

2009 fall

Advanced Physical Metallurgy
“Phase Equilibria in Materials”

10. 06. 2009

Eun Soo Park

Office: 33-316

Telephone: 880-7221

Email: espark@snu.ac.kr

Office hours: by an appointment

Contents for previous class

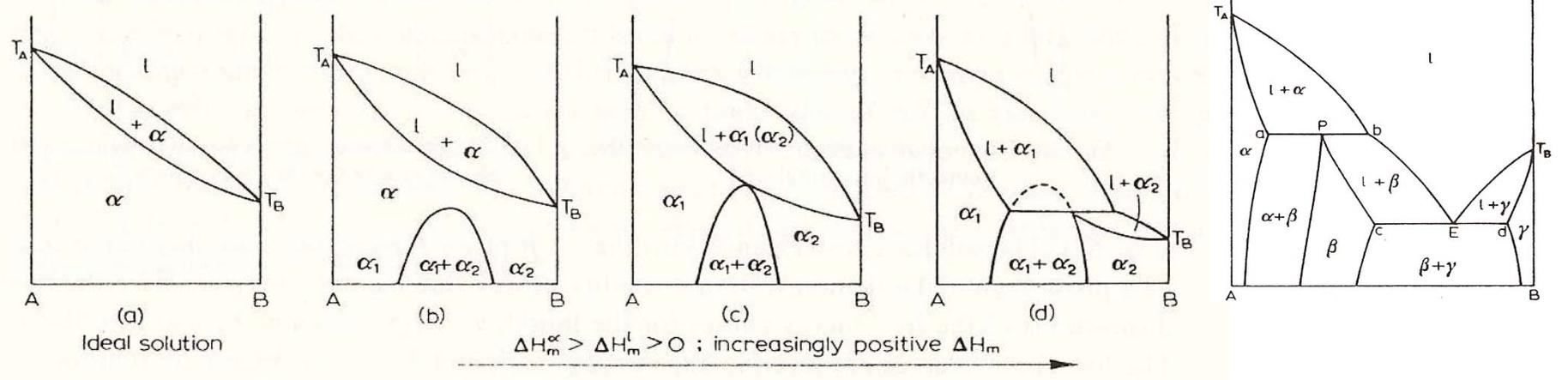
- Gibbs Phase Rule

$$F = C - P + 2 \quad (\text{from } T, P)$$

- Invariant reaction

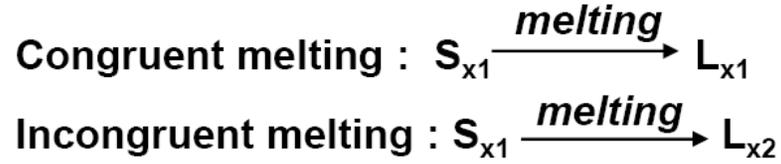
$l \rightleftharpoons \alpha + \beta$	eutectic reaction	(e.g. Ag-Cu system)
$\gamma \rightleftharpoons \alpha + \beta$	eutectoid reaction	(e.g. C-Fe system)
$l_1 \rightleftharpoons \alpha + l_2$	monotectic reaction	(e.g. Cu-Pb system)
$\alpha \rightleftharpoons \beta + l$	metatectic reaction	(e.g. Ag-Li system)
$l + \alpha \rightleftharpoons \beta$	peritectic reaction	(e.g. Cu-Zn system)
$\alpha + \beta \rightleftharpoons \gamma$	peritectoid reaction	(e.g. Al-Cu system)
$l_1 + l_2 \rightleftharpoons \alpha$	syntectic reaction	(e.g. K-Zn system)

- Peritectic reaction

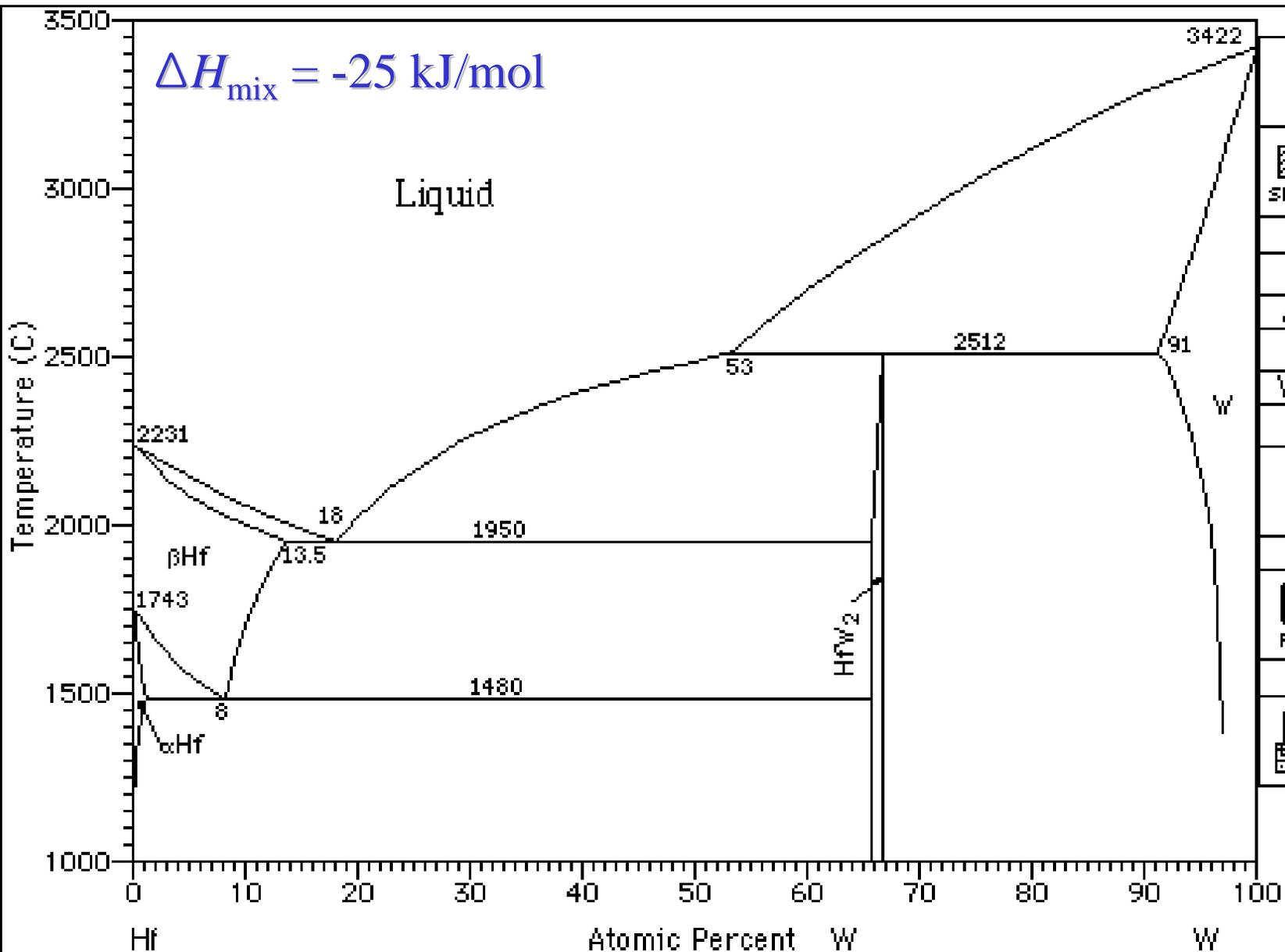


- Congruent transformation

a melting point minimum, a melting point maximum, and a critical temperature associated with a order-disorder transformation



$$\Delta H_{\text{mix}} = -25 \text{ kJ/mol}$$



Screens



SELECT



Temp [°C]

At% W

Wt% W

Windows



REFS

Export



Peritectic solidification

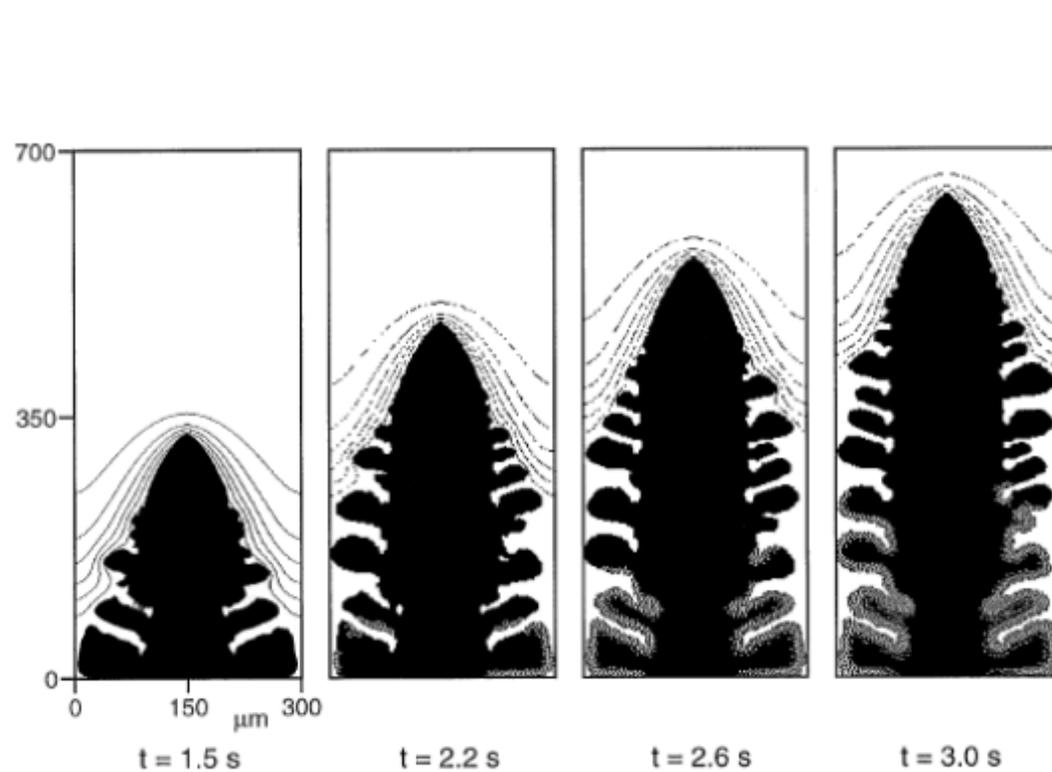
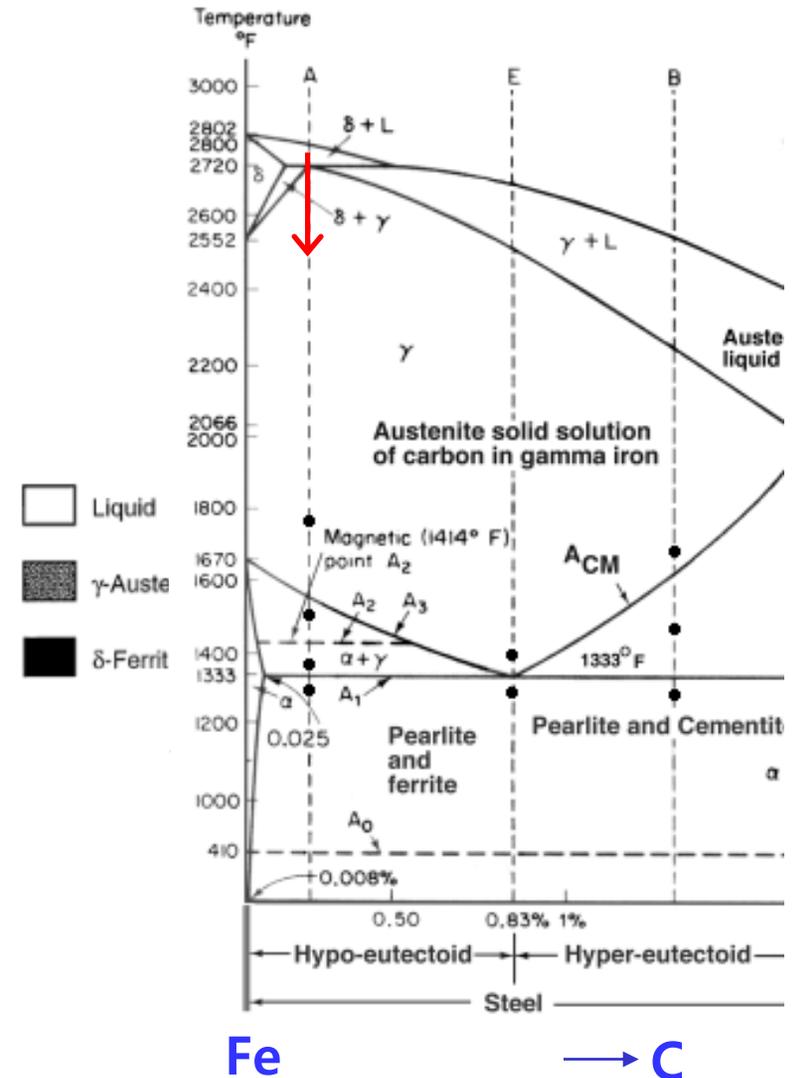


