# 14 Bravais Lattices 

## Read

## Ott Chapter 7, 8

Krawitz Chapter 1.1 ~ 1.5
Hammond Chapter 2.1 ~ 2.4; 3.1 ~ 3.3; 5.1 ~ 5.6
Sherwood \& Cooper Chapter 3.1~3.7

Brief but very good summary can be found here
https://unlcms.unl.edu/cas/physics/tsymbal/teaching/SSP-927/Section\ 01_Crystal\ Structure.pdf

> Primitive lattice; one lattice point per unit cell
> Non-primitive lattice; more than one lattice point per unit cell
> In geometry and crystallography, a Bravais lattice is an infinite array of discrete points generated by a set of discrete translation operations described by: $\mathbf{R}=$ $n_{1} \mathbf{a}_{1}+n_{2} \mathbf{a}_{2}+n_{3} \mathbf{a}_{3}$. This discrete set of vectors must be closed under vector addition and subtraction. For any choice of position vector $\mathbf{R}$, the lattice looks exactly the same. ( $n_{i} ;$ any integer. $\mathbf{a}_{\mathbf{i}}$; primitive vectors which lie in different directions and span the lattice.)
> A crystal is made up of a periodic arrangement of one or more atoms (basis, motif) repeated at each lattice point. Consequently, the crystal looks the same when viewed from any of the lattice points.
> Two Bravais lattices are often considered equivalent if they have isomorphic symmetry groups. In this sense, there are only 14 possible Bravais lattices in 3dimensional space.

## 2D Bravais lattice (Plane lattice)

> In two dimensions, there are five Bravais lattices.

rectangular


$$
a \neq b
$$

$$
\gamma=90^{\circ}
$$

hexagonal


$$
\begin{gathered}
a=b \\
\gamma=120^{\circ}
\end{gathered}
$$

square


$$
\begin{gathered}
a=b \\
\gamma=90^{\circ}
\end{gathered}
$$



## 3D Bravais lattices

> The 14 Bravais lattices in 3 dimensions are obtained by coupling one of the 7 lattice systems (or axial systems) with one of lattice centerings. Each Bravais lattice refers to a distinct lattice type.
$>$ The lattice centerings are
$\checkmark$ Body (I): one additional lattice point at center of the cell.

$\checkmark$ Face (F): additional lattice points at centers of all the faces of the cell.
$\checkmark$ Base (A, B or C): additional lattice points at centers of each pair of cell faces.
> Not all the combinations of crystal systems and lattice centerings are needed to describe the possible lattices.
$>$ There are in total $7 \times 5(\mathrm{P}, \mathrm{I}, \mathrm{F}, \mathrm{C}, \mathrm{R})=35$ possible combinations, but many of these are in fact equivalent to each other.
$\checkmark$ For example, the tetragonal F lattice can be described by a tetragonal I lattice by different choice of crystal axes.
$\rightarrow$ This reduces the number of combinations to $14 . \rightarrow 14$ Bravais lattices


## Symmetry group

$>$ Complete set of symmetry elements $\rightarrow$ symmetry group
> Limited \# of symmetry elements (ten) \& all valid combination among them $\rightarrow 32$ crystallographic symmetry groups $\rightarrow 32$ point groups
$>$ Limited \# of symmetry elements (ten) + the way in which they interact with each other $\rightarrow$ limited \# of completed sets of symmetry elements (32 symmetry groups $=32$ point groups)
$>$ Point group $\leqslant$ symmetry elements in these groups have at least one common point and, as a result, they leave at least one point of an object unmoved

When a symmetry operation has a locus (a point, a line, or a plane) that is left unchanged by the operation, this locus is referred to as the symmetry element.

## 7 Crystal systems

$>$ Combination of symmetry elements \& their orientations w.r.t. one another defines the crystallographic axes $\rightarrow 7$ crystal systems
> Axes can be chosen arbitrarily, but are usually chosen w.r.t. specific symmetry elements present in a group
$\checkmark / /$ rotation axes or $\perp \mathrm{m}$
> All possible 3-D crystallographic point groups can be divided into a total of $\underline{7}$ crystal systems based on the presence of a specific symmetry element or specific combination of them present in the point group symmetry
$>7 \times 5$ types of lattices $\rightarrow 14$ different types of unit cells are required to describe all lattices (14 Bravais lattice)

## 7 Crystal systems, 6 Crystal family

Table 2.6 Seven crystal systems and the corresponding characteristic symmetry elements.

| Crystal system | Characteristic symmetry element or combination of symmetry <br> elements |
| :--- | :--- |
| Triclinic <br> Monoclinic <br> Orthorhombic | $\underline{\text { No axes other than onefold rotation or onefold inversion }}$ <br> Three mutually perpendicular twofold axes, either rotation or <br> Trigersion <br> Tetragonal <br> Hexagonal <br> Cubic |

