

# Nano Materials

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- Basics
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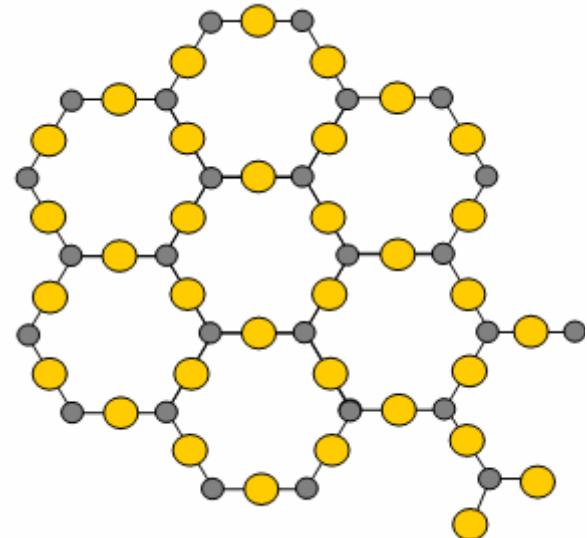
## Basics

- Crystal structure
- Surface
- Kinetics
- Surface chemistry
- Consolidation
- Quantum confinement

## Basics- Crystal structure

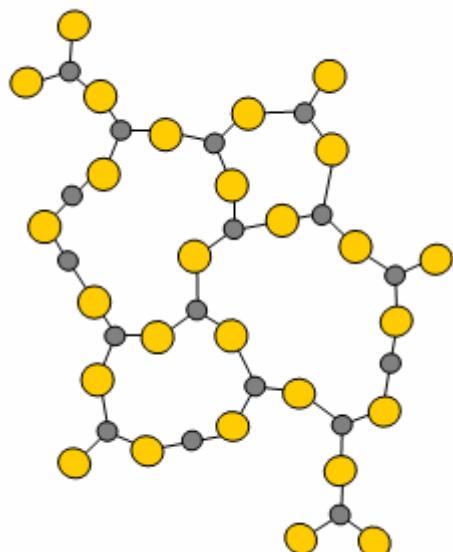
### □ Crystalline vs. amorphous

Crystalline  $\text{SiO}_2$



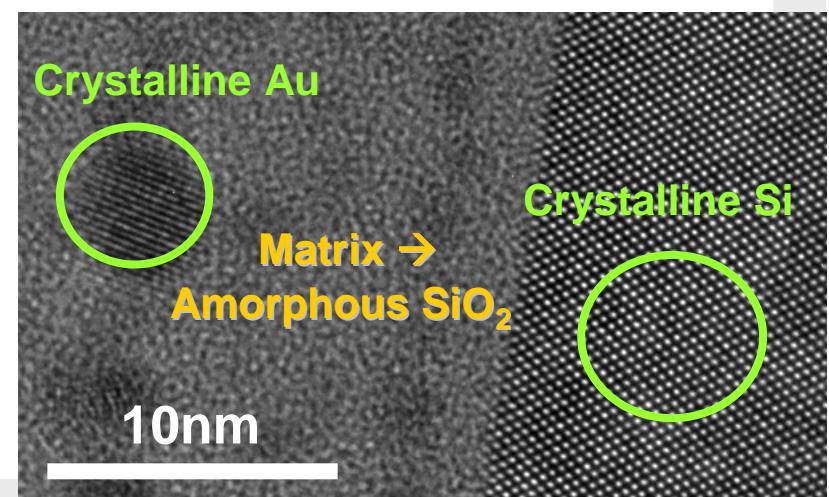
Long range order

Amorphous  $\text{SiO}_2$



Short range order

Oxygen atom  
Silicon atom



## Basics- Crystal structure

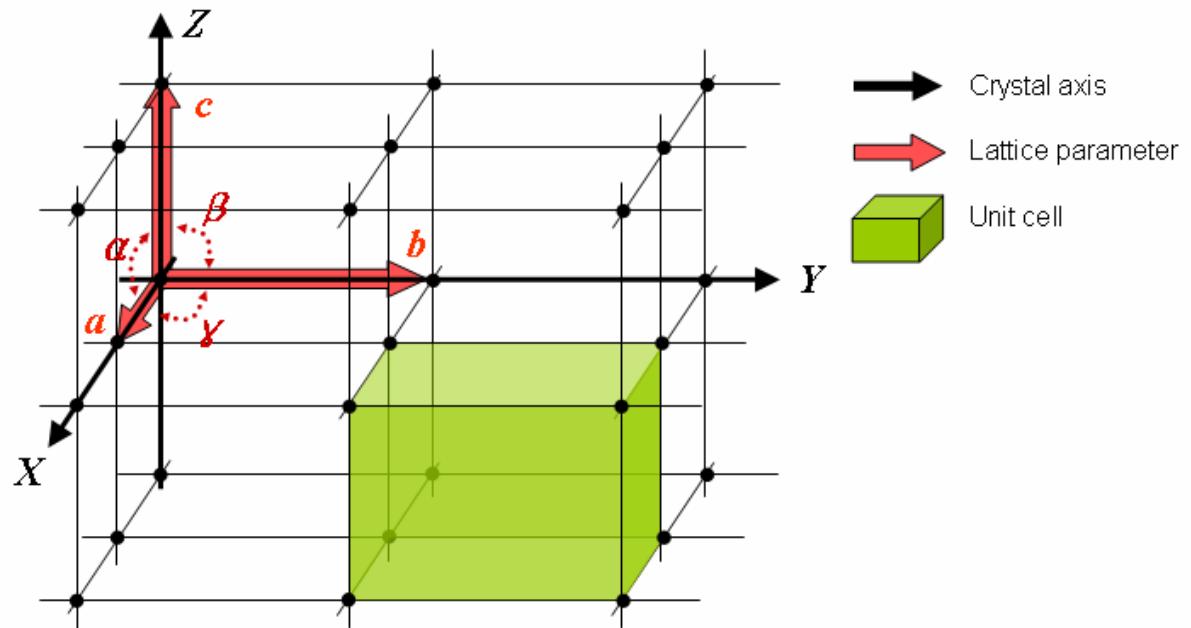
### □ Definition of Crystal

A crystal is an anisotropic, homogeneous body consisting of a three-dimensional periodic ordering of atoms, ions or molecules

### □ Definition of Lattice

A lattice means a three-dimensional array of points coinciding with atom positions (the space arrangement of equivalent sites in a crystal )

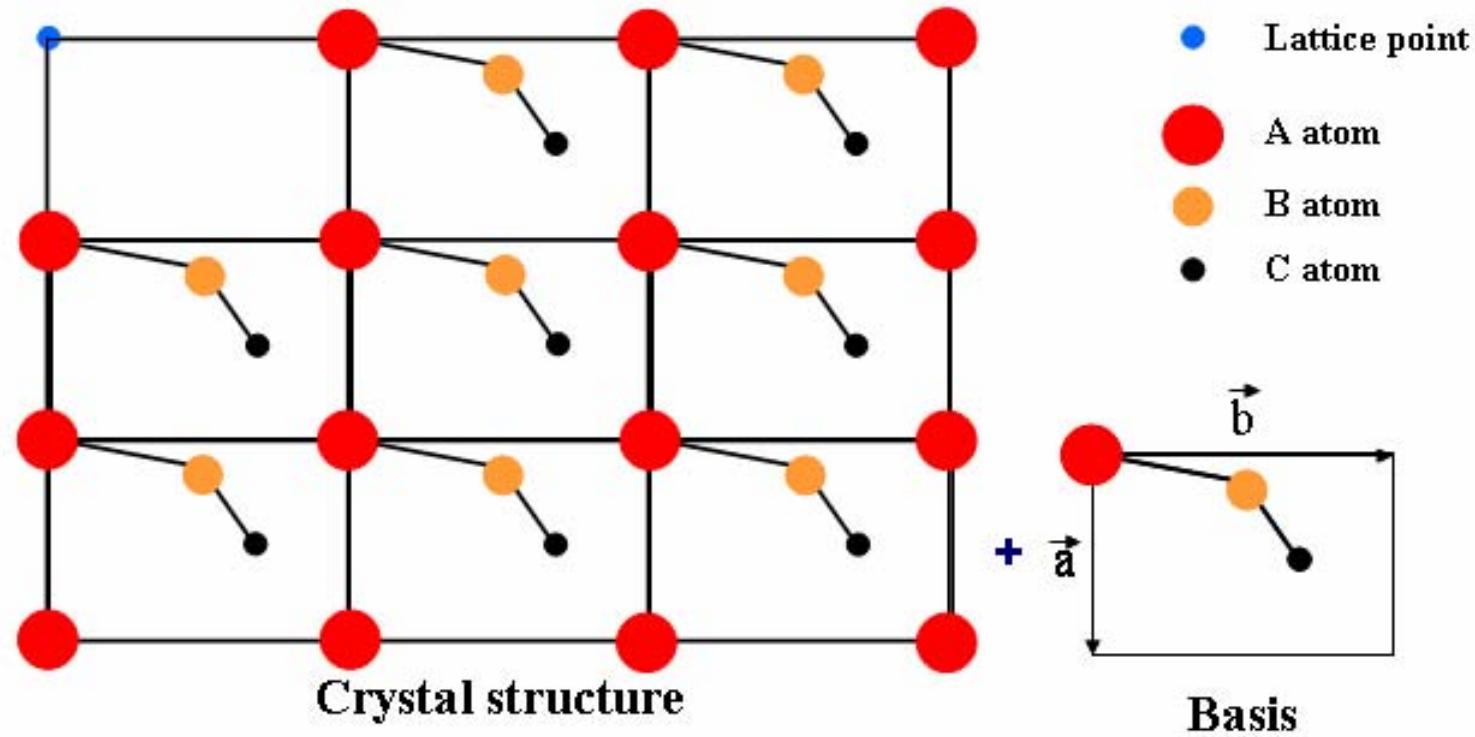
### □ Lattice Parameters



## Basics- Crystal structure

### □ Crystal Structure

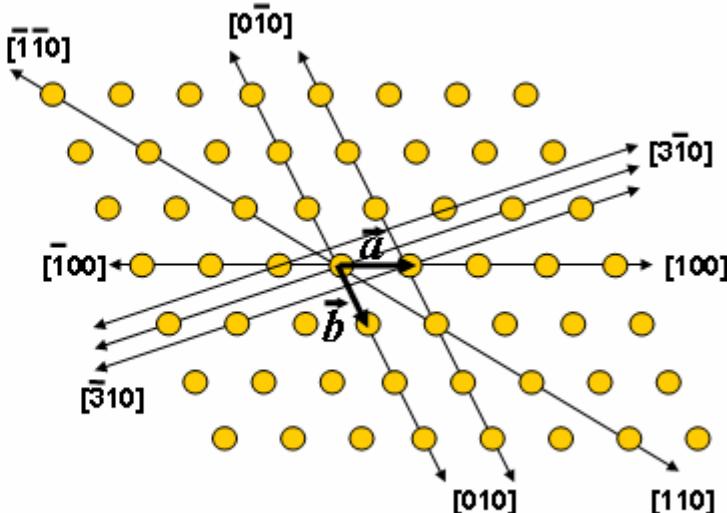
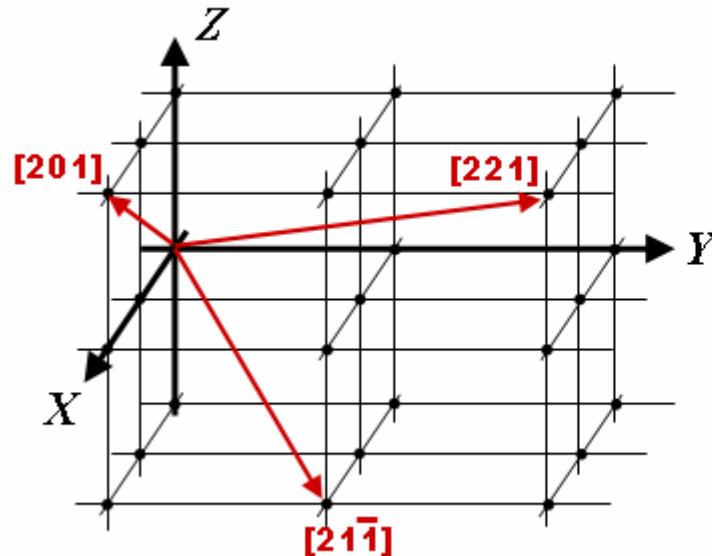
Crystal structure = Lattice + Basis



## Basics- Crystal structure

### □ Definition of Crystallographic direction

Crystallographic direction is defined as a line between two points, or a vector.

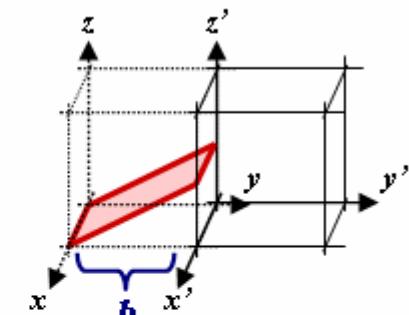
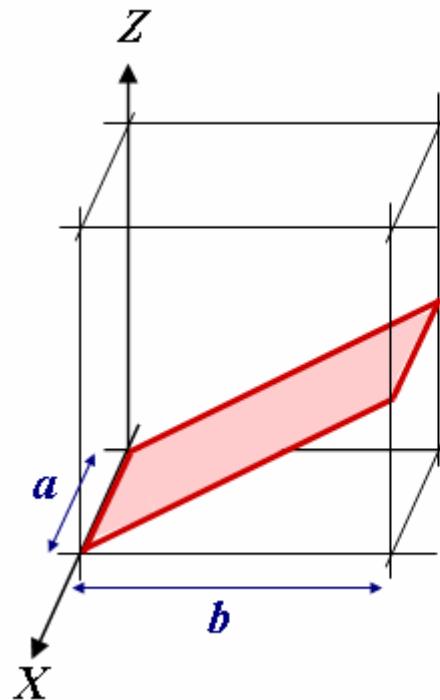


Note that the direction of  $[uvw]$  describes not only a line through the origin and the point  $uvw$ , but the infinite set of lines which are parallel to it.

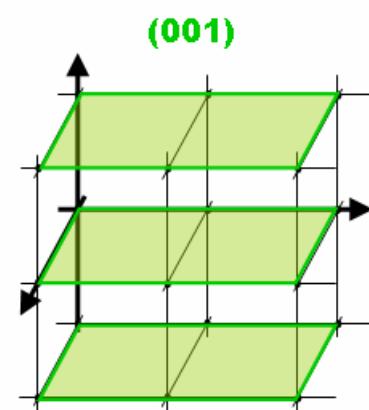
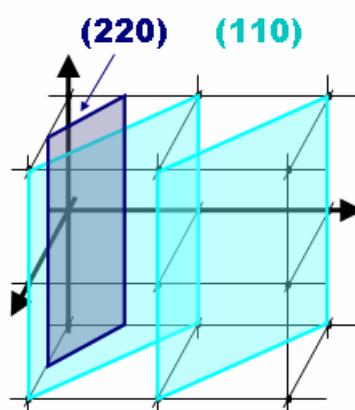
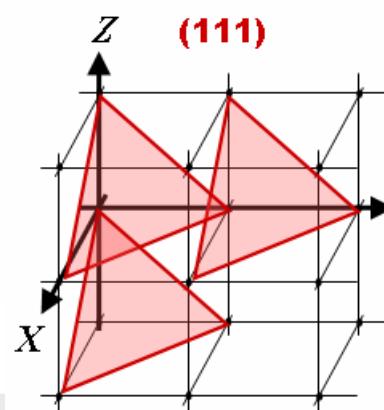
## Basics- Crystal structure

### □ Definition of Crystallographic plane

The values ( $h k l$ ) are called Miller indices, and they are defined as the smallest integral multiples of the reciprocals of the plane intercepts of axis.



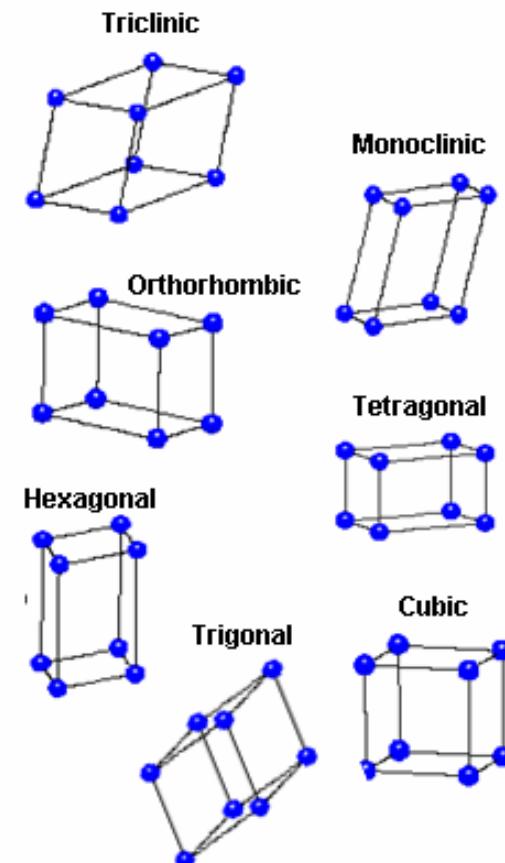
	$x$	$y$	$z$
Projections	$\infty a$	$-b$	$c/2$
Projections (in terms of $a$ , $b$ , and $c$ )	$\infty$	$-1$	$1/2$
Reciprocal	$0$	$-1$	$2$
Reduction (unnecessary)			
Enclosure	$[0\bar{1}2]$		



# Basics- Crystal structure

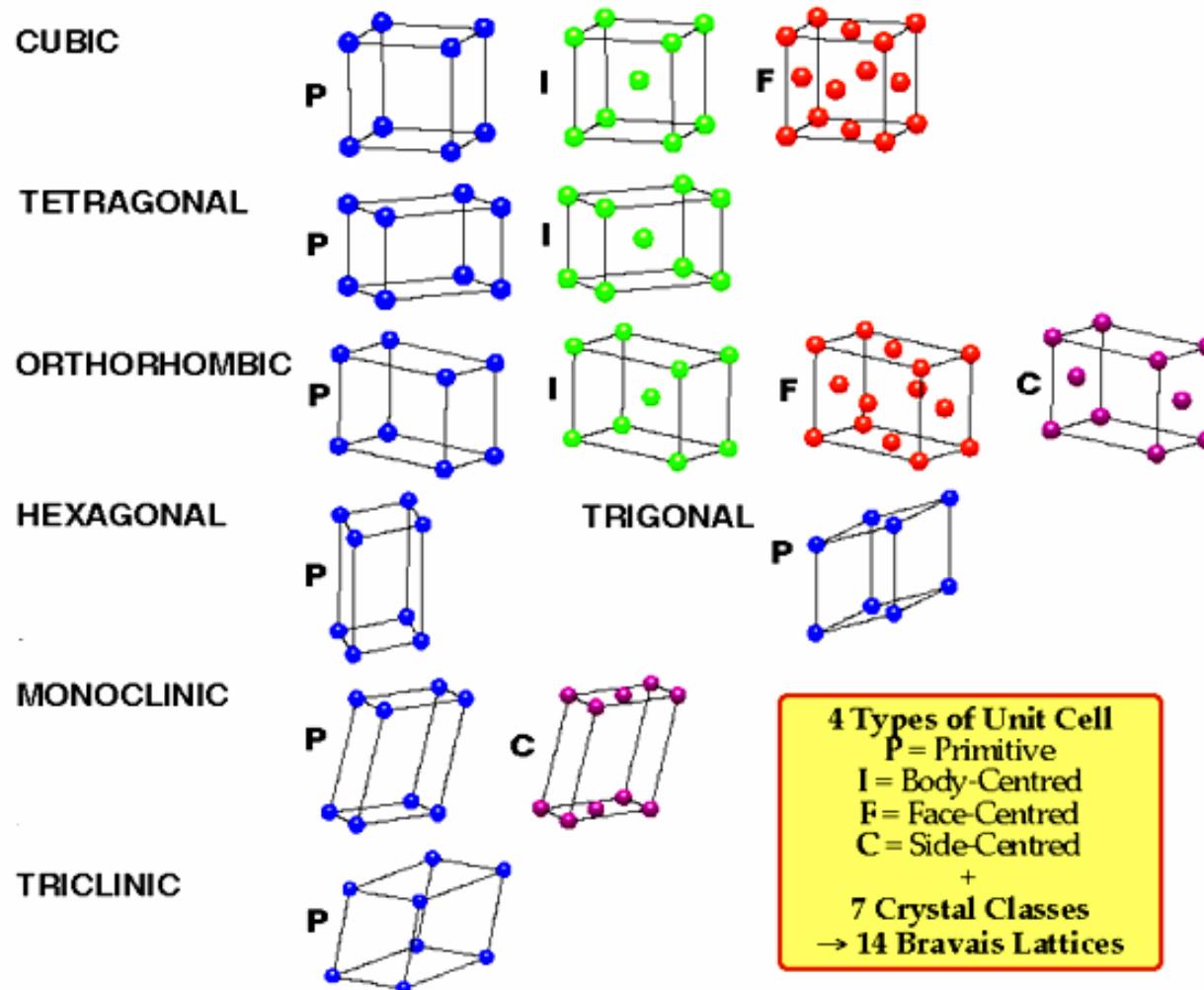
## □ Seven Crystal Systems

Crystal system	Restriction on the axial system
<b>Triclinic</b>	$a \neq b \neq c \quad \alpha \neq \beta \neq \gamma$
<b>Monoclinic</b>	$a \neq b \neq c \quad \alpha = \gamma = 90^\circ, \quad \beta > 90^\circ$
<b>Orthorhombic</b>	$a \neq b \neq c \quad \alpha = \beta = \gamma = 90^\circ$
<b>Tetragonal</b>	$a = b \neq c \quad \alpha = \beta = \gamma = 90^\circ$
<b>Trigonal</b>	$a = b = c \quad \alpha = \beta = \gamma \neq 90^\circ$
<b>Hexagonal</b>	$a = b \neq c \quad \alpha = \beta = 90^\circ, \quad \gamma = 120^\circ$
<b>Cubic</b>	$a = b = c \quad \alpha = \beta = \gamma = 90^\circ$



