

Prob. 8.4 Evaporation rate under varying wall temperature for a pulsed reactor: If 0.1 MJ of plasma energy is dumped on 1 m^2 of a 1000 K vanadium reactor wall in 0.1 ms, estimate the peak temperature, number of atoms evaporated by the thermal spike per m^2 , and the wall thickness loss.

Ans. The surface temperature rise for an energy dump onto a surface over a brief interval of time is given by
 $\Delta T = 2W/S (\pi C_p K \rho_m \tau)^{1/2} \text{ (K)}$, (cf. lecture note)

where $W = \text{energy dump} = 0.1 \text{ MJ} = 1 \times 10^5 \text{ J}$
 $S = \text{wall surface area for dump} = 1 \text{ m}^2$
 $C_p = \text{specific heat for vanadium @ } 1000 \text{ K} = 640 \frac{\text{J}}{\text{kg K}}$
 $K = \text{thermal conductivity " @ } 1000 \text{ K} = 38.6 \text{ W/(m-K)}$ (Table 8.10)
 $\rho_m = \text{density} = 5870 \text{ kg/m}^3$ (Table 8.10)
 $\tau = \text{time interval of dump} = 0.1 \text{ ms} = 1 \times 10^{-4} \text{ s}$

$\therefore \Delta T = 937 \text{ K}$, then the peak temperature is $1000 + \Delta T = \underline{1937 \text{ K}}$
 The number of wall atoms evaporated is given by (cf. lecture note)

$$\Delta n/S \approx 0.1 \tau \phi_n(T_{\text{max}}) \text{ (atoms/m}^2\text{)},$$

where $\phi_n = \text{evaporation flux at the peak temperature}$
 $= 2.6 \times 10^{24} \alpha p / (AT)^{1/2} \text{ (atoms/m}^2\text{.s)}$ (Eq. 8.24)

with $\alpha = \text{sticking coefficient} \approx 1$, $p = \text{vapor pressure} = 1 \times 10^1 \text{ Pa}$,
 $A = \text{atomic weight} = 50.942 \text{ g/mole}$, $T = 1937 \text{ K}$. (Fig. 8.25)

then $\phi_n(T_{\text{max}}) = 8.277 \times 10^{20} \text{ atoms/(m}^2\text{.s)}$

$$\therefore \Delta n/S = 0.1 (1 \times 10^{-4} \text{ s}) (8.277 \times 10^{20} \text{ atoms/(m}^2\text{.s)}) = \underline{8.28 \times 10^{15} \text{ atoms/m}^2}$$

The wall thickness loss due to the evaporation of these atoms is

$$\Delta x = \frac{\Delta n/S}{n_w} = \frac{\# \text{ of atoms evaporated per unit area}}{\# \text{ of wall atoms per unit volume}}$$

where $n_w = 6.93 \times 10^{28} \text{ atoms/m}^3$ for vanadium (cf. Table 8.10)

$$\therefore \Delta x = \frac{8.28 \times 10^{15} \text{ atoms/m}^2}{6.93 \times 10^{28} \text{ atoms/m}^3} = \underline{1.19 \times 10^{-13} \text{ m}}$$

[Note for a constant wall temperature @ 1000 K for vanadium wall (cf. Example 8.4)
 $dx/dt = 1.7 \times 10^{-11} \text{ m/s} = 0.54 \text{ mm/a}$, too high wall erosion!]

Fig. 8.25 Equilibrium vapor pressure for various wall materials versus temperature. For dashed curves read the top temperature scale (Based on data of Honig and Kramer 1969)

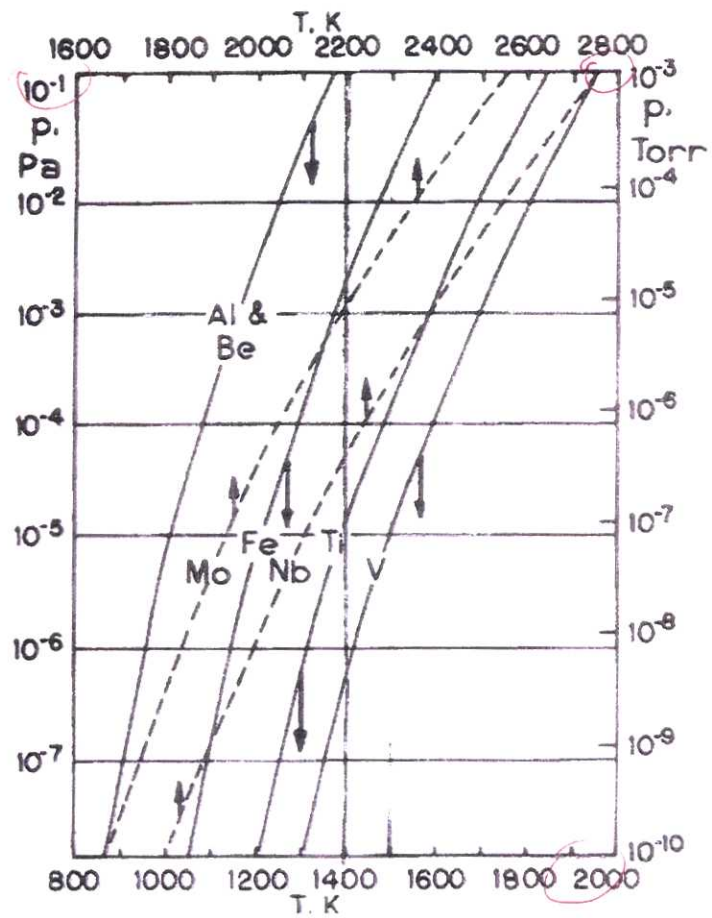


Table 8.10 Thermal properties of various materials. Data are from Behrisch (1972), Table 1