Trigonal \& hexagonal can be described in the same type of the lattice $\rightarrow$ six crystal family
Axial lengths and angles ${ }^{\mathbf{a}}$
Cubic $a=b=c, \alpha=\beta=\gamma=90^{\circ}$
Tetragonal $a=b \neq c, \alpha=\beta=\gamma=90^{\circ}$
Orthorhombic
Rhombohedral $a \neq b \neq c, \alpha=\beta=\gamma=90^{\circ}$
Triclinic
Hexagonal $\quad a=b=c, \alpha=\beta=\gamma \neq 90^{\circ}$
Monoclinic
$a \neq b \neq c, \alpha=\beta=90^{\circ}, \gamma=120^{\circ}$
System $\quad$ Axial lengths and angles ${ }^{a} \quad$ Unit cell geometry AL university $^{\text {a }}$

Cubic

$$
a=b=c, \alpha=\beta=\gamma=90^{\circ}
$$



Tetragonal

$$
a=b \neq c, \alpha=\beta=\gamma=90^{\circ}
$$



Orthorhombic

$$
a \neq b \neq c, \alpha=\beta=\gamma=90^{\circ}
$$


${ }^{a}$ The lattice parameters $a, b$, and $c$ are unit-cell edge lengths. The lattice parameters $\alpha$, $\beta$, and $\gamma$ are angles between adjacent unit-cell axes, where $\alpha$ is the angle viewed along the $a$ axis (i.e., the angle between the $b$ and $c$ axes). The inequality $\operatorname{sign}(\neq)$ means that equality is not required. Accidental equality occasionally occurs in some structures.

Rhombohedral

$$
a=b=c, \alpha=\beta=\gamma \neq 90^{\circ}
$$

Hexagonal

$$
a=b \neq c, \alpha=\beta=90^{\circ}, \gamma=120^{\circ}
$$

Monoclinic

$$
a \neq b \neq c, \alpha=\gamma=90^{\circ} \neq \beta
$$



Triclinic

$$
a \neq b \neq c, \alpha \neq \beta \neq \gamma \neq 90^{\circ}
$$



## Selection of a unit cell

$>$ Trigonal \& hexagonal can be described in the same type of the lattice $\rightarrow$ six crystal family
> Different types of crystal systems (lattices) can be identified by the presence of specific symmetry elements and their relative orientation
Table 2.10 Lattice symmetry and unit cell shapes.

| Crystal family | Unit cell symmetry | Unit cell shape/parameters |
| :--- | :--- | :--- |
| Triclinic | 1 | $a \neq b \neq c ; \alpha \neq \beta \neq \gamma \neq 90^{\circ}$ |
| Monoclinic | $2 / \mathrm{m}$ | $a \neq b \neq c ; \alpha=\gamma=90^{\circ}, \beta \neq 90^{\circ}$ |
| Orthorhombic | $4 / \mathrm{mmm}$ | $a \neq b \neq c ; \alpha=\beta=\gamma=90^{\circ}$ |
| Tetragonal | $6=b \neq c ; \alpha=\beta=\gamma=90^{\circ}$ |  |
| Hexagonal and Trigonal | $6 / \mathrm{mmm}$ | $a=b \neq c ; \alpha=\beta=90^{\circ}, \gamma=120^{\circ}$ |
| Cubic | $\mathrm{m} \overline{3} \mathrm{~m}$ | $A-b=c ; \alpha=\beta=\gamma=90^{\circ}$ |


| Crystal system | Characteristic symmetry element or combination of symmetry <br> elements |
| :--- | :--- |
| Triclinic <br> Monoclinic <br> Orthorhombic | No axes other than onefold rotation or onefold inversion <br> Unique twofold axis and/or single mirror plane |
| Three mutually perpendicular twofold axes, either rotation or <br> Trigonal <br> Tetragonal <br> Hexagonal <br> Cubic | Unique threefold axis, either rotation or inversion |
|  | Unique fourfold axis e either rotation or inversion <br> Four threefold axes, either rotation or inversion, along four body <br> diagonals of a cube |

Rule \#1 --- symmetry of the unit cell should be identical to the symmetry of the lattice, except for translation

## Choice of unit cell is arbitrary

$\checkmark$ It is not always possible to select a primitive cell


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## Selection of a unit cell-2