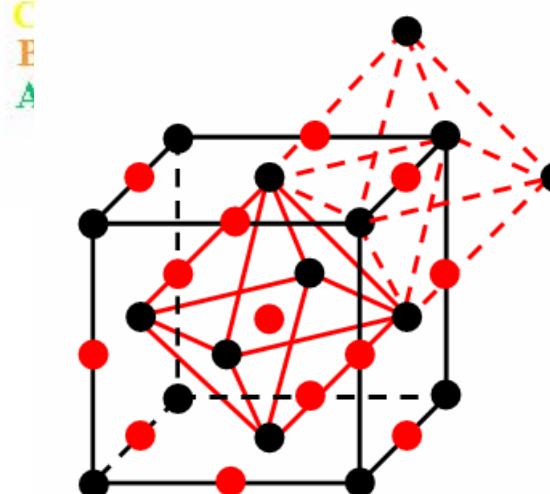
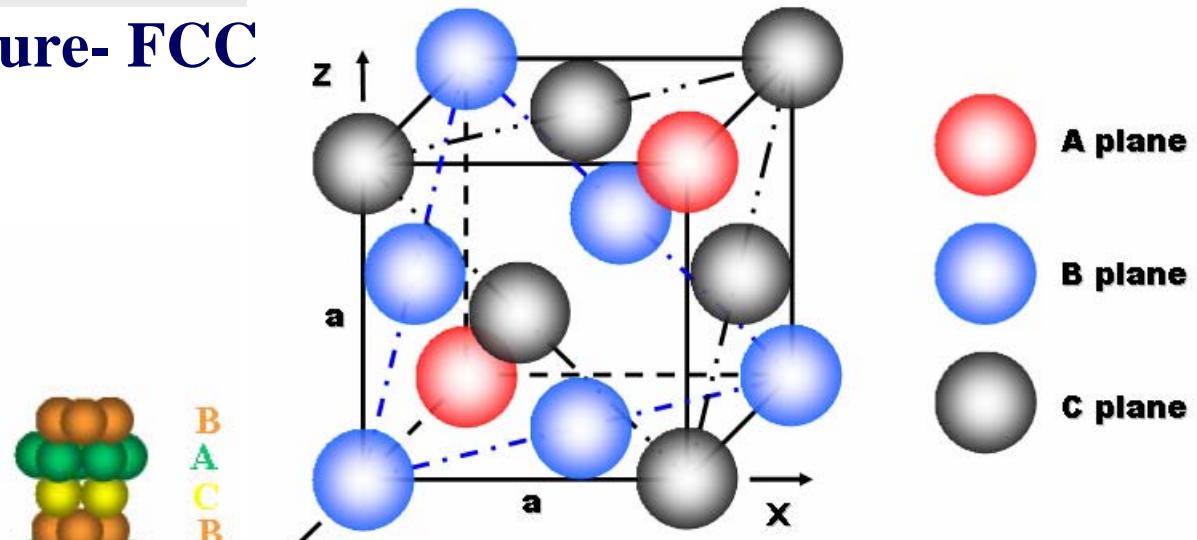
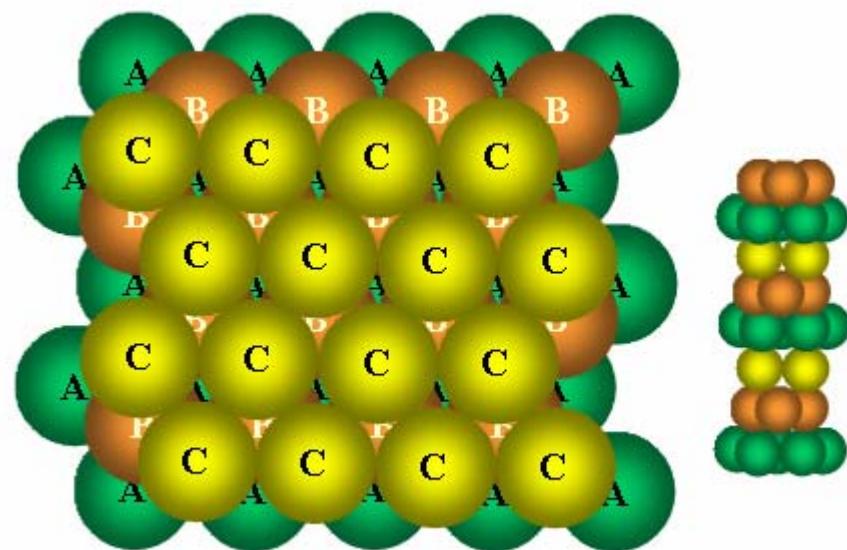
# Basics- Crystal structure

## □ 14 Bravais Lattice

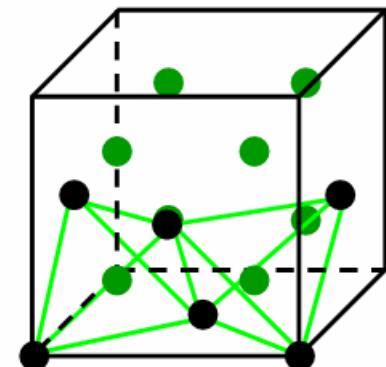


## Basics- Crystal structure

### □ Close Packed Structure- FCC



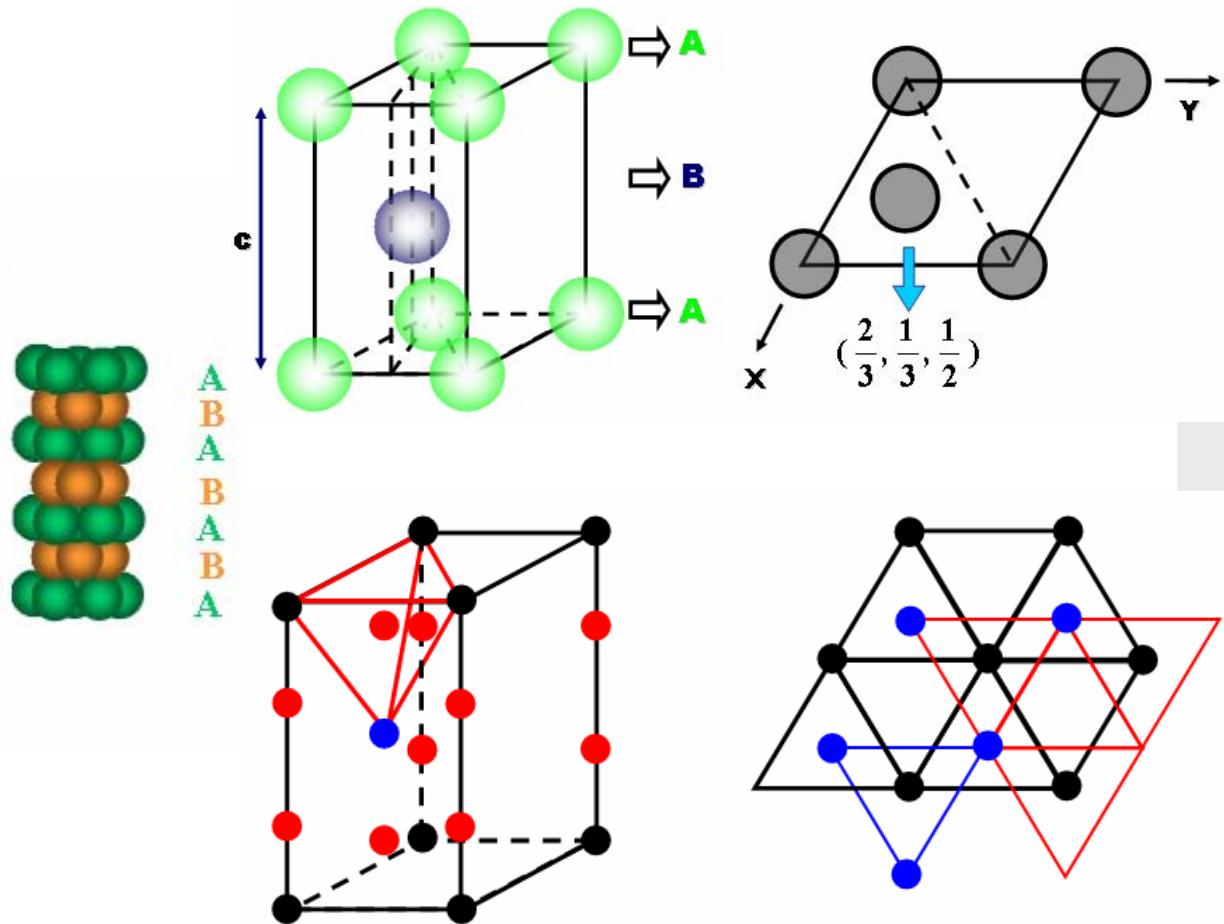
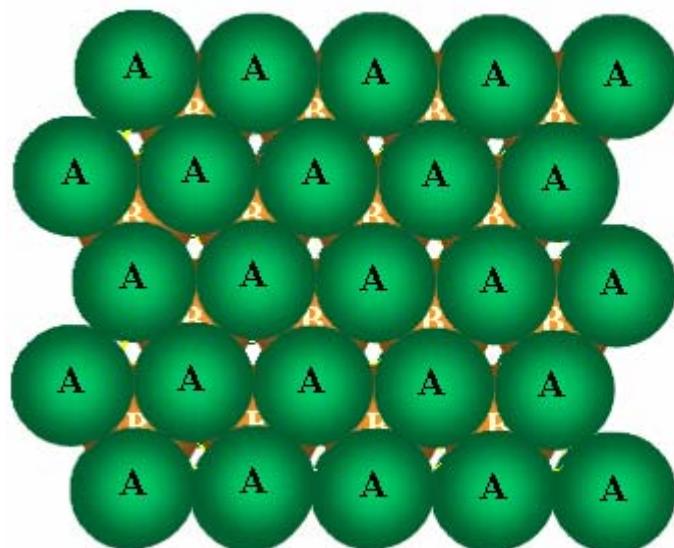
Octahedral sites ; 4



Tetrahedral sites ; 8

## Basics- Crystal structure

### □ Close Packed Structure- HCP

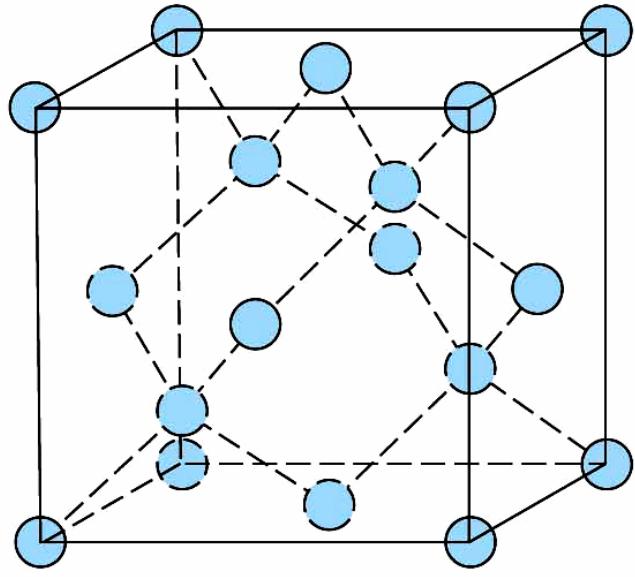


Tetrahedral sites ; 4

$$(0, 0, \frac{3}{8}) \quad (0, 0, \frac{5}{8}) \quad (\frac{1}{3}, \frac{2}{3}, \frac{1}{8}) \quad (\frac{1}{3}, \frac{2}{3}, \frac{7}{8})$$

## Basics- Crystal structure

### □ Diamond



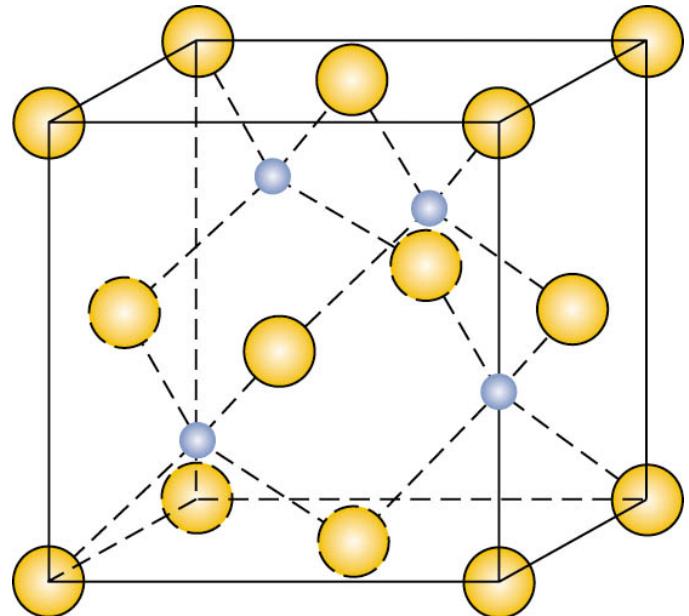
● c Si, Ge

$\frac{1}{2}$  tet. sites  
by cations

- 8 atoms/unit cell
- coordination #: 4:4
- FCC with two atoms per lattice point,  $(0,0,0)$  and  $(\frac{1}{4}, \frac{1}{4}, \frac{1}{4})$
- two interwoven FCC lattices

## Basics- Crystal structure

### □ Zinc Blende



● Zn

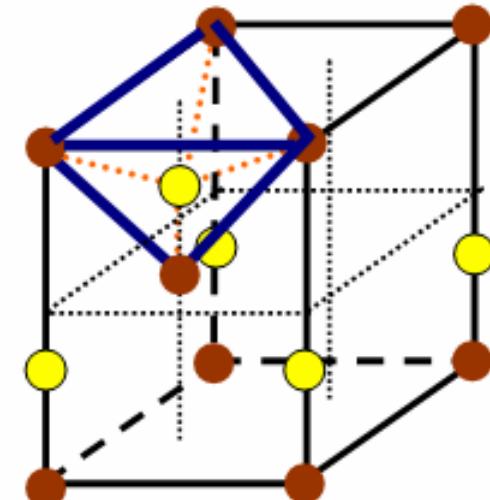
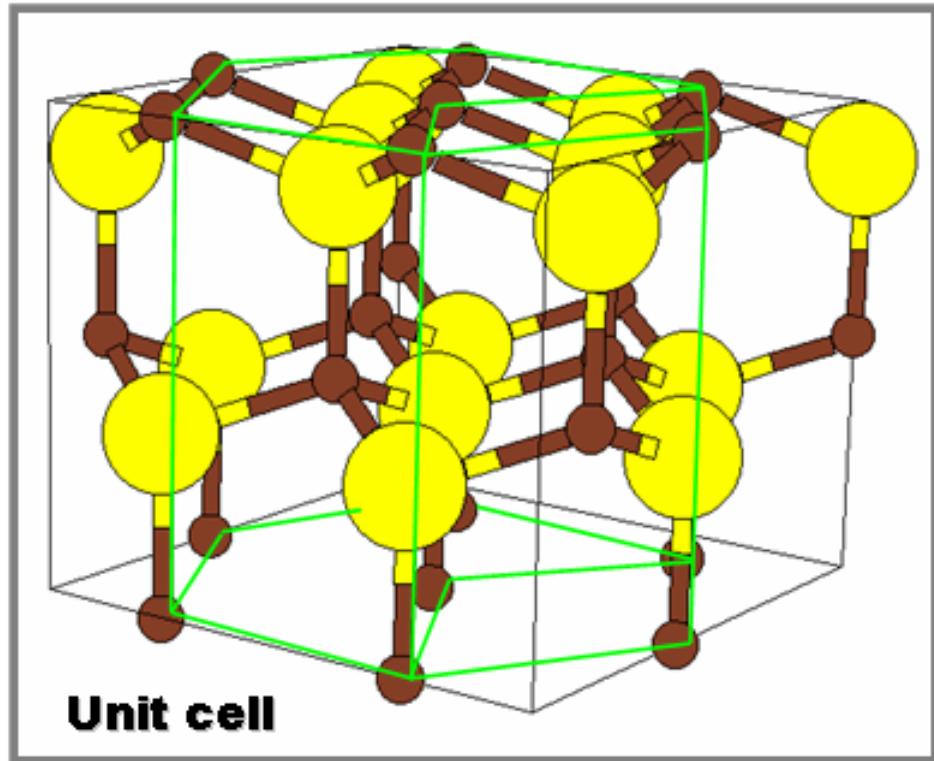
○<sup>s</sup> GaAs

$\frac{1}{2}$  tet. sites  
by cations

- 4 molecules/unit cell
- coordination #: 4:4
- FCC with two different atoms per lattice point,  $(0,0,0)$   $(\frac{1}{4}, \frac{1}{4}, \frac{1}{4})$
- two interwoven FCC lattices

## Basics- Crystal structure

### □ Wurtzite

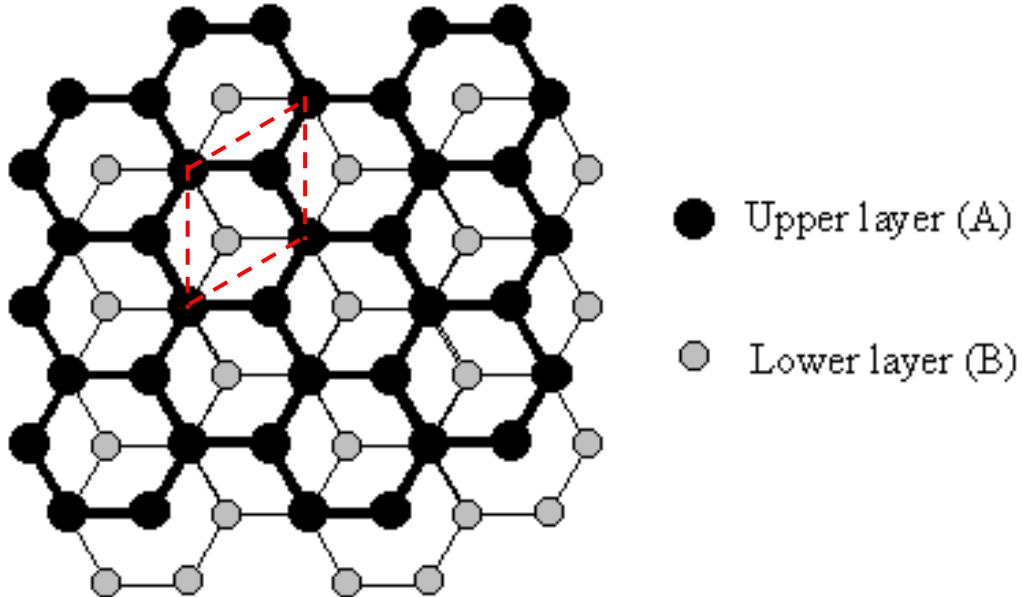
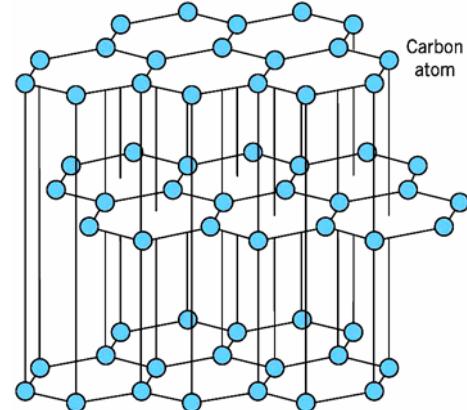


HCP array of S &  
Zn occupying of  $\frac{1}{2}$  tetrahedral sites

# of tetrahedral sites : 4  
Upright 2 ;  $(\frac{2}{3}, \frac{1}{3}, \frac{1}{8})$   $(0, 0, \frac{5}{8})$  or  
Inverted 2 ;  $(\frac{2}{3}, \frac{1}{3}, \frac{7}{8})$   $(0, 0, \frac{3}{8})$

## Basics- Crystal structure

- Carbon
  - diamond
  - graphite



- covalent/van der Waals

- $1.48/3.4 \text{ \AA}^{\circ}$

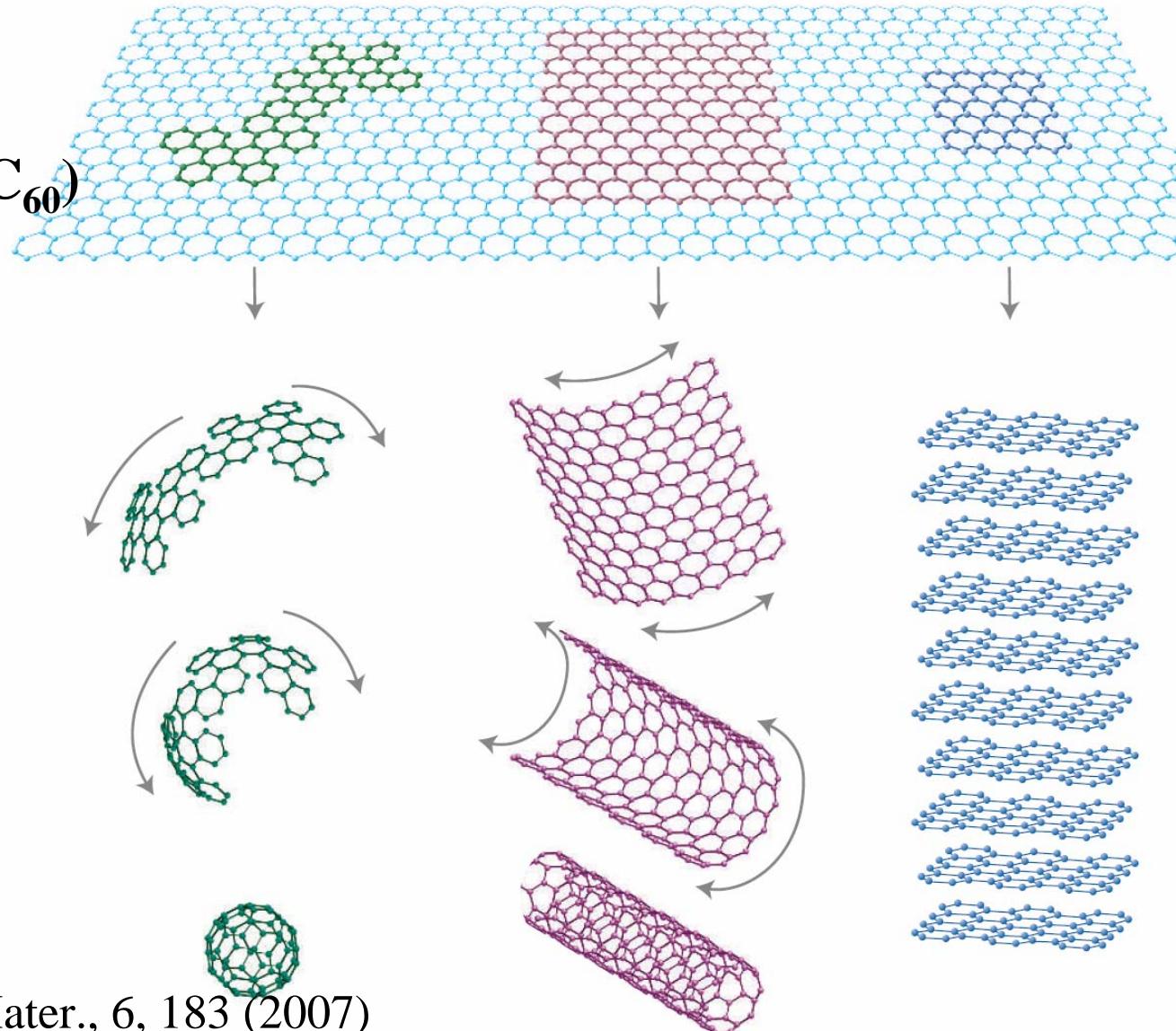
- 4 atoms/unit cell

- coordination #: 3

## Basics- Crystal structure

### □ Carbon

- graphene
- fullerenes ( $C_{60}$ )
- nanotube

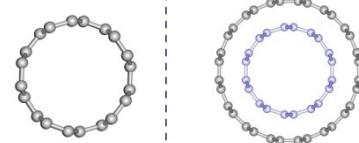
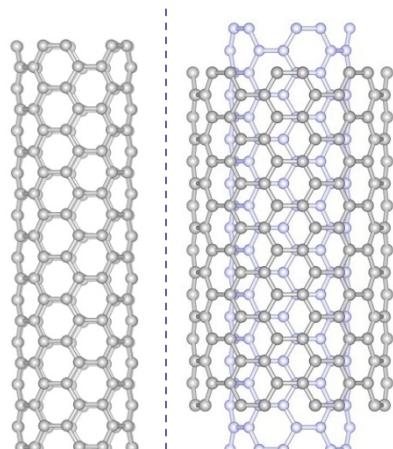


A.K.Geim, Nature Mater., 6, 183 (2007)

