XRD-4 Peak broadening

Size broadening, Strain broadening

Size/strain broadening

- XRD-4 Read

Cullity Chapter 5-1, 5-2, 5-4

Jenkins & Snyder chap 3.9.2; 3.9.3 (p89~p94)

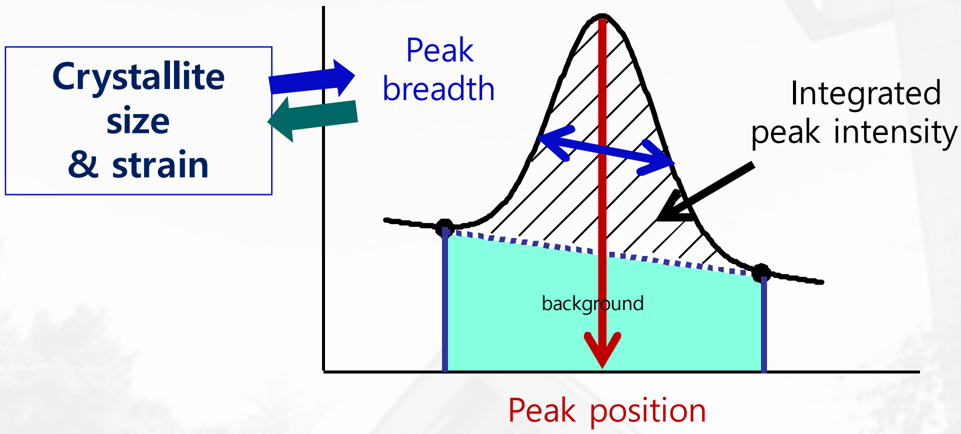
Hammond chap 9.3

Cullity Chapter 5-5, 5-6

Krawitz chap 11.6 (p343~p346)

Cullity Chapter 14-1. 14-2, 14-3, 14-4, 14-6

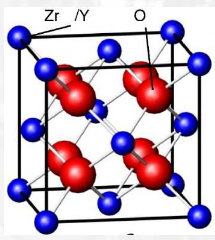
Shape of Peak



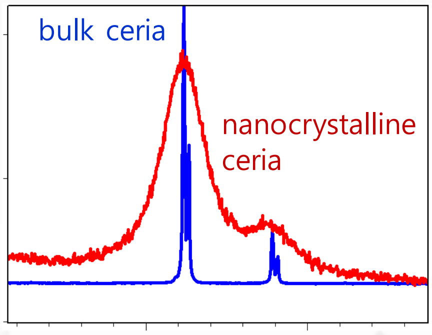
Crystal Structure of "cubic" "ZrO2"

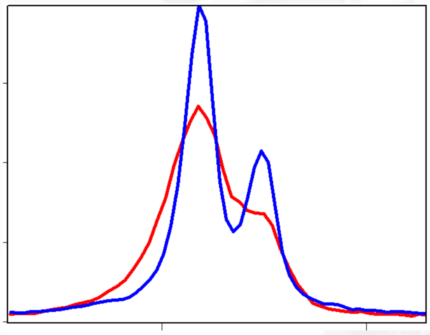
Space Group $Fm\bar{3}m$ (225)
→ cubic
Lattice Parameter a=5.11

Atom	x	y	z	B_{iso}	occupancy
Zr	0	0	0	1.14	1
O	0.25	0.25	0.25	2.4	1



Peak Broadening





same sample run on two different instruments

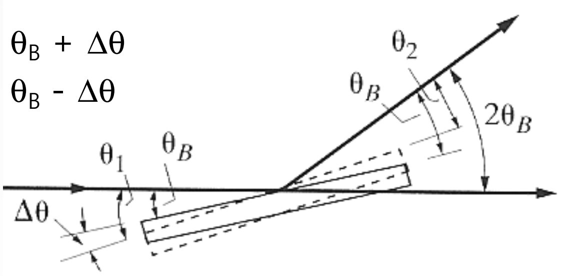
- ➢ Peak broadening ←
  - ✓ Small crystallite size
  - ✓ Stacking faults, Microstrain, and other Defects in the crystal structure
  - ✓ An inhomogeneous composition in a solid solution or alloy

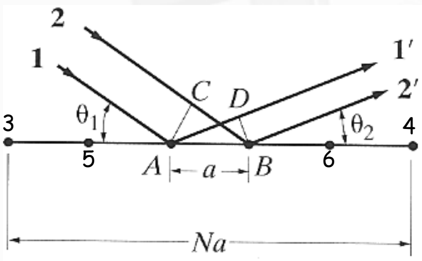
- ➢ Different instrument configurations can change the peak width, too.
  - ← Instrument contribution

Geometrical factor – 1 of Lorenz Factor

$\theta_1 = \theta_B + \Delta\theta$

$\theta_2 = \theta_B - \Delta\theta$

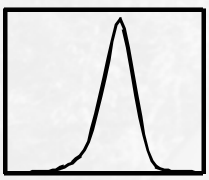




Total length of the plane

$\delta_{1'2'} = 2a\Delta\theta \sin\theta_B,$ → $2Na\ \Delta\theta \sin\theta_B$ Path difference b/w rays scattered by atoms at either end of the plane (3 & 4)

Diffracted intensity = zero when $\boxed{2Na\ \Delta\theta \sin\theta_B = \lambda}$