Figure 7. Phase-field simulation of peritectic reaction [3].



Fe → C

Peritectic solidification

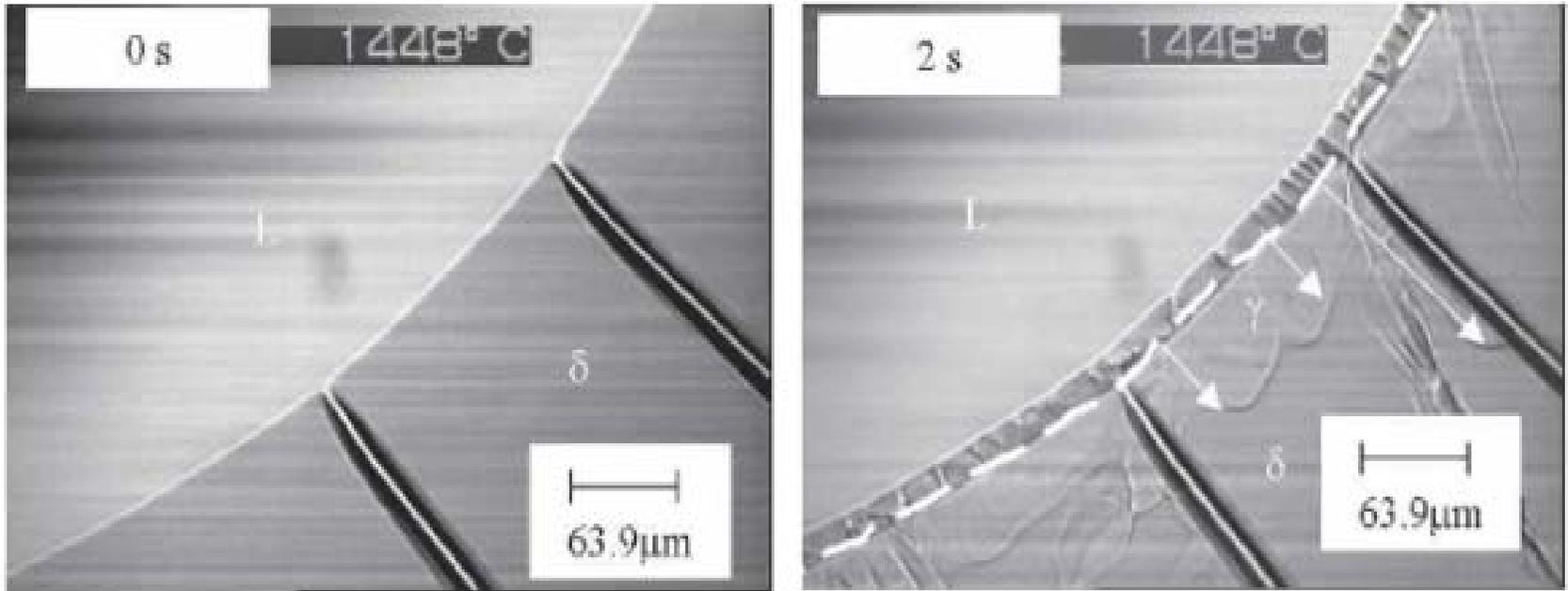


Figure 5. Peritectic reaction in Fe-0.18 pct C alloy: cooling rate =10 K/min [2].

Peritectic solidification

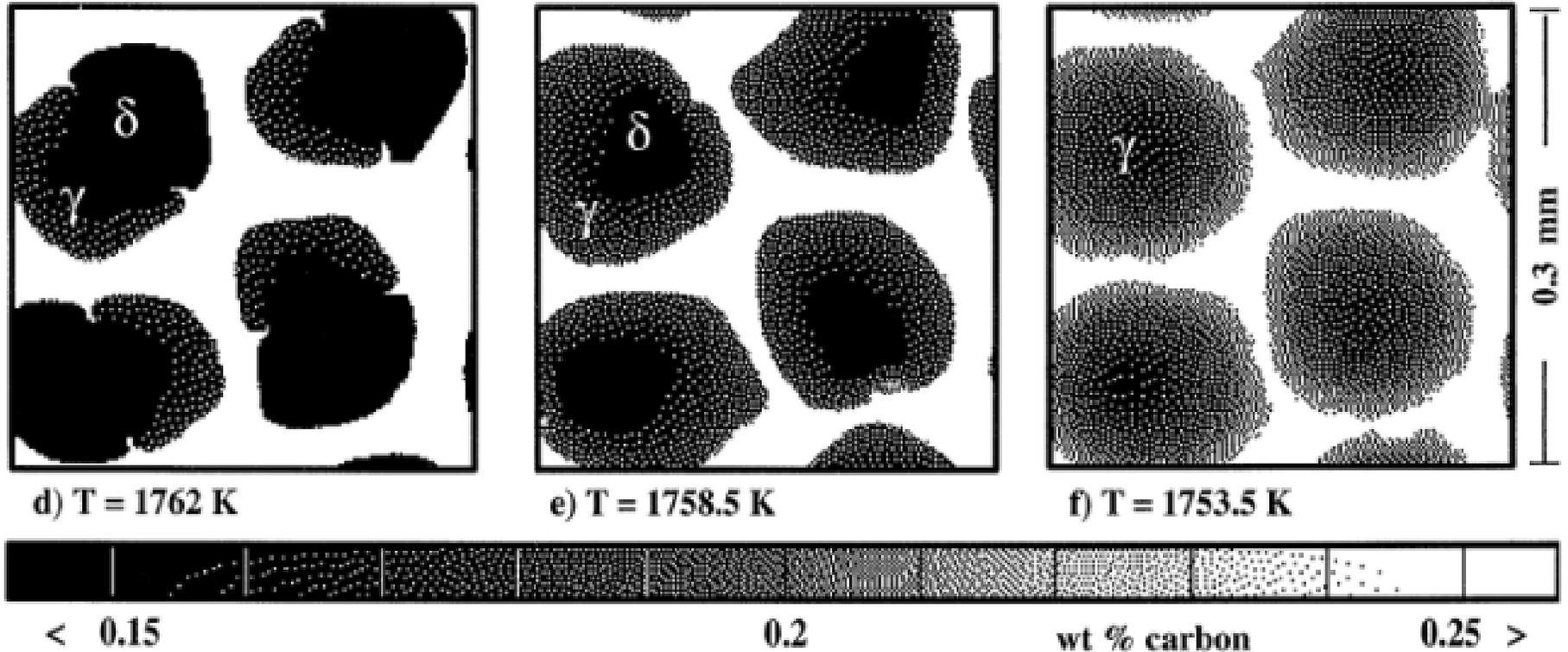


Figure 6. Phase-field simulation of peritectic reaction [3].

Congruent vs Incongruent

Congruent phase transformation: no compositional change associated with transformation

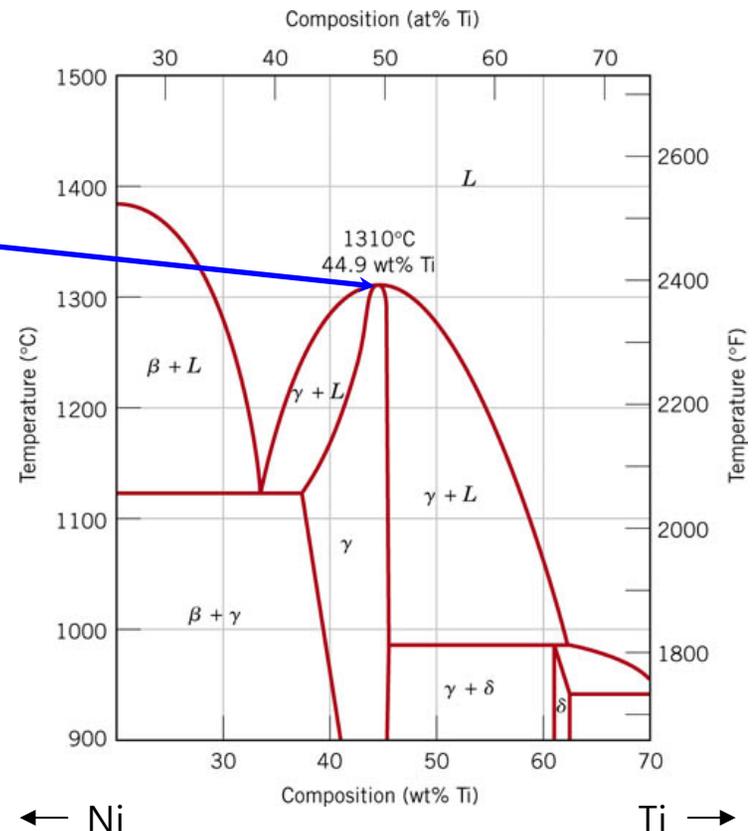
Examples:

- Allotropic phase transformations
- Melting points of pure metals
- **Congruent Melting Point**

Incongruent phase transformation: at least one phase will experience change in composition