Table 2.11 Rules for selecting the unit cell in different crystal systems.
$\left.\begin{array}{lll|}\hline \text { Crystal family } & \begin{array}{l}\text { Standard unit cell choice }\end{array} & \begin{array}{l}\text { Alternative unit cell choice }\end{array} \\ \hline \text { Triclinic } & \begin{array}{l}\text { Angles between crystallographic axes should be } \\ \text { as close to } 90^{\circ} \text { as possible but greater than or } \\ \text { equal to } 90^{\circ}\end{array} & \begin{array}{l}\text { Angle(s) less than or equal } \\ \text { to } 90^{\circ} \text { are allowed }\end{array} \\ \hline \begin{array}{ll}\text { Monoclinic } \\ \hline \text {-axis is chosen parallel to the unique twofold } \\ \text { rotation axis (or perpendicular to the mirror } \\ \text { plane) and angle } \beta \text { should be greater than but as } \\ \text { close to } 90^{\circ} \text { as possible }\end{array} & \begin{array}{l}\text { Same as the standard } \\ \text { choice, but } Z \text {-axis in place } \\ \text { of } Y, \text { and angle } \gamma \text { in place of }\end{array} \\ \beta \text { are allowed }\end{array}\right]$
> Rule \#3 --- minimum volume (or min \# of lattice points inside the unit cell)

Rule \# 1, 2, 3, $\rightarrow 5$ types of lattices (P, I, F, C, R)

Table 2.12 Possible lattice centering.

| Centering of the <br> lattice | Lattice points <br> per unit cell | International <br> symbol | Lattice translation(s) <br> due to centering |
| :--- | :---: | :---: | :--- |
| Primitive | 1 | P | None |
| Base-centered | 2 | A | $1 / 2(\mathbf{b}+\mathbf{c})$ |
| Base-centered | 2 | B | $1 / 2(\mathbf{a}+\mathbf{c})$ |
| Base-centered | 2 | C | $1 / 2(\mathbf{a}+\mathbf{b})$ |
| Body-centered | 2 | I | $1 / 2(\mathbf{a}+\mathbf{b}+\mathbf{c})$ |
| Face-centered | 4 | F | $1 / 2(\mathbf{b}+\mathbf{c}) ; 1 / 2(\mathbf{a}+\mathbf{c}) ; 1 / 2(\mathbf{a}+\mathbf{b})$ |
| Rhombohedral | 3 | R | $1 / 3 \mathbf{a}+2 / 3 \mathbf{b}+2 / 3 \mathbf{c} ; 2 / 3 \mathbf{a}+1 / 3 \mathbf{b}+1 / 3 \mathbf{c}$ |

## 14 Bravais lattice

7 crystal systems (6 crystal families) X 5 types of lattices
$\rightarrow$ only 14 different types of unit cells are required to describe all lattices using conventional crystallographic symmetry $\rightarrow 14$ Bravais lattice
C

Why tetragonal F lattice is not one of 14 ?
$\checkmark$ Because that lattice can be reduced to a lattice with different centering and/or a smaller unit cell (rule \#3)
$\checkmark$ Or Because they do not satisfy rule \# 1 or \#2


Fig. 2.21 The reduction of the tetragonal face-centered lattice (left) to the tetragonal body-centered lattice with half the volume of the unit cell (right). Small circles indicate lattice points.

14 Bravais lattice

## general space lattice

$\checkmark$ no symmetry elements except inversion center

## special space lattice

All lattices are centrosymmteric
$\checkmark$ rotation axis and mirror plane $\checkmark$ restriction on the cell parameters

- ex) $4_{z}$-fold rotation axis $\rightarrow a=b, \gamma=90^{\circ}$
$\checkmark$ simplifications in the crystal morphology and in the physical properties

plane lattices $\rightarrow$ space lattices
$>$ general (oblique) lattice
$\checkmark 2$ fold axis: $1 \rightarrow 2$
$\checkmark$ lattice translation $\vec{a}: 1 \rightarrow 3$
$\checkmark 2$ fold axis: $3 \rightarrow 4$
$\rightarrow$ oblique parallelogram, $a_{0} \neq b_{0}, \gamma \neq 60^{\circ}, 90^{\circ}$, $120^{\circ}$
(1)!
a)

(3) ${ }^{2}$
(4) ${ }^{7}$
0
$\dot{2}$

c)

Position of point 3 is general

> | 7 ; need not be equivalent |
| :--- |
| $=$; are required to be equivalent by symmetry |

5 plane lattices > general \& special plane lattices


Position of point 3 is special

right triangle $a_{0} \neq b_{0}$
$\gamma=90^{\circ}$
S1


Isosceles triangle

$$
\begin{gathered}
a_{0}=b_{01} \\
\gamma \neq 60^{\circ}, 90^{\circ}, 120^{\circ}
\end{gathered}
$$

S2


Isosceles right triangle

$$
\begin{aligned}
& a_{0}=b_{0} \\
& \gamma=90^{\circ}
\end{aligned}
$$

S3

equilateral triangle $a_{0}=b_{0}$
$\gamma=120^{\circ}$
$>$ special lattice ; $\mathrm{a}_{0} \neq \mathrm{b}_{0}, \gamma=90^{\circ}$
$\checkmark$ rectangular unit mesh
$\checkmark$ two perpendicular mirror planes // 2 fold axes

right triangle
$\mathrm{a}_{0} \neq \mathrm{b}_{0}$
$\gamma=90^{\circ}$
(a)