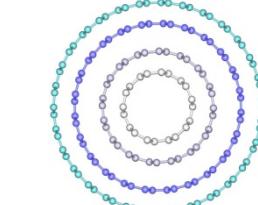
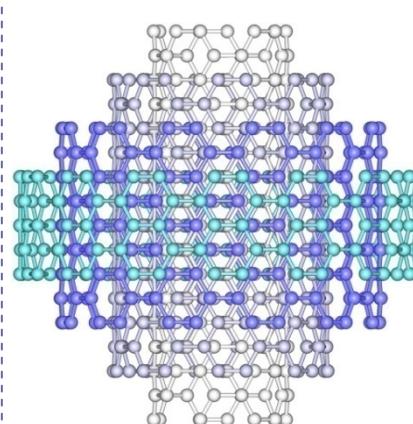
## Basics- Crystal structure

### □ Carbon-Classification of carbon nanotubes

- Single-wall CNT, double-wall CNT, multi-wall CNTs
- Zigzag and armchair nanotubes, achiral nanotube



**SWCNT**



**DWCNT**

**Thin-MWCNT**

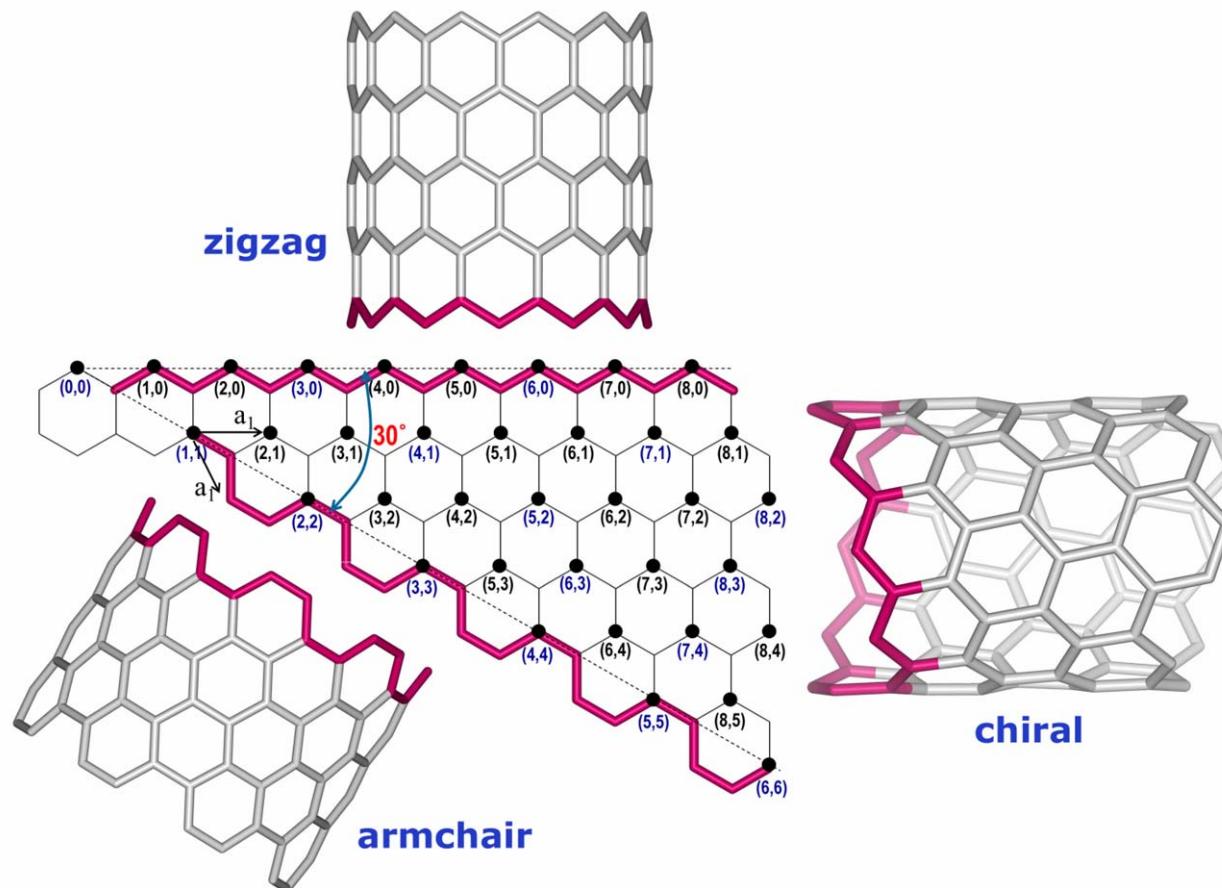


**MWNCT**

## Basics- Crystal structure

### □ Carbon-Classification of carbon nanotubes

- Zigzag and armchair nanotubes, achiral nanotube



## Basics- Defects

### □ Point Defect

- Vacancy
- Interstitial
- Impurities (Dopants)

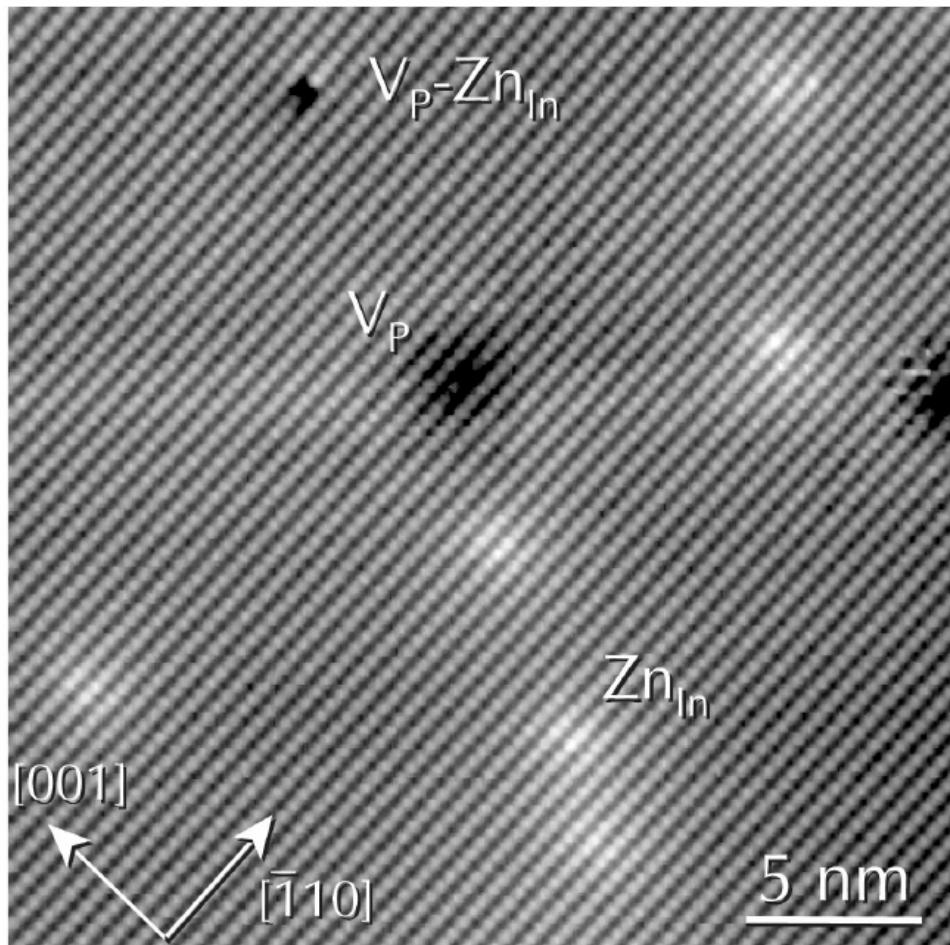


Fig. 1 Constant-current STM images of a p-doped InP(110) surface. Six  $Zn_{In}$  dopant atoms in different subsurface layers, one P vacancy ( $V_p$ ), and one vacancy-dopant complex ( $V_p-Zn_{In}$ ) are visible. The image shows the occupied states at -2.2 V sample voltage. (Adapted with permission from<sup>6</sup>. © 1996 American Physical Society.)

## Basics- Defects

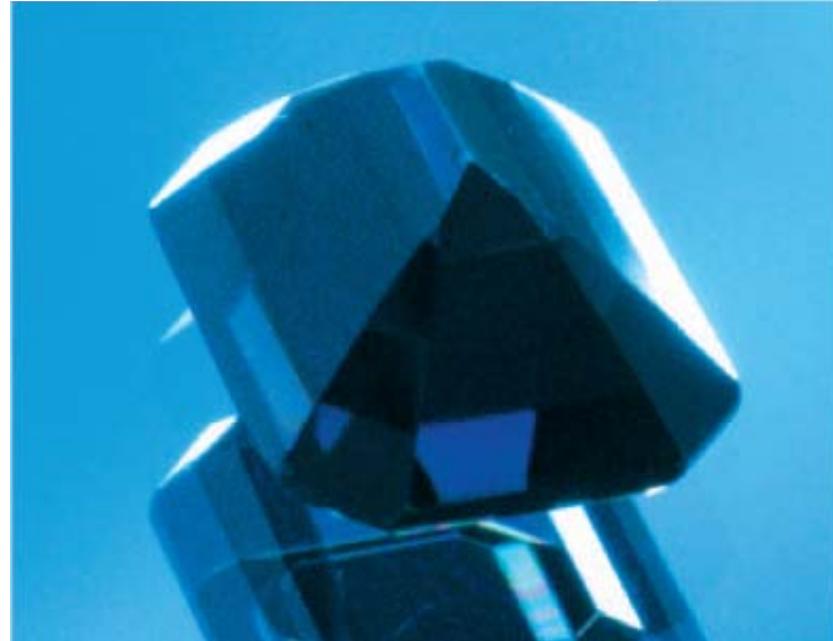
# The value of seeing nothing

Jochen Mannhart and Darrell G. Schlom



Figure 1 Now you see it, now you don't. These micrographs of a SrTiO<sub>3</sub> crystal show the effect of removing oxygen atoms, leaving vacancies in

oxygen vacancy: V<sub>O</sub><sup>III</sup>+2e<sup>-</sup>



the crystal lattice: the glistening oxidized gem (top) is transformed into a dull blue, conductive crystal (bottom).

NATURE | VOL 430 | 5 AUGUST 2004 | www.nature.com/nature

## Basics- Defects

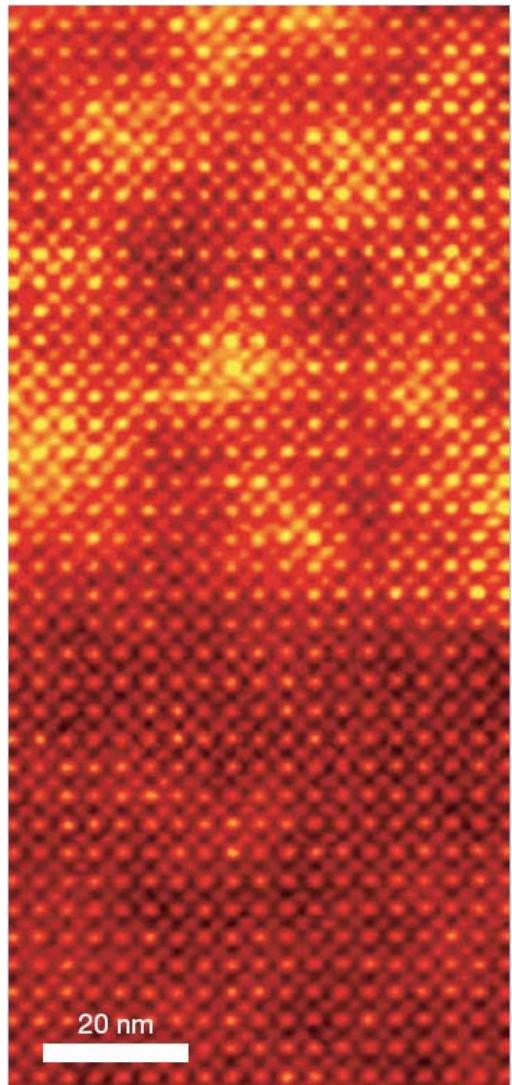
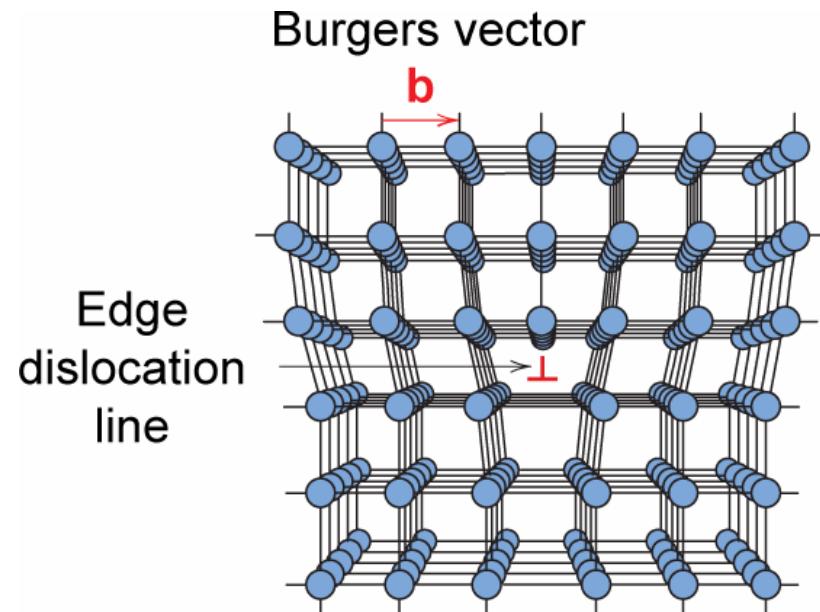


Figure 2 What a difference a layer makes.  
The abrupt junction between layers of  $\text{SrTiO}_3$  (bottom) and  $\text{SrTiO}_{3-\delta}$  (top) is clear in this image created by Muller *et al.*<sup>3</sup> using a scanning transmission electron microscope. Each bright-orange blob is a cluster of oxygen vacancies.

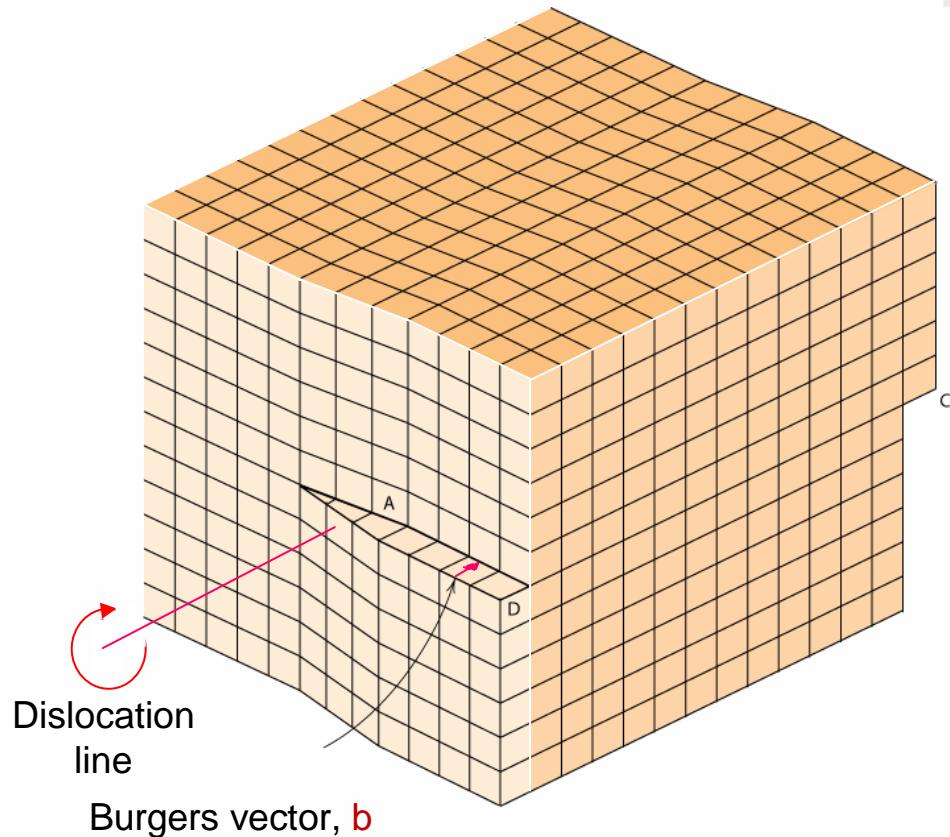
## Basics- Defects

### □ Line Defect (dislocation)

- edge



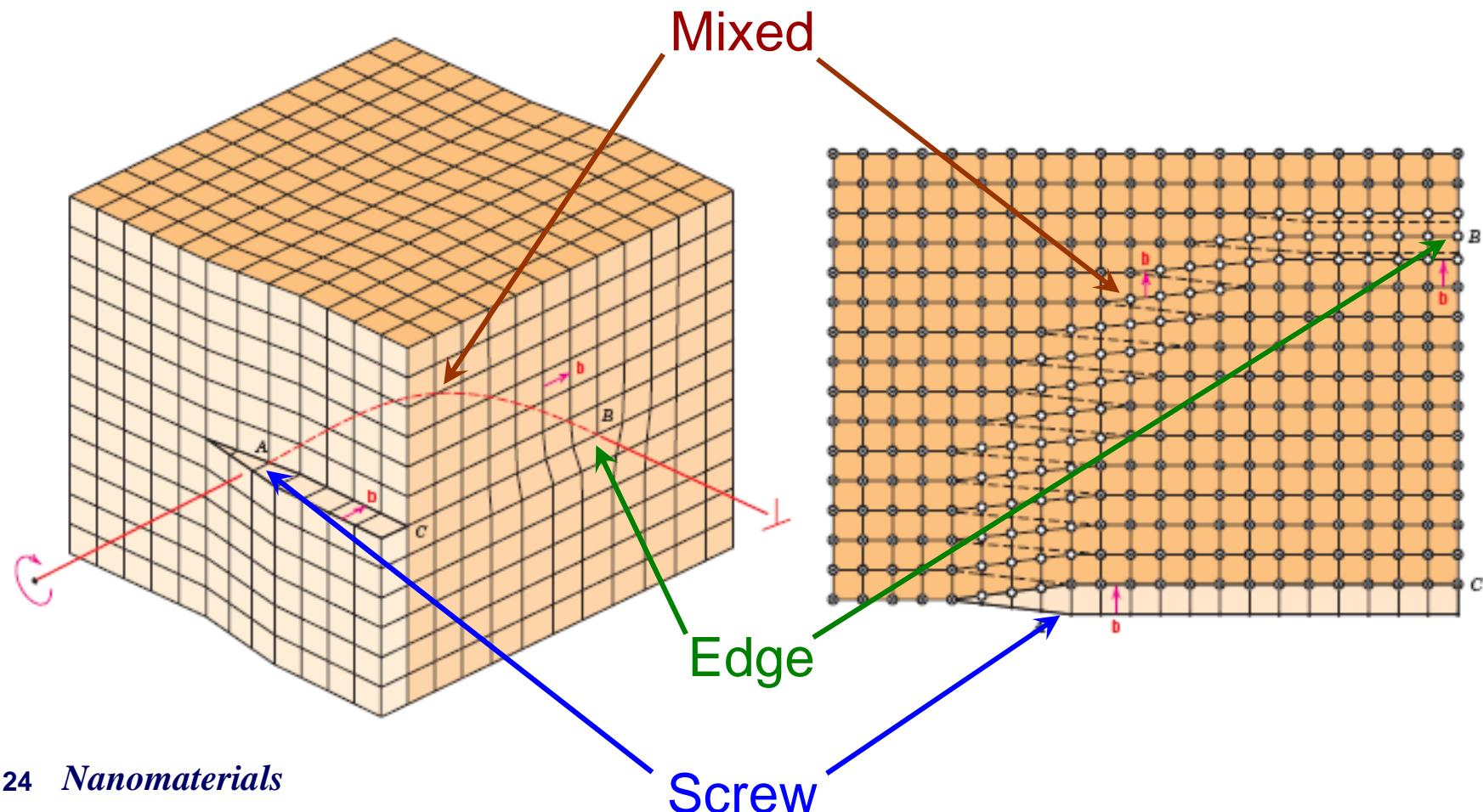
- screw



## Basics- Defects

### □ Line Defect (dislocation)

- mixed

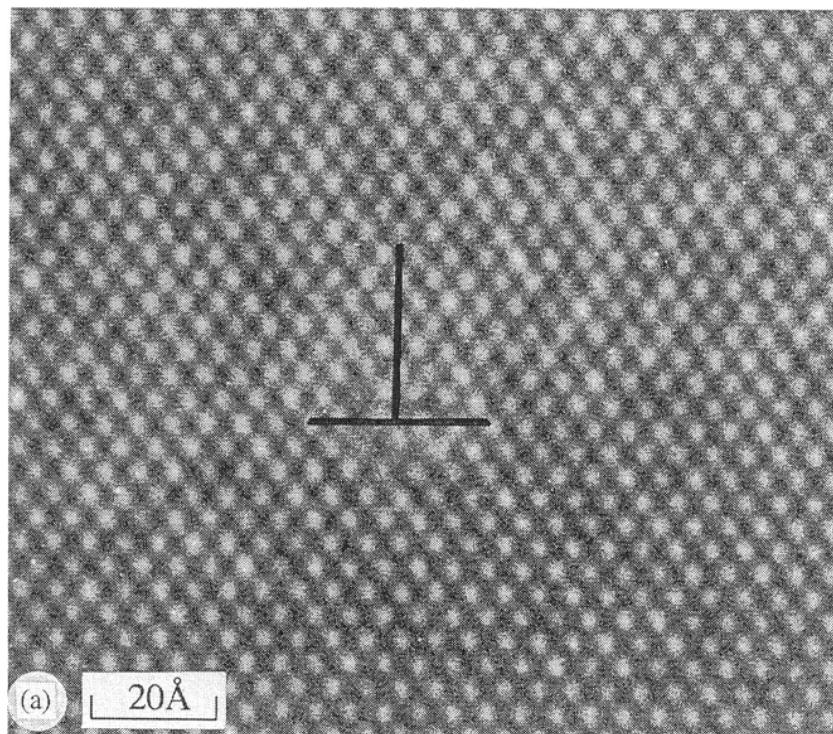


## Basics- Defects

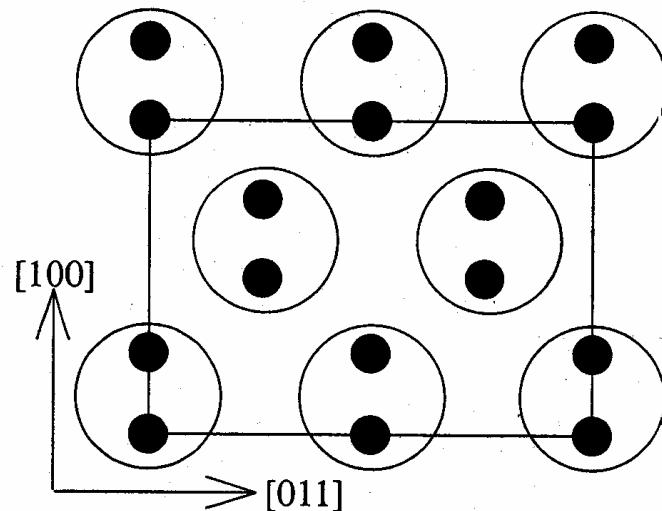
### □ Line Defect (dislocation)

