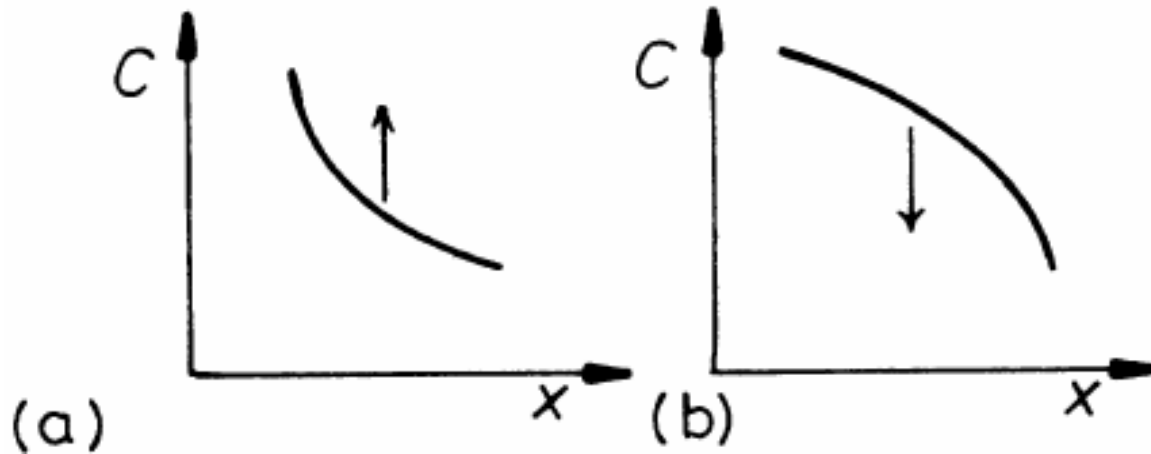
$$\frac{\partial C_B}{\partial t} = D_B \frac{\partial^2 C_B}{\partial x^2}$$

➔ Concentration varies with time and position.



$$\frac{\partial^2 C_B}{\partial x^2} = ?$$

Note that $\frac{\partial^2 C_B}{\partial x^2}$ is the curvature.



Diffusion

Diffusion is the net movement of particles from an area of high concentration to an area of low concentration.

Diffusion occurs in liquids and gases, but not in solids.

Diffusion is a passive process, meaning it does not require energy.

Diffusion is a random process, meaning the movement of particles is unpredictable.

Diffusion is a slow process, meaning it takes a long time to reach equilibrium.

Diffusion is a reversible process, meaning it can be reversed.

Diffusion is a natural process, meaning it occurs in nature.

Diffusion is a physical process, meaning it does not involve a chemical change.

Diffusion is a transport process, meaning it moves particles from one place to another.

Diffusion is a concentration process, meaning it equalizes the concentration of particles.

Diffusion is a dynamic process, meaning it is always occurring.

Diffusion is a continuous process, meaning it never stops.

Diffusion is a universal process, meaning it occurs everywhere.

Diffusion is a fundamental process, meaning it is essential for life.

Diffusion is a key process, meaning it is one of the most important processes in nature.

Diffusion is a basic process, meaning it is one of the most fundamental processes in science.

Diffusion is a simple process, meaning it is easy to understand.

Diffusion is a clear process, meaning it is easy to observe.

Diffusion is a direct process, meaning it is straightforward.

Diffusion is a logical process, meaning it makes sense.

Diffusion is a rational process, meaning it is based on reason.

Diffusion is a scientific process, meaning it is based on evidence.

Diffusion is a practical process, meaning it is useful in the real world.

Diffusion is a valuable process, meaning it is worth studying.

Diffusion is a beautiful process, meaning it is a wonder of nature.