5 plane lattices > special plane lattice - 2 (S2)
Position of point 3 is special $(13=32)$
$a_{0}=b_{0}, \gamma \neq 60^{\circ}, 90^{\circ}, 120^{\circ}$
extension of the edges 1-3 \& 1-4
$\rightarrow$ alternative unit mesh, $\mathrm{a}_{0}^{\prime} \neq \mathrm{b}^{\prime}{ }_{0}, \gamma^{\prime}=90^{\circ}$
$\rightarrow$ centered rectangular ; 2-fold axes, mirror plane


Isosceles triangle $a_{0}=b_{0}$,
$\gamma \neq 60^{\circ}, 90^{\circ}, 120^{\circ}$


b)
(b)

c)


5 plane lattices > special plane lattice - 3 (S3), 4 (S4)

Position of point 3 is special ( $1,2, \&$ 3 make an isosceles right triangle)

$$
\begin{aligned}
& \overline{13}=\overline{23} \\
& a_{0}=b_{0}, \gamma=90^{\circ} \text { square mesh }
\end{aligned}
$$

4-fold axes, 4 mirror planes

(c)


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Position of point 3 is special $(1,2, \& 3$ make an equilateral triangle)
$\overline{13}=\overline{23}$
$a_{0}=b_{0}, \gamma=120^{\circ}$ hexagonal mesh
2, 6, 3-fold axes, mirror planes

(d)


## 5 plane lattices

|  | Shape of <br> unit mesh | Lattice <br> parameters | Characteristic <br> symmetry <br> elements |  |
| :--- | :--- | :--- | :--- | :---: |
| General <br> plane lattices | Parallelogram | $\mathrm{a}_{0} \neq \mathrm{b}_{0}$ <br> $\gamma \neq 90^{\circ}$ | 2 |  |
| Special <br> plane lattice | a | Rectangle <br> (primitive) | $\mathrm{a}_{0} \neq \mathrm{b}_{0}$ <br> $\gamma=90^{\circ}$ | m |
|  | b | Rectangle <br> (centred) | $\mathrm{a}_{0} \neq \mathrm{b}_{0}$ <br> $\gamma=90^{\circ}$ | m |
|  | c | Square | $\mathrm{a}_{0}=\mathrm{b}_{0}$ <br> $\gamma=90^{\circ}$ | 4 |
|  | d | $120^{\circ}$ Rhombus | $\mathrm{a}_{0}=\mathrm{b}_{0}$ <br> $\gamma=120^{\circ}$ | $6(3)$ |

Any two-dimensionally periodic array can be assigned to one of the 5 lattice types

Five 2-D lattice types

| Cell | Name | Axial Parameters | Point groups |
| :---: | :---: | :---: | :---: |
| Oblique | $a \neq b$ <br> $\gamma \neq 90^{\circ}$ | 1,2 |  |



Rectangular

$$
\begin{gathered}
a \neq b \\
\gamma=90^{\circ}
\end{gathered}
$$

$$
\mathrm{m}, 2 \mathrm{~mm}
$$



Hexagonal

Square

$$
\begin{gathered}
a=b \\
\gamma=90^{\circ}
\end{gathered}
$$

4, 4mm
Ten 2-D point groups Ten plane point groups

See Hammond 2.3

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Centered Rectangular

$$
\begin{gathered}
a=b \\
\gamma=120^{\circ}
\end{gathered}
$$

3, 3 m
6, 6 mm


$$
a=b
$$

$\gamma \neq 90^{\circ}$
m, 2 mm

## >Stack plane lattices $\rightarrow$ space lattice

5 space lattices with primitive unit cells from the 5 plane lattices
$\rightarrow$ primitive space lattices

- Congruent lattice planes are stacked above one another

| Shape of unit mesh <br> in stacked layers | Interplanar spacing | Lattice |
| :--- | :---: | :--- |
| Parallelogram <br> $\left(\mathrm{a}_{0} \neq \mathrm{c}_{0}\right)$ | $\mathrm{b}_{0}$ | Monoclinic P |
| Rectangle <br> $\left(\mathrm{a}_{0} \neq \mathrm{b}_{0}\right)$ | $\mathrm{c}_{0}$ | Orthorhombic P |
| Square <br> $\left(\mathrm{a}_{0}=\mathrm{b}_{0}\right)$ | $\mathrm{c}_{0} \neq\left(\mathrm{a}_{0}=\mathrm{b}_{0}\right)$ | Tetragonal P |
| Square <br> $\left(\mathrm{a}_{0}=\mathrm{b}_{0}\right)$ | $\mathrm{c}_{0}=\left(\mathrm{a}_{0}=\mathrm{b}_{0}\right)$ | Cubic P |
| $120^{\circ}-$ Rhombus <br> $\left(\mathrm{a}_{0}=\mathrm{b}_{0}\right)$ | $\mathrm{c}_{0}$ | Hexagonal P |