- observation of dislocation

high resolution transmission electron microscope image electron beam incident along an  $<011>$  zone of Si



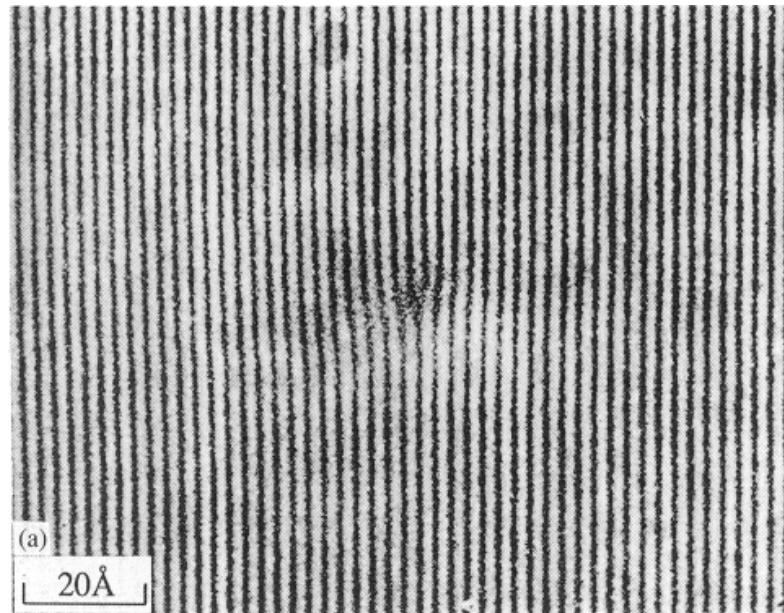
$$\vec{b} = \frac{1}{2}[011]$$



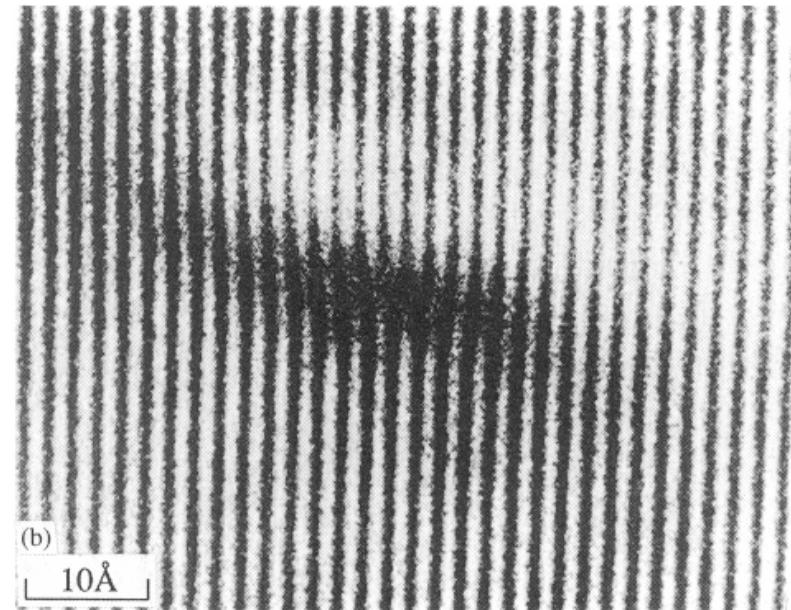
## Basics- Defects

### □ Line Defect (dislocation)

- lattice fringe image of dislocation



narrow edge dislocation

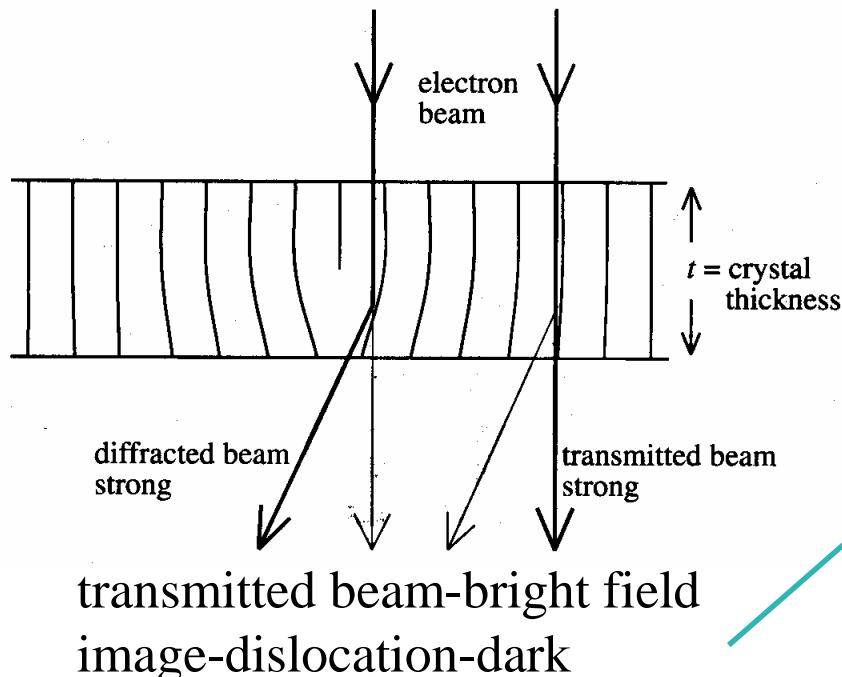


wide edge dislocation

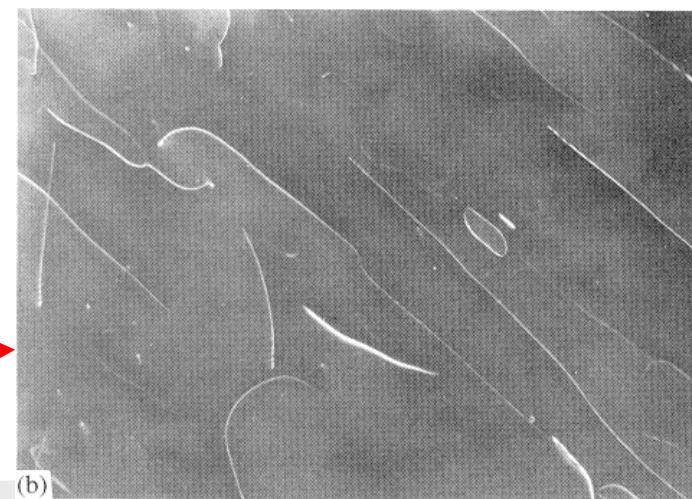
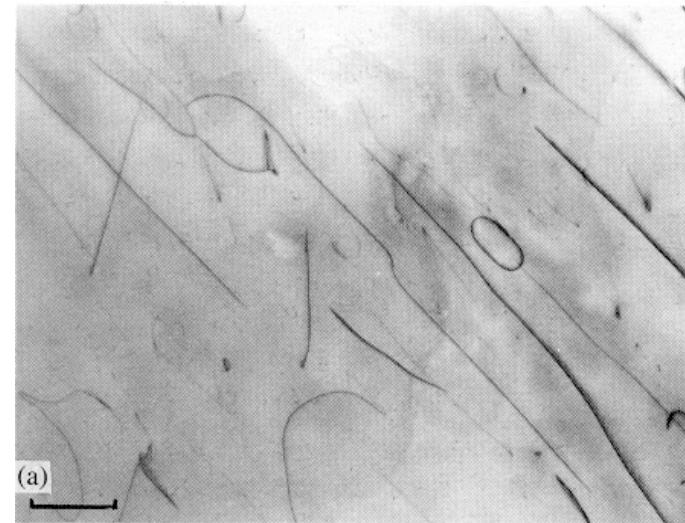
## Basics- Defects

### □ Line Defect (dislocation)

- diffraction contrast



diffracted beam-dark field image  
-dislocation-bright



## Basics- Surface

### □ Surface Energy

$$dG = -SdT + VdP + \sum_i \mu_i dn_i + \gamma dA$$

$$\gamma = \left( \frac{\partial G}{\partial A} \right)_{P,T,n_i} : \text{surface energy } [\text{Jm}^{-2}]$$

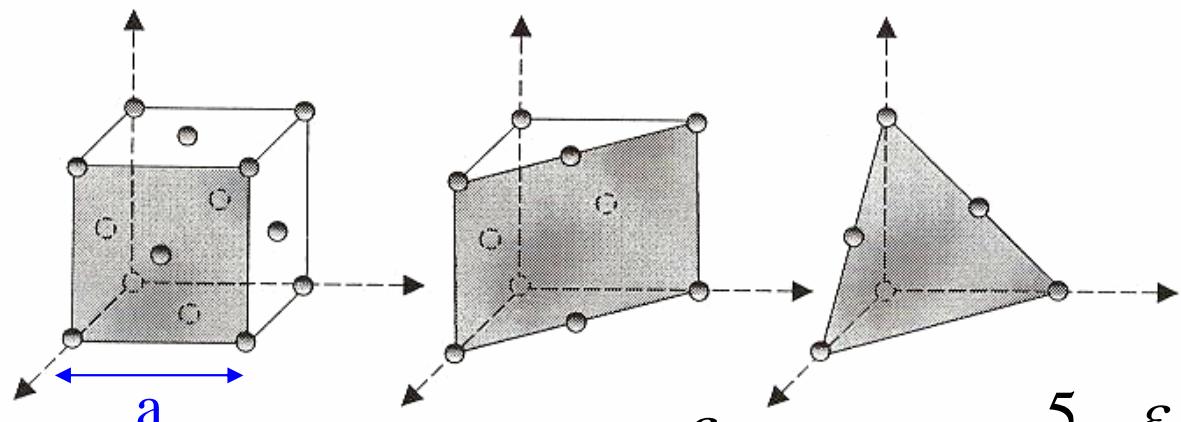
### □ Estimation of surface energy

$$\gamma = \frac{1}{2} N_b \varepsilon \rho_a \quad N_b : \# \text{ of broken bonds, } \rho_a : \text{surface atomic density}$$

### □ Anisotropy

FCC

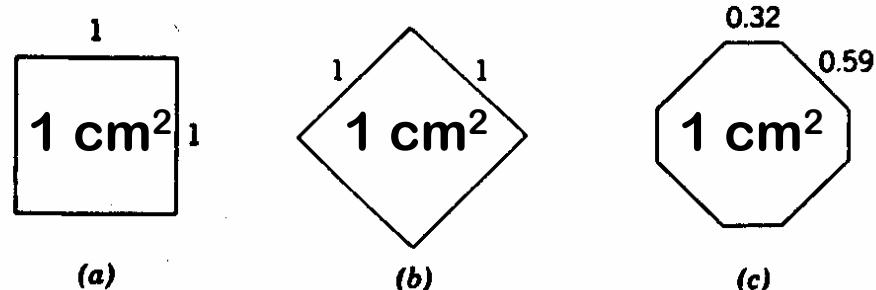
$$\gamma_{\{100\}} = \frac{1}{2} \frac{2}{a^2} 4\varepsilon = \frac{4\varepsilon}{a^2}$$



$$\gamma_{\{111\}} = 2\sqrt{3} \frac{\varepsilon}{a^2} \quad \gamma_{\{110\}} = \frac{5}{\sqrt{2}} \frac{\varepsilon}{a^2}$$

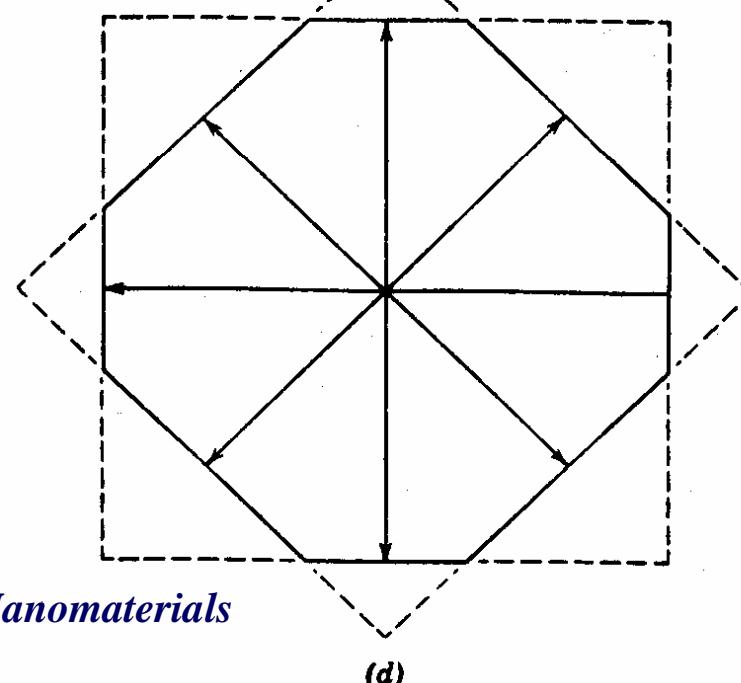
## Basics- Surface

### □ Anisotropy of Surface Energy



$$(4 \times 0.32 \times 250) + (4 \times 0.59 \times 225) = 851 \text{ erg}$$

$$4 \times 1 \times 250 = 1000 \text{ erg} \quad 4 \times 1 \times 225 = 900 \text{ erg}$$



### □ Equilibrium shape

- total surface energy minimization

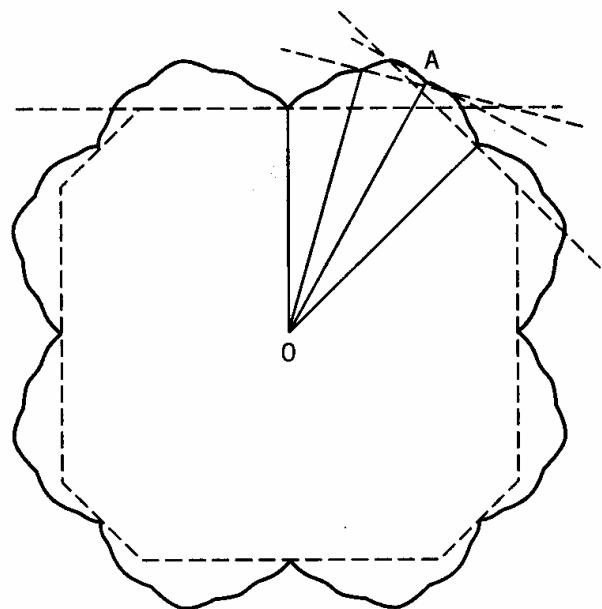
## Basics- Surface

### □ Wulff plot

-  $\gamma$ -plot- curved outer line

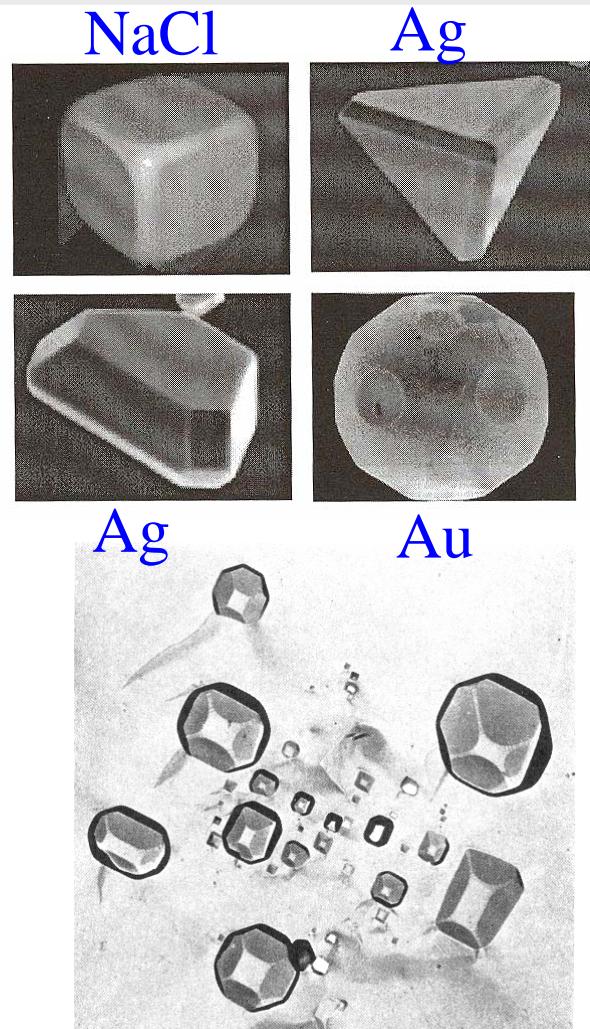
$$\gamma_i = Ch_i$$

- crystal shape- inner envelope



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G. CaO, Nanostructures & Nanomaterials (2004)



Negative crystal  $\text{UO}_2$

Y.-M Chiang, Physical Ceramics (1997)

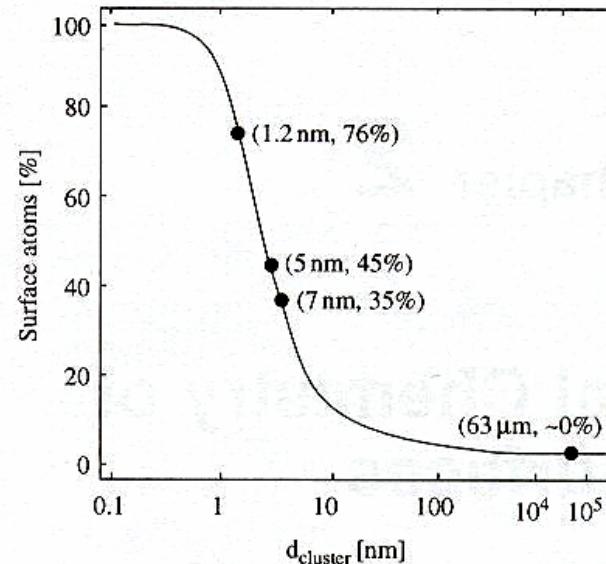
## Basics- Surface

### □ Surface Energy

**Table 5.2** Measured Surface Energies of Various Materials in Vacuum or Inert Atmospheres

	Temperature (°C)	Surface Energy (ergs/cm <sup>2</sup> )
<b>Solids</b>		
Copper	1080	1430
Silver	750	1140
Iron ( $\gamma$ phase)	1350	2100
Platinum	1300	2200
$\text{Al}_2\text{O}_3$	1850	905
TiC	1100	1190
0.20 $\text{Na}_2\text{O}$ -0.80 $\text{SiO}_2$	1350	380
$\text{NaCl}$ (100)	25	300
$\text{LiF}$ (100)	-196 (77 K)	340
$\text{CaF}_2$ (111)	-196 (77 K)	500
$\text{BaF}_2$ (111)	-196 (77 K)	280
$\text{CaCO}_3$ (1010)	25	230
$\text{MgO}$ (100)	-196 (77 K)	1500
$\text{Al}_2\text{O}_3$ (101̄2) (10̄10)	25	6000
(0001)	25	7300
$\text{MgAl}_2\text{O}_4$ (100) (111)	25	>40,000
$\text{SiC}$ (11̄20)	25	3000
Si (111)	—	5000
Ge (111)	—	20,000
		1230
		1060

### □ Surface Energy vs. size



**Table 2.1.** Variation of surface energy with particle size.<sup>22</sup>

Side (cm)	Total surface area (cm <sup>2</sup> )	Total edge (cm)	Surface energy (J/g)	Edge energy (J/g)
0.77	3.6	9.3	$7.2 \times 10^{-5}$	$2.8 \times 10^{-12}$
0.1	28	550	$5.6 \times 10^{-4}$	$1.7 \times 10^{-10}$
0.01	280	$5.5 \times 10^4$	$5.6 \times 10^{-3}$	$1.7 \times 10^{-8}$
0.001	$2.8 \times 10^3$	$5.5 \times 10^6$	$5.6 \times 10^{-2}$	$1.7 \times 10^{-6}$
$10^{-4}$ (1 μm)	$2.8 \times 10^4$	$5.5 \times 10^8$	0.56	$1.7 \times 10^{-4}$
$10^{-7}$ (1 nm)	$2.8 \times 10^7$	$5.5 \times 10^{14}$	560	170

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Y.-M Chiang, Physical Ceramics (1997) G. CaO, Nanostructures & Nanomaterials (2004)

## Basics- Surface

### □ Curved Surface

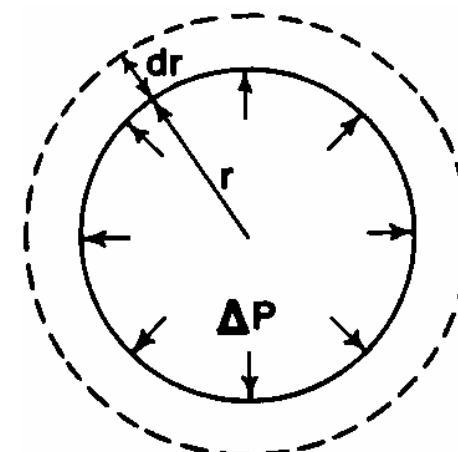
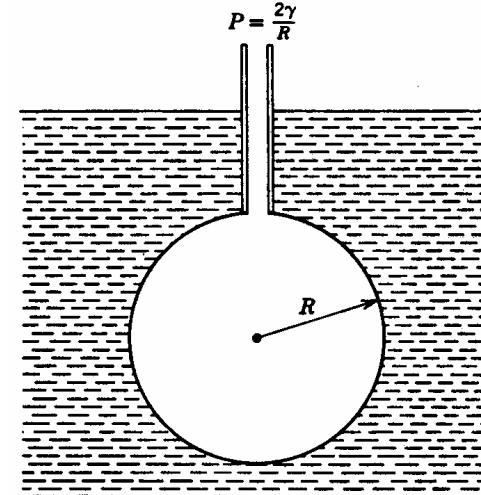
$$\Delta P dV = \gamma dA$$

$$dV = 4\pi r^2 dr \quad dA = 8\pi r dr \text{ for sphere}$$

$$\Delta P = \gamma \frac{dA}{dV} = \frac{2\gamma}{r}$$

$$\text{Laplace equation } \Delta P = \gamma \left( \frac{1}{r_1} + \frac{1}{r_2} \right)$$

$r_1, r_2$ : principal radii of curvature



## Basics- Surface

### □ Curved Surface

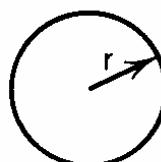
- pressure difference across a curved surface → a change in solubility or vapor pressure
- at constant T, P, n<sub>i</sub>, transfer one mole from flat to curved

$$\mu = \mu_o + RT \ln a = \mu_o + RT \ln c = \mu_o + RT \ln P$$

$$\Delta\mu = RT \ln \frac{P(r)}{P(r=\infty)} = \gamma dA = \gamma 8\pi r dr, V_m = \frac{4}{3}\pi r^3$$

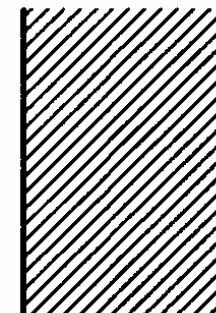
$$P(r) = P(r=\infty) \exp\left[\frac{2\gamma V_m}{RT} \left(\frac{1}{r}\right)\right]$$

: Kelvin equation



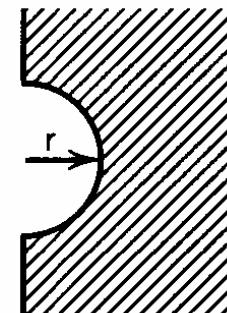
(a)

$$r > 0$$



(b)

$$r = \infty$$



(c)

$$r < 0$$

$$p_1$$

&gt;

$$p_2$$

&gt;

$$p_3$$

## Basics- Surface

### □ Solubility

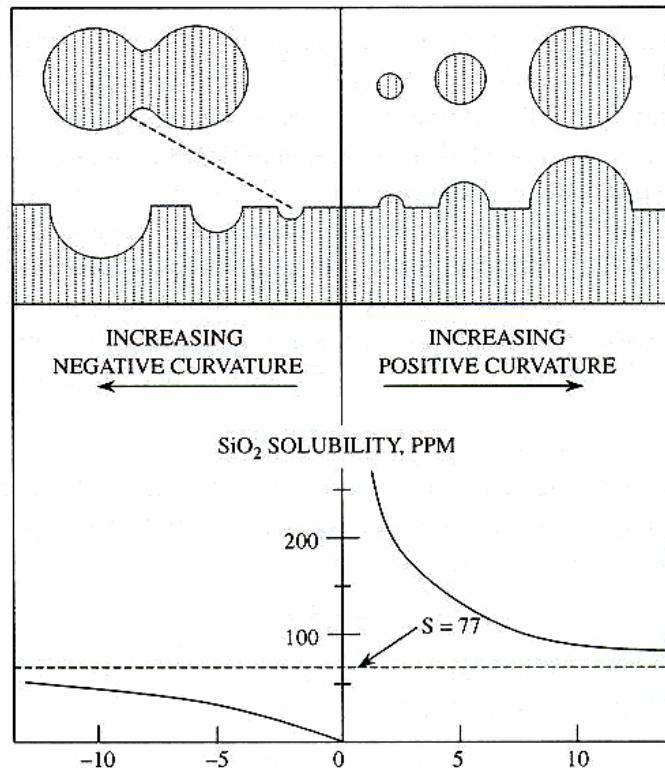


Fig. 2.11. Variation in solubility of silica with radius of curvature of surface. The positive radii of curvature are shown in cross-section as particles and projections from a planar surface; negative radii are shown as depressions or holes in the surface, and in the crevice between two particles. [R.K. Iler, *The Chemistry of Silica*, Wiley, New York, 1979.]