	V	Nb	Mo	C	SS 304
Z, atomic number	23	41	42	6	18 % Cr
A, atomic mass (gm/mole)	50.94	92.9	95.94	12	8 % Ni
ρ_m , mass density (kg/m ³)	5,870	8,570	9,010	2,250	7,900
n_w , atomic density (10 ²⁸ atoms/m ³)	6.93	5.56	5.65	11.3	8.6
T_m , melting temperature (K)	2,192	2,688	2,883	(vap.) 3,925	1,400
ΔH , heat of sublimation (eV/atom)					
	300 K	5.33	7.5	6.83	8.2-10.4
	500 K	5.27	7.43	6.78	
	1,000 K	5.16	7.28	6.63	
	1,500 K	4.96	7.14	6.48	
	2,000 K	4.75	6.97	6.3	
k, thermal conductivity (W/m K)					
	500 K	33.1	56.7	130	80-100
	1,000 K	38.6	64.4	112	49-64
	1,500 K	44.7	72.1	97	40-50
	2,000 K	50.9	79.1	88	30-40
c_p , specific heat (J/kg K)					
	500 K	500	280	254	1,200
	1,000 K	640	300	290	500
	1,500 K	700	330	330	1,800
	2,000 K	850	370	380	2,000

C. Plasma Materials Interactions (cont'd)

iii) Near-surface wall modifications

- Phase changes - can occur in surface films (e.g., as oxides, carbides, and hydrides) and also occur internally (e.g., strain-induced austenitic-martensitic transformation in stainless steels). Flaking and hydrogen cracking may result from the changes of thermal conductivity and the steel's mechanical properties, respectively.
- Alloy composition changes - nuclear transmutations (in both gaseous and solid transmutants), preferential sputtering (with high sputtering yield elements preferentially removed from the surface of an alloy), and diffusion + surface segregation (impurities like carbon diffuse to the surface where they are desorbed or sputtered away; in stainless steel, chromium segregates to the surface as an oxide.)
- microstructural changes - in addition to neutron-induced radiation damage, ion impact produces vacancies, interstitials, and dislocation loops in the first 10 nm of the surface; these defects lead to internal stress, swelling, creep, recrystallization and grain growth, and tritium trapping in the near-surface.
- macrostructural changes - include nonuniform erosion and redeposition (cf. building up flakes), cracking, topological changes (cf. blistering, exfoliation (intergranular corrosion), sputtering (yielding a honeycomb-like surface)), and tungsten fuzz (creating a fuzzy surface under high heat flux)
 ↳ may have lower sputtering rate
- Property changes - may affect physical properties (electrical and thermal conductivity, emissivity, density, optical reflectivity, radioactivity, work function, and magnetic permeability) and mechanical properties (ductility, creep rate, crack growth rate)

D. PFC (Plasma Facing Components)

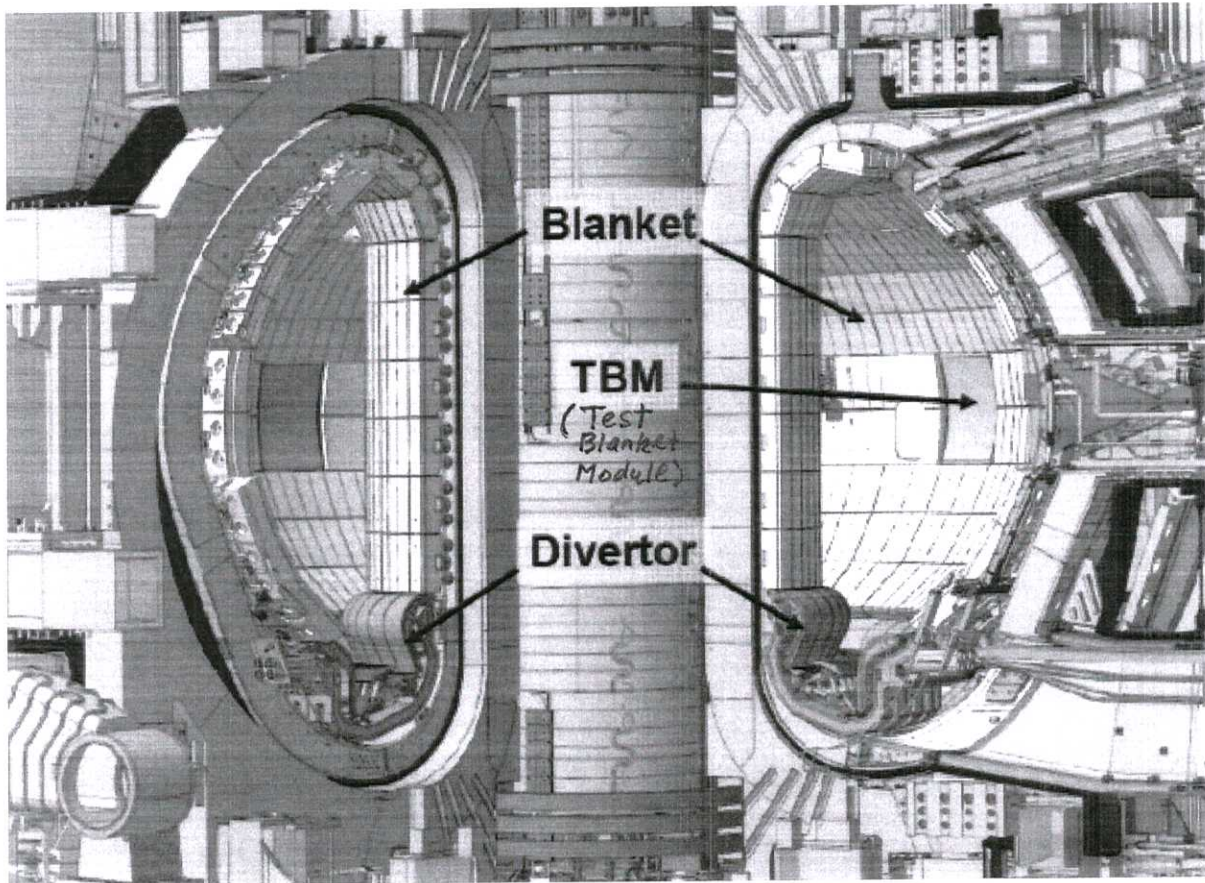
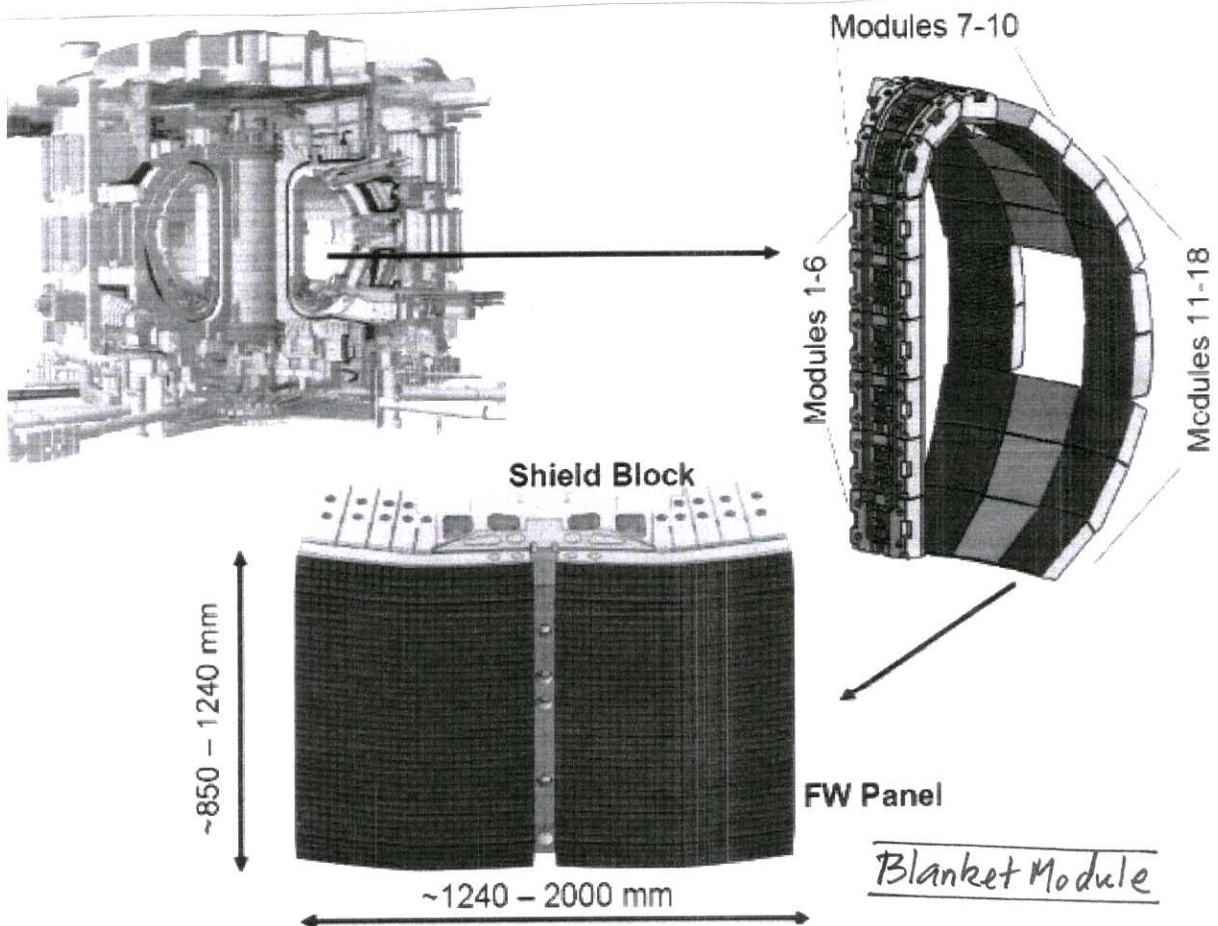