← Δθ →

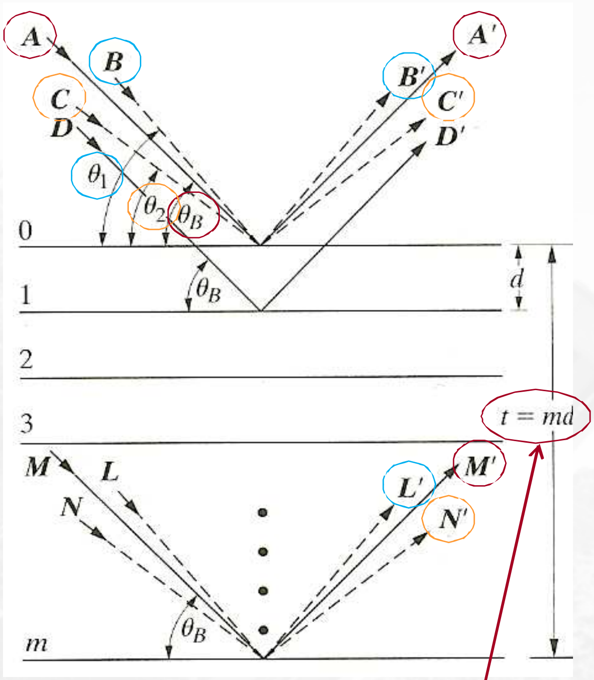
$$\boxed{\Delta\theta = \frac{\lambda}{2\ Na \sin\theta_B}}$$ $I_{max} \propto 1/\sin\theta$

⇩

Max angular range of crystal rotation over which appreciable energy can be diffracted in the direction $2\theta_B$

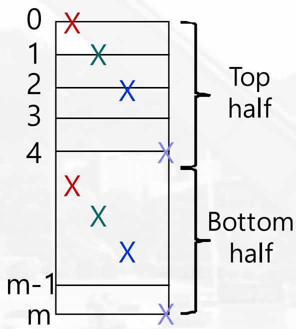
Crystallite size broadening

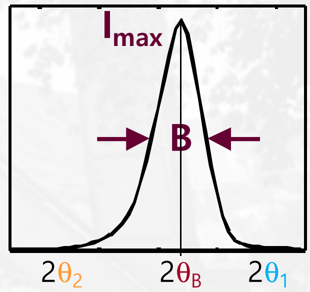
$\theta_1 = \theta_B + \Delta\theta$ (B, B′) $\theta_2 = \theta_B - \Delta\theta$ (C, C′)



t = thickness ~ size

- m + 1 planes
- $\delta(A'D') = 1\lambda$, $\delta(A'M') = m\lambda$
- @ θ_1 or $\theta_2 \neq \theta_B$, incomplete destructive interference
- If $\delta(B'L') = (m+1)\lambda$, intensity zero
- If $\delta(C'N') = (m-1)\lambda$, intensity zero
- @ $2\theta_2 < 2\theta < 2\theta_1$, intensity is not zero

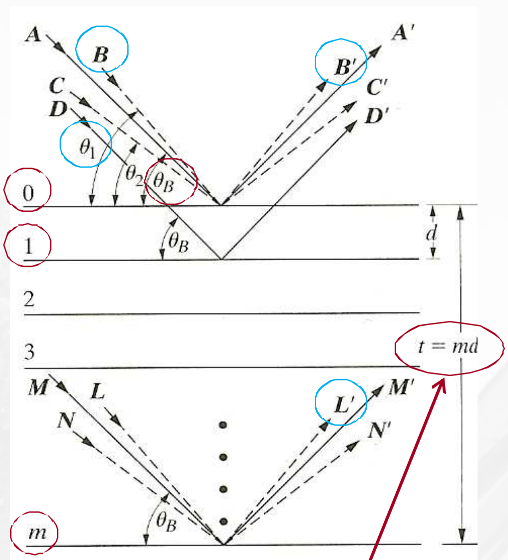




MUST READ Cullity Chap 5-2

Crystallite size broadening

$\theta_1 = \theta_B + \Delta\theta$ (B, B′) $\theta_2 = \theta_B - \Delta\theta$ (C, C′)

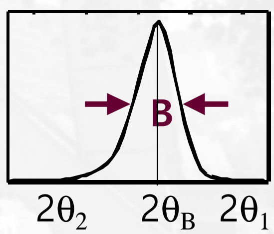


t = md

- Ray B; $\theta_1 = \theta_B + \Delta\theta$, $\delta(B'L') = (m+1)\lambda$
- Compare when m = 10 and m = 10,000
- m = 10 → @ θ_1, $\delta(B'L') = \delta(0m) = 11\lambda$, $\delta(01) = 1.1\lambda$
- m = 10,000 → @ θ_1, $\delta(B'L') = \delta(0m) = 10{,}001\lambda$, $\delta(01) = 1.0001\lambda$
- θ_1(m = 10) >>> θ_1(m = 10,000)
- θ_1 ↑ as m ↓ θ_2 ↓ as m ↓
- $(2\theta_1 - 2\theta_2)$ ↑ as m ↓
- B ↑ as thickness ↓
- Peak width ↑ as size ↓