### □ Vapor Pressure

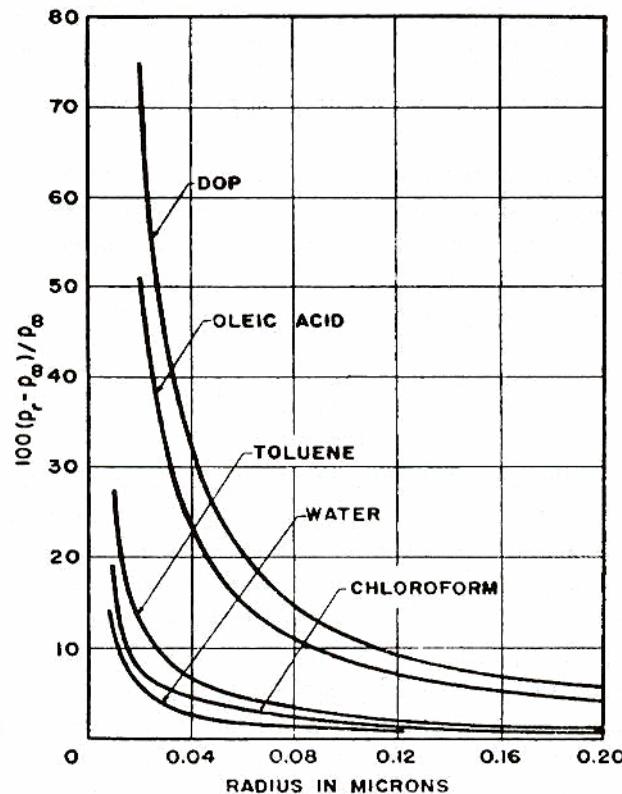


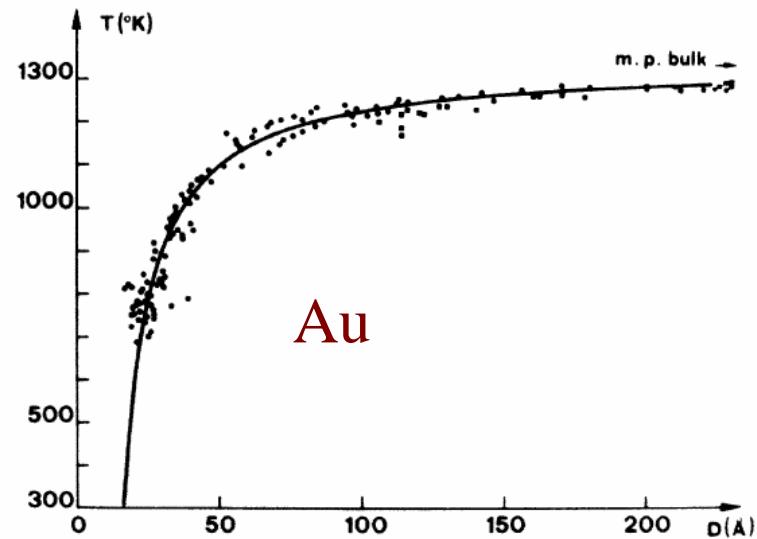
Fig. 2.12. Vapor pressure of a number of liquids as a function of droplet radius. [V.K. La Mer and R. Gruen, *Trans. Faraday Soc.* **48**, 410 (1952).]

## Basics- Surface

### □ Melting Point

- melting point  $\downarrow \Leftrightarrow$  particle size  $\downarrow$

$$T_b - T_m = \left[ \frac{2T_b}{\Delta H \rho_s r_s} \right] \left[ \gamma_s - \gamma_l \left( \frac{\rho_s}{\rho_l} \right)^{2/3} \right]$$

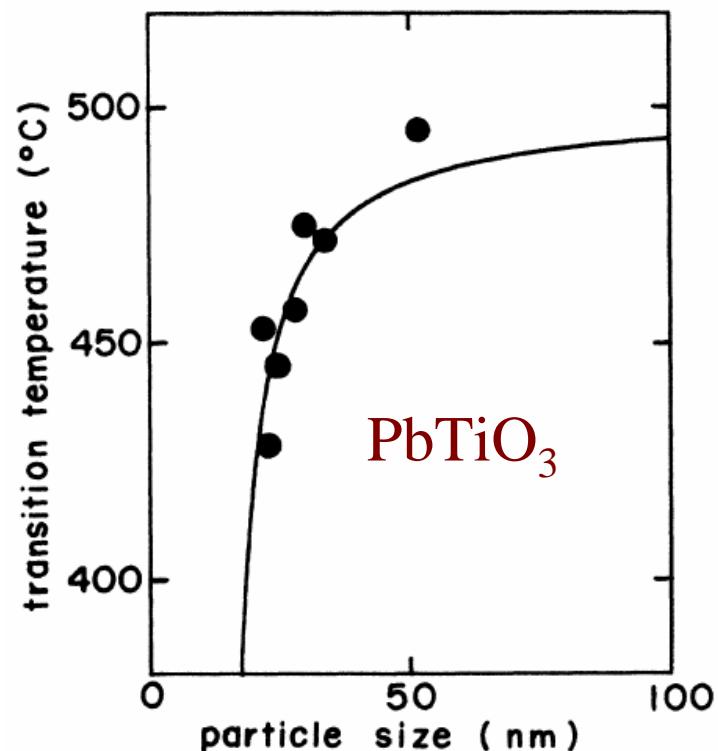


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Ph. Buffat, Phys. Rev., A13 (1976) 2287

### □ Transition Temperature

$$T_c = 500 - 588.5/(D - 12.6) \text{ (°C)}$$

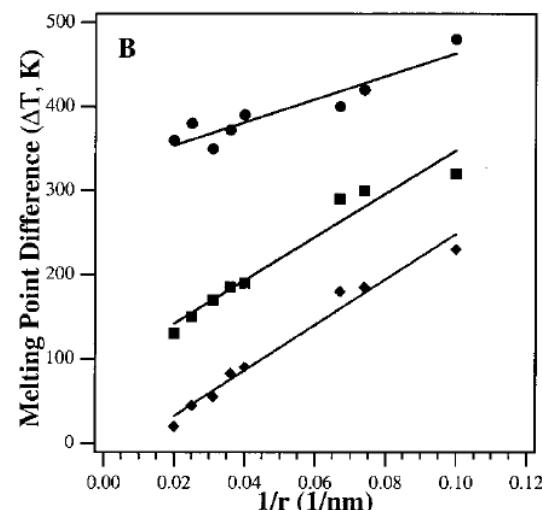
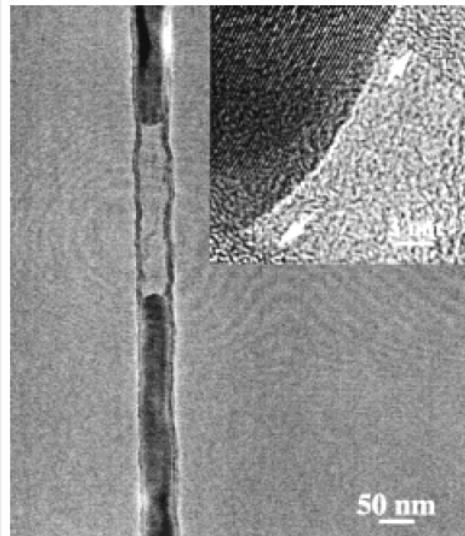


K. Ishikawa, Phys. Rev., B37 (1976) 5852

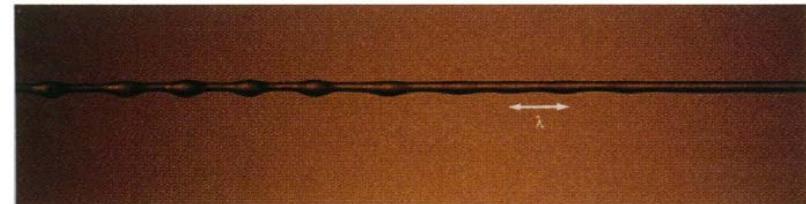
## Basics- Surface

### □ Melting Point of nano-wire

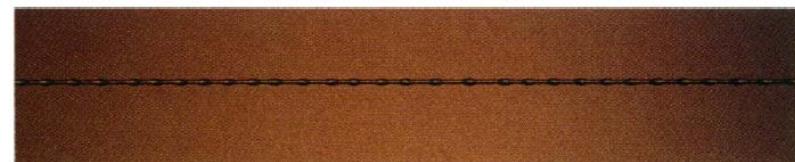
- Ge nano-wire



- Rayleigh instability



**Fig. 1.** Film of honey on a fishing line of radius  $b = 140 \mu\text{m}$ . The film is made by drawing the thread horizontally out of a large drop of honey. The instability is fully developed on the left, but still not on the right where the optimal wavelength  $\lambda$  has not yet been selected.

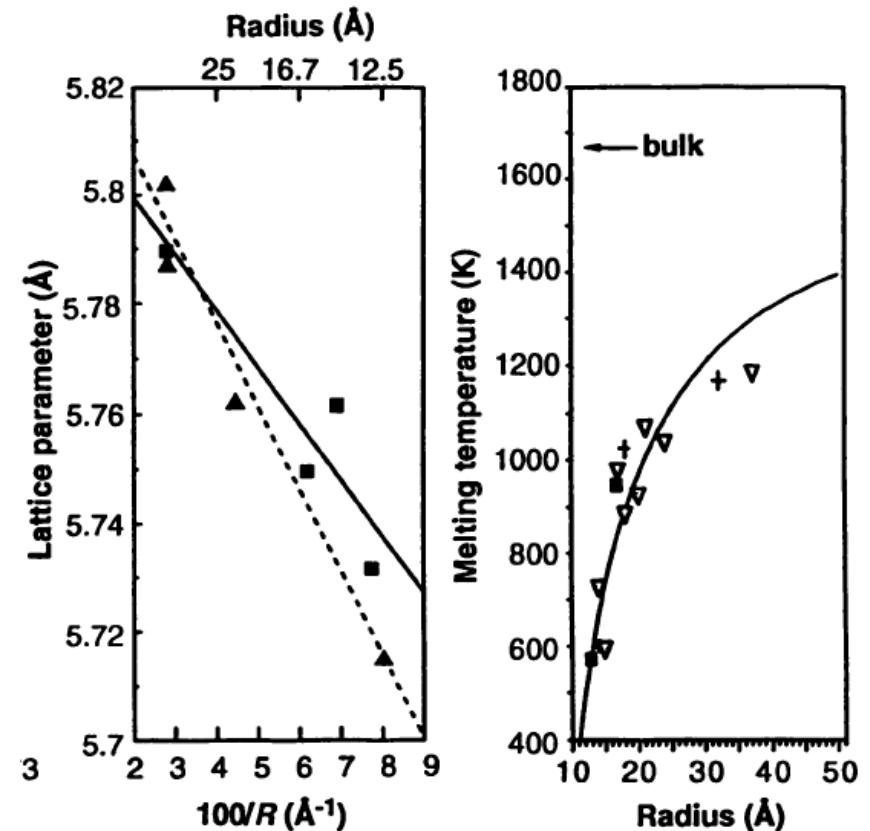
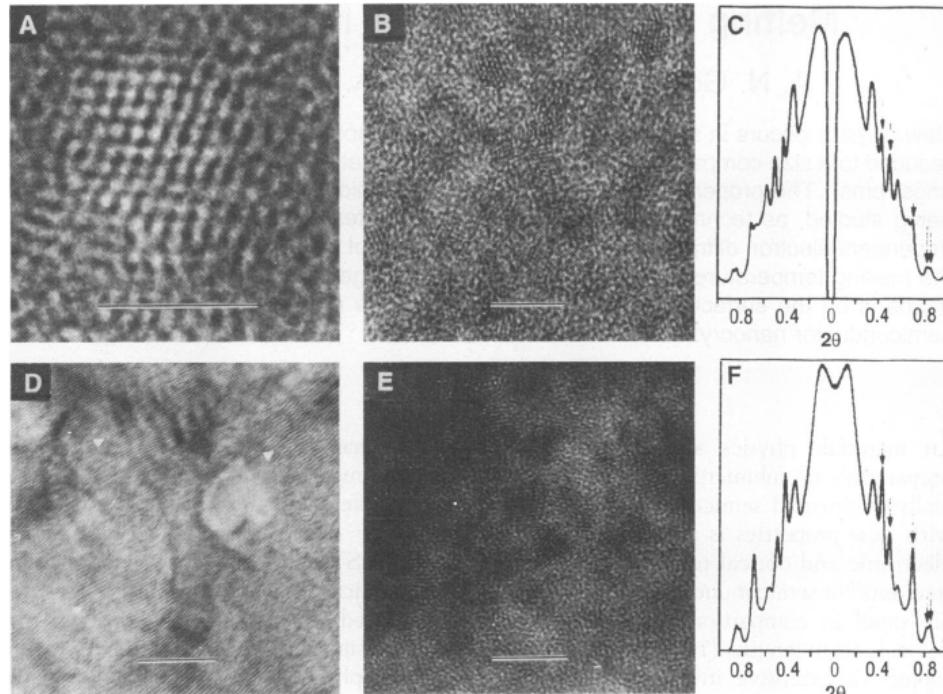


**Fig. 2.** Drops resulting from a thin film of silicone oil coating a nylon fiber (radius  $b = 80 \mu\text{m}$ ). The spacing between the droplets is regular and is about  $10b$  (as shown by Rayleigh) for thin films on fibers.

## Basics- Surface

### □ Melting Point & Lattice Parameter

- CdS nanocrystal



## Basics- Surface

### □ Ostwald Ripening

$$C_r = C_o \left(1 + \frac{2\sigma V_m}{RT} \frac{1}{r}\right)$$

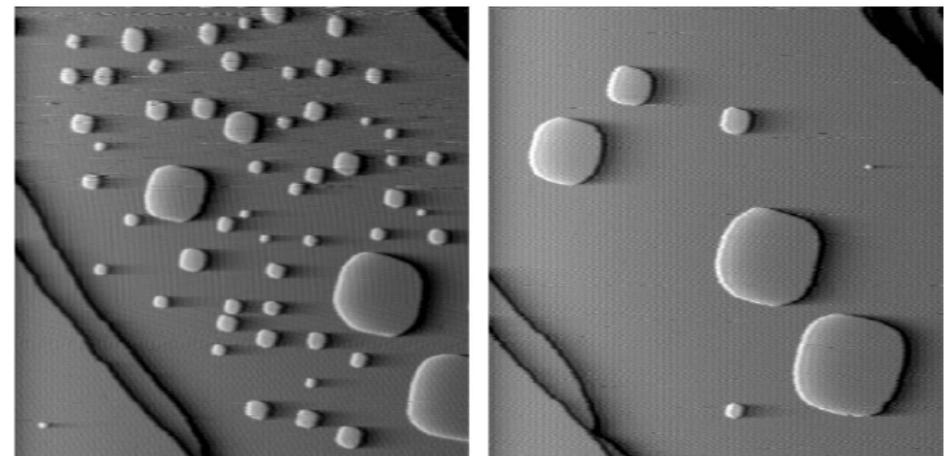
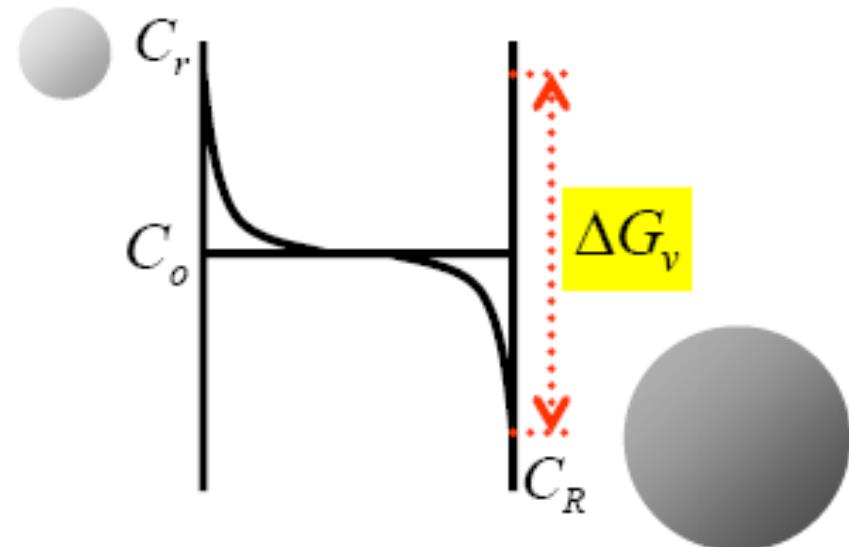


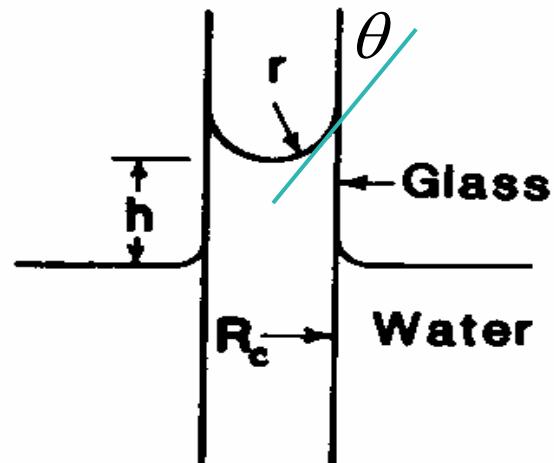
FIG. 2. Two  $300 \text{ nm} \times 300 \text{ nm}$  STM images, separated in time by 20 000 s, showing island ripening on Cu(001) at 343 K.

## Basics- Surface

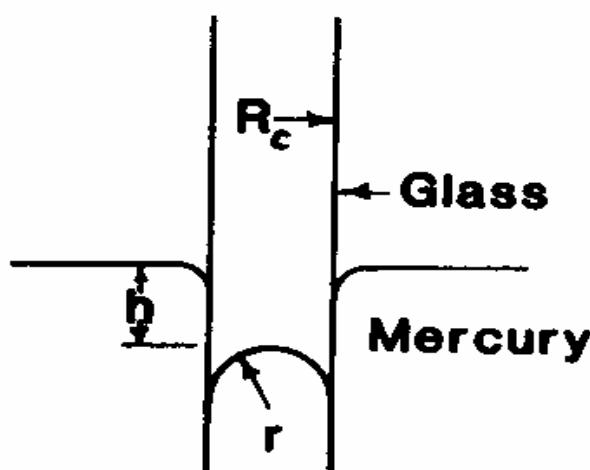
### □ Capillary Rise

$$\Delta P = \gamma \left( \frac{2}{r} \right) = \gamma \left( \frac{2 \cos \theta}{R_c} \right) = \rho g h \quad \gamma = \frac{R \rho g h}{2 \cos \theta}$$

$0 < \theta < 90$



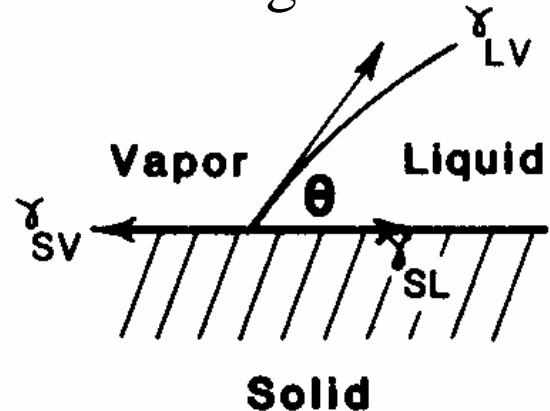
$90 < \theta < 180$



## Basics- Surface

### □ Wetting & Spreading

- contact angle

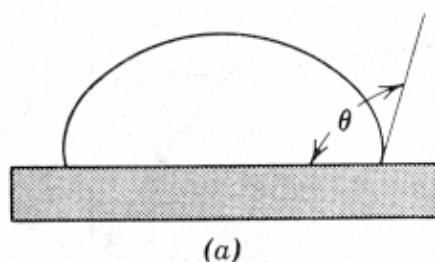


- wetting

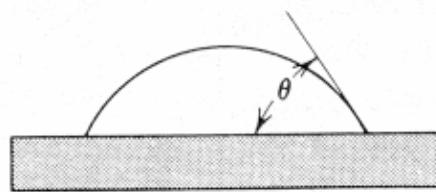
$$\gamma_{SV} = \gamma_{SL} + \gamma_{LV} \cos \theta$$

θ: contact angle

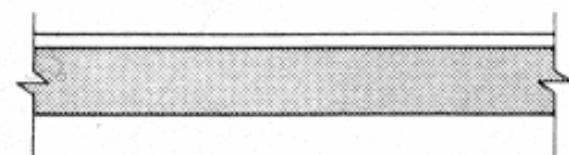
Young equation



nonwetting  $\theta > 90$



wetting  $\theta < 90$

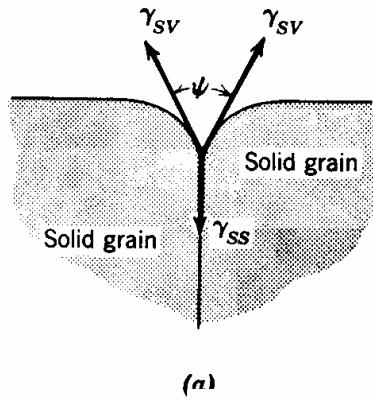


spreading  $\theta = 90$

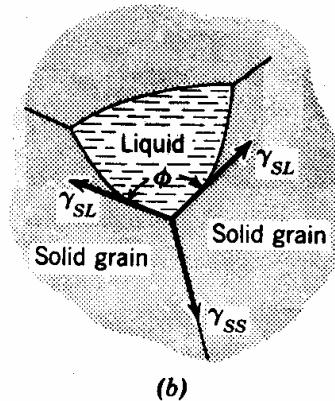
## Basics- Surface

### □ Wetting

- grain boundary



(a)