${ }^{\text {a }}$ Note that for historical reasons, the description $\mathrm{a}_{0} \neq \mathrm{b}_{0}, \gamma \neq 90^{\circ}$ has been changed in this case to $\mathrm{a}_{0} \neq \mathrm{c}_{0}, \beta \neq 90^{\circ}$. (See slide \# 14; standard unit cell choice of monoclinic)

## Symmetry of P-lattices

The presence of any two of the following symmetry elements implies the presence of the third
$>$ Rule 1: A rotation axis of even order $\left(X_{e}=2,4,6\right)$, a mirror plane normal to $X_{e}$ and an inversion center at the point of intersection of $X_{e}$ and $m$
$\rightarrow$ Rule 2 : Two mutually perpendicular mirror planes and a 2 -fold axis along their line of intersection


- Every lattice has inversion centers on the lattice point and midway between any two of them

Space group


All lattices are centrosymmteric
$\checkmark$ the complete set of symmetry operations in a lattice or a crystal structure
$\checkmark$ a group of symmetry operations including lattice translations
$\checkmark 230$ space groups

Space lattice > Triclinic P-lattice (general lattice)
$>$ When stacked directly above one another $\rightarrow$ monoclinic
 P lattice
$>$ When lattice points of stacked plane do not coincide $\rightarrow$ lose 2-fold axis $\rightarrow$ triclinic P lattice $\rightarrow$ The only point symmetry elements are inversion centres

set of 2-fold axes // b
$>$ mirror planes normal to $b$ at $x, 0, z$ and $x, 1 / 2, z$
> inversion center

unit mesh

unit cell

( $\mathrm{x}, \mathrm{0}, \mathrm{z}$ projection)
」 indicates a mirror plane // to the plane of the page at heights of 0 and $1 / 2$
stack with inter-planar spacing $b_{0}$


The b -axis (// to 2 and $\perp$ to m ) is called symmetry direction ("2nd setting")


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$$
\begin{array}{ll}
a_{o} \neq b_{o} \neq \mathrm{c}_{\mathrm{o}} & \text { point group: } \frac{2}{m} \\
\alpha=\gamma=90^{\circ}, \beta>90^{\circ} & \text { space group: } \mathrm{P} \frac{2}{\mathrm{~m}}
\end{array}
$$

## Space lattice > Monoclinic P-lattice

Highest symmetry point group in monoclinic system

along $b$ direction
along symmetry direction

monoclinic point groups of lower symmetry
> rectangular plane lattice
> set of 2-fold axes // c
$>$ symmetry of stacked plane + (mirror planes $\perp c$ at $x, y, 0$ and $x, y, 1 / 2)+$ (inversion centres)
$>$ rule ${ }^{*}(\overline{1}$ on $\mathrm{m} \rightarrow 2 \perp \mathrm{~m})$ or rule II* $(\mathrm{m} \perp \mathrm{m} \rightarrow 2)$ generates
2-fold axes at $x, 0,0 ; x, 0,1 / 2 ; x, 1 / 2,0 ; x, 1 / 2,1 / 2 ; 0, y, 0 ; 0, y, 1 / 2 ; 1 / 2, y, 0 ; 1 / 2, y, 1 / 2$


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* rule I or rule II ; see slide \# 28
 Ott Chap 7


## Space lattice > Orthorhombic P-lattice



$$
\begin{aligned}
& a_{o} \neq b_{o} \neq \mathrm{c}_{\mathrm{o}} \\
& \alpha=\beta=\gamma=90^{\circ}
\end{aligned}
$$

point group: $\frac{2}{m} \frac{2}{m} \frac{2}{m}$
space group: $\mathrm{P} \frac{2}{\mathrm{~m}} \frac{2}{m} \frac{2}{m}$

$2 / \mathrm{m} \mathrm{2/m} \mathrm{2/m}-\mathrm{D}_{2 \mathrm{~h}}$

orthorhombic point groups of lower symmetry

## Space lattice > Tetragonal P-lattice

symmetry of stacked plane + (mirror planes $\perp c$ at $x, y, 0$ and $x, y, 1 / 2)+$ (inversion centres)
$\Rightarrow$ rule ${ }^{*}(\overline{1}$ on $m \rightarrow 2 \perp \mathrm{~m})$ or rule II* $(\mathrm{m} \perp \mathrm{m} \rightarrow 2)$ generates several 2-fold axes

unit mesh


* rule I or rule II ; see slide \# 28

point group: $\frac{4}{m} \frac{2}{m} \frac{2}{m}$
space group: $\mathrm{P} \frac{4}{\mathrm{~m}} \frac{2}{m} \frac{2}{m}$
Ott Chap 7