Solutions to the diffusion equations

Ex1. Homogenization

of sinusoidal varying composition
in the elimination of segregation in casting

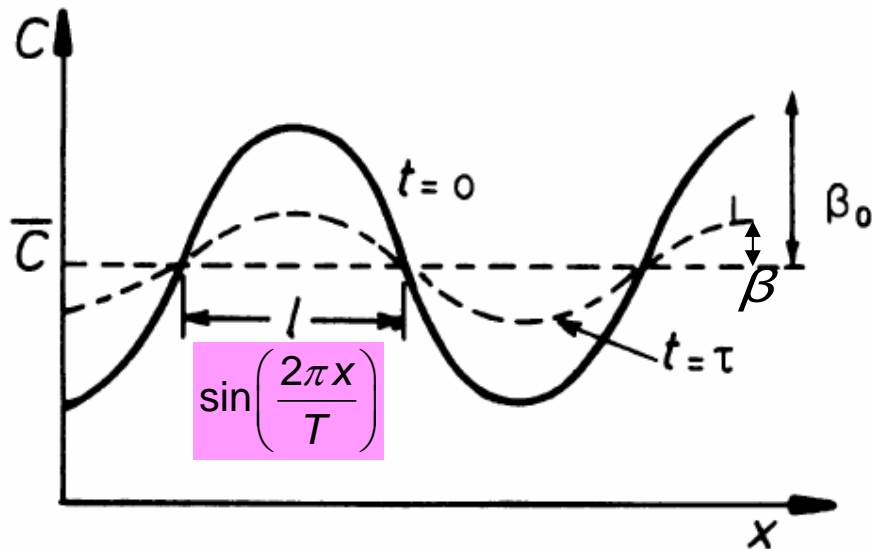


Fig. 2.10 The effect of diffusion on a sinusoidal variation of composition.

\bar{C} : the mean composition l : half wavelength

β_0 : the amplitude of the initial concentration profile

Initial or Boundary Cond.?

$$C = \bar{C} + \beta_0 \sin \frac{\pi x}{l} \quad \text{at } t=0$$

$$C = \bar{C} + \beta_0 \sin \frac{\pi x}{l} \exp\left(\frac{-t}{\tau}\right)$$

$$\tau = \frac{l^2}{\pi^2 D} \quad \tau : \text{relaxation time}$$

$$\beta = \beta_0 \exp(-t / \tau) \quad \text{at } x = \frac{l}{2}$$

Rigorous solution of $\frac{\partial C}{\partial t} = D \frac{\partial^2 C}{\partial x^2}$ for $C(x,0) = \bar{C} + \beta_0 \sin \frac{\pi x}{l}$

변수분리법이용

Let $C = XT$

$$X \frac{dT}{dt} = DT \frac{d^2 X}{dx^2}$$

$$\frac{1}{DT} \frac{dT}{dt} = \frac{1}{X} \frac{d^2 X}{dx^2} \equiv -\lambda^2$$

$$\frac{1}{T} \frac{dT}{dt} = -\lambda^2 D$$

$$\frac{d \ln T}{dt} = -\lambda^2 D$$

$$T = T_0 e^{-\lambda^2 D t}$$

$$\frac{d^2 X}{dx^2} + \lambda^2 X = 0$$

$$X = A' \cos \lambda x + B' \sin \lambda x$$

$$X(x) \equiv \bar{C} + \beta_0 \sin \frac{\pi x}{l}$$

$$\therefore C = (A \cos \lambda x + B \sin \lambda x) e^{-\lambda^2 D t}$$

$$\therefore C = A_0 + \sum_{n=1}^{\infty} (A_n \sin \lambda_n x + B_n \cos \lambda_n x)$$

$$t = 0 \rightarrow C \equiv \bar{C} + \beta_0 \sin \frac{\pi x}{l}$$

$$; A_0 = \bar{C}, B_n = 0, A_1 = \beta_0$$

($A_n = 0$ for all others)

$$\therefore C \equiv \bar{C} + \beta_0 \sin \frac{\pi x}{l} e^{-\frac{t}{l^2/\pi^2 D}}$$

$$C = \bar{C} + \beta_0 \sin \frac{\pi x}{l} \exp\left(\frac{-t}{\tau}\right)$$

Solutions to the diffusion equations

Ex2. Carburization of Steel

The aim of carburization is to increase the carbon concentration in the surface layers of a steel product in order to achieve a harder wear-resistant surface.

1. Holding the steel in CH_4 and/or CO at an austenitic temperature.
2. By controlling gases the concentration of C at the surface of the steel can be maintained at a suitable constant value.
3. At the same time carbon continually diffuses from the surface into the steel.



Carburizing of steel

Depth of Carburization?

