6. Insertion reaction electrodes (Huggins, ch. 6)

- 1. Insertion of guest species into layer structures
- 2. Floating and pillared layer structure
- 3. Terminology
- 4. Types of inserted guest species configurations
- 5. Sequential insertion reactions
- 6. Coinsertion of solvent species
- 7. Insertion into materials with parallel linear tunnels
- 8. Changes in host structure
- 9. Potential variation with composition in insertion reaction

-Mobile guest species can be inserted into, or removed (extracted) from, a stable host crystal structure -In many cases the *insertion* of guest species is reversible -The terms "intercalation" and "de-intercalation" are often used for reactions involving the insertion and extraction of guest species for the specific case of host materials that have layertype crystal structures \rightarrow "insertion" and "extraction" are more general terms \rightarrow since such reactions involve a change in the

chemical composition of the host material, they can also be called solid solution reactions

-Insertion reactions are generally topotactic, with the guest species moving into, and residing in, specific sites within the host lattice structure. These sites can often be thought of as interstitial sites in the host crystal lattice. The occurrence of a topotactic reaction implies some 3-D correspondence between the crystal structures of the parent and the product. cf. the term epitaxy relates to 2-D, such as on a surface -hydrogen can be inserted into, and extracted from, palladium and some of its alloys. Palladium–silver alloys are commonly used as hydrogen-pass filters

-graphite and some clays, used to remove contaminants from water by absorbing them into their crystal structures -most common guest cationic species: H⁺, Li⁺, and Na⁺ -electrostatic energy considerations dictate that only neutral species, or neutral combinations of species, can be added to, or deleted from, solids \rightarrow the addition of cations requires the concurrent addition of electrons, and the extraction of cations is accompanied by either the insertion of holes or the deletion of electrons \rightarrow this phenomenon almost always involves materials that have at least some modicum of electronic conductivity -In some cases, however, the insertion or extraction of mobile atomic or ionic species causes irreversible changes in the structure of the host material. In extreme cases, the structure may be so distorted that it becomes amorphous

1. Examples of the Insertion of Guest Species into Layer Structures

-covalently bonded slabs containing several layers of atoms \rightarrow these slabs are held together by relatively strong, e.g., covalent, bonds. But adjacent slabs are bound to each other by relatively weak van der Waals forces \rightarrow the space between the tightly bound slabs is called the *gallery space*, and additional species can reside there.

-Depending upon the identity, size, and charge of any inserted species present, the interslab dimensions can be varied.

e.g. CdI_2 structure \rightarrow stoichiometry MX_2 , close-packed layers of negatively charged X ions held together by strong covalent bonding to positive M cations. In this case, the cations are octahedrally coordinated by six X neighbors, and the stacking of the X layers is hexagonal, with alternate layers directly above and below each other. This is generally described as ABABAB stacking (Fig. 6.1)

Cdl₂, Mg(OH)₂, Fe(OH)₂, Ni(OH)₂, and TiS₂



Fig. 6.1 Simple schematic model of a layer-type crystal structure with hexagonal ABABAB stacking. The empty areas between the covalently bonded slabs are called galleries



Fig. 6.2 Another type of model of a layer-type crystal structure. The example is TiS₂

2. Floating and Pillared Layer Structures

-mobile species move into and through sites in the previously empty gallery space between slabs of host material that are held together only by relatively weak van der Waals forces. \rightarrow floating \rightarrow guest results in a significant change in the interslab spacing

-in other cases the slabs are already rigidly connected by *pillars*, which partially fill the galleries through which the mobile species move \rightarrow the mobile species move through a 2-D network of interconnected tunnels, instead of through a sheet of available sites



Fig. 6.3 Schematic model of pillared layer structure

3. More on Terminology Related to the Insertion of Species into Solids

-Sheets: Single layers of atoms or ions. In the case of graphite, individual sheets are called graphene layers

-Stacks: Parallel sheets of chemically identical species

-Slabs or Blocks: Multilayer structures tightly bound together, but separated from other structural features. e.g. covalently bonded MX_2 slabs such CdI_2 -Galleries: The spaces between slabs in which the bonding is relatively weak, and in which guest species typically reside

-Pillars: Immobile species within the galleries that serve to support the adjacent slabs and to hold them together

-Tunnels: Connected interstitial space within the host structure in which the guest species can move and reside. Tunnels can be empty, partly occupied, or fully occupied by guest species