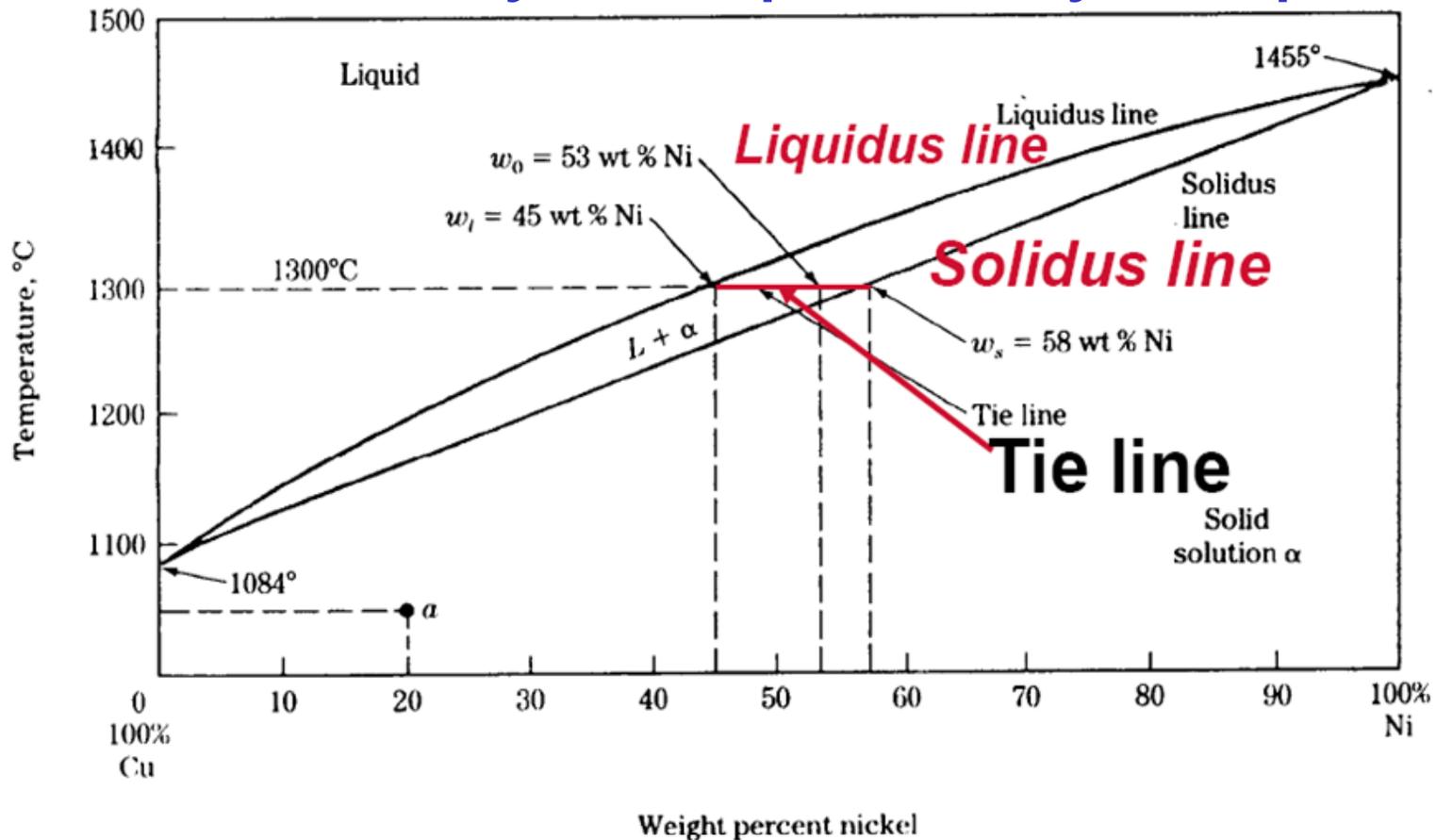
Examples:

- Melting in isomorphous alloys
- Eutectic reactions
- Peritectic Reactions
- Eutectoid reactions



Isomorphous systems contain metals which are **completely soluble** in each other and have a **single type of crystal structure**.

Cu-Ni: Binary Isomorphous Alloy Example



4.3.5. Non-stoichiometric compounds

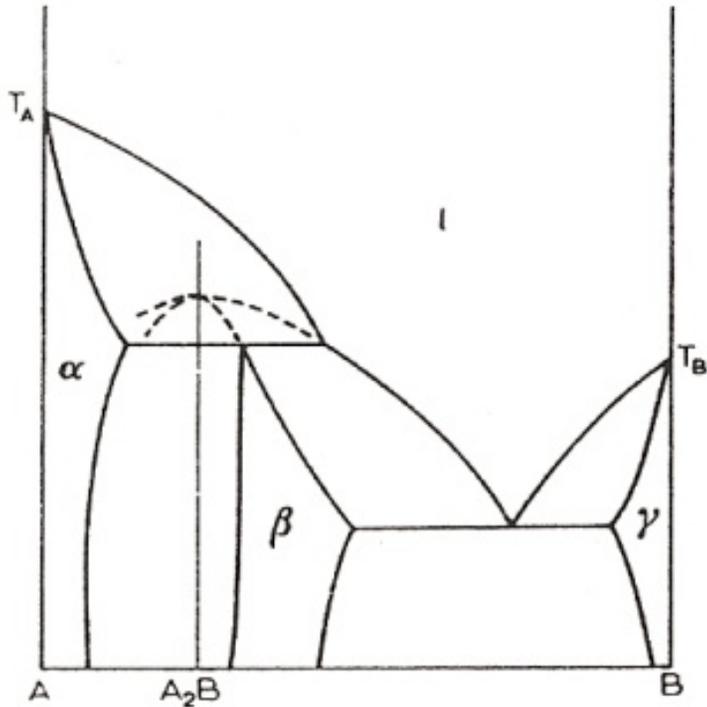


Fig. 74. A non-stoichiometric β phase based on the intermediate phase A_2B .

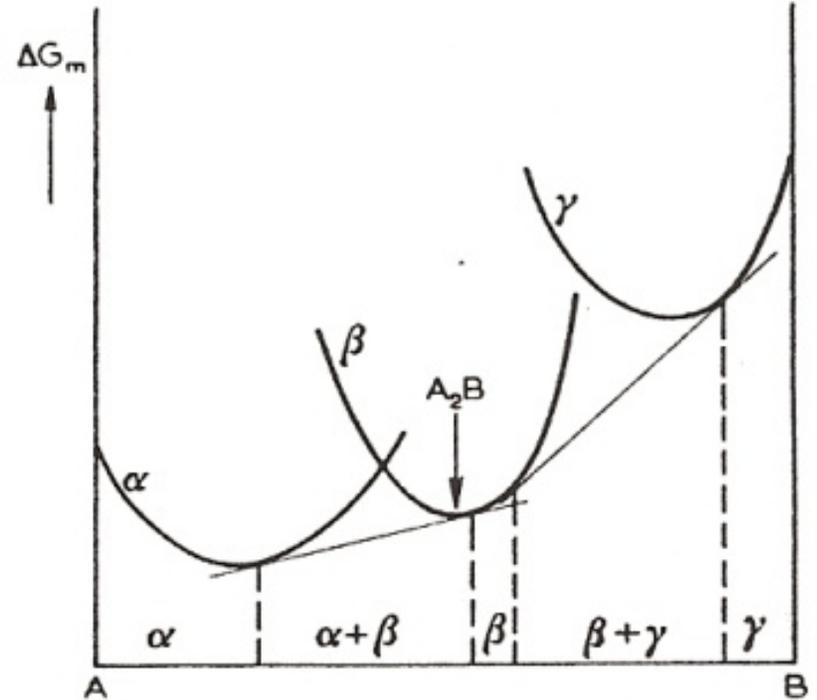
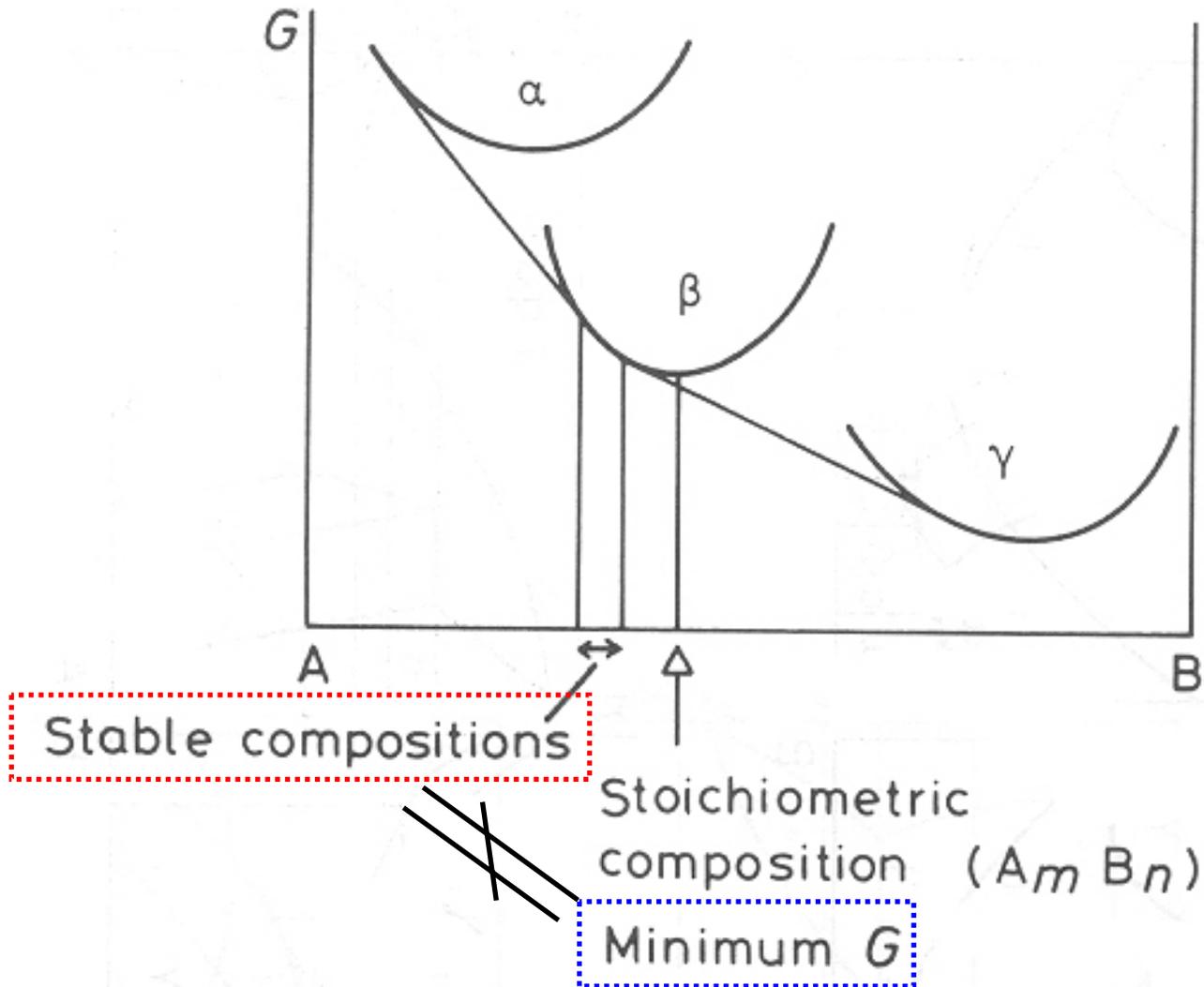


Fig. 75. Use of free energy curves to illustrate the occurrence of non-stoichiometric phases.