## Space lattice > Tetragonal P-lattice

$a_{o}=b_{o} \neq \mathrm{c}_{\mathrm{o}}$
$\alpha=\beta=\gamma=90^{\circ}$
point group: $\frac{4}{m} \frac{2}{m} \frac{2}{m}$
space group: $\mathrm{P} \frac{4}{\mathrm{~m}} \frac{2}{m} \frac{2}{m}$

tetragonal point groups of lower symmetry
$>\mathrm{a}_{0}=\mathrm{b}_{0}=\mathrm{c}_{0} \rightarrow$ four 3-fold axes along the body diagonals of unit cell as well as inversion center $\rightarrow \overline{3}$
> Rule I* or II* generates 2-fold axes // [110] and equivalent directions


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symmetry directions of cubic
d Space group


Symmetry elements (incomplete) of the cubic P -lattice. This is one of the space groups of highest symmetry in the cubic system

* rule I or rule II ; see slide \# 28 Ott Chap 7


$$
\begin{aligned}
& a_{o}=b_{o}=c_{o} \\
& a=b=g=90^{\circ}
\end{aligned}
$$

$$
\text { point group: } \frac{4}{m} \frac{2}{m}
$$

$$
\text { space group: } \mathrm{P} \frac{4}{m} \overline{3} \frac{2}{m}
$$


$4 / \mathrm{m} \overline{3} 2 / \mathrm{m}-\mathrm{O}_{\mathrm{h}}$
Highest symmetry point group in cubic system

Space lattice > Cubic P-lattice

cubic point groups of lower symmetry

## Space lattice > Hexagonal P-lattice

symmetry of stacked plane + (mirror planes $\perp c$ at $x, y, 0$ and $x, y, 1 / 2)+$ (inversion centres)
$>$ rule I* ( $\overline{1}$ on $m \rightarrow 2 \perp \mathrm{~m}$ ) or rule II* $(\mathrm{m} \perp \mathrm{m} \rightarrow 2)$ generates several 2-fold axes

unit mesh
stack with inter-planar spacing $\mathrm{c}_{0}$


* rule I or rule II ; see slide \# 28 Ott Chap 7

$a_{o}=b_{o} \neq \mathrm{c}_{\mathrm{o}}$
$\alpha=\beta=90^{\circ}, \gamma=120^{\circ}$
point group: $\frac{6}{m} \frac{2}{m} \frac{2}{m}$
space group: $\mathrm{P} \frac{6}{\mathrm{~m}} \frac{2}{m} \frac{2}{m}$

hexagonal point groups of lower symmetry


## Space lattice > Hexagonal P-lattice

hexagonal point groups of lower symmetry

start with hexagonal plane lattice
$\checkmark$ second plane $\frac{1}{3} c_{o}$ with a lattice point on 3 -fold at $\frac{2}{3}, \frac{1}{3}, c_{o}$
$\checkmark$ third plane $\frac{2}{3} c_{o}$ with a lattice point on 3 -fold at $\frac{1}{3}, \frac{2}{3}, z$
$\checkmark \rightarrow$ reduce 6 -fold to 3 -fold axis,

- remove mirrors $x, 0, z ; 0, y, z ; x, x, z$
- remove 2 -fold axis // c


What is the difference between hexagonal and trigonal (rhombohedaral) lattice? Starting 2D mesh is same, but difference in stacking.

## Space lattice > Trigonal, Rhombohedral

> Two unit cells


## Trigonal R-lattice

$$
\begin{aligned}
& a_{o}=b_{o} \neq \mathrm{c}_{\mathrm{o}} \\
& \alpha=\beta=90^{\circ} \quad \gamma=120^{\circ}
\end{aligned}
$$

Rhombohedral P-lattice

$$
a_{o}=b_{o}=\mathrm{c}_{\mathrm{o}}
$$



$$
\alpha=\beta=\gamma
$$


directions
$>$ Primitive cells of cubic I and cubic F are rhombohedral (Fig 3.2 of Hammond)
$>\alpha=90^{\circ} \rightarrow$ cubic P lattice
$>\alpha=60^{\circ} \rightarrow$ cubic $F$ lattice
$>\alpha=109.47^{\circ} \rightarrow$ cubic I lattice



Fig. 3.2. (a) The cubic $I$ and (b) the cubic $F$ lattices with the primitive rhombohedral cells and inter-axial angles indicated.
> When hexagonal layers of lattice points in the rhombohedral lattice are spaced apart in such a way that $\alpha=90^{\circ}, 60^{\circ}$, or $109.47^{\circ}$, then cubic symmetry results.
> What is the difference between hexagonal and trigonal (rhombohedaral) lattice?
$\checkmark$ Hexagonal --- AAAAAA
$\checkmark$ Rhombohedral --- ABCABC

## Space lattice > Trigonal


trigonal point groups of lower symmetry


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## 14 Bravais lattice > Non-primitive lattice

What happens when we add extra lattice planes into monoclinic P-lattice?
> monoclinic P-lattice
$\checkmark$ lattice point: $\frac{2}{m}$
$\checkmark$ new lattice points to be added also should have $2 / \mathrm{m}$