-Cavities: Empty space larger than the size of a single atom vacancy

-Windows: Locations within the host structure through which the guest species have to move in order to go from one site to another. Windows are typically defined by structural units of the host structure

4. Types of Inserted Guest Species Configurations

-several types of insertion reactions. In one case the mobile guest species randomly occupy sites, gradually filling them all up as the guest population increases \rightarrow the variation of the electric potential with composition indicates a single-phase solid solution reaction

-if the presence of the guest species causes a modification of the host structure, the insertion process can occur by the motion of an interface that separates the region into which the guest species have moved from the area in which there are no, or fewer, guest species. Thermodynamically, this has the characteristics of a polyphase reconstitution reaction and occurs at a constant potential.

-there can be two or more types of sites in the gallery space, with different energies, and the guest species can occupy an ordered array of sites, rather than all of them. When this is the case, changes in the overall concentration of mobile species

require the translation of the interface separating the occupied regions from those that are not occupied, again characteristic of a constant-potential reconstitution reaction. These moving interfaces can remain planar, or they can develop geometrical roughness. Several possibilities are illustrated schematically in Figs. 6.4–6.6.



Fig. 6.4 Random diffusion of guest species into gallery space



Fig. 6.5 Motion of two-phase interface when guest species is not ordered upon possible sites



Fig. 6.6 Motion of two-phase interface when guest species is ordered upon possible sites in the gallery space

5. Sequential Insertion Reactions

-If there are several different types of sites with different energies, insertion generally occurs on one type of site first, followed by the occupation of the other type of site. Figure 6.7: insertion of lithium into NiPS₃ \rightarrow two types of sites available (random occupation in both cases) \rightarrow 1.4 Li



Fig. 6.7 Coulometric titration curve related to the insertion of lithium into NiPS₃. There is random filling of the first two types of sites. A reconstitution reaction occurs above about 1.4 Li

Another example: different types of sites available for the insertion of Li ions involves the host $K_xV_2O_5$ structure \rightarrow different types of sites for guest ions



Fig. 6.8 (010) projection of the $K_x V_2 O_5$ structure, showing the different types of sites for the guest species

Coulometric titration curve for the insertion of Li ions into a member of this group of materials \rightarrow the reaction involves three sequential steps.

- (i) Up to about 0.4 Li can be incorporated into the first set of sites randomly.
- (ii) This is followed by the insertion of another 0.4 Li into another set of sites in an ordered arrangement → there are two different lithium arrangements, with a moving interface between them → two phases present, so this corresponds to a reconstitution reaction
- (iii) This is then followed by another reconstitution reaction, the insertion of about one additional Li into another ordered structure.



Fig. 6.9 Coulometric titration curve for the insertion of Li into K_{0.27}V₂O₅

A different type of ordered reaction involves selective occupation of particular galleries, and not others \rightarrow "staging" If alternate galleries are occupied and intervening ones are not \rightarrow "secondstage" structure If every third gallery is occupied, the structure is "third stage" and so forth.

Stage 1	Stage 2	Stage 3	Stage 4	
0000				
	0 0 0 0			
0000		0000		
			0000	
0 0 0 0	0000			
		0000		
0000	0000			
			0000	
0000				
	0000	0000		
0000				
	Gron	hana lavor		
	Grap	Graphene layer		
0 0	○ ○ Lithi	Lithium layer		

Fig. 6.10 Simple model depicting staging

6. Coinsertion of Solvent Species

In some cases it is found that species from the electrolyte can also move into the gallery space. This tends to be the case when the electrolyte solvent molecules are relatively small, so that they can enter without causing major disruption of the host structure.

7. Insertion into Materials with Parallel Linear Tunnels -mobile guest species in the material Hollandite, that has a crystal structure with parallel linear tunnels, rather than slabs. At low temperatures the interstitial ions within the tunnels are in an ordered arrangement upon the available sites. In addition, there is coordination between the arrangement in one tunnel with that of other nearby tunnels. Thus there is 3-D ordering of the guest species.

Fig. 6.11



Fig. 6.11 Hollandite structure. Viewed along the c-axis

As the temperature is raised somewhat, increased thermal energy causes the ordered interaction between the mobile ion distributions in nearby tunnels to relax, although the ordering within tunnels is maintained.

At even higher temperatures the in-tunnel ordering breaks down, so that the species are distributed randomly inside the tunnels, as well.