4.3.5. Non-stoichiometric compounds

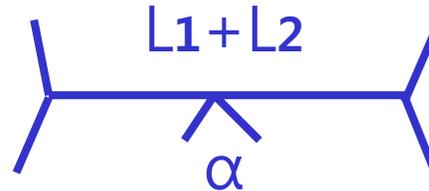


Contents for today's class

- **Monotectic reaction: Liquid1 \leftrightarrow Liquid2 + solid**

- **Synthetic reaction:**

Liquid1 + Liquid2 \leftrightarrow α



K-Zn, Na-Zn,
K-Pb, Pb-U, Ca-Cd

- **Binary Phase Diagrams: Reactions in the Solid State**

* **Eutectoid reaction: $\alpha \leftrightarrow \beta + \gamma$**

* **Monotectoid reaction: $\alpha_1 \leftrightarrow \beta + \alpha_2$**

* **Peritectoid reaction: $\alpha + \beta \leftrightarrow \gamma$**

- **Allotropy of the Components**

* SYSTEMS IN WHICH ONE PHASE IS IN EQUILIBRIUM WITH THE LIQUID PHASE

* SYSTEMS IN WHICH TWO PHASES ARE IN EQUILIBRIUM WITH THE LIQUID PHASE

Metatectic reaction: $\beta \leftrightarrow L + \alpha$ Ex. Co-Os, Co-Re and Co-Ru₁₁

$$\Delta G_m = NCX_A(1-X_A) + NkT[X_A \ln X_A + (1-X_A) \ln(1-X_A)] \quad \text{where, } C = Z[H_{AB} - (H_{AA} + H_{BB})/2].$$

Here, C is an energy term, and k is bolzmann's constant (energy/temperature).

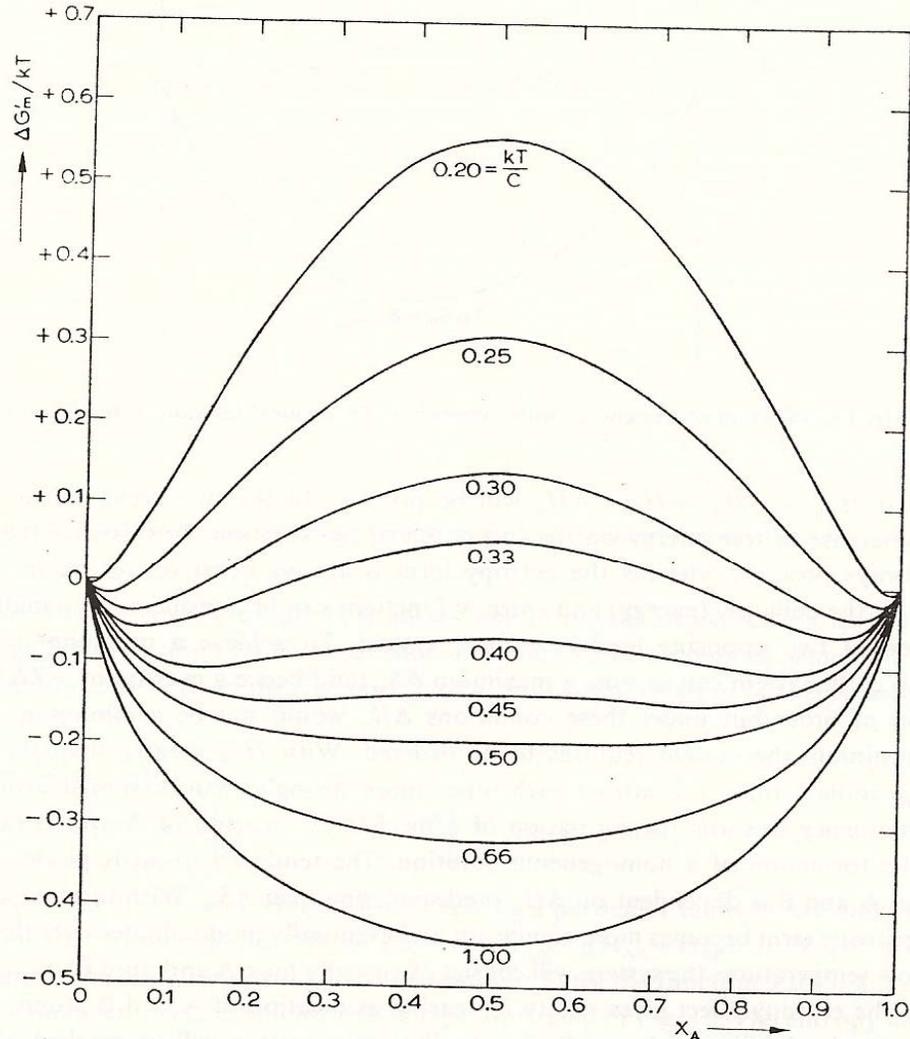
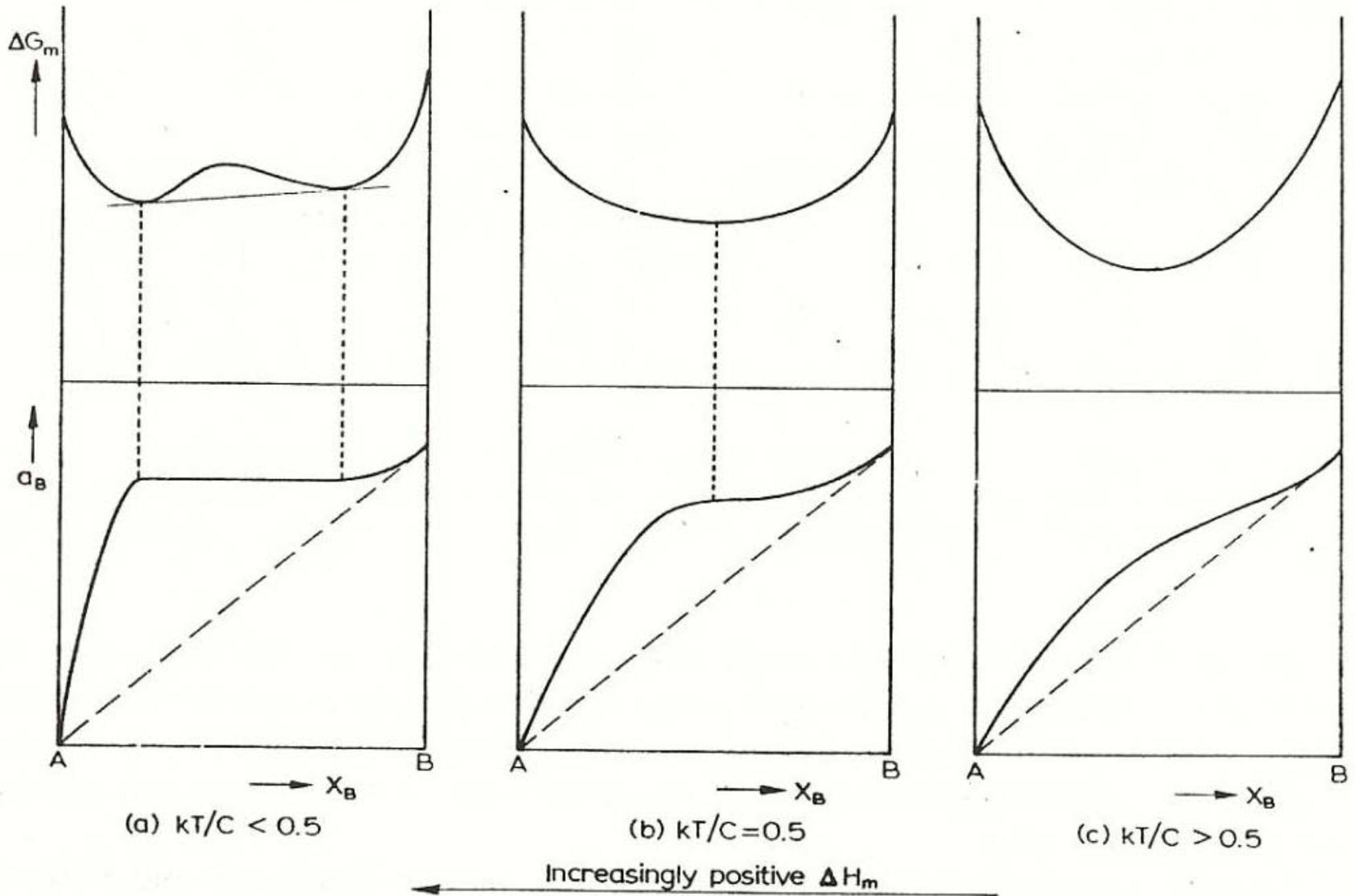
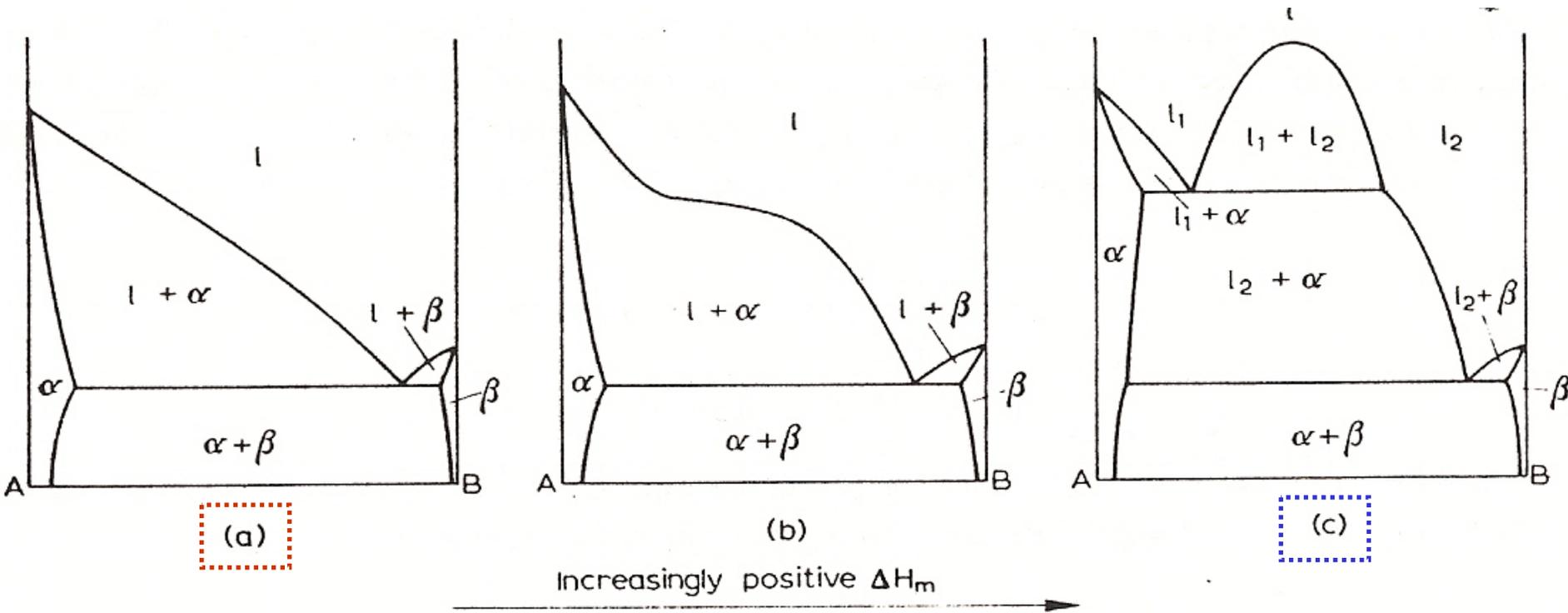


Fig. 14. Variation of free energy with composition for a homogeneous solution with $\Delta H_m > 0$. Free energy-composition curves are given for various values of the parameter kT/C .