- $1 / 2,0,0 ; 0,1 / 2,0 ; 0,0,1 / 2 ; 1 / 2,1 / 2,0 ; 1 / 2,0,1 / 2 ; 0,1 / 2,1 / 2 ; 1 / 2,1 / 2,1 / 2$


Consider all the possibilities for introducing extra lattice planes into the monoclinic $P$ lattice (next 3 slides) $\rightarrow$ all these can be represented as $P$ or $C$ lattice
> Introduce extra lattice planes into monoclinic P-lattice
(a) new lattice point @ $1 / 2,1 / 2,0 \rightarrow$ C-centered lattice or C-lattice

(b) @ $0,1 / 2,1 / 2 \rightarrow$ A-lattice $\rightarrow$ C-lattice (a \& c can be swapped)


14 Bravais lattice > Add extra lattice planes into Monoclinic P-lattice
(c) at $1 / 2,0,1 / 2 \rightarrow$ B-lattice $\rightarrow$ smaller primitive unit cell (P-lattice)

(d) at $1 / 2,1 / 2,1 / 2 \rightarrow$ - lattice $\rightarrow$ C-lattice
(e) at $1 / 2,0,0 ; 0,1 / 2,0$ or $0,0,1 / 2 \rightarrow$ half the cell $\rightarrow$ P-lattice
(f) at $1 / 2,1 / 2,0 \& 01 / 2,1 / 2 \rightarrow$ further lattice point @ $1 / 2,0,1 / 2$ (because all lattice points should have the same environment) $\rightarrow$ F-lattice $\rightarrow$ can be reduced to C-lattice of half the volume

(e)



In last 3 slides, considered all the possibilities for having non-primitive monoclinic lattice $\rightarrow \mathrm{P}$ or C monoclinic lattice can exist $>$ monoclinic lattice can only be either P or C

## 14 Bravais lattice > space group symbols

|  | $P$ | $C$ | $I$ | $F$ |
| :--- | :--- | :--- | :--- | :--- |
| Triclinic | $P \overline{1}$ |  |  |  |
| Monoclinic | $P 2 / \mathrm{m}$ | $\mathrm{C} 2 / \mathrm{m}$ |  |  |
| Orthorhombic | $P 2 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}$ | $\mathrm{C} 2 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}$ | $\mathrm{I} 2 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}$ | $\mathrm{F} 2 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}$ |
| Tetragonal | $P 4 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}$ |  | $\mathrm{I} 4 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}$ |  |
| Trigonal | $\mathrm{R} \overline{3} 2 / \mathrm{m}$ |  |  |  |
| Hexagonal |  |  |  |  |
| Cubic |  |  |  |  |

> The 14 Bravais lattice represent the 14 and only way in which it is possible to fill space by a 3-D periodic array of points.
> All crystals are built up on one of 14 Bravais lattices.

- Any crystal structure has only one Bravais lattice.
$>$ Number of lattice is fixed at 14.
$>$ Infinite number of arranging atoms in a cell $\leftarrow$ basis

| Xtal systems | Symmetry directions |  | Axial system |  |
| :---: | :---: | :---: | :---: | :---: |
| Triclinic |  |  | $a 1 \neq a 2 \neq a 3, \alpha \neq \beta \neq \gamma \neq 90^{\circ}$ |  |
| Monoclionic | b | $a 1 \neq a 2 \neq a 3, \alpha=\gamma=90^{\circ} \neq \beta$ |  |  |
| Orthorhombic | $a$ | $b$ | $c$ | $a 1 \neq a 2 \neq a 3, \alpha=\beta=\gamma=90^{\circ}$ |
| Tetragonal | $c$ | $<a>$ | $<110>$ | $a 1=a 2 \neq a 3, \alpha=\beta=\gamma=90^{\circ}$ |
| Trigonal | $c$ | $<a>$ |  | $a 1=a 2=a 3, \alpha=\beta=\gamma<120^{\circ} \neq 90^{\circ}$ |
| Hexagonal | $c$ | $<a>$ | $<210>$ | $a 1=a 2 \neq a 3, \alpha=\beta=90^{\circ}, \gamma=120^{\circ}$ |
| Cubic | $<a>$ | $<111>$ | $<110>$ | $a 1=a 2=a 3, \alpha=\beta=\gamma=90^{\circ}$ |