Fig. 1. ITER Plasma facing components
[Ref. M. Merola, et al., ITER plasma-facing Components, *Fusion Engrg & Design* 85 (2010) 2312-2322.]



Blanket design

Table 1
Summary of main blanket design parameters.

Parameter	Value
Number of BMs	440
Typical module dimension	1.4 m × 1.0 m × 0.45 m
Max allowable mass of one module	4.5 t
Total mass on BMs	1530 t
Coolant type	Water
Inlet temperature (°C)	100
Inlet pressure at chimney bulk head (MPa)	3.0
Minimum FW panel flow velocity (m/s)	1.5
Minimum SB flow velocity (m/s)	0.8
Materials	
Heat sink of FW panels	CuCrZr-IG alloy
Armour	Beryllium (S-65C or equivalent)
Pipes and hydraulic connections	316L
Shield block	316L(N)-IG
Design heat flux on FW panels	1–2 and 5 MW/m ² depending on location
Max neutron damage in Be/heat sink/steel pipe/SB	1.6/5.3/3.4/2.3 dpa

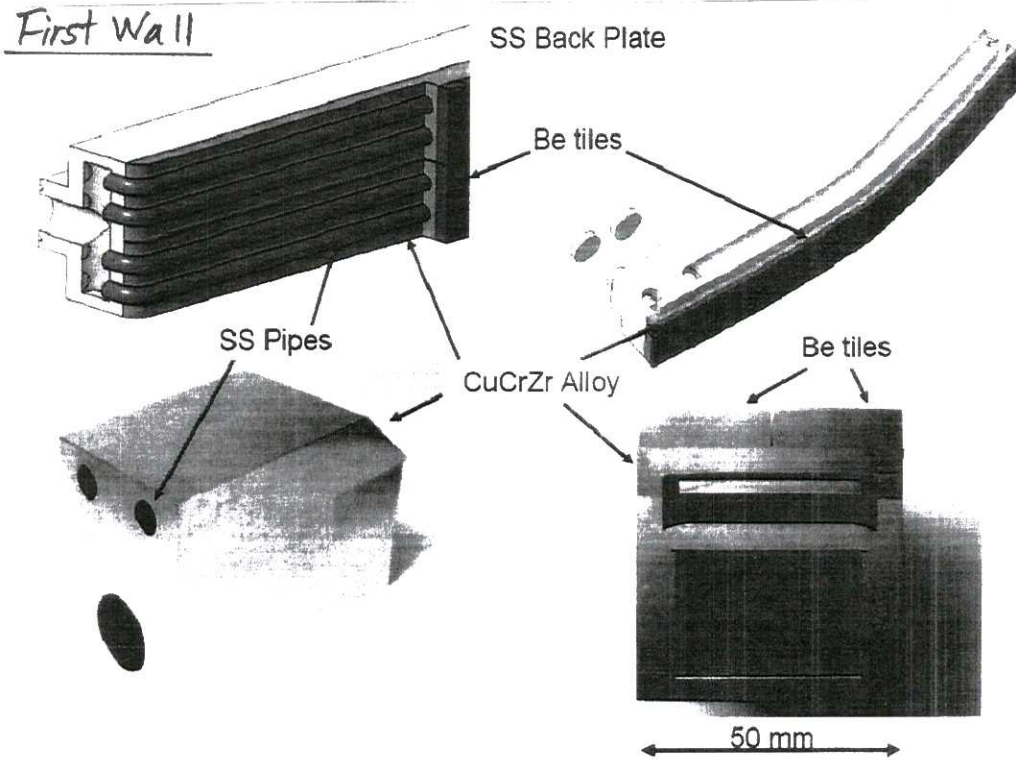


Fig. 3. Illustration of the basic FW panel structure and fingers. Left: normal heat flux fingers (concept with steel cooling pipes). Right: enhanced heat flux fingers (concept with rectangular channels).

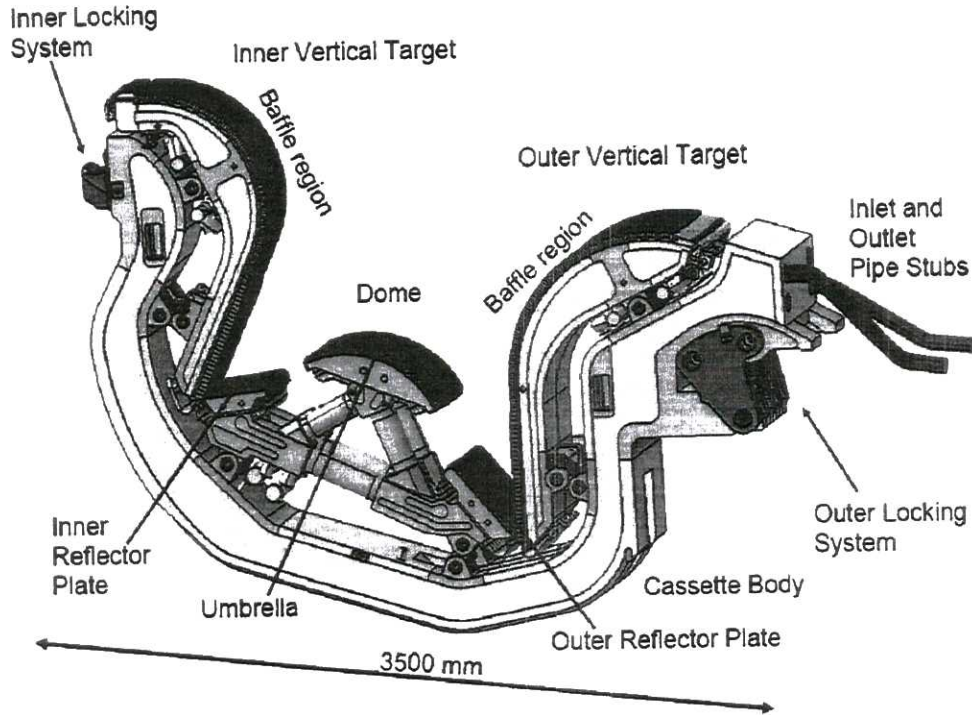


Fig. 6. Divertor cassette assembly.

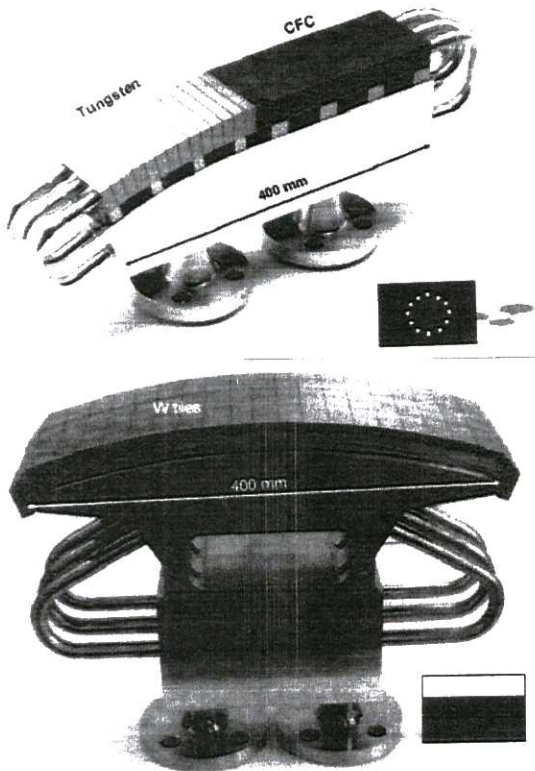
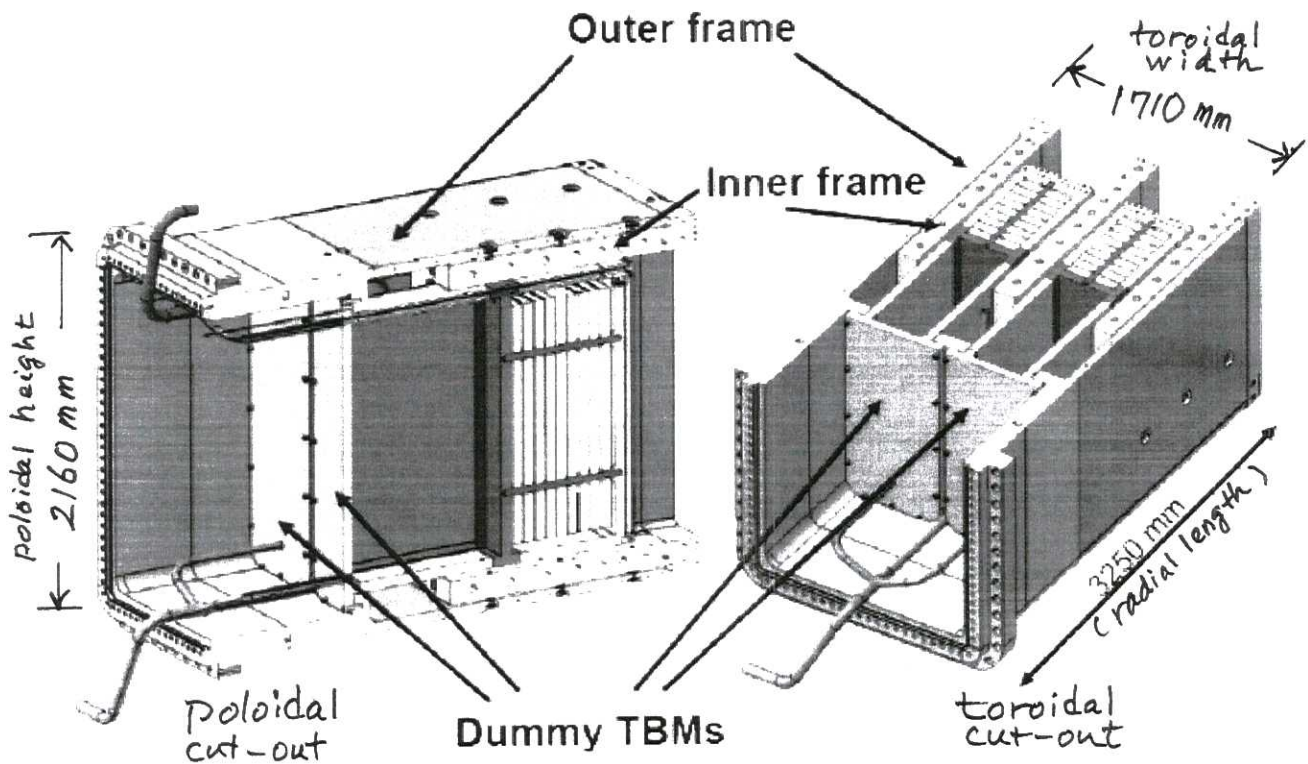