Free energy and activity curves for (a) $kT/C < 0.5$, (b) $kT/C = 0.5$ (c) $kT/C > 0.5$



Effect of very large positive deviations from ideality in changing the phase diagram from a eutectic to a monotectic reaction

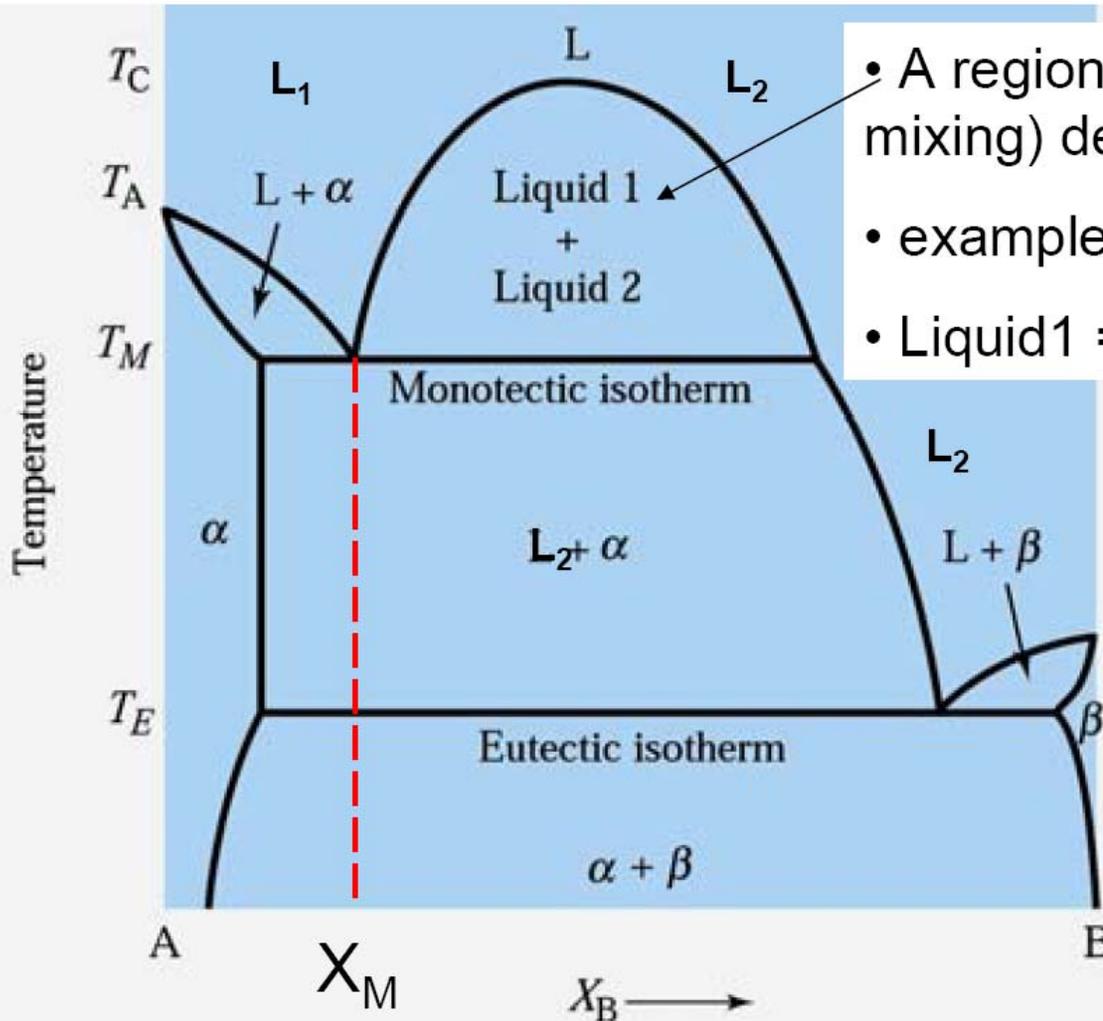


Eutectic reaction: Liquid $\leftrightarrow \alpha + \beta$

Monotectic reaction: Liquid₁ \leftrightarrow Liquid₂ + solid

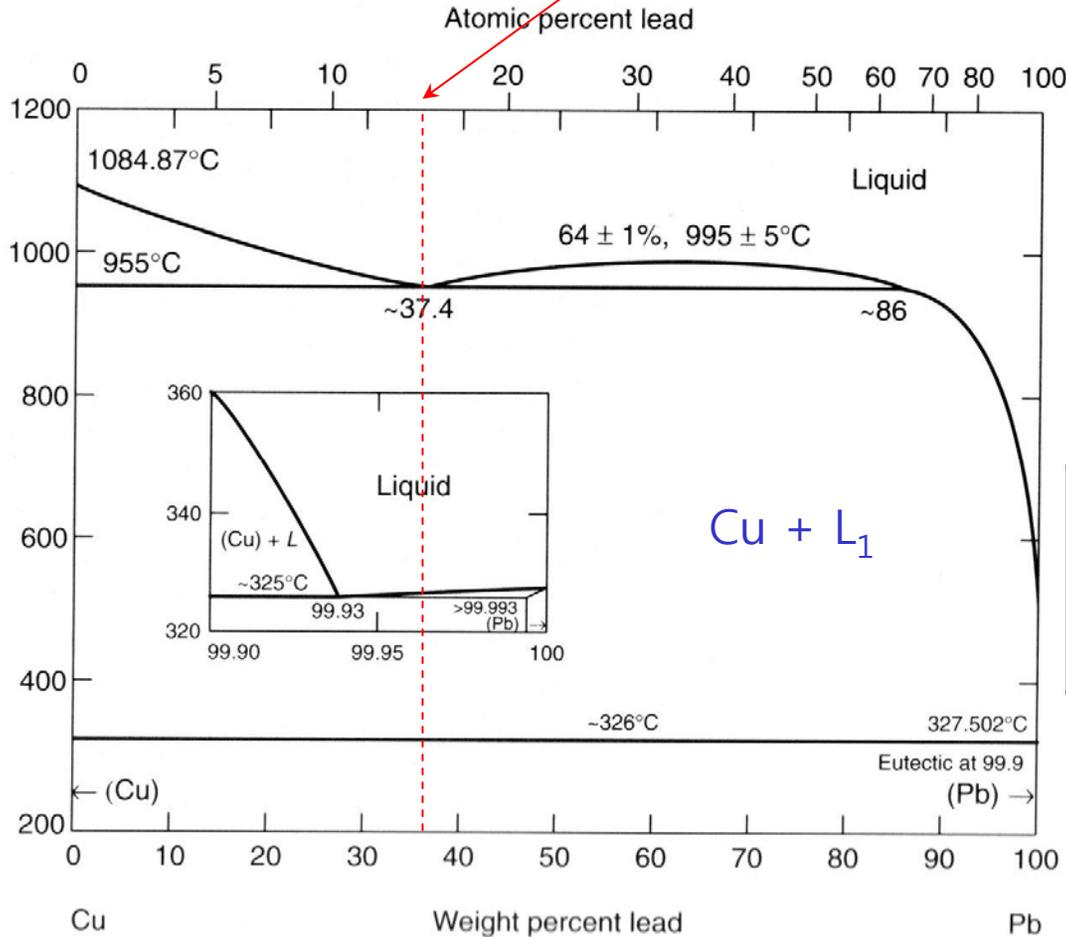
The reversible transition, on cooling, of a liquid to a mixture of a second liquid and a solid

Monotectic Phase Diagrams



- A region of immiscibility (non-mixing) develops in the L phase
- example: oil and water
- Liquid1 = Liquid2 + α (solid)

Monotectic



Pb and Zn do not mix in solid state:

- RT: Cu in Pb < 0.007%
- RT: Pb in Cu ~ 0.002 – 0.005%

	Crystal Structure	electroneg	r (nm)
Pb	FCC	1.8	0.175
Cu	FCC	1.9	0.128

26.8%

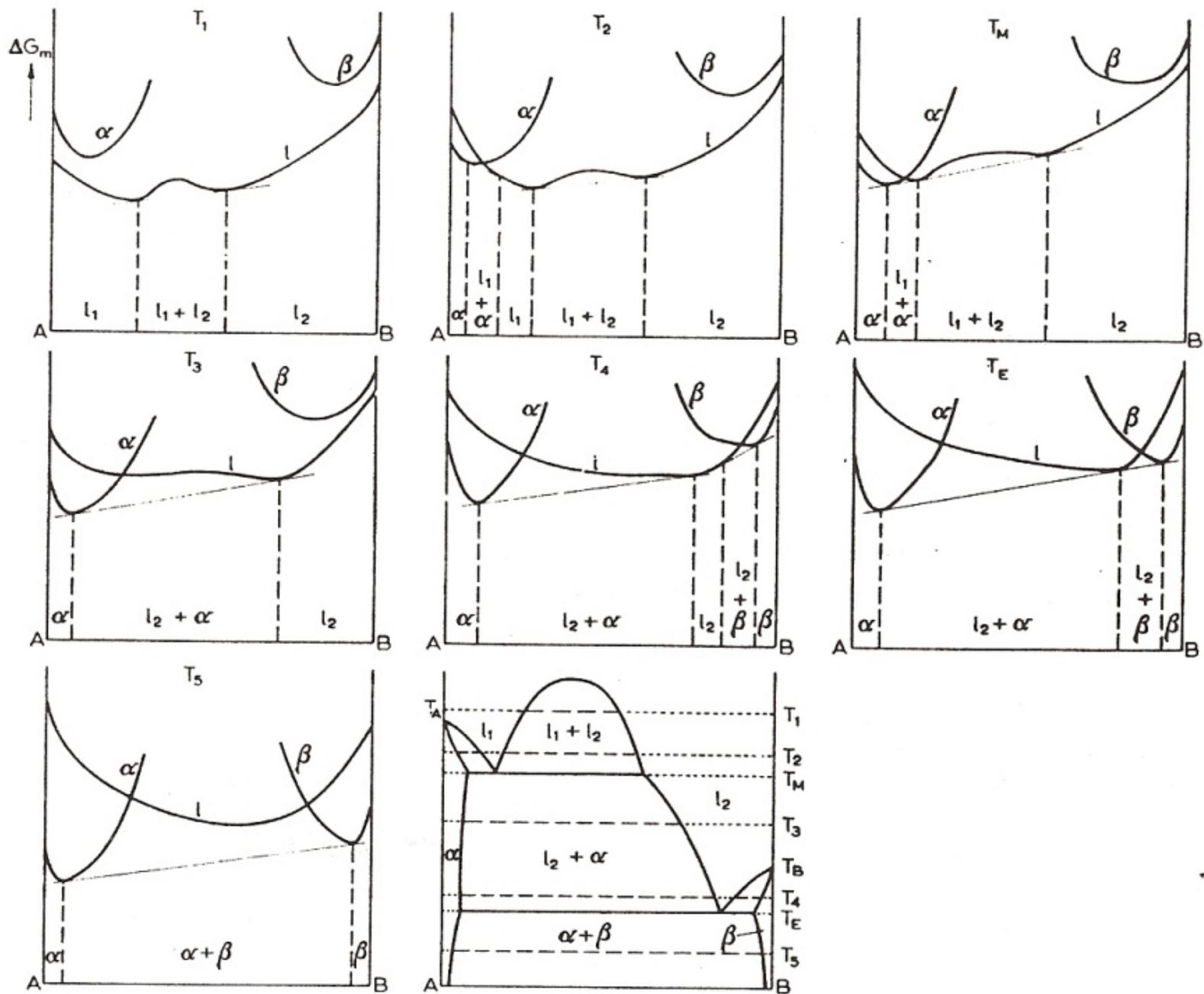


Fig. 85. Derivation of the monotectic phase diagram from the free energy curves for the liquid, α and β phases.