$$2t \sin\theta_1 = (m+1)\lambda$$
$$2t \sin\theta_2 = (m-1)\lambda$$

→ $t(\sin\theta_1 - \sin\theta_2) = \lambda$



Cullity page 168

Crystallite size broadening

Assume diffraction line is triangular in shape

$$B = \frac{1}{2}(2\theta_1 - 2\theta_2) = \theta_1 - \theta_2.$$

$$2t \sin \theta_1 = (m + 1)\lambda$$

$$2t \sin \theta_2 = (m - 1)\lambda$$

$$t(\sin \theta_1 - \sin \theta_2) = \lambda,$$

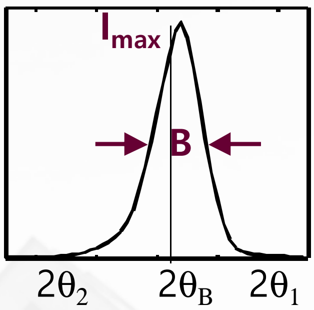
$$2t \cos\left(\frac{\theta_1 + \theta_2}{2}\right) \sin\left(\frac{\theta_1 - \theta_2}{2}\right) = \lambda$$

$$\theta_1 + \theta_2 = 2\theta_B \text{(approx.)}$$

$$\sin\left(\frac{\theta_1 - \theta_2}{2}\right) = \left(\frac{\theta_1 - \theta_2}{2}\right) \text{(approx.)}$$

$$2t\left(\frac{\theta_1 - \theta_2}{2}\right) \cos \theta_B = \lambda$$

$$t = \frac{\lambda}{B \cos \theta_B}$$



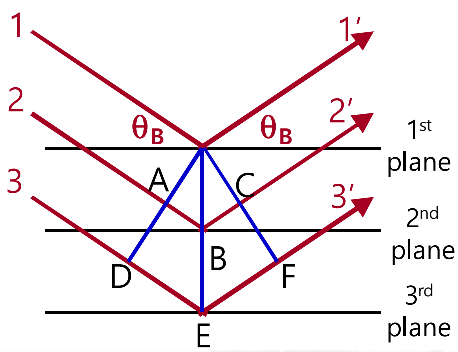
B; an angular width, in terms of 2θ (not a linear width)

Shape factor; depends on the shape of the crystallites

$$t = \frac{0.9\lambda}{B \cos \theta_B}$$

Scherrer equation

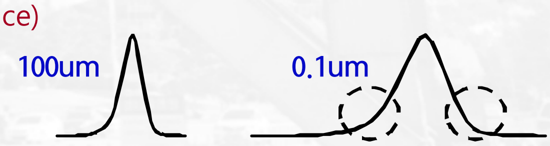
Crystallite size broadening



- @ θ_B; ABC = λ, DEF = 2λ → diffraction peak
- ABC = 0.5λ, DEF = 1λ → no diffraction peak
- ABC = 1.1λ, DEF = 2.2λ

  → PD (path diff.) in 6th plane = 5.5λ → 1′ & 6′ out of phase → no net diffraction
- ABC = 1.001λ → 1′ & 501′ out of phase; ABC = 1.00001λ → 1′ & 50001′ out of phase

  →→ Sharp diffraction peak @ θ_B
- When crystal is only 100nm in size, 5000′ or 50000′ are not present
- Peak begins to show intensity at a lower θ and ends at a higher θ than θ_B → particle size broadening
- Crystallites smaller than 1um can cause broadening → size can be determined using the peak width (← incomplete destructive interference)



Crystallite size broadening

size broadening ; degree of being “out-of-phase” that can be tolerate

- In case λ = 1.5 Å, d= 1.0 Å, θ = 49 ° ,

$$t = \frac{0.9\lambda}{B \cos \theta_B}$$

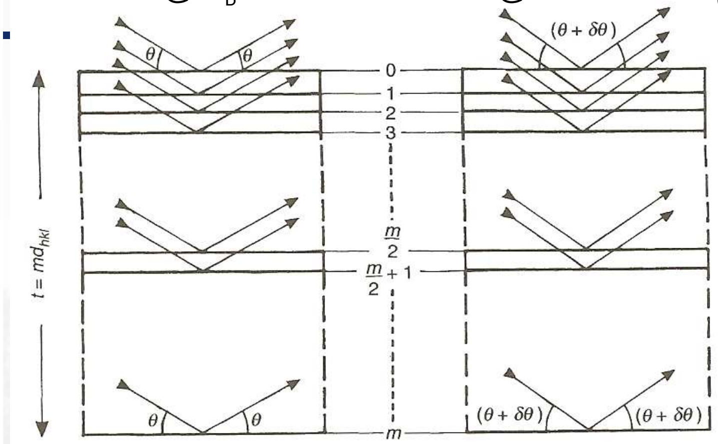
- 1mm(millimeter) diameter crystal → 10^7 parallel lattice planes, ~10^{-7} radian*, ~10^{-5} degree → too small to observe
- 500 Å diameter crystal → 500 parallel lattice planes, ~10^{-3} radian, ~0.2 degree → measurable

- Non-parallel incident beam, non-monochromatic incident beam → diffraction @ angles not exactly satisfying Bragg’s law → line broadening

* B = $(0.9 \times 1.5 \times 10^{-10})/(10^{-3} \times \cos 49°)$ ~ 2×10^{-7} rad