The error function solution:

$$C = C_s - (C_s - C_0) \operatorname{erf}\left(\frac{x}{2\sqrt{Dt}}\right)$$

$$\operatorname{erf}(Z) = \frac{2}{\sqrt{\pi}} \int_0^Z e^{-y^2} dy$$

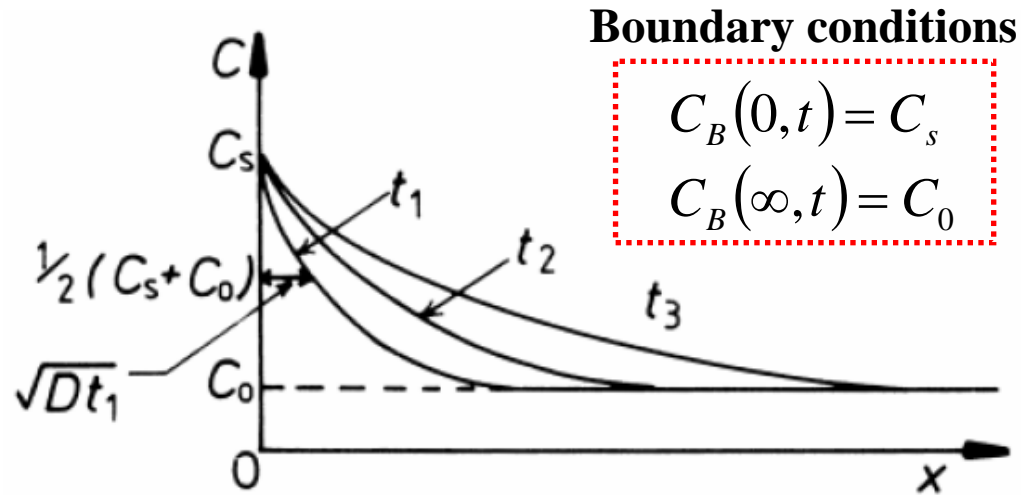


Fig. 2.11 Concentration profiles at successive times ($t_3 > t_2 > t_1$) for diffusion into a semi-infinite bar when the surface concentration C_s is maintained constant.

$$\frac{C_s - C}{C_s - C_0} = \operatorname{erf}\left(\frac{x}{2\sqrt{Dt}}\right) \quad \operatorname{erf}(0.5) \approx 0.5 \quad C = \frac{C_s + C_0}{2}$$

- Since $\operatorname{erf}(0.5) \approx 0.5$ the depth at which the carbon concentration is midway between C_s and C_0 is given $(x / 2\sqrt{Dt}) \cong 0.5$