Fig. 6.12 Influence of temperature upon various types of order in structure with parallel tunnels

8. Changes in the Host Structure Induced by Guest Insertion or Extraction

(1) Conversion of host structure from crystalline to amorphous -examples in which an initially crystalline material becomes amorphous as a result of the insertion of guest species and the corresponding mechanical strains in the lattice \rightarrow this often occurs gradually as the insertion/extraction reaction is repeated, e.g., upon electrochemical cycling



Fig. 6.13 Discharge curve observed during the initial insertion of lithium into a material that was initially V_6O_{13}

Fig. 6.13. the first insertion of lithium into crystalline $V_{6}O_{13}$ shows that a sequence of reconstitution reactions take place that give rise to a series of different phases

Fig. 6.14. After a number of cycles \rightarrow a single-phase insertion reaction. X-ray diffraction experiments confirmed amorphous



Fig. 6.14 Discharge curve observed during the 20th insertion of lithium into a material that was initially V_6O_{13}

Fig.6.15. lithium inserted into initially V_2O_5 Formation of a series of different phases as lithium added

 \rightarrow forming the ϵ and δ structures, if only up to about 1 Li was inserted into αV_2O_5 . The addition of more lithium resulted in the formation of different structural modifications, called the y and ω structures, which have nominal compositions of Li₂V₂O₅ and $Li_3V_2O_5$, respectively. These two reactions are not reversible, however. When lithium was extracted from the ω phase, its charge-discharge curve became very different, exhibiting the characteristics of a single phase with a wide range of solid solution. The amount of lithium that could be extracted from this phase was quite large, down to a composition of about $Li_{0.4}V_2O_5$. Upon the reinsertion of lithium, the discharge curve maintained the same general form, indicating that a reversible amorphous structure had been formed during the first insertion process.



Fig. 6.15 The variation of the potential as lithium is added to V_2O_5 . When the composition reached $Li_3V_2O_5$ an amorphous phase was formed

(2) Dependence of the Product upon the Potential -Si or other materials: reacted with lithium to a low potential (high lithium activity) \rightarrow irreversible reaction occurs that results in the formation of fine particles of amorphous silicon in an inert matrix. Upon cycling, the amorphous Li–Si structure shows both good capacity and reversibility. However, if further lithium is inserted, going to a potential below 50mV, a crystalline Li–Si phase forms instead of the amorphous one. -Because of the light weight of silicon, the large amount of lithium that can react with it, and the attractive potential range, it appears as though silicon or its alloys may play an important role as a negative electrode reactant in lithium batteries in the future.

(3) Changes upon the Initial Extraction of the Mobile Species -initial extraction of a mobile species that is already present in a solid \rightarrow initial composition of about $\text{Li}_{0.6}\text{V}_2\text{O}_4 \rightarrow$ the potential starts between 3.0 and 3.5V vs. pure Li The initial lithium could be essentially completely deleted from the structure, causing the potential to rise to over 4V. When lithium was subsequently reinserted, the discharge curve had a quite different shape, indicating the presence of a reconstitution reaction resulting in the formation of an intermediate phase.



Fig. 6.16 Initial charging and discharge curves of a material with a composition about Li_{0.6}V₂O₄

9. The Variation of the Potential with Composition in Insertion Reaction Electrodes

(1) Introduction

The externally measured electrical potential of an electrode is determined by the electrochemical potential of the electrons within it, η_{e^-} , called Fermi level, $E_F \rightarrow$ not absolute value, always measured as differences \rightarrow voltage of electrochemical cell is the electrically measured difference between the Fermi levels of the two electrodes

 $\Delta E = \Delta \eta_{e^-}$

the measured potential of an electrode often varies with its composition, e.g., as guests species are added to, or deleted from, a host material. The relevant compositional parameter is the chemical potential of the electrically neutral electroactive species. If M⁺ within the electrode, the chemical potential

 $\mu_M = \eta_{M+} + \eta_{e-}$

Under open circuit conditions there is no flux of ions through the cell

$$d\eta_{M+}/dx = \Delta \eta_{M+} = 0$$

the measured voltage across the cell is simply related to the difference in the chemical potential of the neutral electroactive species in the two electrodes by

$$\Delta E = \Delta \eta_{e^-} = (-1/z_{M^+}q)\Delta \mu_M$$

-to understand the potentials of electrons is based upon the electron energy band model Metal: free-electron theory Semiconductor: CB, VB, energy gap, dopants.... (2) The Variation of the Electrical Potential with Composition in Simple Metallic Solid Solutions insertion of mobile guest species \rightarrow a solid solution of the guest species in the host crystal structure \rightarrow potential is the variation of the chemical potential of the neutral guest species as a function of its concentration \rightarrow two components, electron concentration in the host material and concentration of the ionic guest species, M⁺

(3) Configurational Entropy of the Guest Ions

-guest \rightarrow host

-There will be a contribution to the total free energy due to the configurational entropy S_c related to the random distribution of the guest atoms over the available sites.