@ θ_B @ $\theta + \Delta\theta$

Crystallite size broadening



between planes 0 & (m/2)

Constructive interference at angle θ

$$(m/2)\lambda = (m/2)2d_{hkl} \sin\theta$$

Destructive interference at angle $\theta + \delta\theta$

$$(m/2)\lambda + \lambda/2 = (m/2)2d_{hkl} \sin(\theta + \delta\theta)$$

δ(1, (m/2)+1) = 0.5λ

$\cos\delta\theta = 1$ and $\sin\delta\theta \approx \delta\theta$

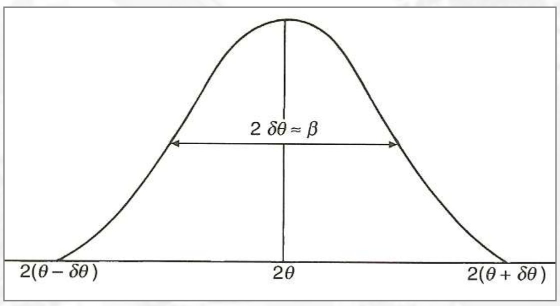
$$(m/2)\lambda + \lambda/2 = (m/2)2d_{hkl} \sin\theta + (m/2)2d_{hkl} \cos\theta\, \delta\theta$$

$$md_{hkl} = t$$

$$2\,\delta\theta = \frac{\lambda}{t\cos\theta} = \beta$$

$$B = \beta = \frac{\lambda}{t\cos\theta} = \frac{\lambda \sec\theta}{t}$$

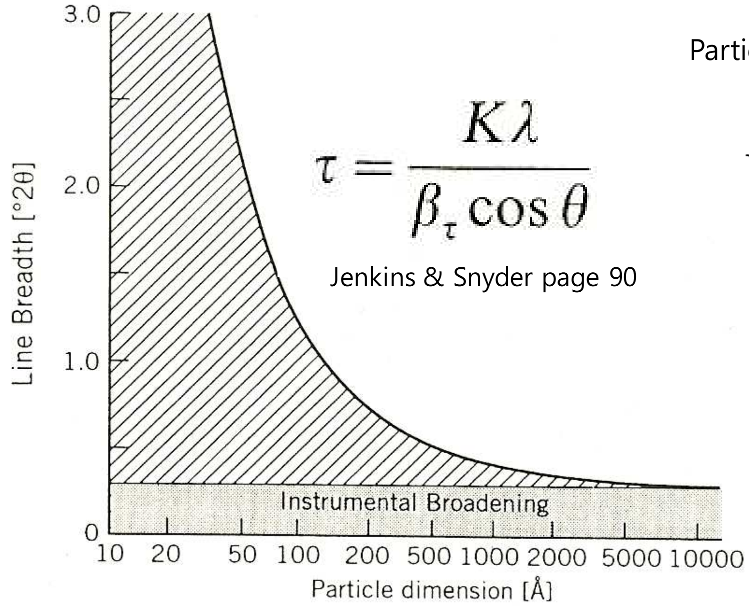
Scherrer equation



Crystallite size broadening

Scherrer equation

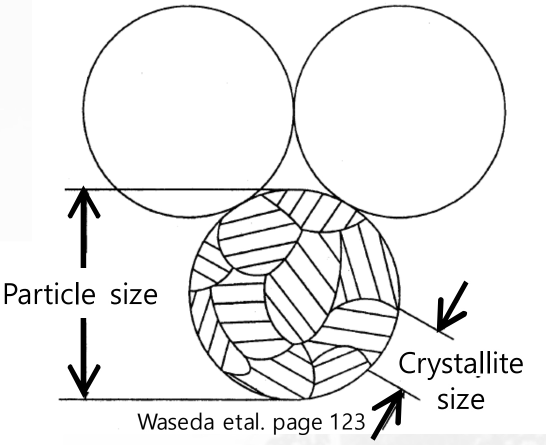
$$t = \frac{0.9\lambda}{B \cos \theta_B}$$



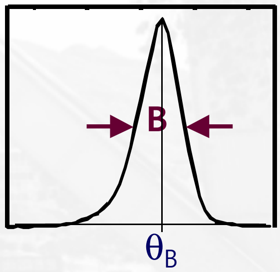
$$\tau = \frac{K\lambda}{\beta_\tau \cos\theta}$$

Jenkins & Snyder page 90

Figure 3.21. Line width as a function of particle dimension.



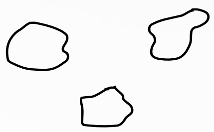
Waseda etal. page 123



Waseda & Matsubara, X-ray diffraction Crystallography, Springer, 2011

Particle Size vs. Crystallite Size

Particles can be individual crystallites.



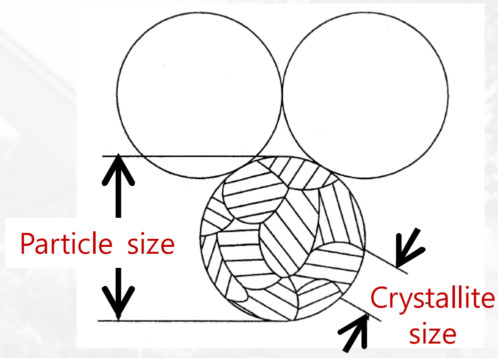
Particle size = crystallite size

Particles may be imperfect single crystals.