that is $x \cong \sqrt{Dt}$ → **Depth of Carburization**

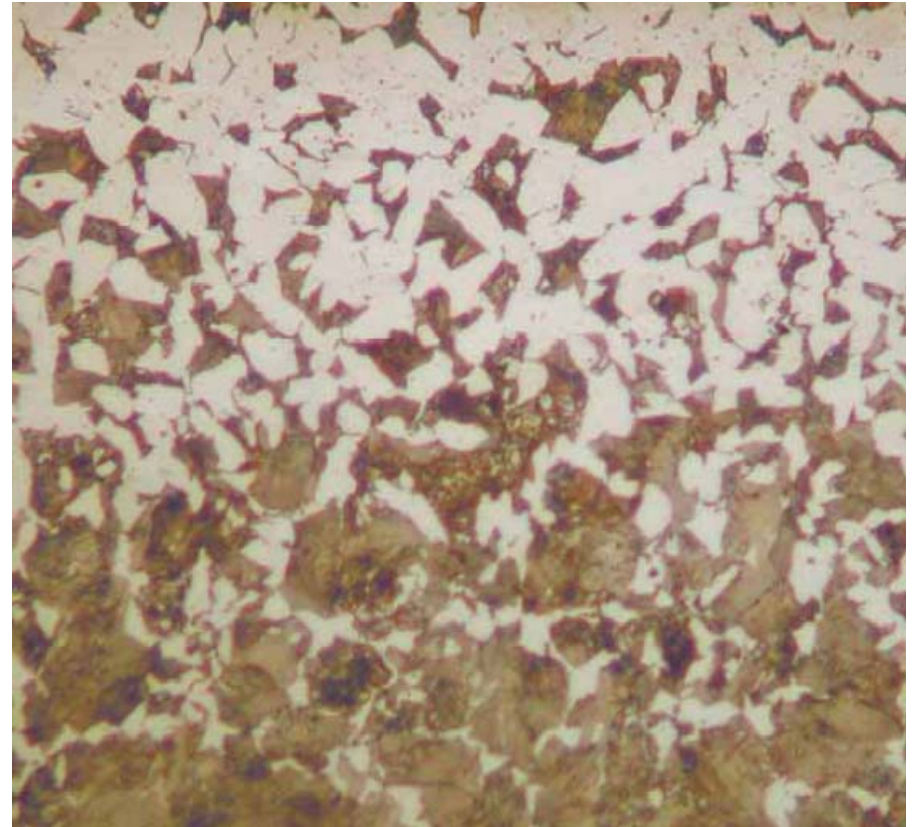
Carburizing of steel

Thus the thickness of the carburized layer is $\cong \sqrt{Dt}$.

Note also that the depth of any isoconcentration line is directly proportion to \sqrt{Dt} , i.e. *to obtain a twofold increase in penetration requires a fourfold increase in time.*

Decarburization of Steel?

$$C = C_0 \operatorname{erf} \left(\frac{x}{2\sqrt{Dt}} \right)$$



Error function

In mathematics, the error function (also called the Gauss error function) is a non-elementary function which occurs in probability, statistics and partial differential equations.

It is defined as:

$$\operatorname{erf}(x) = \frac{2}{\sqrt{\pi}} \int_0^x e^{-t^2} dt.$$

By expanding the right-hand side in a Taylor series and integrating, one can express it in the form

$$\operatorname{erf}(x) = \frac{2}{\sqrt{\pi}} \sum_{n=0}^{\infty} \frac{(-1)^n x^{2n+1}}{(2n+1)n!} = \frac{2}{\sqrt{\pi}} \left(x - \frac{x^3}{3} + \frac{x^5}{10} - \frac{x^7}{42} + \frac{x^9}{216} - \dots \right)$$

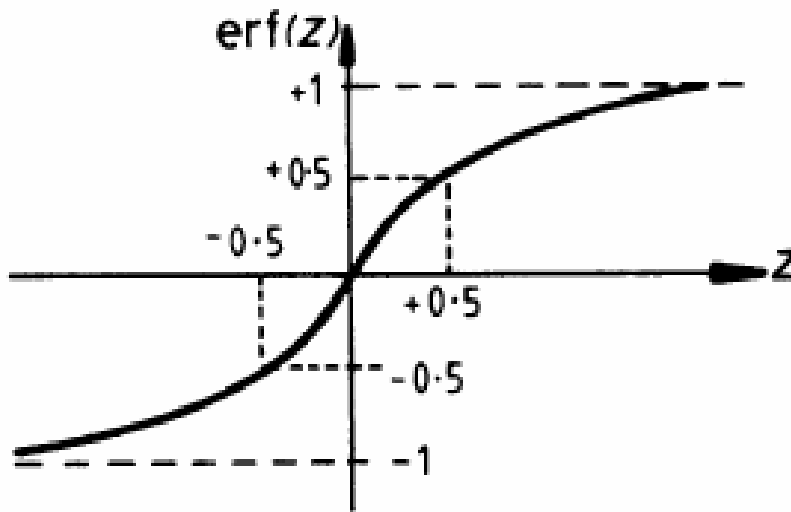
for every real number x .

(From Wikipedia, the free encyclopedia)

Error function

$$\text{erf}(z) = \frac{2}{\sqrt{\pi}} \int_0^z \exp(-y^2) dy$$

Table 1-1. The Error Function



(a)

Fig. 2.12 (a) Schematic diagram illustrating the main features of the error function.

z	$\text{erf}(z)$	z	$\text{erf}(z)$
0	0	0.85	0.7707
0.025	0.0282	0.90	0.7969
0.05	0.0564	0.95	0.8209
0.10	0.1125	1.0	0.8427
0.15	0.1680	1.1	0.8802
0.20	0.2227	1.2	0.9103
0.25	0.2763	1.3	0.9340
0.30	0.3286	1.4	0.9523
0.35	0.3794	1.5	0.9661
0.40	0.4284	1.6	0.9763
0.45	0.4755	1.7	0.9838
0.50	0.5205	1.8	0.9891
0.55	0.5633	1.9	0.9928
0.60	0.6039	2.0	0.9953
0.65	0.6420	2.2	0.9981
0.70	0.6778	2.4	0.9993
0.75	0.7112	2.6	0.9998
0.80	0.7421	2.8	0.9999