## 14 Bravais lattice



Numbers \& coordinates of the lattice points in the unit cells of the Bravais lattices

| Lattice | No. of lattice points <br> in unit cell | Coordinates of lattice points <br> in unit cell |
| :--- | :--- | :--- |
| P | 1 | $0,0,0$ |
| A | 2 | $0,0,0 ; 0, \frac{1}{2}, \frac{1}{2}$ |
| B | 2 | $0,0,0 ; \frac{1}{2}, 0, \frac{1}{2}$ |
| C | 2 | $0,0,0 ; \frac{1}{2}, \frac{1}{2}, 0$ |
| I | 2 | $0,0,0 ; \frac{1}{2}, \frac{1}{2}, \frac{1}{2}$ |
| R | 3 | $0,0,0 ; \frac{2}{3}, \frac{1}{3}, \frac{1}{3}, \frac{2}{3}, \frac{2}{3}$, |
| F | 4 | $0,0,0 ; \frac{1}{2}, \frac{1}{2}, 0 ; \frac{1}{2}, 0, \frac{1}{2} ; 0, \frac{1}{2}, \frac{1}{2}$ |


| 7 Xtal systems | $a, b, c, \alpha, \beta, \gamma$ | 14 Bravais lattice | Lattice symbol |
| :---: | :---: | :---: | :---: |
| Cubic | $\begin{aligned} & a=b=c \\ & \alpha=\beta=\gamma=90^{\circ} \end{aligned}$ | Simple | P |
|  |  | Body-centered | 1 |
|  |  | Face-centered | F |
| Tetragonal | $\begin{aligned} & a=b \neq c \\ & \alpha=\beta=\gamma=90^{\circ} \end{aligned}$ | Simple | P |
|  |  | Body-centered | 1 |
| Orthorhombic | $\begin{aligned} & a \neq b \neq c \\ & \alpha=\beta=\gamma=90^{\circ} \end{aligned}$ | Simple | P |
|  |  | Body-centered | 1 |
|  |  | Base-centered | C |
|  |  | Face-centered | F |
| Rhombohedral | $a=b=c, \alpha=\beta=\gamma<120^{\circ}, \neq 90^{\circ}$ | Simple | R |
| Hexagonal | $a=b \neq c, \alpha=\beta=90^{\circ}, \gamma=120^{\circ}$ | Simple | P |
| Monoclinic | $a \neq b \neq c, \alpha=\gamma=90^{\circ} \neq \beta$ | Simple | P |
|  |  | Base-centered | C |
| triclinic | $a \neq b \neq c, \alpha \neq \beta \neq \gamma \neq 90^{\circ}$ | Simple | P |

Table 8.9. Characteristic symmetry elements of the seven crystal systems

|  | Crystal system | 32 Point groups ${ }^{\text {a }}$ | Characteristic symmetry elements |
| :---: | :---: | :---: | :---: |
|  | Cubic | $\begin{gathered} 4 / \mathrm{m} \overline{3} 2 / \mathrm{m} \\ \overline{4} \mathrm{~m}, 4 \underline{3} 2,2 / \mathrm{m} \underline{3}, 2 \underline{3} \end{gathered}$ | 4 - |
|  | Hexagonal | $\begin{gathered} \underline{6} / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m} \\ \underline{6} \mathrm{~m} 2, \underline{\mathrm{~mm}}, \underline{6} 22, \\ \underline{6} / \mathrm{m}, \underline{6}, \underline{6} \end{gathered}$ | - or ${ }^{\text {d }}$ |
|  | Tetragonal | $\begin{gathered} \frac{4}{4} / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m} \\ \underline{4} \mathrm{~m}, 4 \mathrm{~mm}, 422, \\ \underline{4} / \mathrm{m}, \underline{4}, \underline{4} \end{gathered}$ | $\underset{(3 \mathbf{~ o r ~} 3 \rrbracket \Rightarrow \text { cubic })}{1 \llbracket}$ |
|  | Trigonal | $\begin{gathered} \frac{\overline{3}}{\underline{3}} 2 / \mathrm{m} \\ \underline{3}, \underline{32}, \underline{3}, \underline{3} \end{gathered}$ | (remember that m normal to 3 gives $\overline{6} \Rightarrow$ hexagonal |
|  | Orthorhombic | $\frac{2 / \mathrm{m} 2 / \mathrm{m} 2 / \mathrm{m}}{\mathrm{~mm} 2,222}$ | 2 and/or m in three orthogonal directions |
|  | Monoclinic | $\frac{2 / \mathrm{m}}{\underline{m}, \underline{2}}$ | 2 and/or m in one direction |
|  | Triclinic | $\begin{aligned} & \overline{1} \\ & \underline{1} \end{aligned}$ | $\overline{1}$ or 1 only |
| Chan Par ${ }^{\text {a }}$. Characteric symmetry elements are underlined. |  |  |  |

## todos

$>$ Read
$\checkmark$ Ott Chapter 7, 8
$\checkmark$ Krawitz Chapter 1.1 ~ 1.5
$\checkmark$ Hammond Chapter $2.1 \sim 2.4 ; 3.1 \sim 3.3 ; 5.1 \sim 5.6$
$\checkmark$ Sherwood \& Cooper Chapter 3.1~3.7
>Bravais HW (due in 1 week)
$\checkmark$ Hammond 1.9; 3.1, 3.2