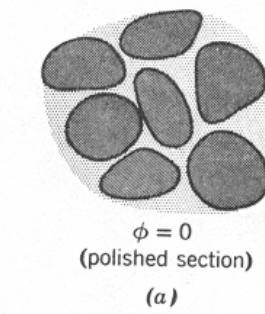
(b)

$$\gamma_{ss} = 2\gamma_{sv} \cos \frac{\psi}{2}$$

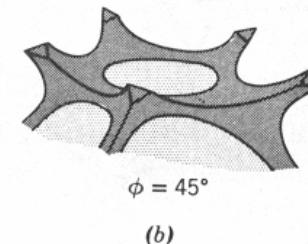
$$\gamma_{ss} = 2\gamma_{sl} \cos \frac{\phi}{2}$$

$$\cos \frac{\phi}{2} = \frac{1}{2} \frac{\gamma_{ss}}{\gamma_{sl}}$$

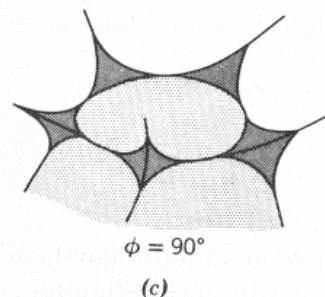
$\phi$ : dihedral angle



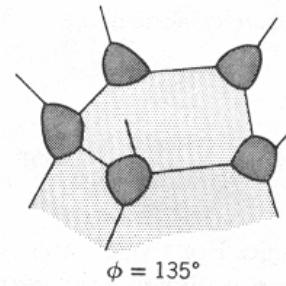
$\phi = 0$   
(polished section)  
(a)



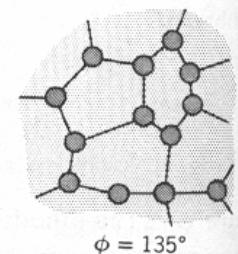
$\phi = 45^\circ$   
(b)



$\phi = 90^\circ$   
(c)



$\phi = 135^\circ$   
(d)



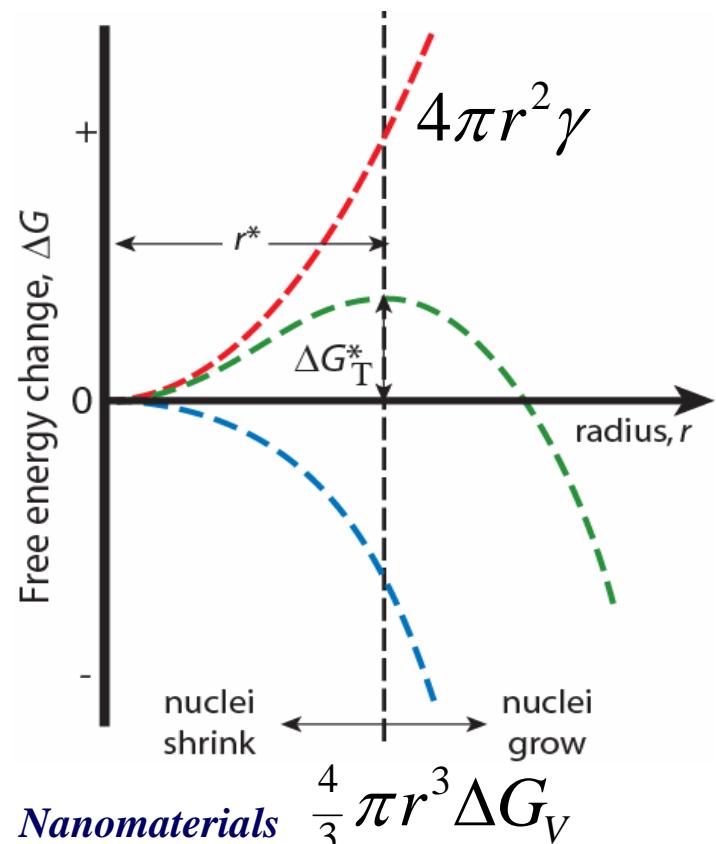
$\phi = 135^\circ$   
(polished section)  
(e)

Fig. 5.27. Second-phase distribution for different values of the dihedral angle.

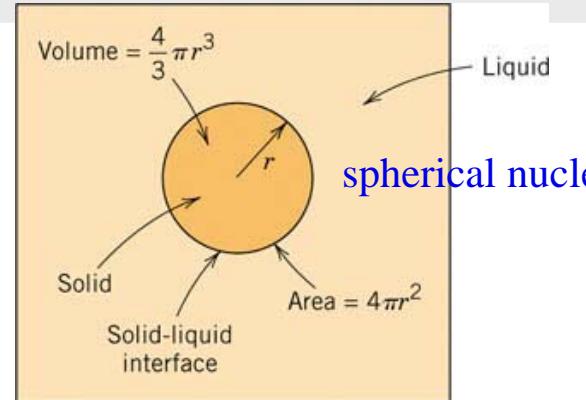
## Basics- Kinetics

### □ Homogeneous Nucleation

- supersaturation ( $\Delta G_V$ )
- surface



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$$\Delta G = \frac{4}{3}\pi r^3 \Delta G_V + 4\pi r^2 \gamma$$

critical radius  $r^*$

$$\left( \frac{d(\Delta G)}{dr} \right)_{r=r^*} = 0 \rightarrow r^* = \frac{-2\gamma}{\Delta G_V}$$

barrier height  $\Delta G^*$

$$\Delta G^* = \frac{16\pi r^3}{3\Delta G_V}$$

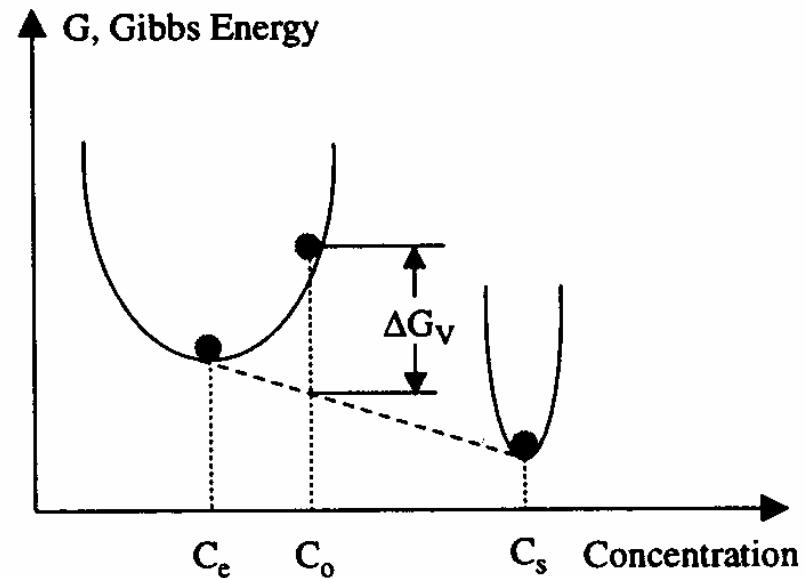
## Basics- Kinetics

### □ Homogeneous Nucleation

ex) supersaturated solution

$$\Delta G_v = -\frac{kT}{\Omega} \ln(C/C_o) = -\frac{kT}{\Omega} \ln(1+\sigma)$$

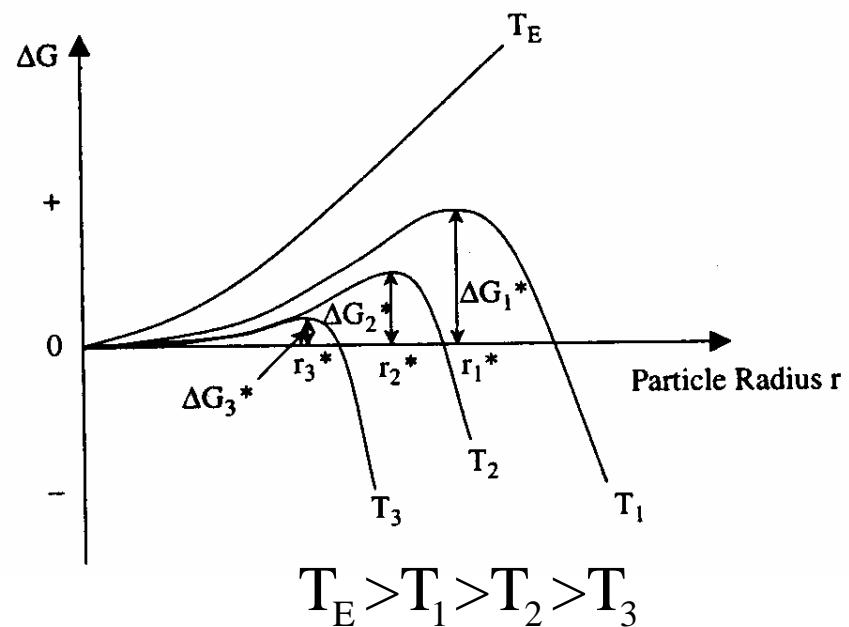
$$\sigma = (C - C_o)/C_o$$



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■ For nanoparticles

- $\Delta G_v \uparrow \leftarrow \sigma \uparrow \leftarrow T \downarrow$
- $\gamma \downarrow$

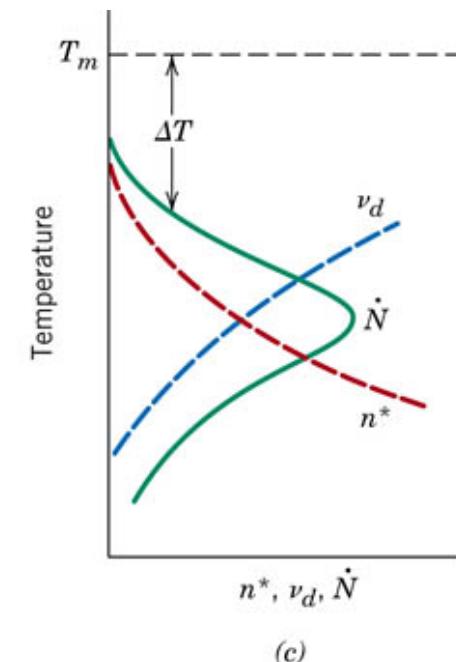
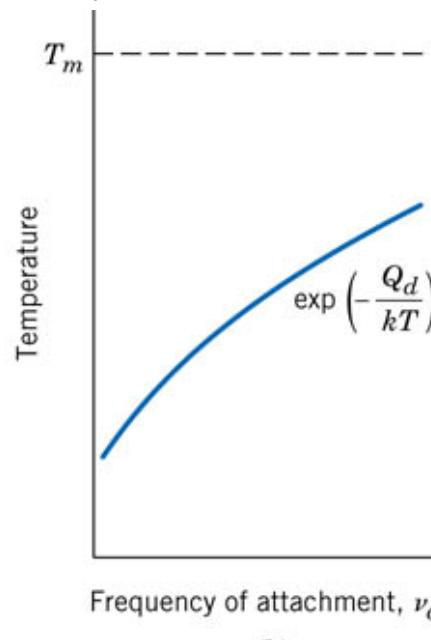
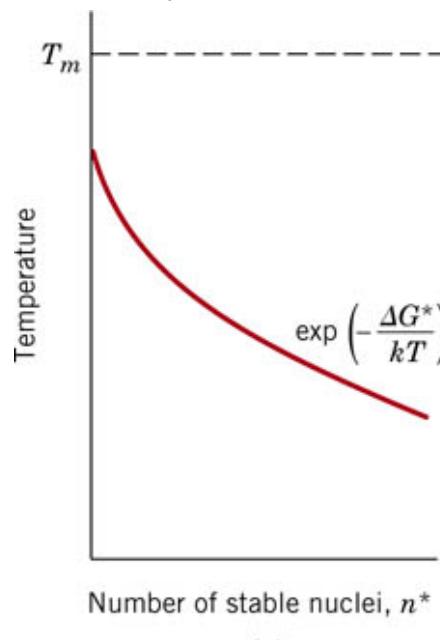


## Basics- Kinetics

### □ Nucleation Rate

$$\dot{N} \sim n^* (\# \text{ of stable nuclei}) \times v_d \text{ (collision frequency)}$$

$$\sim K_1 \exp\left(-\frac{\Delta G^*}{kT}\right) K_2 \exp\left(-\frac{Q_d}{kT}\right)$$



## Basics- Kinetics

### □ Mono-dispersed Particle - Lamar diagram

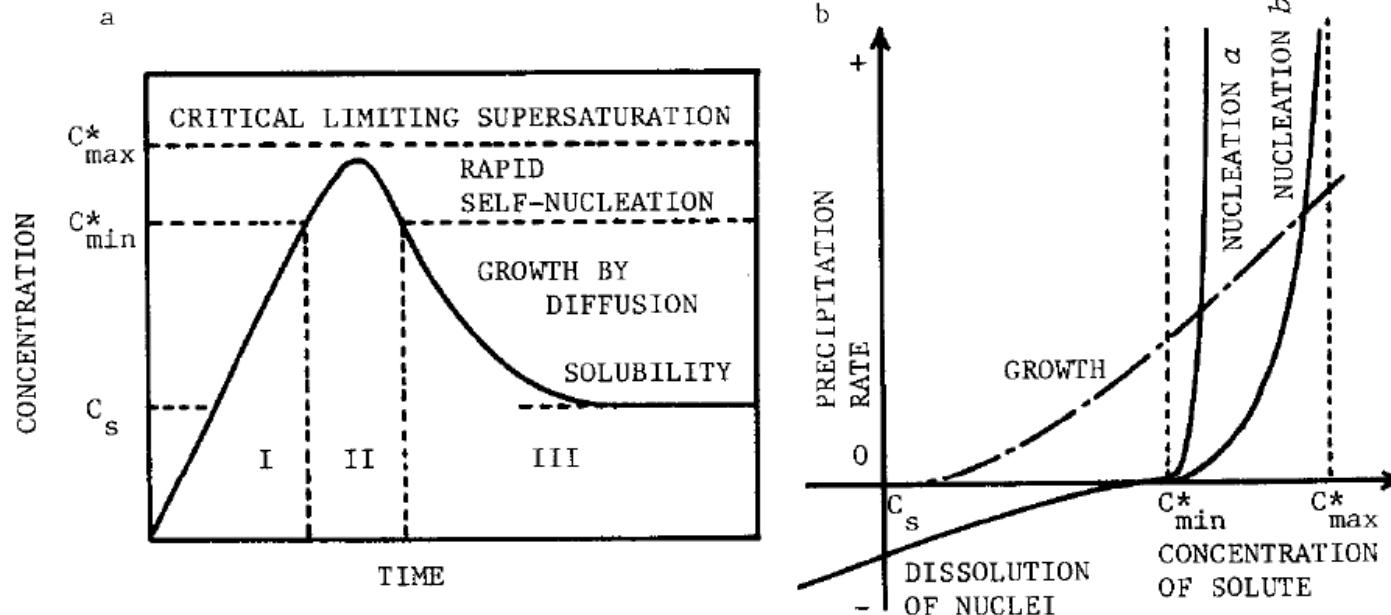


Fig. 1. (a) The LaMer model for monodispersed particle formation ( $C_s$ : solubility;  $C^*_{\min}$ : minimum concentration for nucleation;  $C^*_{\max}$ : maximum concentration for nucleation; I: prenucleation period; II: nucleation period; III: growth period) (ref. 15). (b) Precipitation rate for nucleation and growth as a function of solute concentration, where the growth curve is the one for a given amount of seed particles.

## Basics- Kinetics

### □ Homogeneous Nucleation

- subsequent growth

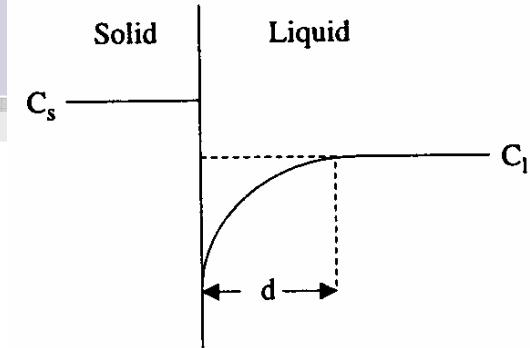
(1) diffusion controlled

$$\frac{dr}{dt} = D(C - C_s) \frac{V_m}{r}$$

$$r^2 = 2D(C - C_s)V_m t + r_o^2$$

$$r^2 = k_D t + r_o^2$$

$$\delta r = \frac{r_o(\delta r_o)}{r} = \frac{r_o(\delta r_o)}{\sqrt{k_D t + r_o^2}}$$



(2) interface controlled

(2-1) monolayer growth

$$\frac{dr}{dt} = k_m r^2; \frac{1}{r} = \frac{1}{r_o} - k_m t$$

$$\delta r = r^2 \left( \frac{\delta r_o}{r_o^2} \right)$$

(2-2) multilayer growth

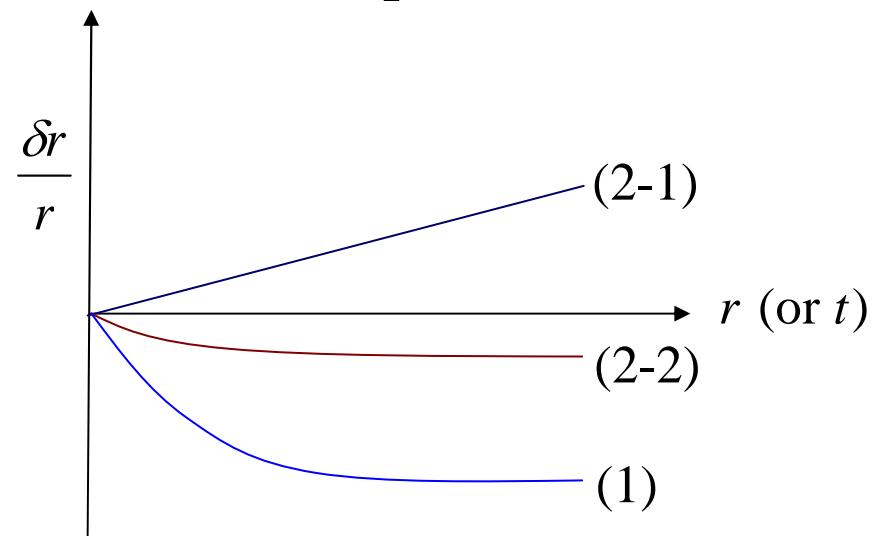
$$\frac{dr}{dt} = k_p; r = k_p t + r_o$$

$$\delta r = \delta r_o$$

## Basics- Kinetics

### □ Homogeneous Nucleation

- for the uniformity in size  
→ diffusion controlled process is desired



- how to achieve it

→ extremely low concentration of growth species  
high viscosity, diffusion barrier, controlled supply of  
growth species

## Basics- Kinetics

### □ Homogeneous Nucleation

ex) ZnS

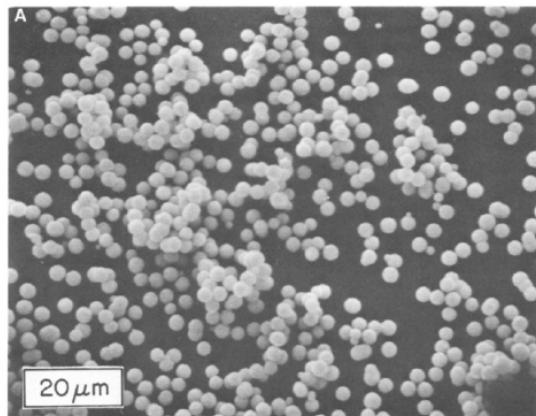
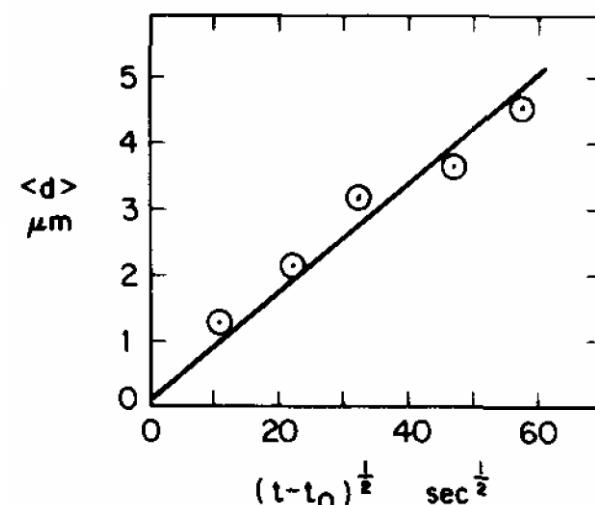
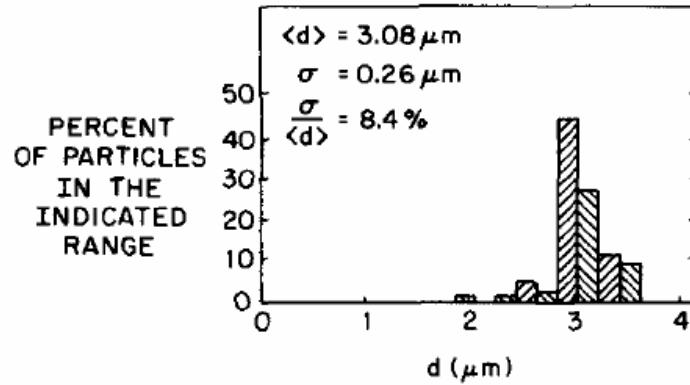


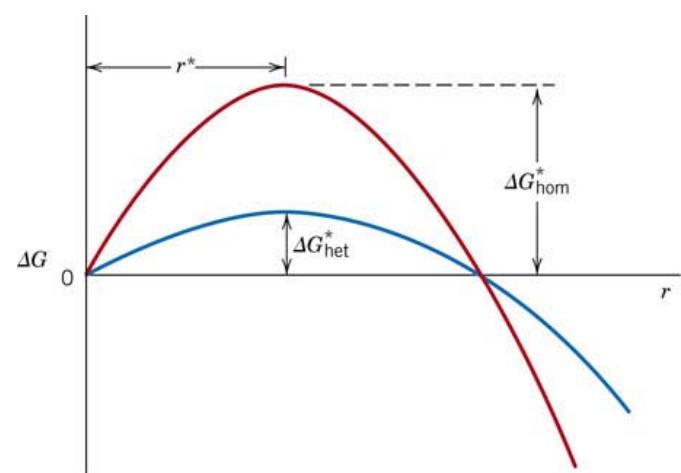
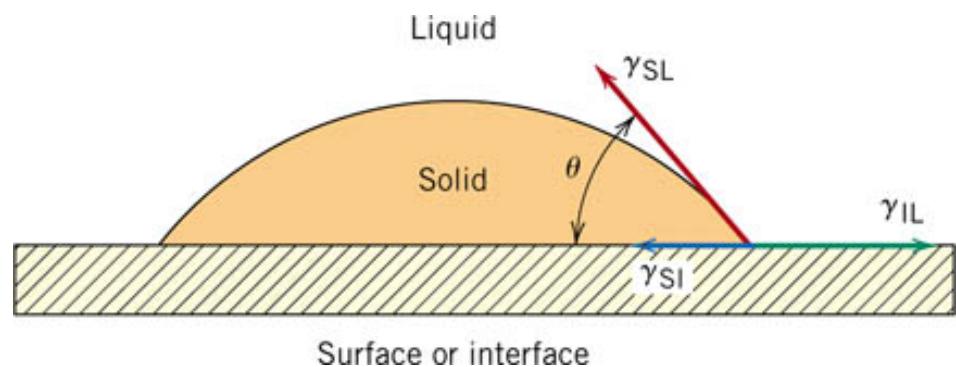
FIG. 1. (A) Particles of ZnS growth under conditions that minimize stirring and convection.  $Zn^{2+}$  initial concentration = 0.05 mole/liter. Thioacetamide initial concentration = 0.21 mole/liter.  $H_2SO_4$  initial concentration = 0.15 mole/liter. Scanning electron microscope photo taken at 1000 $\times$  magnification. Temperature = 80°C.