Fig. 8. Vertical target (top) and dome (bottom) medium-scale prototypes.

Table 2
Summary of main divertor design parameters.

Parameter	Value
Size	
Toroidal extent of a cassette	6.67°
Number of cassettes	54
Coolant type	Water
Inlet temperature (°C)	100
Inlet pressure at pipe stubs (MPa)	4.2
Max permissible pressure drop (MPa)	1.6
Max permissible total flow rate (kg/s)	1000
Materials	
Cassette body and supporting structures of PFCs	SS 316L(N)-IG and XM-19
Heat sink of PFUs	CuCrZr-IG alloy
→ Armour	CFC, pure sintered and deformed W ←
Pipes	316L
Pins of multilink attachments	C63200 Ni-Al bronze
Inner and outer locking system: nose and knuckle	C63200 Ni-Al bronze
Hard cover plates of divertor rails	Steel 660
Shear pins of hard cover plates	C63200 Ni-Al bronze
Design load for target in strike point region	~10 MW/m ²
Design load of target in baffle region	~5 MW/m ²
Min critical heat flux margin	1.4
Thickness of carbon fibre composite (CFC) at end of life	>4 mm
Plasma-facing surface area	~190 m ²

Test Blanket Module (TBM)



The TBM frame is a water-cooled 316L(N)-IG steel port structure, which is based on the same materials and manufacturing process as those used for the blanket shield block (SB).

Note: Attached references on plasma-facing components, i.e., M. Merola, et al., ITER plasma-facing components, Fusion Engineering & Design 85 (2010) 2312-2322; D. Stork, et al., Developing... plasma facing materials; Journal of Nuclear Materials 455 (2014) 277-291; and V. Philipps, Tungsten as material for plasma-facing components in fusion devices, Journal of Nuclear Materials 415 (2011) 52-59.