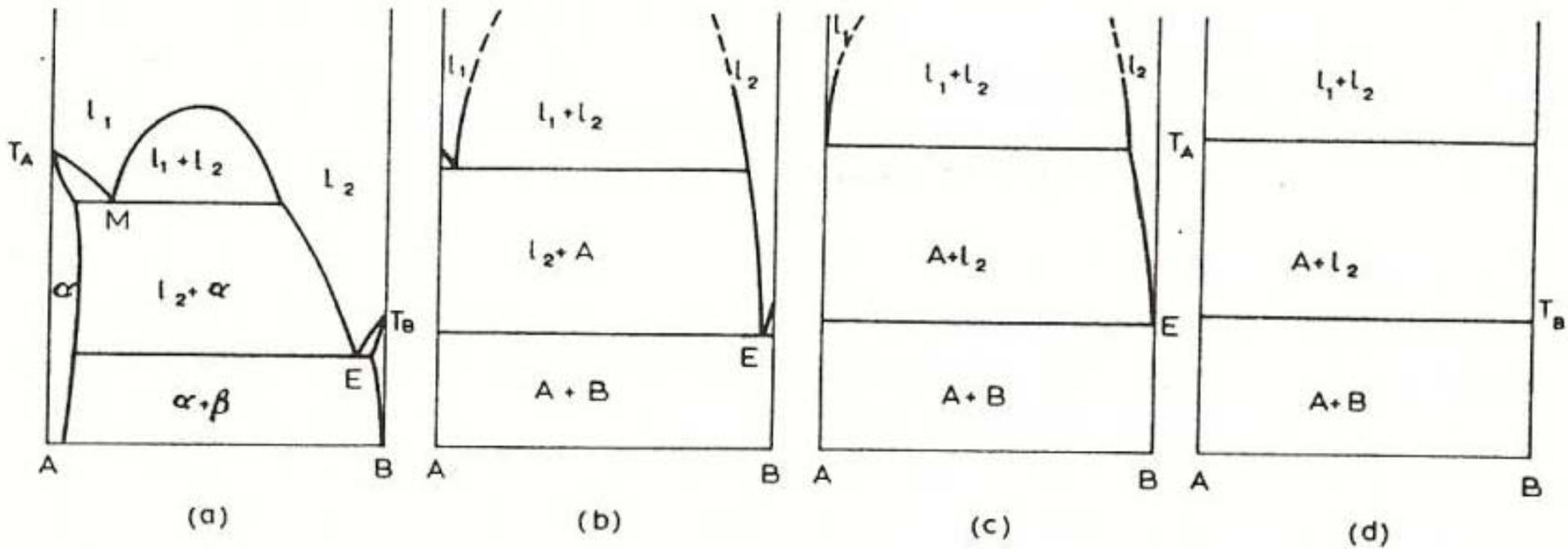
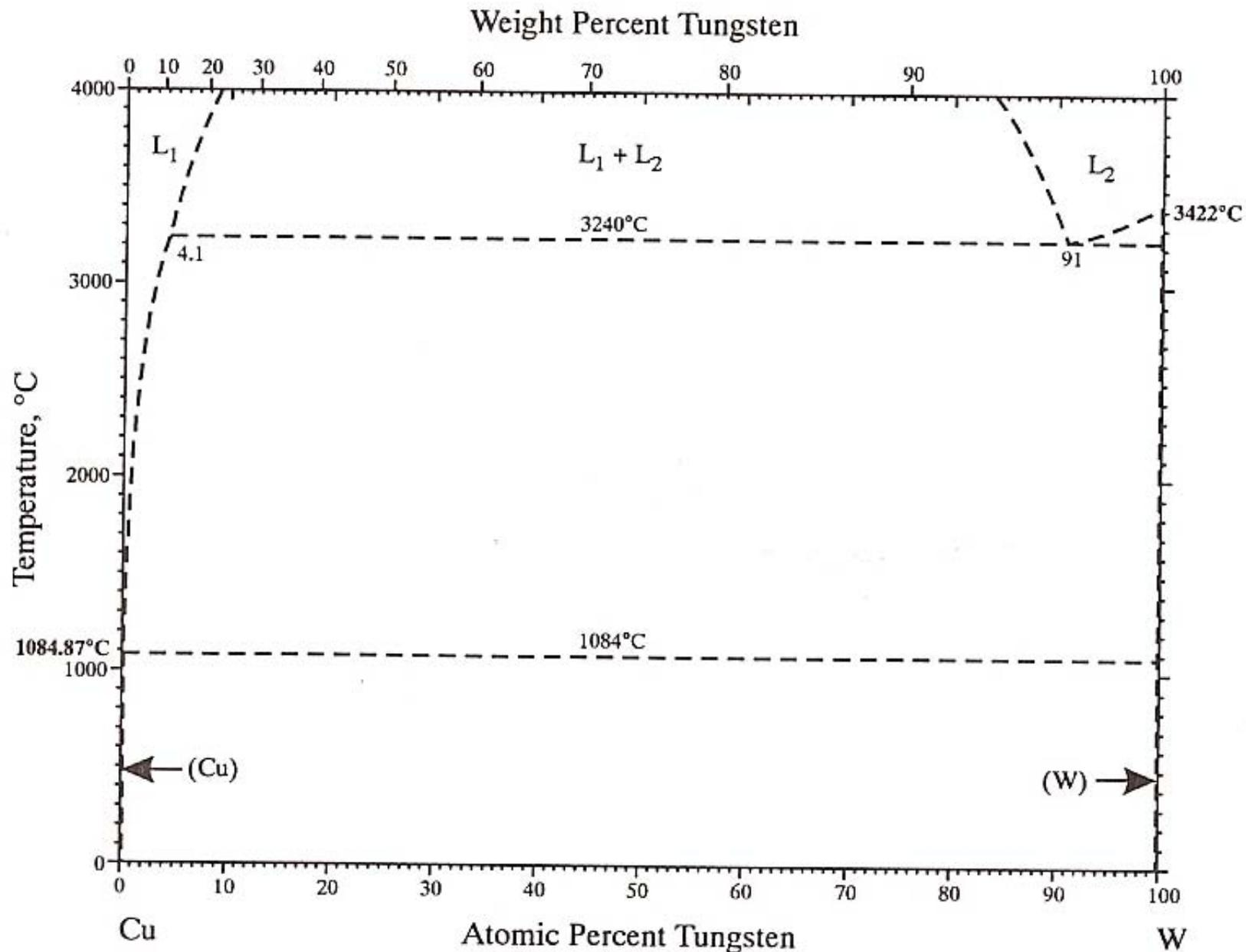


Fig. 86. Limiting form of the monotectic phase diagram.



Morphology in monotectic solidification

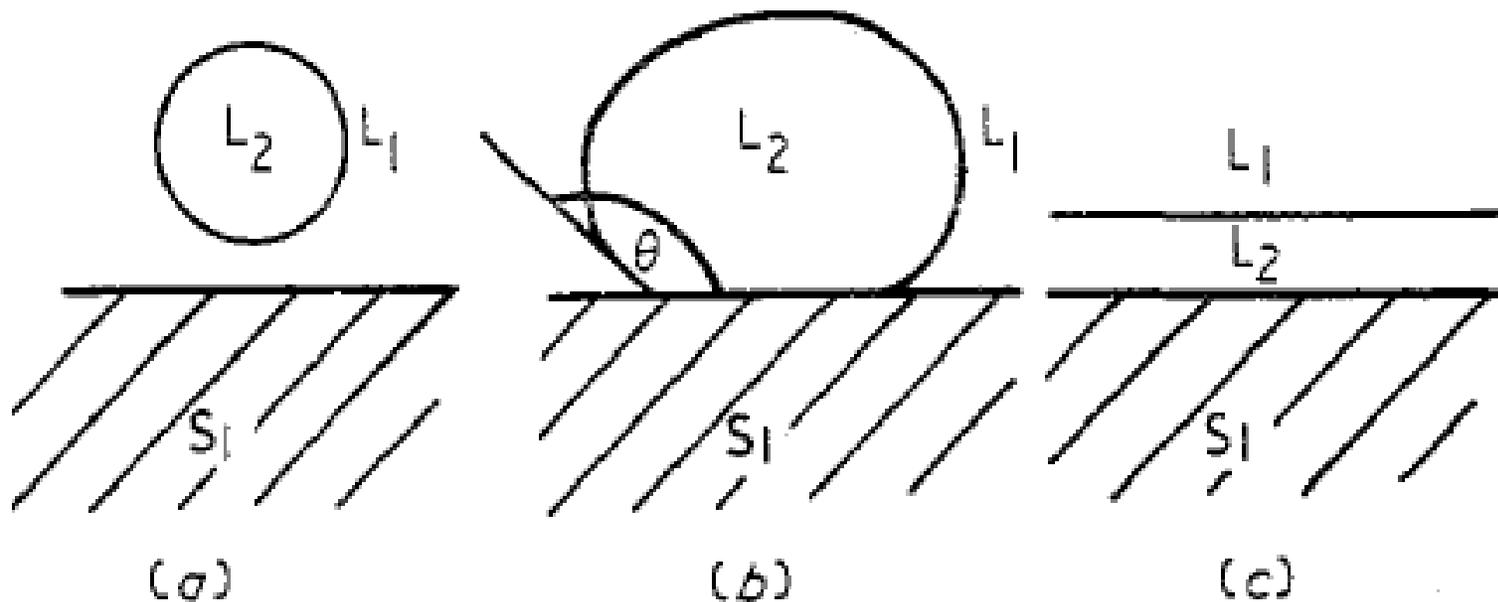
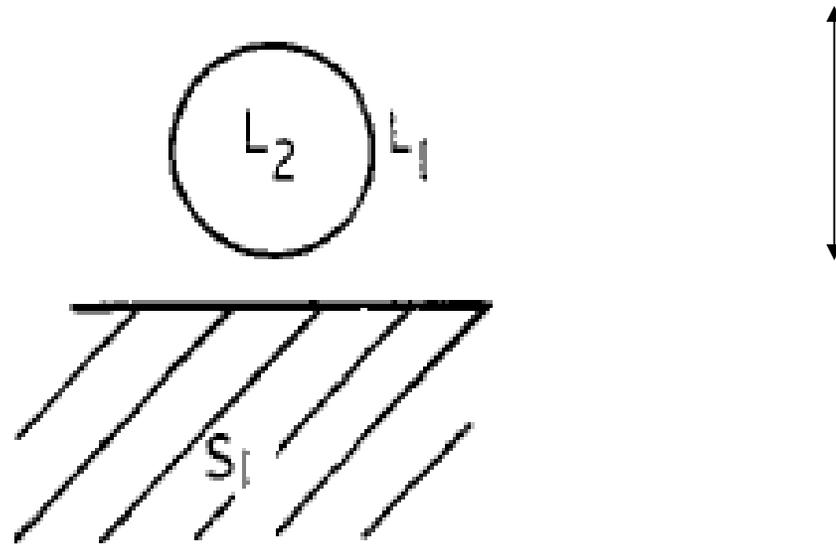


Figure 2. Solid-liquid interface morphology for different interfacial energy conditions: (a) $\gamma_{S_1L_2} > \gamma_{S_1L_1} + \gamma_{L_1L_2}$, (b) $\gamma_{S_1L_2} = \gamma_{S_1L_1} - \gamma_{L_1L_2} \cos \theta$, (c) $\gamma_{S_1L_1} > \gamma_{S_1L_2} + \gamma_{L_1L_2}$.