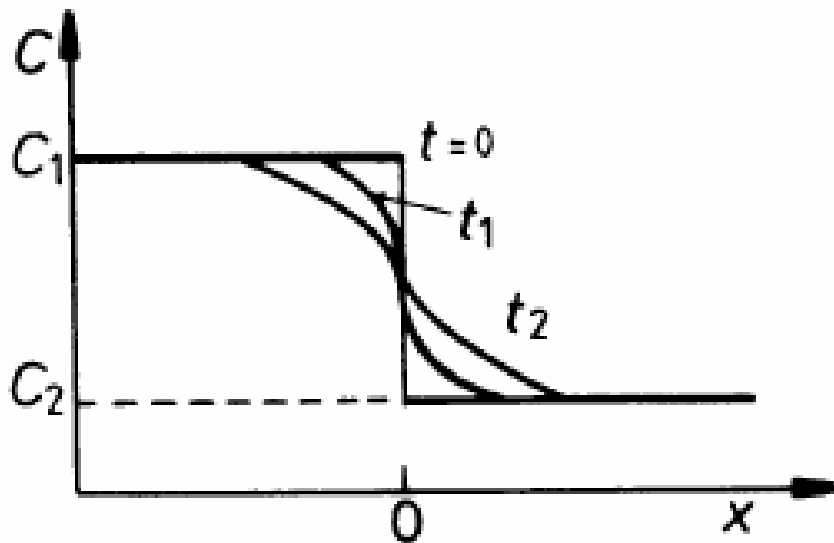
Solutions to the diffusion equations

Ex3. Diffusion Couple

Joining of two semi-infinite specimens of compositions C_1 and C_2 ($C_1 > C_2$)

Draw C vs. x with time $t = 0$ and $t > 0$.

Boundary conditions?



$$C = C_1 \quad t = 0, \quad -\infty < x < 0$$

$$C = C_2 \quad t = 0, \quad 0 < x < \infty$$

$$C = C_1 \quad x = -\infty, \quad 0 < t < \infty$$

$$C = C_2 \quad x = \infty, \quad 0 < t < \infty$$

$$C = \left(\frac{C_1 + C_2}{2} \right) - \left(\frac{C_1 - C_2}{2} \right) \operatorname{erf} \left(\frac{x}{2\sqrt{Dt}} \right)$$

(b)

Fig. 2.12 (b) Concentration profiles at successive times ($t_2 > t_1 > 0$) when two semi-infinite bars of different composition are annealed after welding.

The section is completed with 4 example solutions to Fick's 2nd law:
carburisation, decarburisation, diffusion across a couple and homogenisation.

The solutions given are as follows:

Process	Solution
Carburisation	$C = C_s - (C_s - C_0) \operatorname{erf}\left(\frac{x}{2\sqrt{Dt}}\right)$ $C_s = \text{Surface concentration}$ $C_0 = \text{Initial bulk concentration}$
Decarburisation	$C = C_0 \operatorname{erf}\left(\frac{x}{2\sqrt{Dt}}\right)$ $C_0 = \text{Initial bulk concentration}$
Diffusion Couple	$C = \left(\frac{C_1 + C_2}{2}\right) - \left(\frac{C_1 - C_2}{2}\right) \operatorname{erf}\left(\frac{x}{2\sqrt{Dt}}\right)$ $C_1 = \text{Concentration of steel 1}$ $C_2 = \text{Concentration of steel 2}$
Homogenisation	$C = C_{\text{mean}} + \beta_0 \sin\left(\frac{\pi x}{\lambda}\right) \exp\left(-\frac{t}{\tau}\right)$ $C_{\text{mean}} = \text{Mean concentration}$ $\beta_0 = \text{Initial concentration amplitude}$ $l = \text{half-wavelength of cells}$ $t = \text{relaxation time}$