- diffusion of the HS- ion to the growing particle is the rate-limiting process



## Basics- Kinetics

### □ Heterogeneous Nucleation

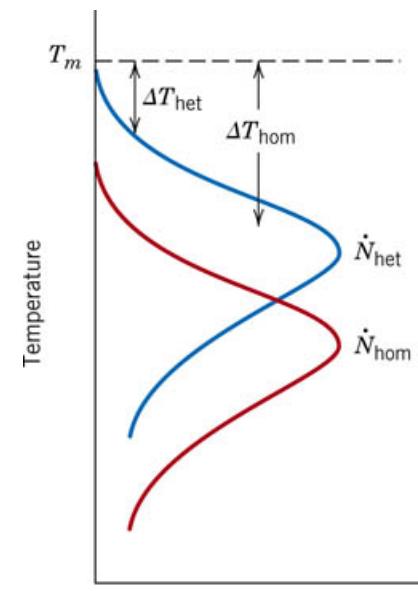


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$$r_{heter}^* = r_{hom}^* \frac{\sin^2 \theta \cos \theta + 2 \cos \theta - 2}{2 - 3 \cos \theta + \cos^3 \theta}$$

$$\Delta G_{heter}^* = \Delta G_{hom}^* f(\theta)$$

$$f(\theta) = \frac{2 - 3 \cos \theta + \cos^3 \theta}{4}$$



W. D. Callister, Materials Science and Engineering

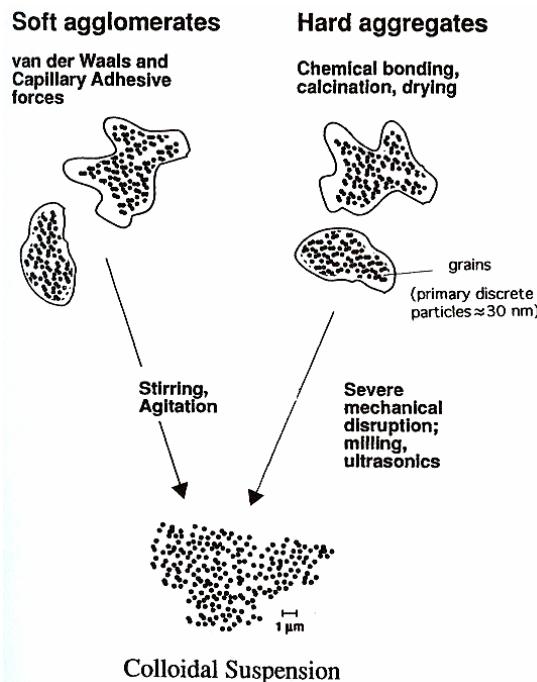
# Basics- Surface chemistry

## □ Dispersion

- a high solids homogeneous suspension with a well-defined rheological behavior

## □ Dispersion in liquid

- wetting - interparticle interaction



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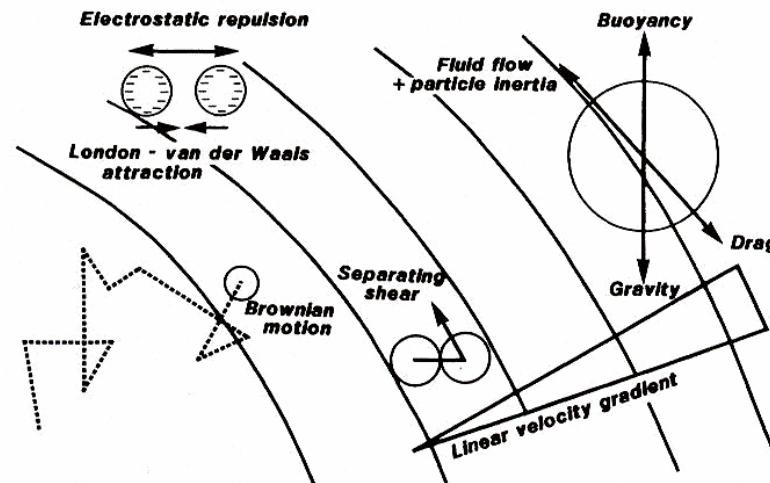
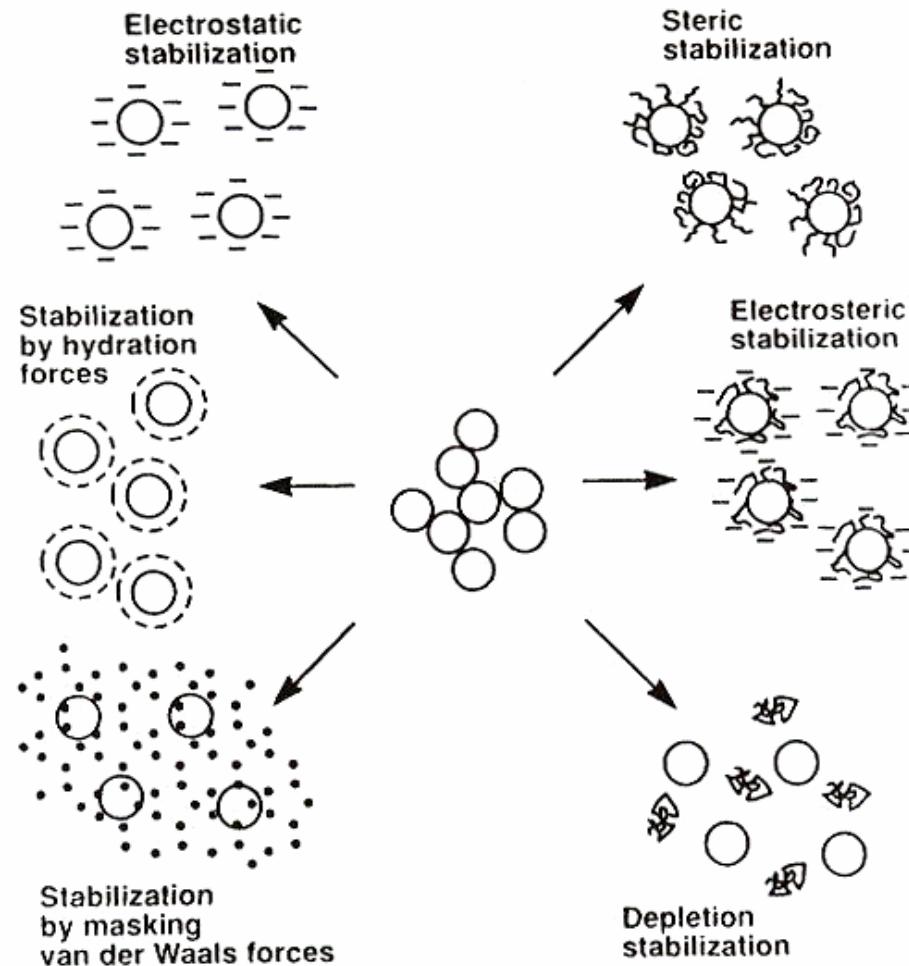


TABLE 2 Energies of Particles in Suspension due to Various Interactions

Type of interaction	Energy (in units of $kT$ ) for particles of given size		
	0.1 μm	1 μm	10 μm
van der Waals attraction	10	100	1000
Electrostatic repulsion	0-100	0-1000	0-10,000
Brownian motion	1	1	1
Kinetic energy of sedimentation	$10^{-13}$	$10^{-6}$	10
Kinetic energy of stirring	1	1000	$10^6$

# Basics- Surface chemistry

## □ Stabilization



## Basics- Surface chemistry

### □ Electrostatic Stabilization

electrical repulsion

van der waals attraction

### □ Electrical double layer

- surface charge ( $\Psi_o$ )

- stern potential ( $\Psi_d$ )

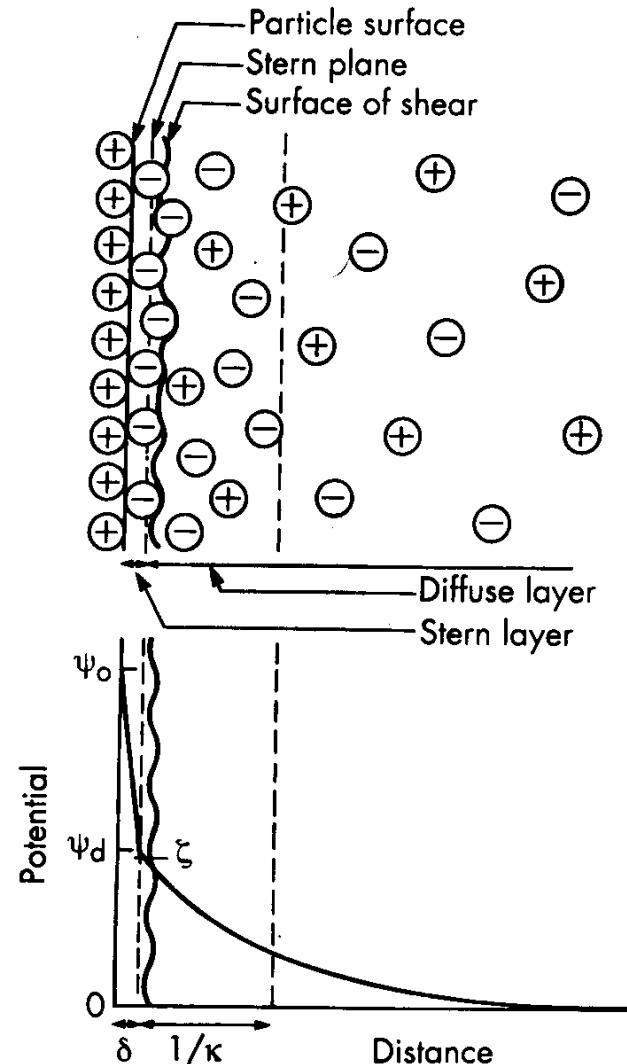
- specific adsorption of counter-ion

- electrokinetic or  $\zeta$ (zeta) potential at plane of shear (slipping plane)

- double layer thickness ( $\kappa^{-1}$ )

$$\kappa = \left( \frac{1000e^2 N_A}{\epsilon kT} \sum_i z_i^2 M_i \right)^{1/2}$$

$M_i$  : molar concentration



## Basics- Surface chemistry

### □ Van der Waals attraction

induced dipole- induced dipole (London)  $\sim x^{-6}$

permanent dipole- induced dipole (Debye)  $\sim x^{-6}$

permanent dipole- permanent dipole (Keesom)  $\sim x^{-6}$

### □ Attraction between two spheres

$$\Phi_A = \frac{-Ar}{12S} \quad A: \text{Hamaker constant}$$

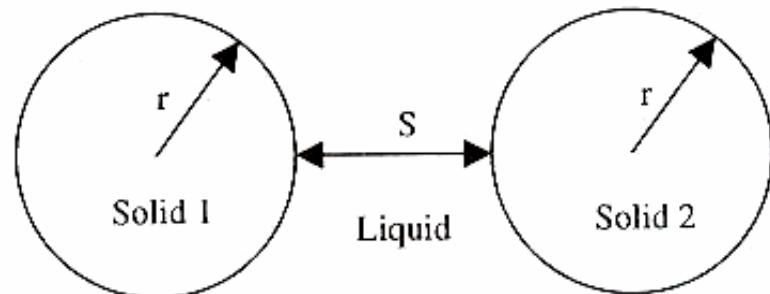


Table 2.3. Hamaker constants for some common materials.<sup>45</sup>

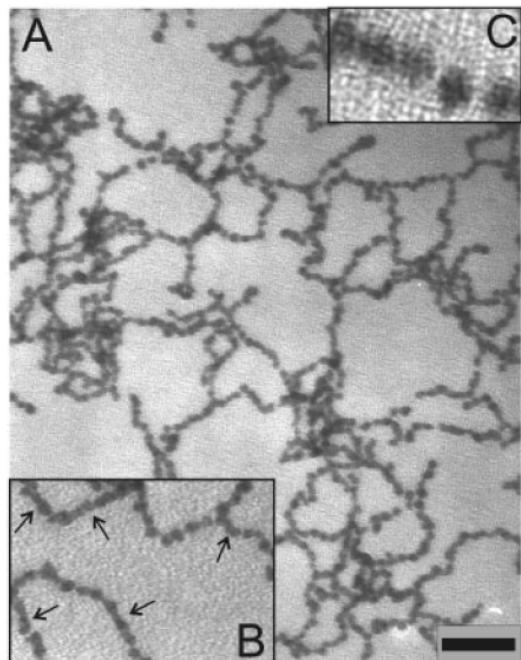
Materials	$A_i(10^{-20} \text{ J})$
Metals	16.2–45.5
Gold	45.3
Oxides	10.5–15.5
$\text{Al}_2\text{O}_3$	15.4
$\text{MgO}$	10.5
$\text{SiO}_2$ (fused)	6.5
$\text{SiO}_2$ (quartz)	8.8
Ionic crystals	6.3–15.3
$\text{CaF}_2$	7.2
Calcite	10.1
Polymers	6.15–6.6
Polyvinyl chloride	10.82
Polyethylene oxide	7.51
Water	4.35
Acetone	4.20
Carbon tetrachloride	4.78
Chlorobenzene	5.89
Ethyl acetate	4.17
Hexane	4.32
Toluene	5.40

## Basics- Surface chemistry

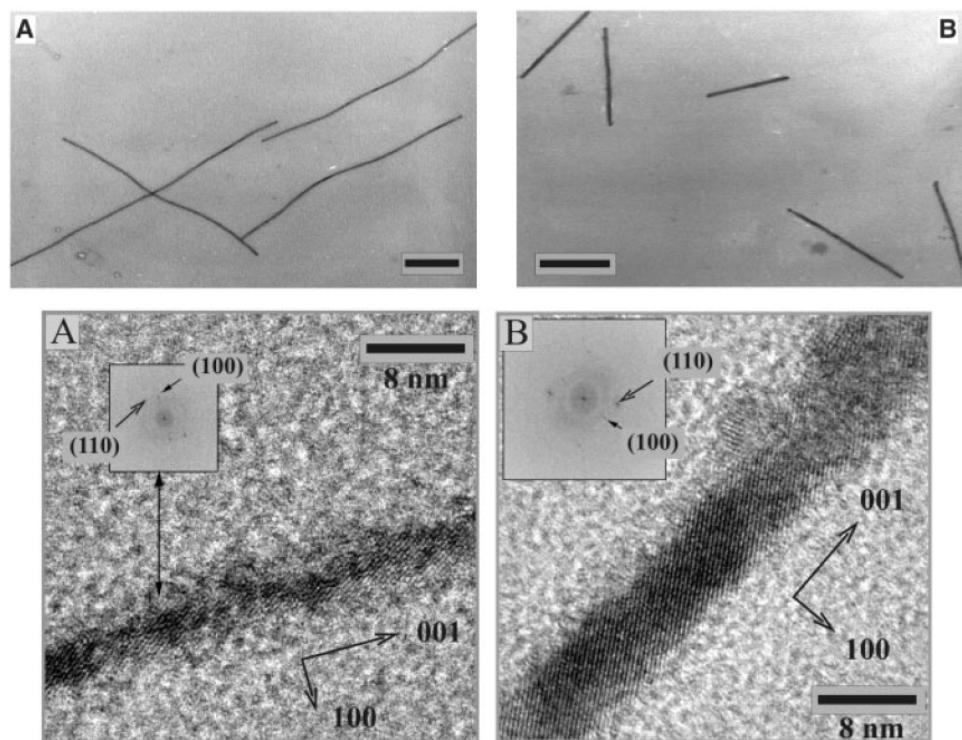
### □ Example- CdTe nanowire

dipole-dipole interaction between nano-particles produce nano-wires with self assembling modes

intermediate stage

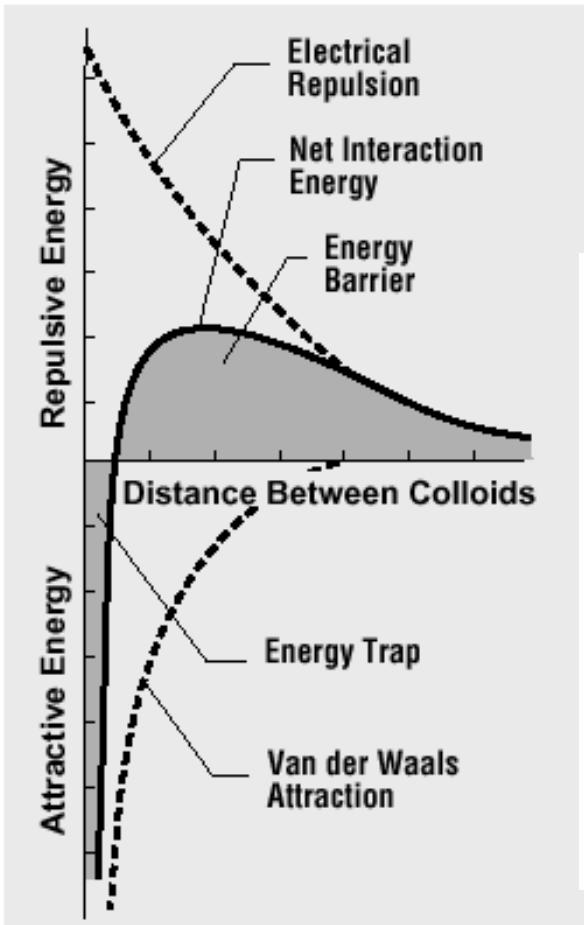


nano-wire

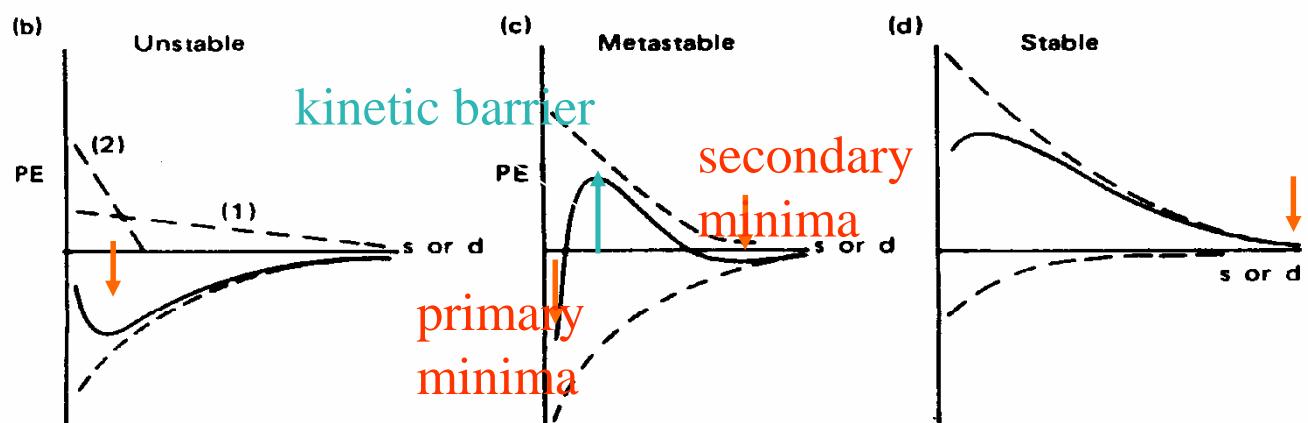


# Basics- Surface chemistry

## □ DLVO theory



$$\Phi = \Phi_R + \Phi_A$$



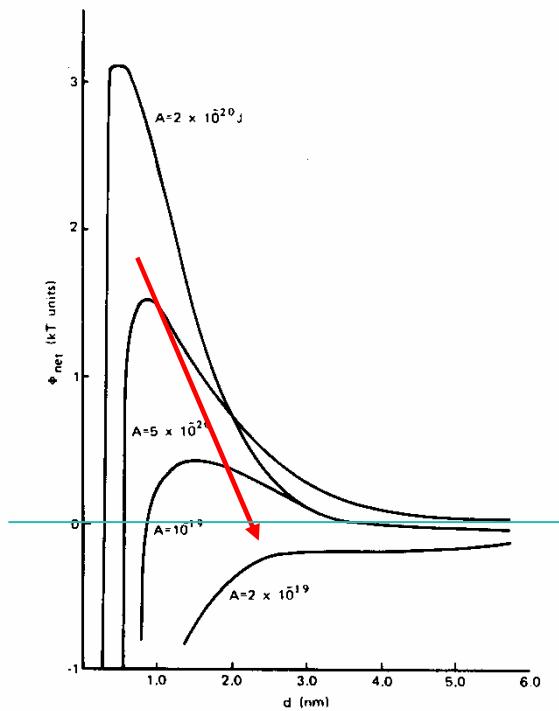
**Figure 11.1** Potential energy curves for the interaction of two colloidal particles. Negative values correspond to attraction, and positive values to repulsion. (a) Definition of variables. (b) Repulsion less than attraction in magnitude and/or range. (c) Repulsion and attraction comparable in magnitude and range. (d) Attraction less than repulsion.

# Basics- Surface chemistry

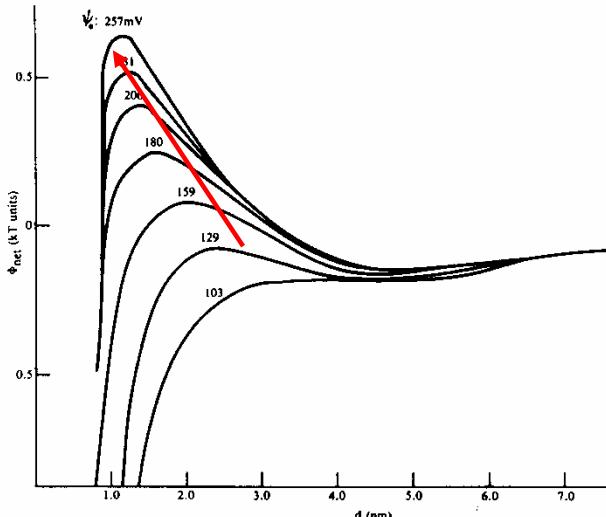
## □ DLVO theory

- effect of  
Hamaker constant      surface potential

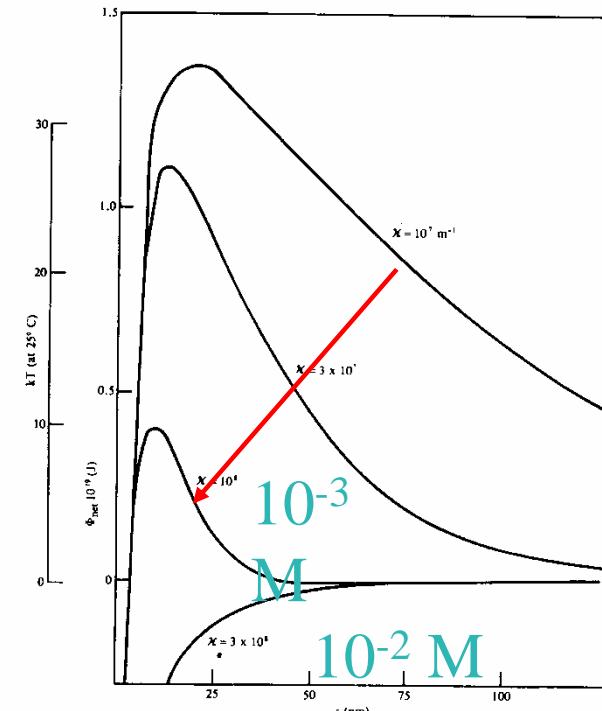
electrolyte conc.