§ 8.10 Irradiation Facilities

REPORT MDC E3016
16 MAY 1986

Assessment of Neutron Requirements and Potential Sources for Fusion Development

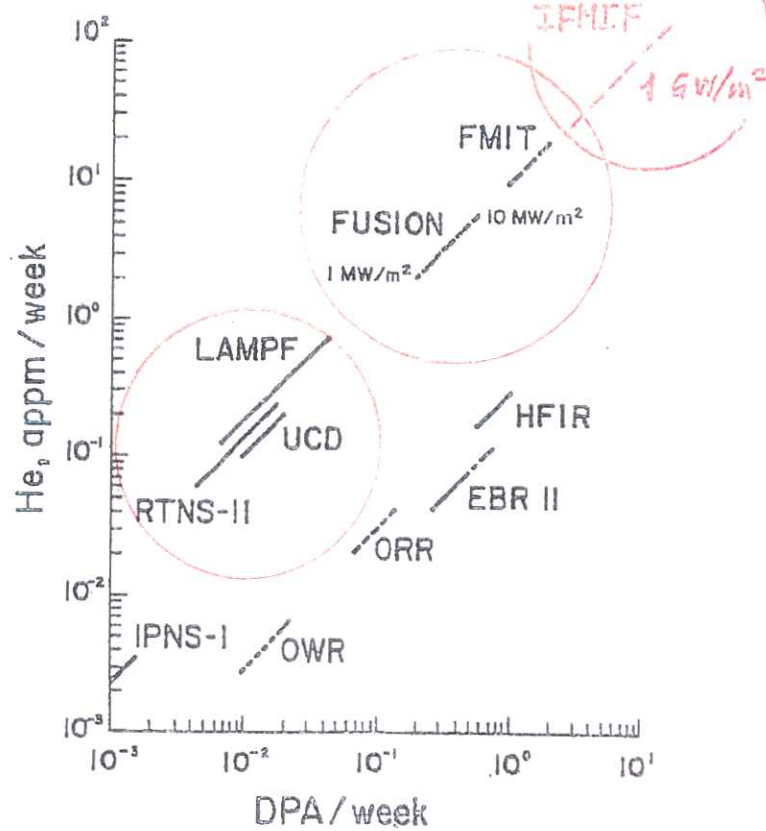


Figure 4-20

Production of He per dpa in Various Neutron Source Devices

Estimate for Neutron Irradiation of 10-50 dpa

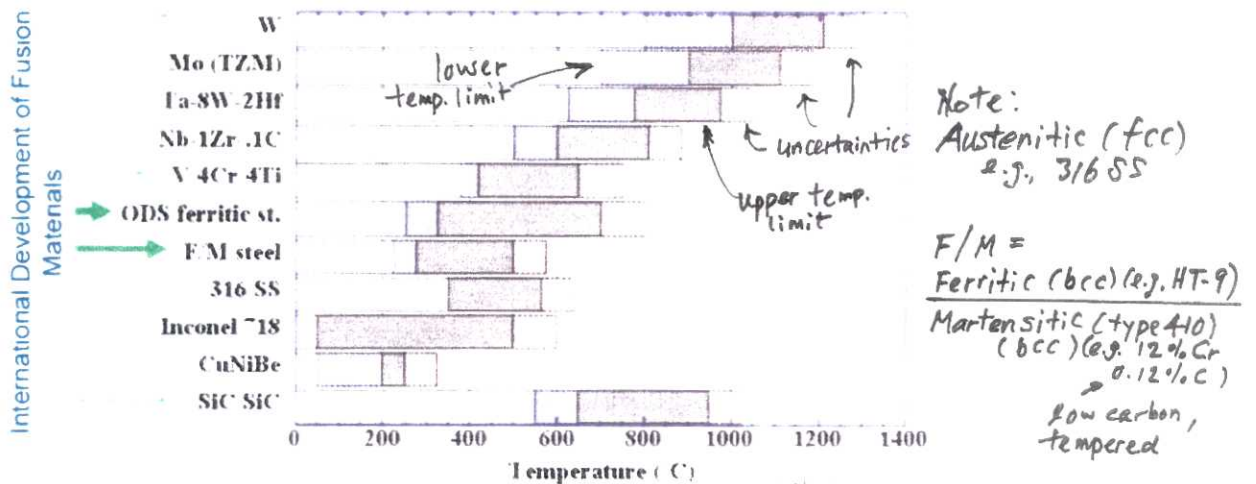


Fig. 8.47 Operating temperatures of some structural materials. Arrows indicate favored materials (Zinkle and Ghoniem 2000)

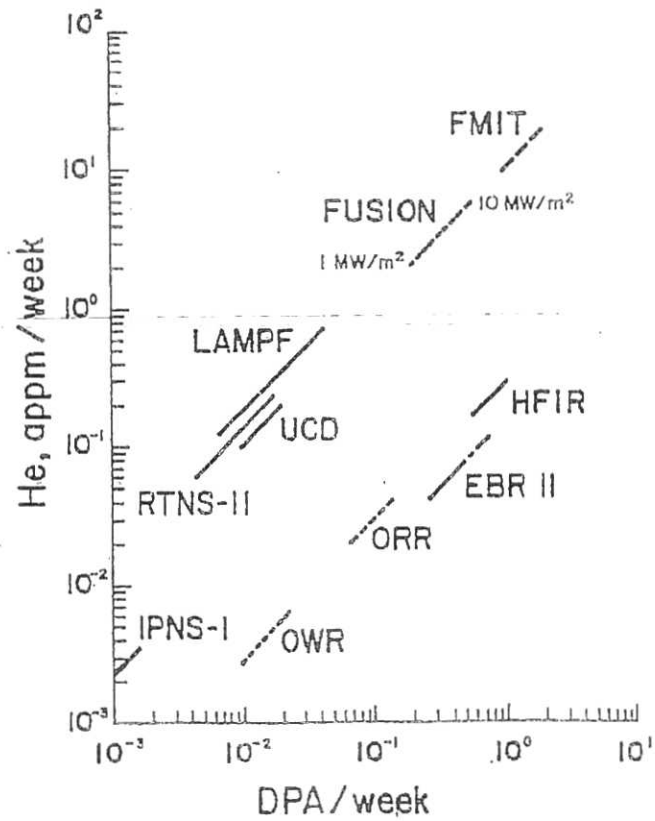


Figure 4-20

Production of He per dpa in Various Neutron Source Devices

ACCELERATOR FLUX ($10^{12}n/cm^2 \cdot sec$)	EXPERIMENTAL VOLUME (LITERS)				
	< 1	1 - 10	10 - 25	25 - 50	> 50
> 1000	FMIT				
500-1000					LAMPF (U target)
100-500	FMIT				
50-100	UCD				LAMPF (Cu target)
10-50	RTNS-II				
5-10					
< 5	IPNS-I (REF)				

Experiment volume (liters)

Fast Flux ($n/cm^2 \cdot sec$)	<1	1-10	10-25	25-50	50-75	75-100	100-200	200-350	>350
$>10^{15}$		FFTF EBR-II		FFTF					
$5 \times 10^{14} - 10^{15}$	HFIR	HIFR							
$10^{14} - 5 \times 10^{14}$	NSBR HFIR HFBR MITR ATR	ETR, ATR GETR ORR, HFIR MITR, HFR	ETR	ETR					ETR
$5 \times 10^{13} - 10^{14}$									
$10^{13} - 5 \times 10^{13}$	MURR	OWR	PBR					ORR	
$5 \times 10^{12} - 10^{13}$		BSR							
$< 5 \times 10^{12}$		ATR	ATR					ATR	