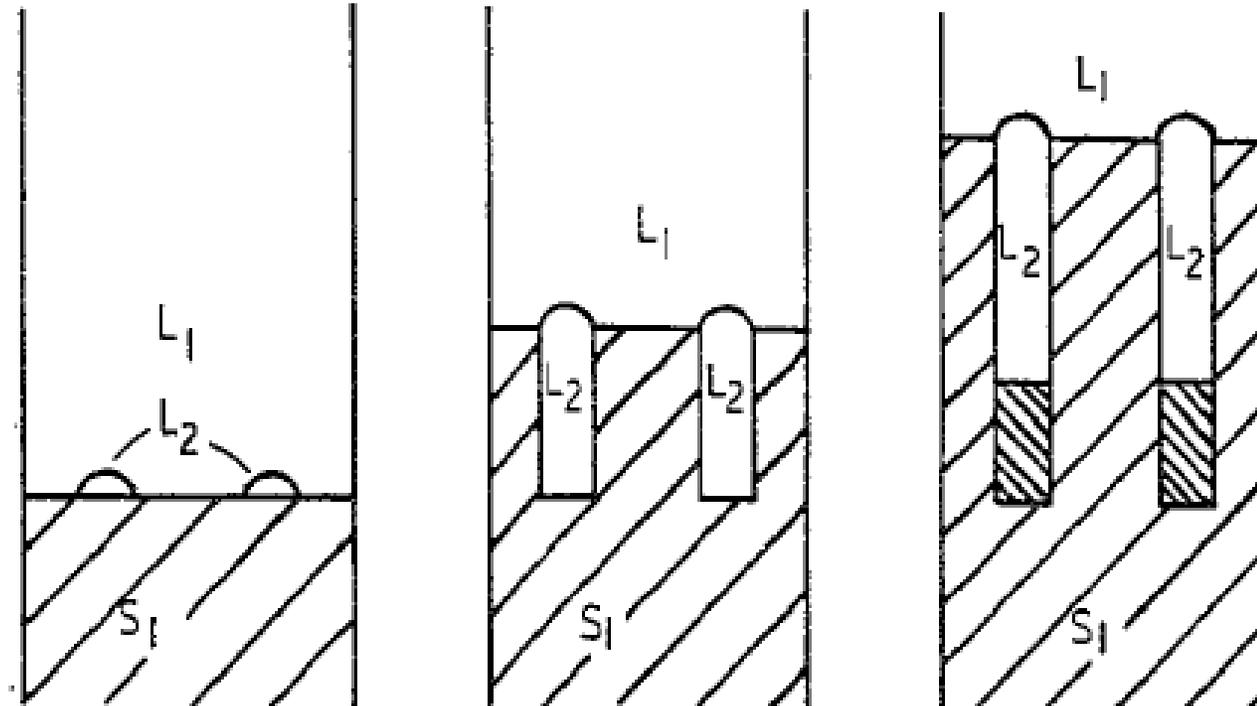
Case 1:

$$\gamma_{\alpha l_1} < \gamma_{\alpha l_2} - \gamma_{l_1 l_2}$$



Hg-Te single crystal

Case 2: $\gamma_{\alpha l_1} = \gamma_{\alpha l_2} + \gamma_{l_1 l_2} \cos \theta$

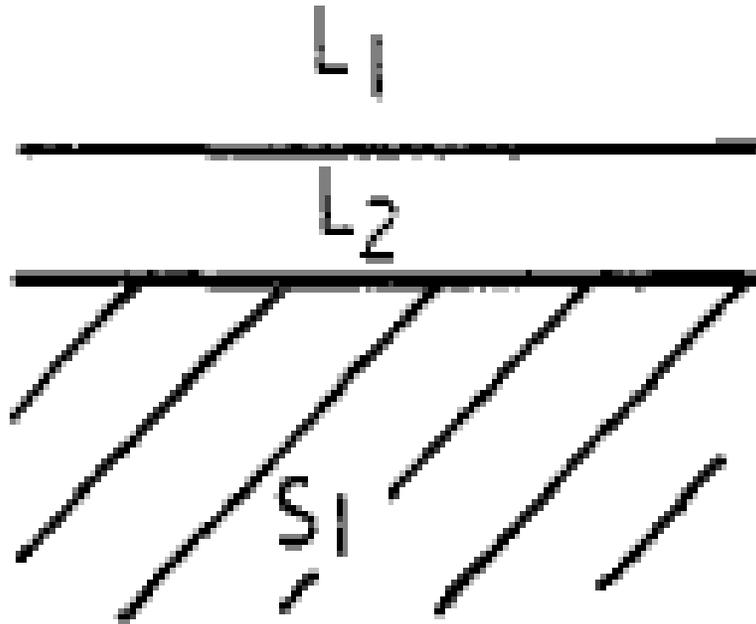


Growth mechanism of alloy of monotectic composition to produce a fibrous structure

$$\lambda \propto V^{-0.5}$$

Case 3:

$$\gamma_{\alpha l_1} > \gamma_{\alpha l_2} + \gamma_{l_1 l_2}$$



Synthetic reaction

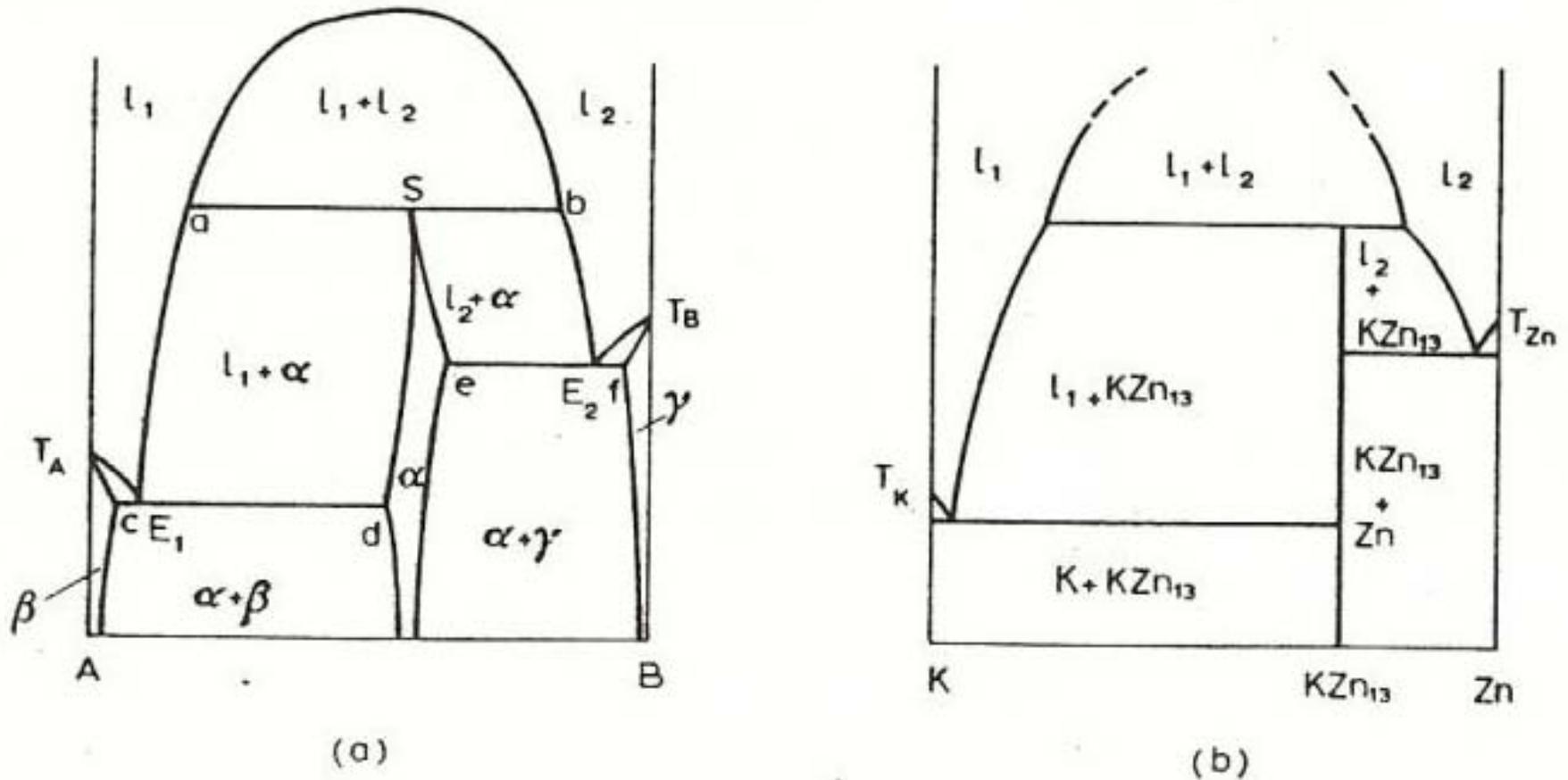
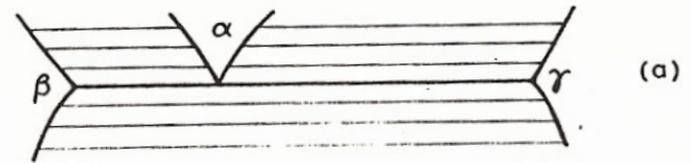


Fig. 87. Syntectic phase diagrams. (a) Schematic; (b) the K-Zn system.

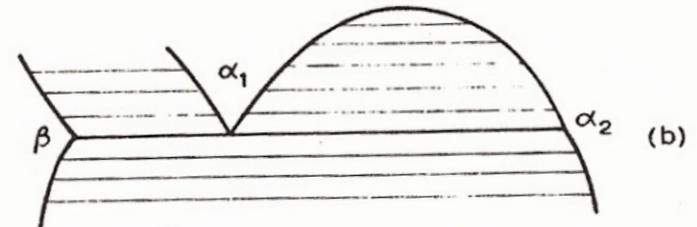
Ex. K-Zn, Na-Zn, K-Pb, Pb-U and Ca-Cd

Binary Phase Diagrams: Reactions in the Solid State

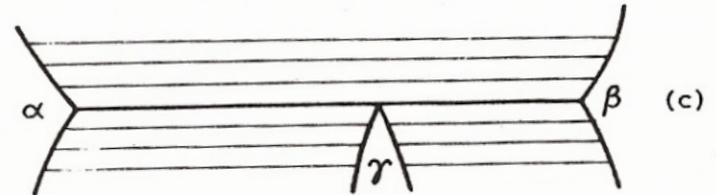
Eutectoid reaction: $\alpha \leftrightarrow \beta + \gamma$



Monotectoid reaction: $\alpha_1 \leftrightarrow \beta + \alpha_2$



Peritectoid reaction: $\alpha + \beta \leftrightarrow \gamma$



Transformation can only proceed if $-\Delta G_{bulk} > +\Delta G_{interface} + \Delta G_{strain}$

Disordered atomic arrangement at grain boundaries will reduce the strain energy factor and the interfacial energy needed to nucleate a new phase.

➔ The finer the grain size, and hence the larger the grain boundary area, the more readily will the transformation proceed.