**Figure 12.7** Plot of  $\Phi_{net}$  versus  $d$  according to Eq. (83) for flat blocks. Curves are drawn for different values of  $A_{212}$  with constant values of  $\kappa$  ( $10^9 \text{ m}^{-1}$ ) and ( $103 \text{ mV}$ ). Units of ordinate: multiples of  $kT$  at  $25^\circ\text{C}$  for an interaction area of  $\text{nm}^2$ .



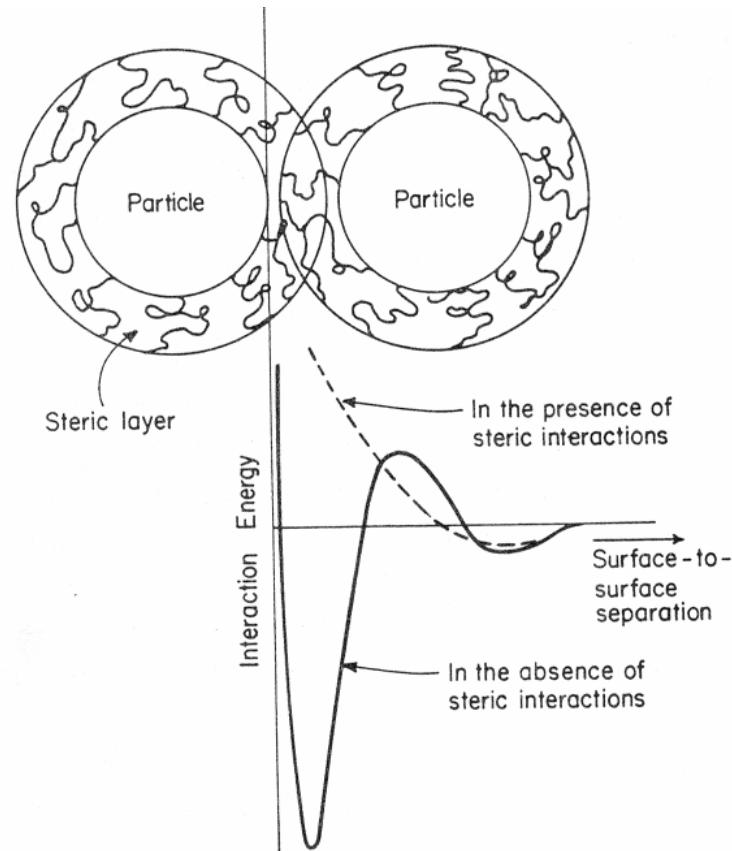
**Figure 12.8** Plot of  $\Phi_{net}$  versus  $d$  according to Eq. (83) for flat blocks. Curves are drawn for different values of  $\psi_0$  with constant values of  $\kappa$  ( $10^9 \text{ m}^{-1}$ ) and  $A$  ( $2 \times 10^{-19} \text{ J}$ ). Units of ordinate: multiples of  $kT$  at  $25^\circ\text{C}$  for an interaction area of  $4.0 \text{ nm}^2$ .



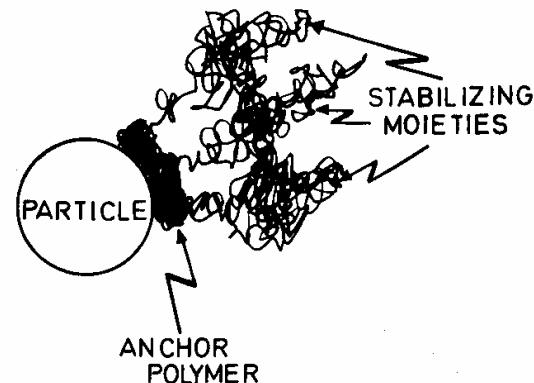
**Figure 12.9** Plot of  $\Phi_{net}$  versus  $s$ , the separation of surfaces, for two spheres of equal radius (100 nm). Curves are drawn for different values of  $\kappa$  with constant values of  $A$  ( $10^{-19} \text{ J}$ ) and  $\psi_0$  (25.7 mV). (From Ref. 8, used with permission.)

## Basics- Surface chemistry

### □ Steric Stabilization



Steric stabilizer : amphipathic block or graft copolymer



Typical stabilizing moieties and anchor polymers for sterically stabilized dispersions

Anchor polymer	Aqueous dispersions	
		Stabilizing moieties
polystyrene		poly(oxyethylene)
poly(vinyl acetate)		poly(vinyl alcohol)
poly(methyl methacrylate)		poly(acrylic acid)
poly(acrylonitrile)		poly(methacrylic acid)
poly(dimethylsiloxane)		poly(acrylamide)
poly(vinyl chloride)		poly(vinyl pyrrolidone)
poly(ethylene)		poly(ethylene imine)
poly(propylene)		poly(vinyl methyl ether)
poly(lauryl methacrylate)		poly(4-vinylpyridine)

D. H. Napper, Polymeric Stabilization of Colloidal Dispersions (1983)

## Basics- Stability of Nanoparticle

### □ Metal oxide nanoparticles in aqueous suspensions

- kinetic stability- energy barrier (DLVO theory)
    - dispersion, aggregation, flocculation
  - thermodynamic stability- surface energy minimization
    - Ostwald ripening (dissolution-reprecipitation)
- \* Is it possible to avoid the ripening of nanoparticles in suspension and to control their dimension by monitoring the precipitation conditions?

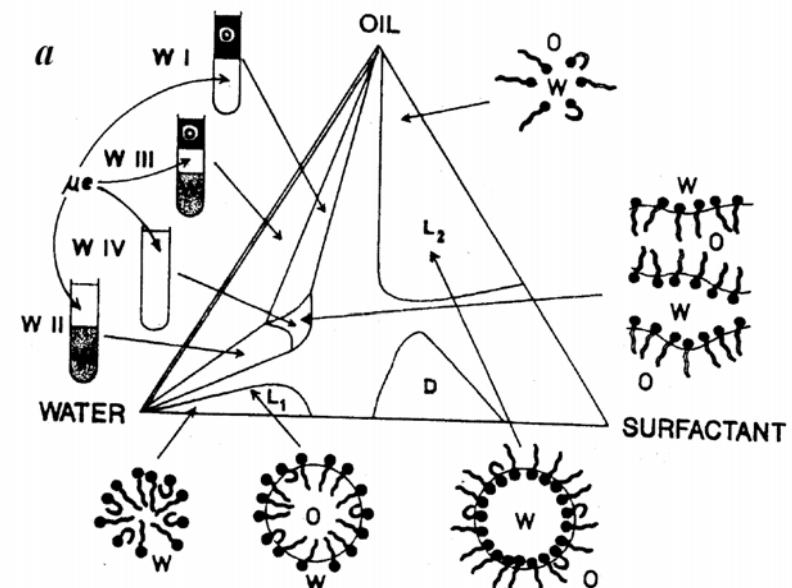
ex) thermodynamically stable dispersed system- microemulsion  
answer) possible

When the pH of precipitation is sufficiently far from the point of zero charge and the ionic strength sufficiently high, the ripening of nanoparticles is avoided. The stability condition, defined by a 'zero' interfacial tension, corresponds to the chemical and electrostatic saturation of the water-oxide interface. In such a condition, the density of charged surface groups reaches its maximum, the interfacial tension its minimum and further adsorption forces the surface area to expand and consequently, the size of nanoparticles to decrease.

## Basics- Stability of Nanoparticle

### □ Microemulsion

- clear, stable, isotropic liquid mixtures of oil, water, and surfactant, frequently in combination with a co-surfactant.
- aqueous phase may contain salt(s) and/or other ingredients, and the “oil” may actually be a complex mixture of different hydrocarbons and olefins.
- **thermodynamically stable**
- interfacial tension is very low ( $10^{-2}\sim10^{-3}$  mN/m)



# Basics- Stability of Nanoparticle

## □ Metal oxide nanoparticles in aqueous suspensions

	Domain of Precipitation	
	Unstable	Stable
Experimental condition	$\text{PZC} \leq \text{pH} < \text{PZIT}$	$\text{pH} \geq \text{PZIT}$
Surface charge	$\sigma < \sigma_{\max}$	$\sigma = \sigma_{\max}$
Interfacial tension	$\gamma > 0$	$\gamma \approx 0$

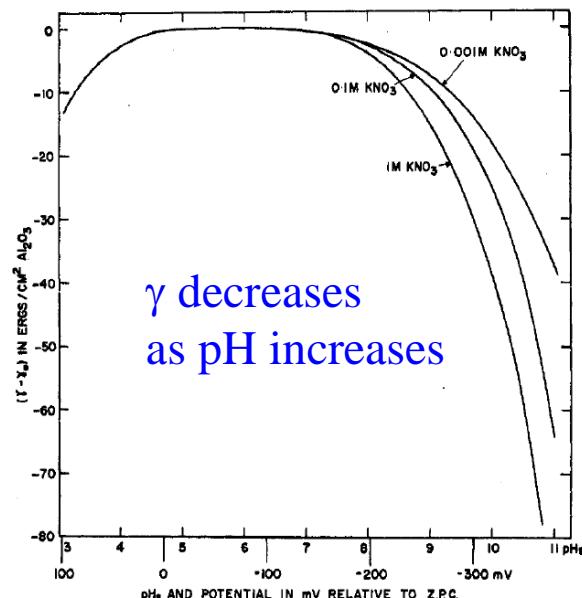


Figure 5. Variation in  $(\gamma - \gamma_0)$  at the  $\text{Al}_2\text{O}_3$ -solution interface with final pH and potential difference relative to the zpc.

S.M.Ahmed, J. Phys. Chem. 73, 3546 (1969)

L. Vayssieres, Int. J. Nanotech. 2, 411 (2005)

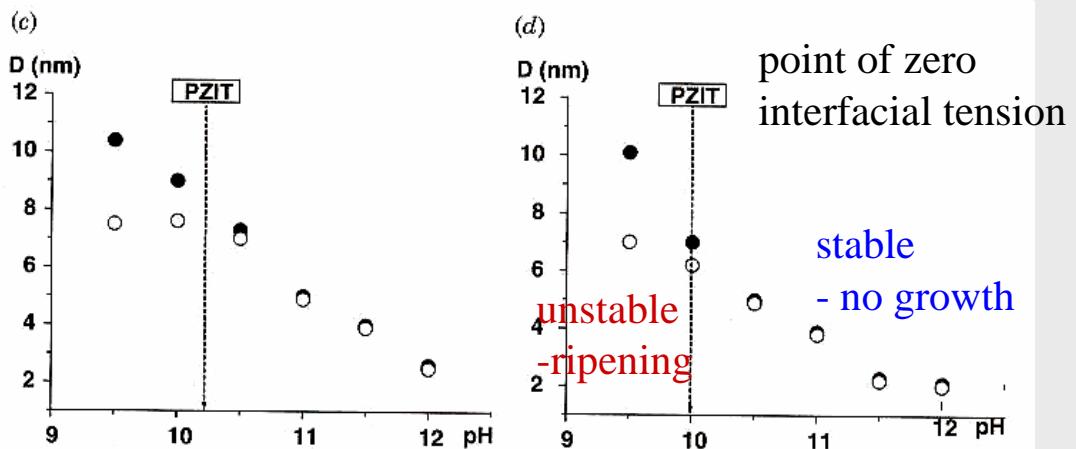


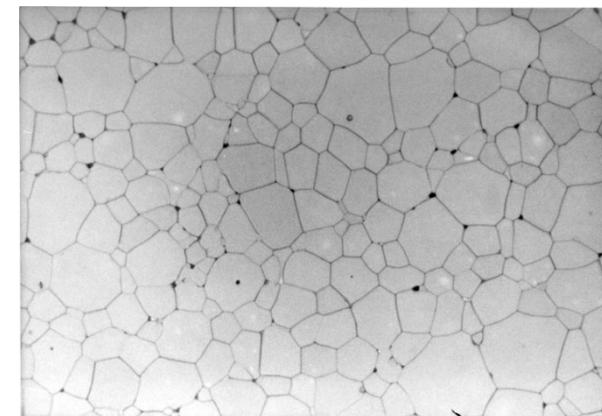
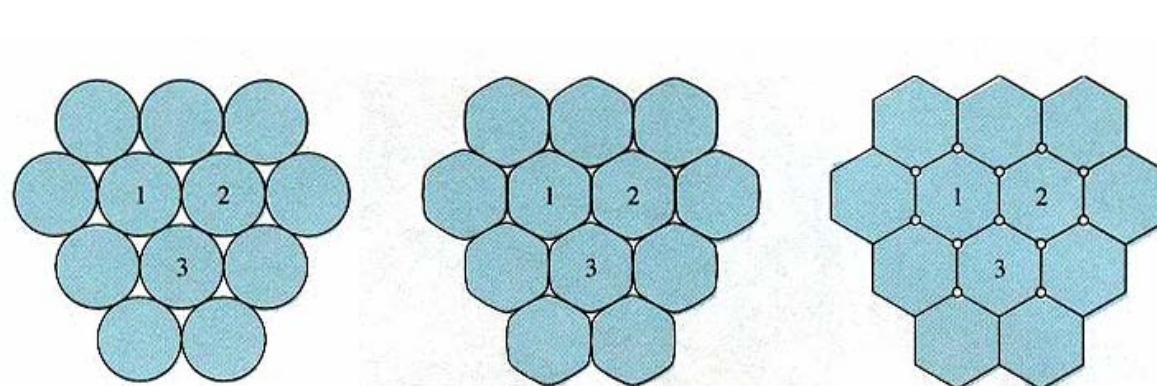
Figure 2.5. Influence of the precipitation medium on the average particle size of magnetite nanoparticles observed by TEM immediately after the synthesis (○) and after 8 days of ageing in suspension (●) for precipitations performed at pH 12.0 and (a)  $I = 0.5 \text{ M}$ , (b)  $I = 1.0 \text{ M}$ , (c)  $I = 1.5 \text{ M}$ , and (d)  $I = 3.0 \text{ M}$ . The PZIT is displayed as dotted lines and indicates the limit between the stable and unstable domains as calculated from Eq. 2.16 with  $\text{PZC} = 8.1$  and  $\sigma_{\max} = 0.84 \text{ C/m}^2$  (such values were obtained by titration methods) from Ref. 100 with permission.

## Basics- Consolidation

### □ Consolidation (sintering)

processes involved in the heat treatment of powder compacts at elevated temperatures, usually at  $T > 0.5T_m$  [K], in the temperature range where diffusional mass transport is appreciable resulting in a dense polycrystalline solid.

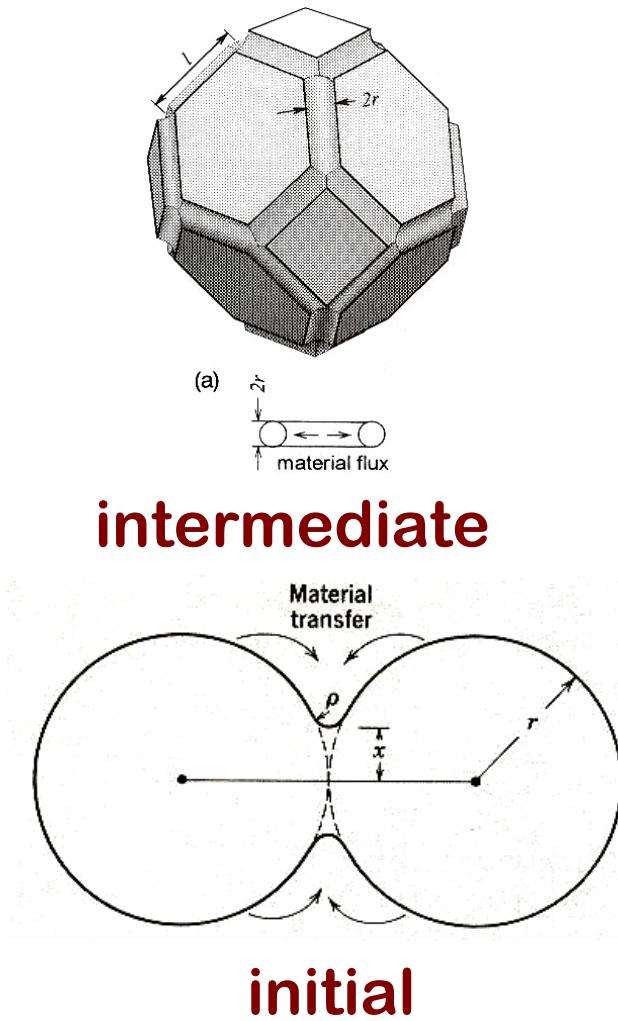
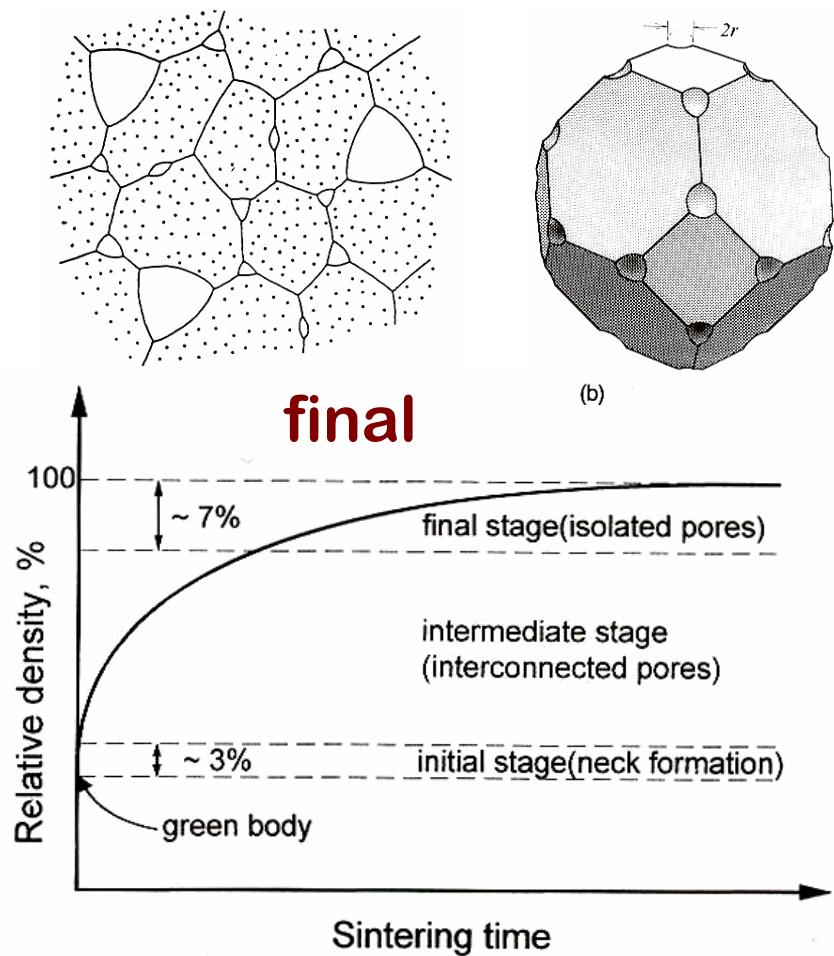
- pore removal
- densification



MgO-doped  $\text{Al}_2\text{O}_3$

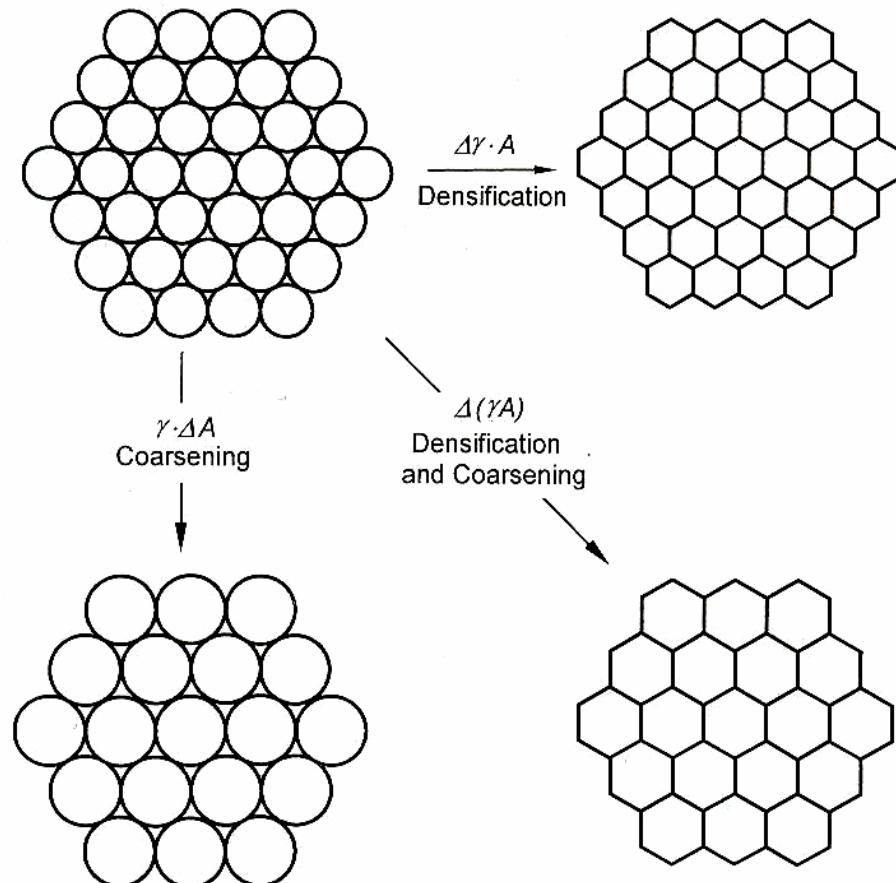
## Basics- Consolidation

### □ Solid state sintering

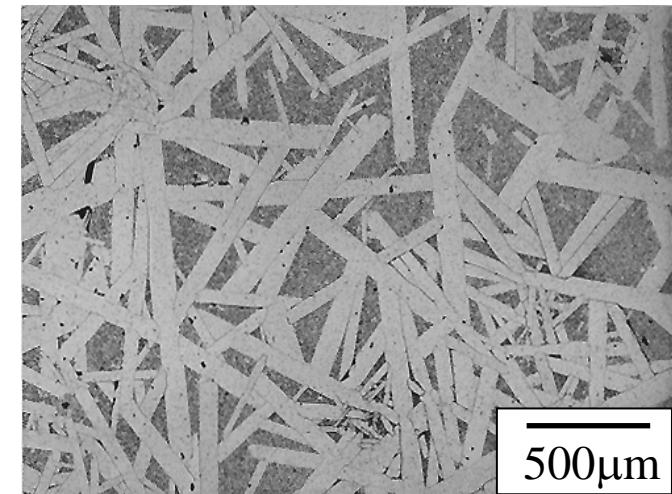


## Basics- Consolidation

### □ Densification vs. Grain Growth



63 Nanomaterials



$\text{TiO}_2$  &  $\text{SiO}_2$ -doped  $\text{Al}_2\text{O}_3$