Particle size > crystallite size

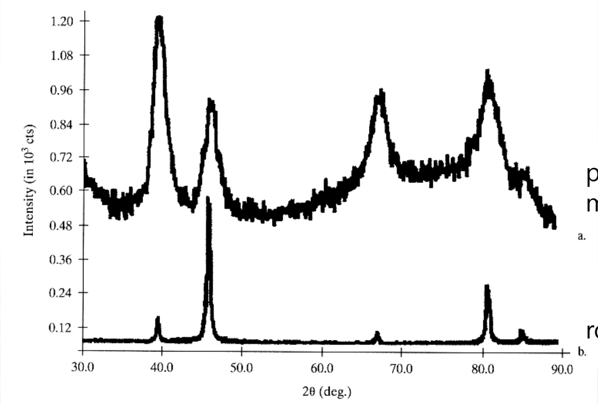
- Individual crystallites are perfect.
- Boundaries
  - Dislocations
  - Twin walls
  - Anti-phase walls
  - Stacking faults

From presentation of Dr. Mark Rodriguez
@ DXC 2017 "What usually causes trouble?"



Waseda et al. page 123

Crystallite Size Broadening



platinum nano-particle in a matrix of amorphous carbon

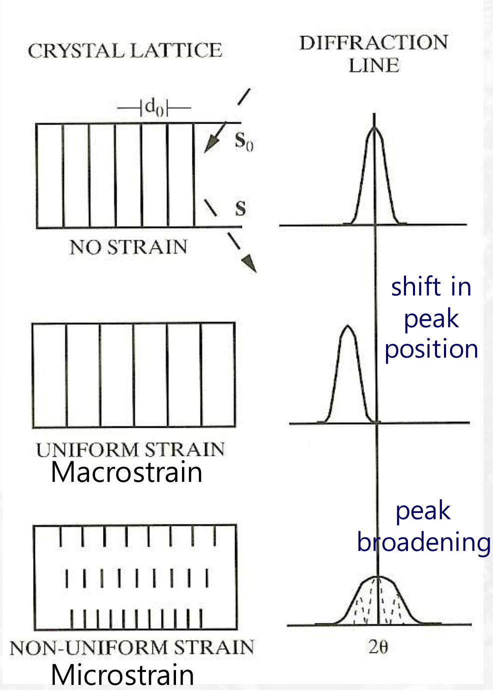
rolled platinum sheet

hkl	FWHM (°2θ)	t (Å)
111	1.9	50
200	1.7	55
220	2.1	50
311	2.5	45-50

Strain broadening

Strain/Stress

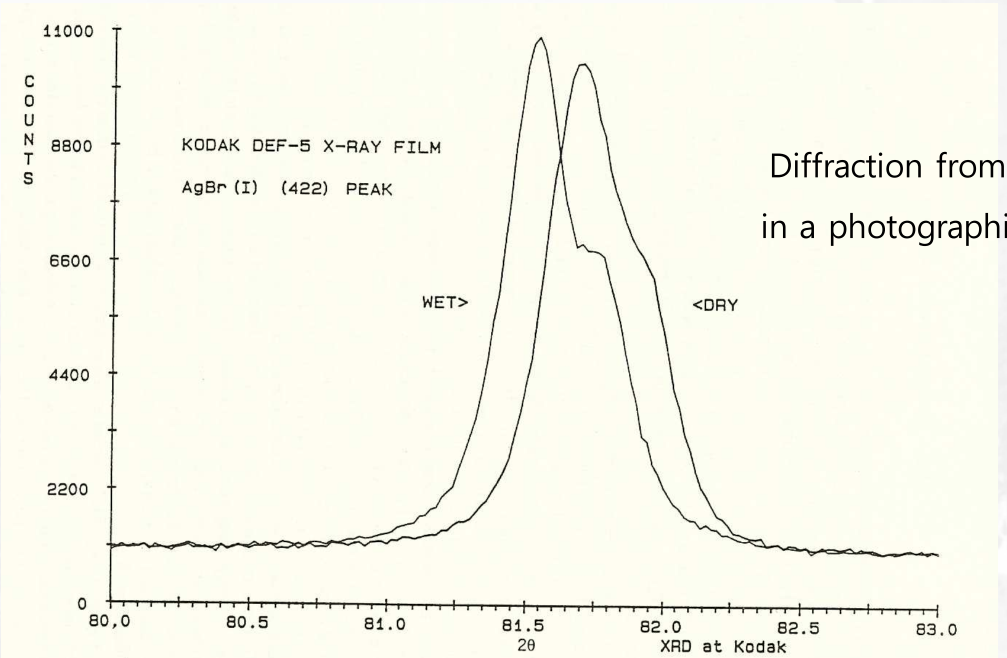
- ➢ Macrostrain/Macrostress → shift in peak position
  - ✓ Stress is uniformly compressive or tensile over large distances. ← lattice parameter measurement
- ➢ Microstrain/Microstress → peak broadening
  - ✓ Distribution of both tensile & compressive stress → distribution of d-values
  - ✓ Can come from dislocations, vacancies, defects, shear planes, thermal expansion/contraction, etc.