Iron-Carbon System

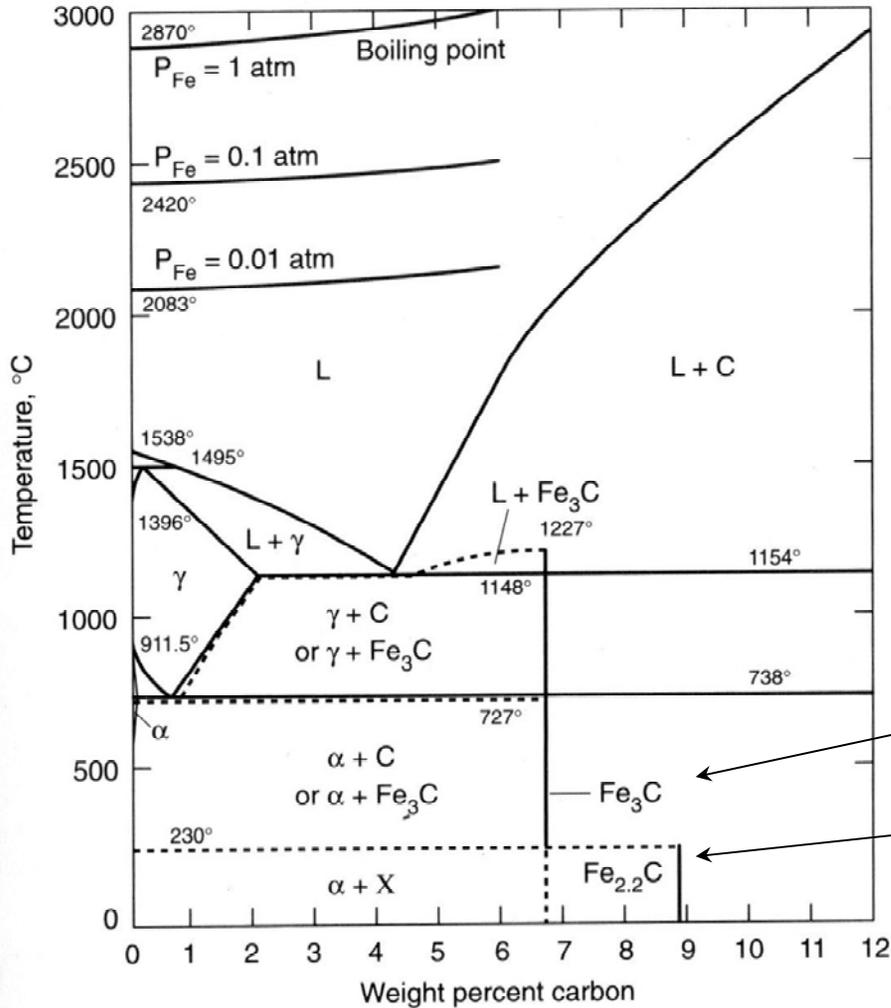
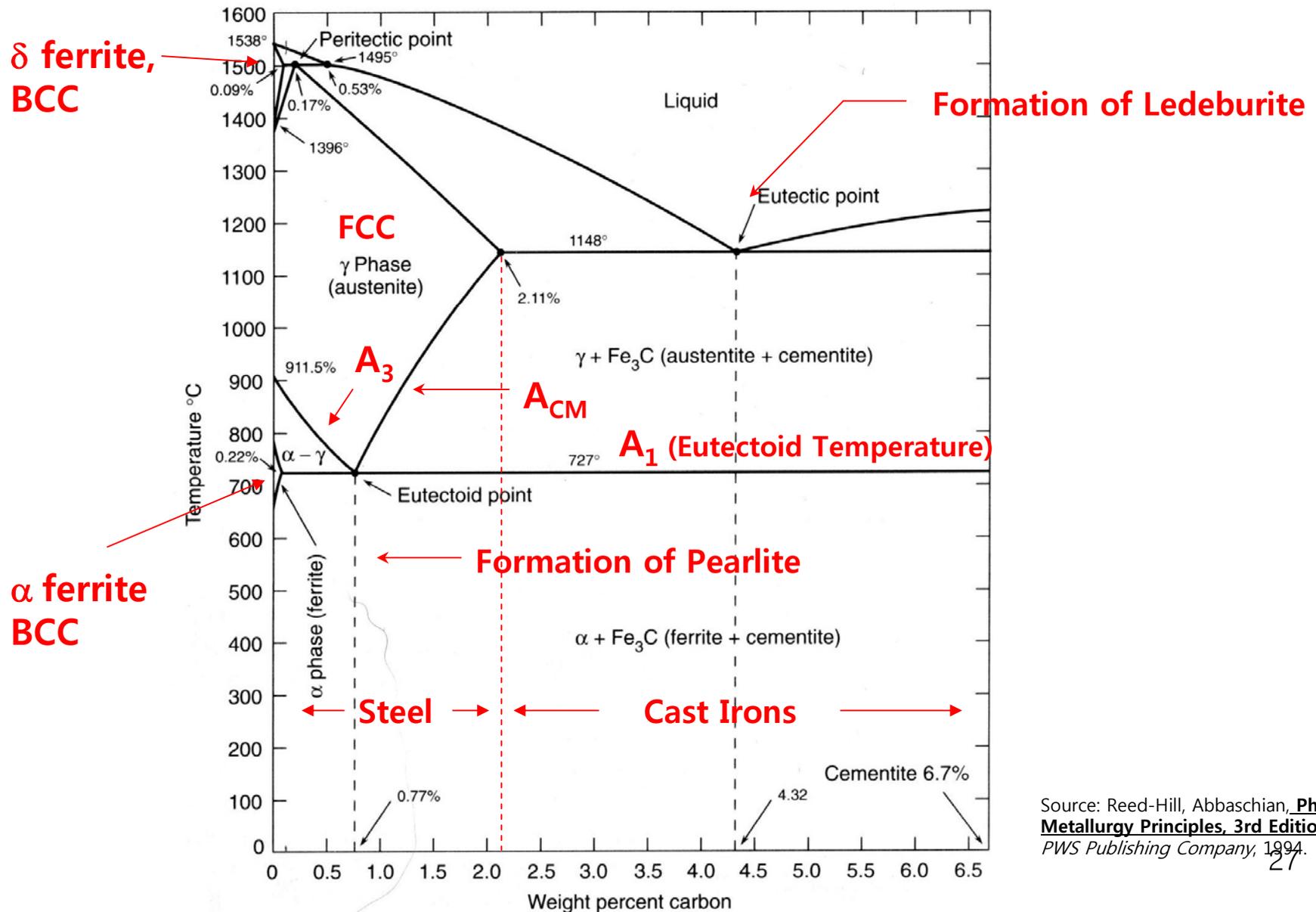


Diagram is not ever plotted past 12 wt%

Cementite

Hägg carbide

Iron Carbon Phase Diagram



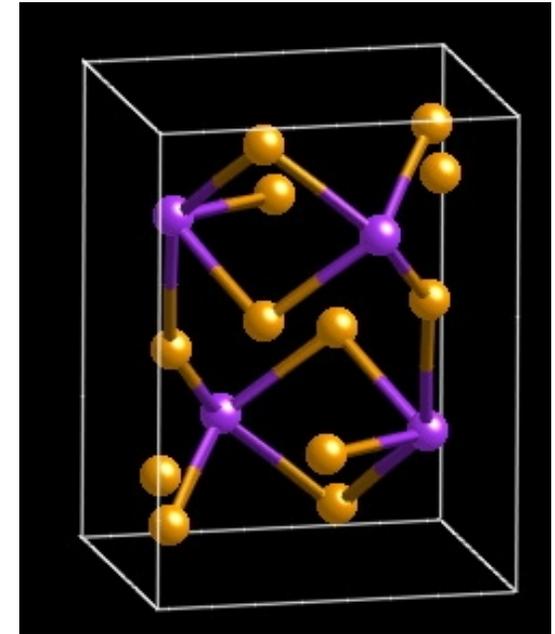
Source: Reed-Hill, Abbaschian, **Physical Metallurgy Principles, 3rd Edition**, PWS Publishing Company, 1994.

Cementite – What is it?

Iron Carbide – Ceramic Compound

Purple: Carbon atoms

Orange: Iron atoms

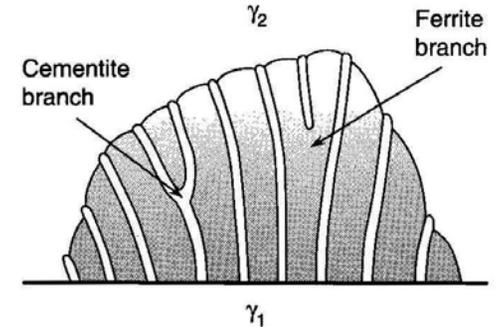


- Cementite has an orthorhombic lattice with approximate parameters 0.45165, 0.50837 and 0.67297 nm.
- There are twelve iron atoms and four carbon atoms per unit cell, corresponding to the formula Fe_3C .

Source: <http://www.msm.cam.ac.uk/phase-trans/2003/Lattices/cementite.html>

H. K. D. H. Bhadeshia

Pearlite: What is it?



- The eutectoid transformation:
 γ (0.77% C) \rightarrow α (0.02%C) + Fe_3C (6.67%C)
- Alternate lamellae of ferrite and cementite as the continuous phase
- Diffusional Transformation
- "Pearlite" name is related to the regular array of the lamellae in colonies. Etching attacks the ferrite phase more than the cementite. The raised and regularly spaced cementite lamellae act as diffraction gratings and a pearl-like luster is produced by the diffraction of light of various wavelengths from different colonies

Pearlite

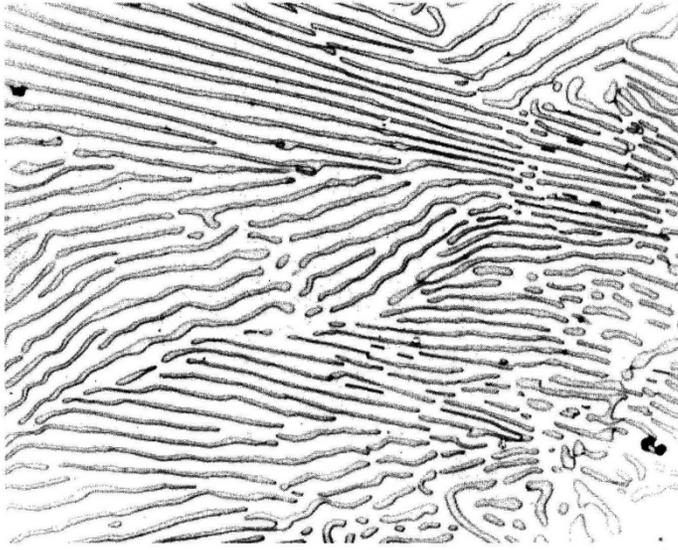


Fig. 18.6 Pearlite consists of plates of Fe_3C in a matrix of ferrite. (Vilella, J. R., *Metallographic Technique for Steel*, ASM Cleveland, 1938.) 2500X.

Reed-Hill, Abbaschian, 1994, [5]

- Two phases appear in definite ratio by the lever rule:

$$\alpha = \frac{6.67 - 0.77}{6.67} \approx 88\%$$

$$\text{cementite} = \frac{0.77 - 0}{6.67} \approx 12\%$$

- Since the densities are same (7.86 and 7.4) lamellae widths are 7:1
- Heterogeneous nucleation and growth of pearlite colonies – but typically grows into only 1 grain