 $S_{c} = -kln[x/(x_{0}-x)]$

where x is the concentration of guest ions and x_0 is the concentration of identical available sites. k is Boltzmann's constant

This configurational entropy contribution to the potential is the product of the absolute temperature and the entropy. Assumption: (i) no interaction between nearby guest species, (ii) only one type of site available for occupation



Fig. 6.17 Contribution to the potential due to the configurational entropy of a random distribution of the guest ions upon the available identical positions in a host crystal structure. The values on the abscissa are the fractional site occupation, and those on the ordinate are mV

(4) The Concentration Dependence of the Chemical Potential of the Electrons in a Metallic Solid Solution electrochemical potential of the electrons can be approximated by the energy of the Fermi level E_F . In the free-electron model

 $E_F =$

 \rightarrow

where m is the electron mass, $N_{\rm A}$ is Avogadro's number, $V_{\rm m}$ is the molar volume, and $E_{\rm F}$ is calculated from the bottom of the conduction band

 \rightarrow the electronic contribution to the total chemical potentialis proportional to the two-third the power of the guest species concentration. More generally the electron mass is replaced by an effective mass m^{*} (5) Sum of the Effect of These Two Components upon the electrical potential of a metallic solid solution \rightarrow composition dependence of the electrode potential in a metallic solid solution

smaller effective masses make the first term larger, and thus results in the potential being more composition – dependent



Fig. 6.18 Calculated influence of the value of the electronic effective mass upon the composition dependence of the potential in an insertion reaction in a simple metal

"tungsten bronze." In this family of oxides the electron energy spectrum approximates that of a free-electron metal.



Fig. 6.19 Variation of the potential as a function of lithium concentration in Li_xNa_{0.4}WO₃

(6) The Composition: Dependence of the Potential in the Case of Insertion Reactions that Involve a Two-Phase Reconstitution Reaction

e.g. potential is composition-independent:

insertion and extraction of lithium in materials with the composition $Li_4Ti_5O_{12}$ (spinel structure). spinel structure: AB_2O_4 , where the A species resides on tetrahedral sites and the B species on octahedral sites within the closepacked face centered cubic oxygen lattice. If one assumes this general stoichiometry, one of the four lithium ions would share the octahedral sites with the titanium ions, and the other ones would reside on tetrahedral sites. Thus the composition can be written as $Li_3[LiTi_5]O_{12}$, or alternatively, $Li[Li_{1/3}Ti_{5/3}]O_4$. additional lithium ion can react with this material,

 $Li + Li[Li_{1/3}Ti_{5/3}]O_4 = Li_2[Li_{1/3}Ti_{5/3}]O_4$

 \rightarrow a moving-interface reconstitution reaction.

Fig. 6.20. Because of the small volume change, negligible hysteresis, and rapid kinetics. The one disadvantage is that its potential is unfortunately about half way between the negative and positive electrode potentials in most lithium batteries. -hysteresis (when charging and discharging): related to mechanical strain energy, i.e., dislocation generation and motion, as a consequence of volume changes that occur due to the insertion and extraction of the guest



Fig. 6.20 Charge and discharge curves for two different Li₄Ti₅O₁₂ cells

lithium with FePO₄

Material is prepared (in air) as $LiFePO_4$, and the initial reaction within the cell involves charging, i.e., deleting lithium from it. This lithium goes across the electrochemical cell and into the carbon material in the negative electrode. The reaction that occurs at the operating potential during the initial charge can be simply written as

 $LiFePO_4 = Li+FePO_4$ 1st-3rc 4 Voltage (V) 3 Cathode: LiFePO₄ Anode: Li Metal 1st-3rd 0.12mA/cm2 PC/DMC/1MLiPF₆ 1 0 2040 80 10012014060 1600 Capacity (mAh/g)

Fig. 6.21 Charge and discharge curves for the reaction of lithium with FePO₄