Cullity page 176

Jenkins & Snyder, page 91~93

Peak shift ← macrostrain



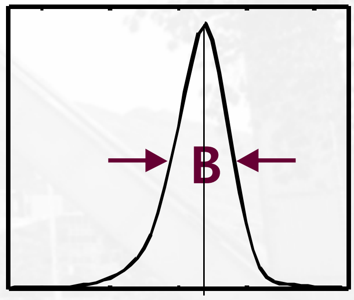
Diffraction from AgBr in a photographic film

Differential expansion between the film substrate & AgBr causes macrostrain → changes lattice parameter → peak shift

Jenkins & Snyder, page 92

Strain broadening

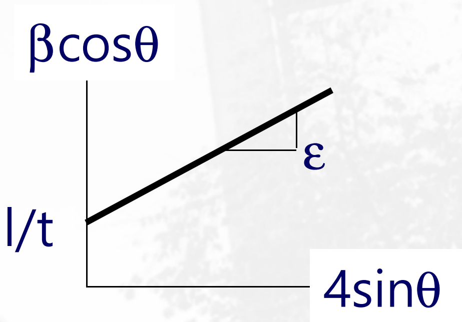
- $\lambda = 2d \sin\theta$
- $0 = 2d \cos\theta\, \delta\theta + 2 \sin\theta\, \delta d$
- $\Delta(2\theta) = -2(\delta d/d)\tan\theta = B$; extra broadening produced by microstrain
- $\beta_\varepsilon = 4\varepsilon \tan\theta$ (Jenkins & Snyder p93)
- $\beta = -2\varepsilon \tan\theta$ (Hammond p266)



Size & Strain broadening

LaB_6 (SRM 660c)

- $\beta_{size} = \lambda/(t\cos\theta)$, $\beta_{strain} = 4\varepsilon \tan\theta$, $\beta_{instrument}$
- $\beta(total) = \lambda/(t\cos\theta) + 4\varepsilon \tan\theta + \beta_{instrument}$
- $\beta = \lambda/(t\cos\theta) + 4\varepsilon (\sin\theta/\cos\theta)$
- $\beta\cos\theta = \lambda/t + 4\varepsilon \sin\theta$
- $\beta\cos\theta/\lambda = 1/t + (4\varepsilon \sin\theta)/\lambda$
- plot $\beta\cos\theta/\lambda$ vs $\sin\theta/\lambda$ (Williamson-Hall plot) → can separate size & strain contributions to line broadening --- semi-quantitative



Cullity page 401

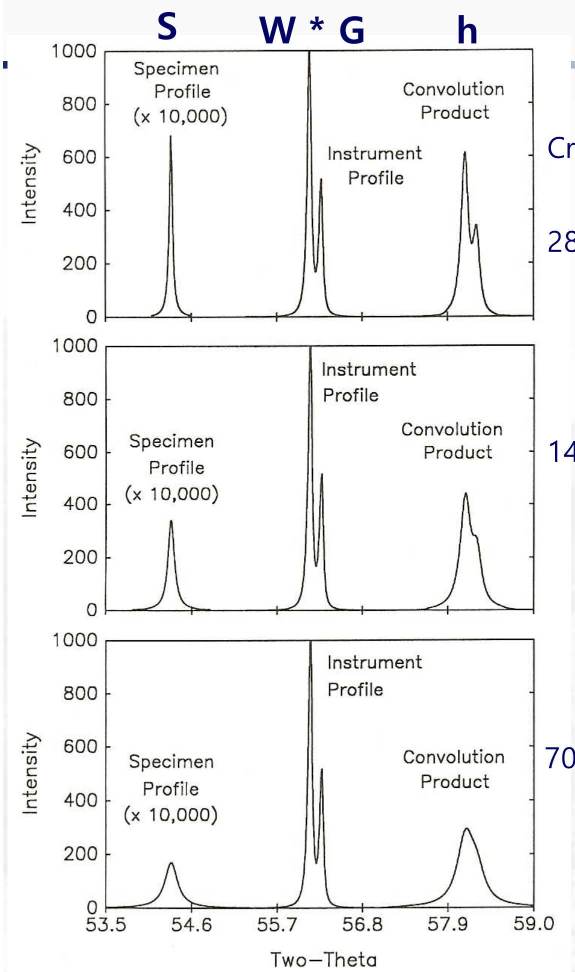
Broadening

- Darwin width
  - ✓ Incident photon is confined to certain volume.
  - ✓ Result of uncertainty principle ($\Delta p \Delta x = h$) --- Location of the photon in a xtal is restricted to a certain volume.
  - ✓ Δp must be finite. → $\Delta\lambda$ must be finite. → finite width of diffraction peak
- Specimen contribution (S)
- Spectral distribution (radiation source contribution) (W)
- Instrumental contribution (G)

- (W * G) ← X-ray source image, flat specimen, axial divergence of incident beam, specimen transparency, receiving slit, etc.
- (W * G) ; fixed for a particular instrument/target system → instrumental profile g(x)
- Overall line profile h(x) = (W * G) * S + background = g(x) * S + BKG

LaB_6 SRM

h(x) = (W * G) * S



Crystallite size

286 nm

143 nm

70 nm

Line shape ← **convolution** of a profile representing instrument (W*G) and specimen (S) contributions