LIST OF ACRONYMS

ACRONYM	DEFINITION
ADIP	Alloy Development for Irradiation Performance → FHIT
ANL	Argonne National Laboratory FWB
ATR	Advance Test Reactor GA
BANG	Brookhaven Accelerator-Based Neutron Generator GETR
BCSS	Blanket Comparison and Selection Study HEDL
BNL	Brookhaven National Laboratory HFBR
CCT	Center Cracked Tension → HFIR
CLIRA	Closed Loop in Reactor Assembly HFR
CRFR	Compact Reversed-Field Pinch Reactor IING
CRNL	Chalk River National Laboratory INGRID
CV	Continuous Wave INS
DAFS	Damage Analysis and Fundamental Studies INTOR
DBTT	Ductile Brittle Transition Temperature → IPNS
d.c.	Direct Current ITR
DEMO	Fusion Demonstration Reactor JAERI
DOE	Department of Energy JET
dpa	Displacement per atom JWTR
DT	Deuterium-tritium KARIN
→ EBR-II	Experimental Breeder Reactor-II → LAMPF
ECRH	Electron Cyclotron Resonance Heating LANL
EOL	End of Life LASREF
e.p.	Electro-Potential LLNL
EPRI	Electric Power Research Institute LOCA
ETEC	Energy Technology Engineering Center MARS
ETF	Engineering Test Facility MCF
ETR	Engineering Test Reactor MDAC
FED	Fusion Engineering Device MFR
FERF	Fusion Engineering Research Facility MFTF m+T
FETF	Fast Flux Test Facility MFTF-B
	MHD
	MIT
	MITR
	MOA
	MTR
	MURR
	NBIHS
	Fusion Materials Irradiation Test Facility
	First Wall/Blanket
	GA Technologies, Inc.
	General Electric Test Reactor
	Hanford Engineering Development Laboratory
	High Flux Beam Research Reactor
	High Flux Isotope Reactor → ORR
	High Flux Reactor → OWR
	Intense Neutron Generator
	Intense Neutron Generator for Radiation-Induced Damage
	Intense Neutron Source
	International Torus
	Intense Pulsed Neutron Source
	Ignition Test Reactor
	Japanese Atomic Energy Research Institute
	Joint European Torus → RTNS
	Japan Materials Testing Reactor
	Karlsruhe Ring-Ion-Source Neutron Generator
	Los Alamos Meson Physics Facility
	Los Alamos National Laboratory
	Los Alamos Spallation Radiation Effects Facility
	Lawrence Livermore National Laboratory
	Loss of Coolant Accident
	Mirror Advanced Reactor Study
	Magnetic Confinement Fusion
	McDonnell Douglas Astronautics Company
	Magnetic Fusion Reactor
	Mirror Fusion Test Facility Upgrade → UCD
	Mirror Fusion Test Facility (Tandem Configuration)
	Magnetohydrodynamic
	Massachusetts Institute of Technology
	Massachusetts Institute of Technology Reactor
	Materials Open Test Assembly
	Materials Test Reactor
	University of Missouri Reactor
	Neutralized-Beam Intense Neutron Source
	NRL
	NSBR
	OHEE
	ORELA
	ORIC
	ORNL
	Oak Ridge National Laboratory
	Oak Ridge Isochronous Cyclotron
	Oak Ridge Research Center
	Omega West Reactor
	Prime Candidate Alloy
	Puised White
	Research and Development
	Rapid Cycling Synchrotron
	Radiation Effects Facility
	Reverse-Field Pinch
	Field-Reversed Theta Pinch
	Rotating Target Neutron Source
	Single Cell Technology Demonstration Facility
	Spallation Neutron Source
	Technology Development Facility
	Transmission Electron Microscopy
	Tokamak Engineering Test Reactor
	Tokamak Fusion Test Reactor
	Tandem Mirror Reactor
	Tandem Mirror Experiment
	The Next Step
	Tritium Systems Test Assembly
	Absolute Temperature of melt (K or R)
	University of California-Davis
	University of California-Los Angeles
	University of New Mexico
	Ultimate Tensile Strength
	University of Wisconsin
	University of Wisconsin Tokamak Conceptual Design
	Vertical Test Assemblies
	Weapons Neutron Research Facility
	Yield Strength

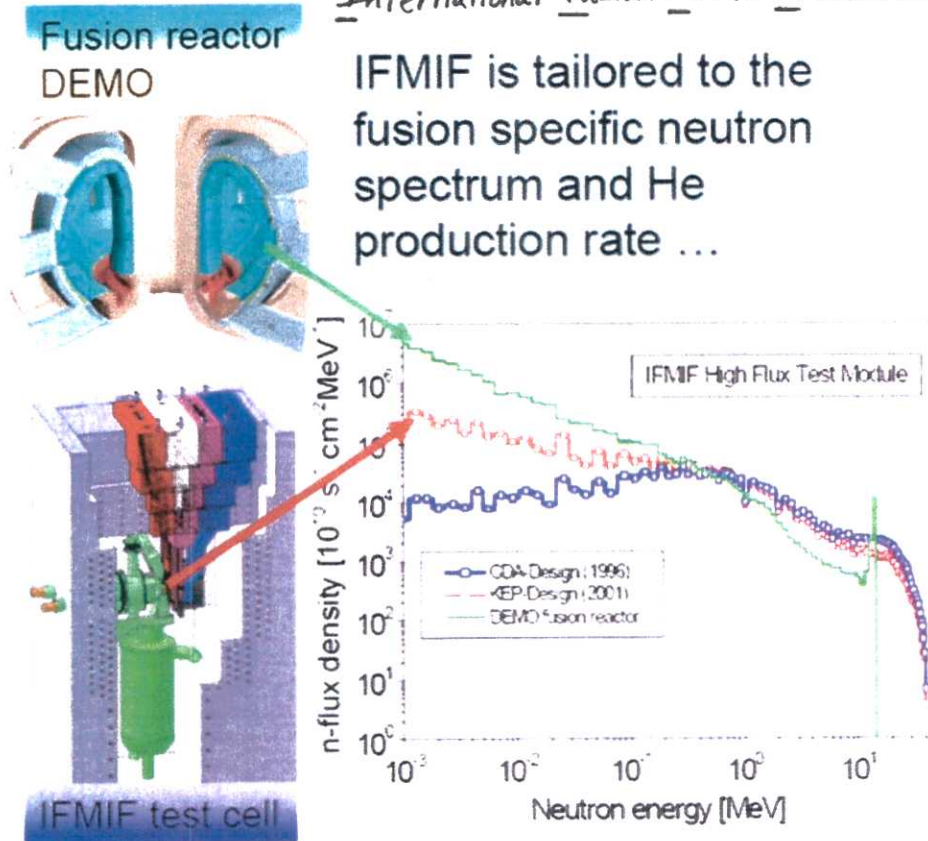
International Fusion Mat'ls Irradiation Facility :

Fig. 8.42 Comparison of predicted IFMIF and DEMO neutron energy spectra (Rieth 2008, From work by A. Möslang, KIT)

The capabilities of IFMIF will be:

- High flux region ($V = 0.5 \text{ L}$) $\geq 20 \text{ dpa/a}$
- Medium-flux region ($V = 6 \text{ L}$) $\geq 1 \text{ dpa/fpy}$ [fpy = full power year]
- Temperature control
- Miniaturized specimens
- Post-irradiation examination (PIE)
- Availability $\geq 70 \%$.

