

Phase Transformation of Materials

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Eun Soo Park

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Office: 33-316 Telephone: 880-7221 Email: espark@snu.ac.kr Office hours: by an appointment

Contents for previous class

< Phase Transformation in Solids >

Diffusional Transformation
Precipitation





Contents for today's class

- Precipitate growth
 - Growth behind Planar Incoherent Interfaces
 - Diffusion Controlled lengthening of Plates or Needles
 - Thickening of Plate-like Precipitates
- Overall Transformation Kinetics TTT Diagram
 - Johnson-Mehl-Avrami Equation
- Precipitation in Age-Hardening Alloys
 - GP Zones
 - Transition phases

5.3 Precipitate Growth



석출물 성장 → 계면의 이동 : 성장하는 동안 석출물 모양 각 계면의 상대적 이동 속도에 의해 좌우됨.

If the nucleus consists of semi-coherent and incoherent interfaces, what would be the growth shape?





 \rightarrow Origin of the Widmanstätten morphology

Incoherent interface \rightarrow similar to rough interface \rightarrow local equilibrium \rightarrow diffusion-controlled

Diffusion-Controlled Thickening: 석출물 성장 속도



Fig. 5.14 Diffusion-controlled thickening of a precipitate plate.

$$\rightarrow v = f(\Delta \mathbf{T} \text{ or } \Delta \mathbf{X}, \mathbf{t})$$



or interstitial diffusion coeff.









Fig. 5.16 The effect of temperature and position on growth rate, v. $_7$

Nucleation and Growth Rates – Poor Glass Formers



- Strong overlap of growth and nucleation rates
- Nucleation rate is high
- Growth rate is high
- Both are high at the same temperature

Nucleation and Growth Rates – Good Glass Formers





Fig. 5.17 (a) Interference of growing precipitates due to overlapping diffusion fields at later stage of growth. (b) Precipitate has stopped growing.



Fig. 5.18 Grain-boundary diffusion can lead to rapid lengthening and thickening of grain boundary precipitates.

치환형 확산이 필요한 경우 상대적으로 중요

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2) Diffusion Controlled lengthening of Plates or Needles



Needle \rightarrow Gibbs-Thomson increase in G = $2\gamma V_m/r$ instead of $\gamma V_m/r$ \rightarrow the same equation but the different value of r^{*}

2) Diffusion Controlled lengthening of Plates or Needles



3) Thickening of Plate-like Precipitates

Thickening of Plate-like Precipitates by Ledge Mechanism



Half Thickness Increase

 $V = \frac{uh}{\lambda}$ u) rate of lateral migrationAssuming the diffusion-controlled growth, $v = \frac{D}{C_{\beta} - C_{r}} \cdot \frac{\Delta C}{kr}$ 돌출맥 모서리가 부정합인 경우

$$u = \frac{D\Delta X_0}{k(X_\beta - X_e)h}$$

- For the diffusion-controlled growth, a monatomic-height ledge should be supplied constantly.
- sources of monatomic-height ledge
 → spiral growth, 2-D nucleation,
 nucleation at the precipitate edges,
 or from intersections with other
 precipitates (heterogeneous 2-D)

$$V = \frac{D\Delta X_0}{k(X_{\beta} - X_e)\lambda^{1/2}}$$

3) Thickening of Plate-like Precipitates



Fig. 5.22 The thickening of a g plate in an Al-15 wt% Ag alloy at 400°C. (From C. Laird and H.I. Aaronson, *Acta Metallurgica* 17 (1969) 505.)