"Intergrated line intensity" of the convolution product remains the same while the peak broadens and the "peak intensity" decreases.

$$B_{obs} = B_{size/strain} + B_{inst}$$ Lorentzian profile

$$B_{obs}^2 = B_{size/strain}^2 + B_{inst}^2$$ Gaussian profile

Bish & Post

Standard Reference Materials (SRMs)

- Powder Line Position + Line Shape Std for Powder Dif
  - ✓ **Silicon (SRM 640f)**; $745/7.5g
- Line position - Fluorophlogopite mica (SRM 675); $809/7.5g
- Line profile - **LaB_6 (SRM 660c)**; $907/6g

No broadening from size & strain

- Intensity
  - ✓ ZnO, TiO_2 (rutile), Cr_2O_3, CeO_2 (SRM 674b); out of stock
- Quantitative phase analysis
  - ✓ Al_2O_3 (SRM 676a); out of stock, Silicon Nitride (SRM 656); $580/ 20g
- Instrument Response Std
  - ✓ Alumina plate (SRM 1976c); $721/1 disc

Gold
$58.66 / gram
(2021-06-17)
goldprice.org

Prices; 2021-06-17
www.nist.gov/srm/index.cfm

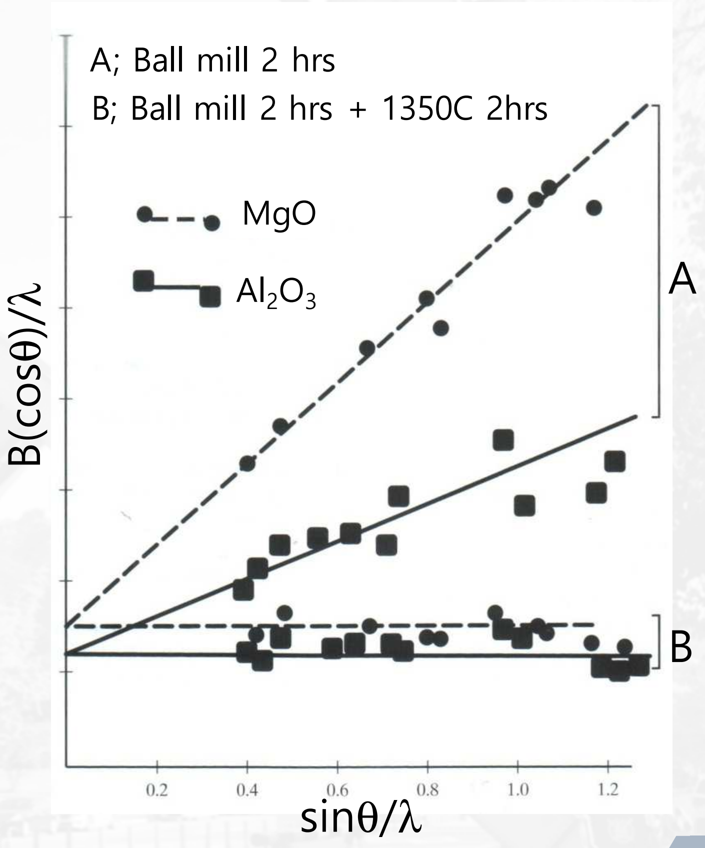
Conventional Williamson-Hall Plot

- Size + Strain

$$\underbrace{\frac{B\cos\theta}{\lambda}}_{y} = \underbrace{\frac{0.9}{d}}_{a} - \underbrace{2\frac{\Delta d}{d}}_{b}\underbrace{\frac{\sin\theta}{\lambda}}_{x}$$

$$y = a + bx$$

- Size >> strain
  - ✓ Horizontal line
- Size << strain
  - ✓ Linear function



Cullity

The effect of dislocation contrast on x-ray line broadening: A new approach to line profile analysis

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(Received 29 May 1996; accepted for publication 6 September 1996)

- explained strain broadening by dislocations.

$$\frac{B\cos\theta}{\lambda} = \frac{0.9}{d} + \Delta K^D \qquad y = a + X$$

Classical $\quad X = -2\dfrac{\Delta d}{d}\dfrac{\sin\theta}{\lambda}$

Modified $\quad X = A(\rho^*)^{1/2} + A'(Q^*)^{1/2}$

$$\frac{B\cos\theta}{\lambda} = \frac{0.9}{d} + A(\rho^*)^{1/2} + A'(Q^*)^{1/2}$$

- ρ^* : (formal) dislocation density
- Q^* : (formal) two-particle correlations in the dislocation ensemble
- A, A' : parameter determined by dislocations

- True values of dislocation density, correlation factor

$$\rho^* = \rho(\pi g^2 b^2 \overline{C})/2 \qquad Q^* = Q(\pi g^2 b^2 \overline{C})^2/4$$

- ✓ $\overline{C}$:average contrast factor of dislocation
- ✓ b :Burgers vector of dislocation
- ✓ Particular reflection $\quad g = \dfrac{2\sin\theta}{\lambda}$