First phase: 3 years, half-intensity

- Screening candidate structural materials
- Calibrating data from fission reactors and ion beams.

Second phase: 20 years, full power test facility

} 30-yr facility

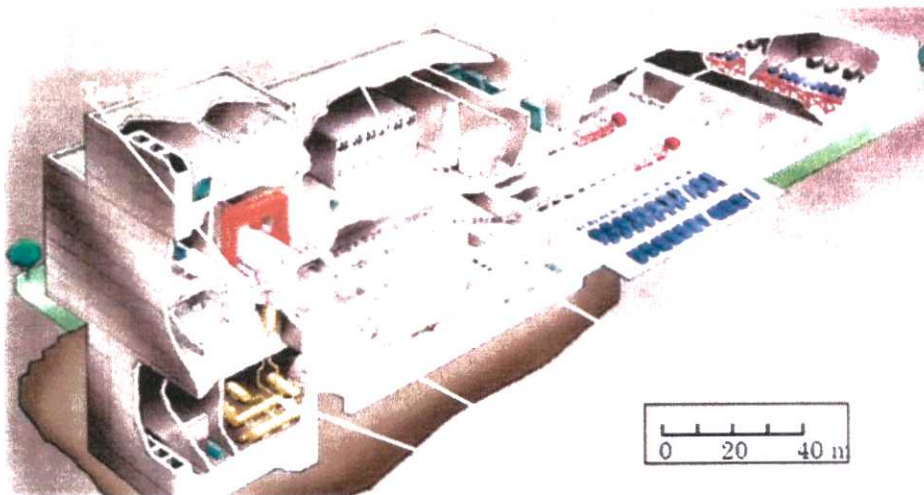


Fig. 8.43 The International Fusion Materials Irradiation Facility (IFMIF 2004)

The main IFMIF parameters :

→ 2 D ⁺ beams each	40 MeV, 125 mA
Beam deposition area on target	0.2 m × 0.05 m
Jet velocity	15 m/s
Average target heat flux	1 GW/m ²
Li flow rate	130 l/s
→ Pressure at Li surface	10 ⁻³ Pa
Hydrogen isotopes content in Li	<10 wppm
Impurity content (each C, N, O)	<10 wppm
Structure	SS-316
Back wall replacement period	11 months
Other components lifetime	30 years
Availability	>95 %

Another future facility -

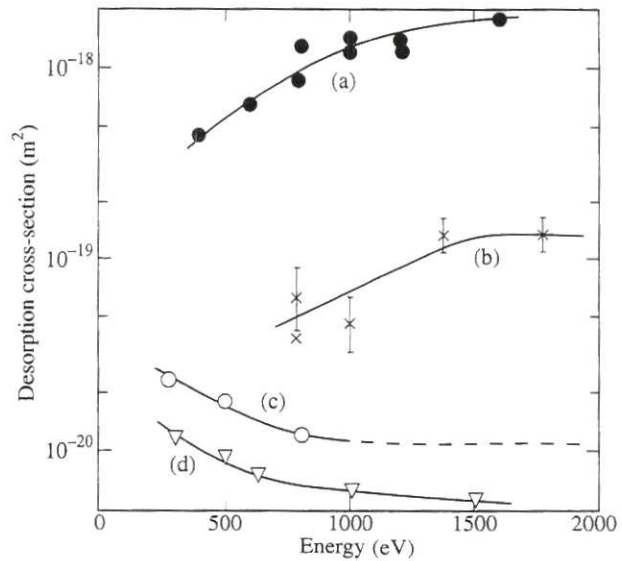
The European Spallation Source (ESS) in Lund, Sweden :

*produce intense neutron source by impact of
a 5 MW, 2.5 GeV proton beam onto a tungsten target
or Pb-Bi eutectic. [Note: the neutron source spectrum
will not match that of a fusion reactor.]*

Wall Conditioning :

- Adsorbed atoms have binding energies - $\sim 0.3 \text{ eV} - \sim 3 \text{ eV}$.
- Impurities (e.g., CO and H₂O) are typical adsorbents, besides hydrogen adsorbed on the surfaces.

Impurities can be desorbed by incident ions, neutrals, electrons, and photons. Desorption cross-section $\sim 10^{-18} \text{ m}^2$.



Ref. Wesson, Tokamaks

Fig. 9.6.1 Energy dependence of experimental desorption cross sections: (a) $^4\text{He}^+$ incident, carbon monoxide on nickel; (b) $^3\text{He}^+$ incident, hydrogen on tungsten; (c) $^4\text{He}^+$ incident, hydrogen on molybdenum; (d) H^+ incident, deuterium on nickel (Taglauer, E., in *Nuclear Fusion Special Issue*, ed. Langley, R.A., I.A.E.A., Vienna (1984).)

- Discharge cleaning \leftarrow 10-20 MHz @ 10^{-1} Pa
glow discharges, pulsed discharges and discharges excited by radiation at electron cyclotron frequency aim to optimize the removal of wall impurities by energetic ions, neutrals, electrons, or photons. [cf. Aleator]
- Thin metallic films ($< 10-100 \mu\text{m}$)
Evaporation of wires (e.g., Ti, Cr, Be) at high vapor pressures at typically $1500-2000 \text{ K}$ produces thin metallic films for improving the vacuum (cf. gettering)
- Non-metallic films ($\sim 1 \mu\text{m}$)
Low Z non-metallic films (e.g., C and B) to coat the walls of tokamaks in order to minimize the release of high Z impurities. (cf. carbonization and boronization)
Optimum wall temperature is about $300 \text{ }^\circ\text{C}$ to form a hard, adherent film, typically containing 40% hydrogen.