Conventional vs. Modified W-H Plot

$$y = a + X$$

Conventional

$$X = -2\underbrace{\frac{\Delta d}{d}}_{b}\underbrace{\frac{\sin\theta}{\lambda}}_{x}$$

$$= bx$$

$$K = 2\frac{\sin\theta}{\lambda}$$

Modified

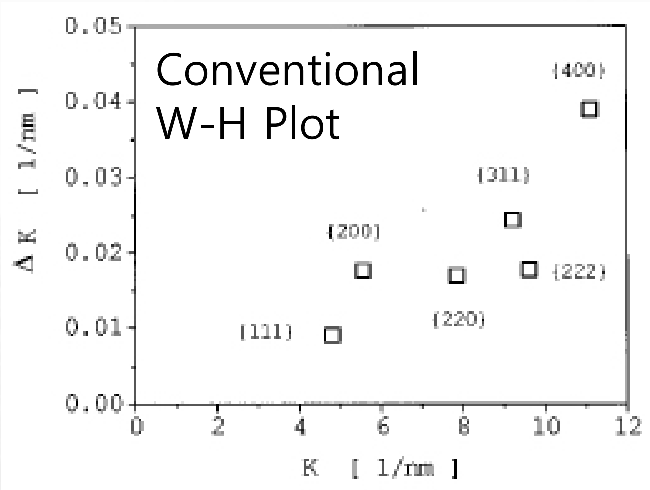
$$X = \underbrace{\left(\pi A b^2/2\right)^{1/2}\rho^{1/2}}_{b'}\underbrace{\frac{2\sin\theta}{\lambda}C^{1/2}}_{x'}$$

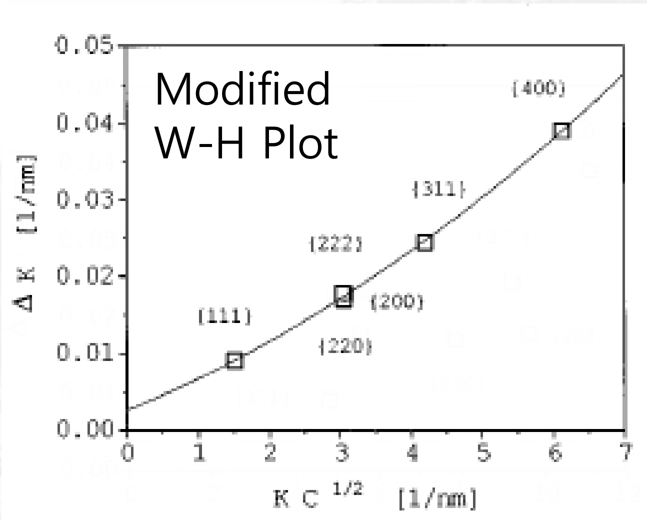
$$+\underbrace{\left(\pi A' b^2/2\right)Q^{1/2}}_{b''}\underbrace{\left(\frac{2\sin\theta}{\lambda}C^{1/2}\right)^2}_{x'^2}$$

$$= b'x' + b''x'^2$$

Appl. Phys. Lett. 69 (21),3173 (1996)

Conventional vs. Modified W-H Plot



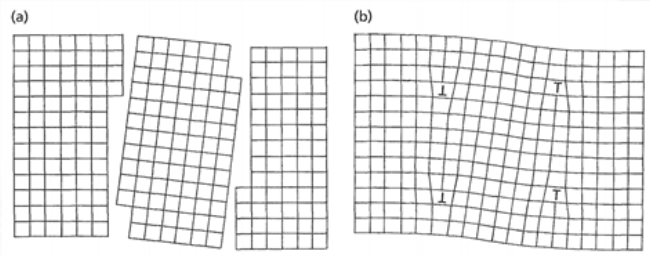


Appl. Phys. Lett. 69 (21),3173 (1996)

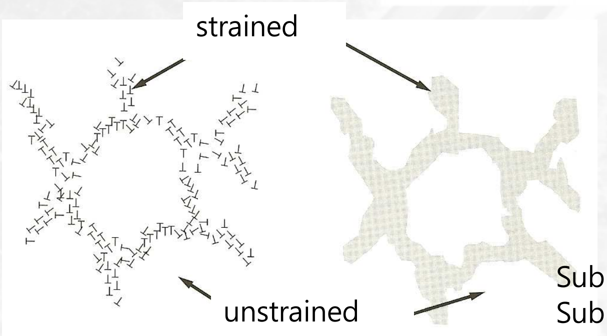
Mosaic structure, mosaic blocks

- Angle of disorientation between the tiny blocks is ε. → diffraction occurs at all angles between θ_B and $\theta_B + \varepsilon$.
- Increases the integrated intensity relative to that obtained (or calculated) for an ideally perfect crystal. ← strains & strain gradients associated with the groups of dislocations

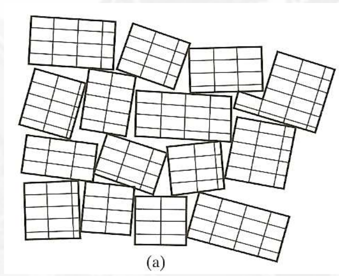
A single xtal divided into 3 mosaic blocks



Boundaries as "walls" of edge dislocations



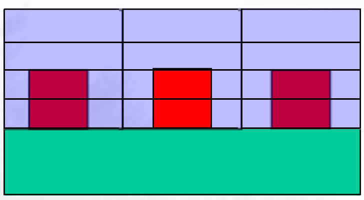
Sub-grain
Sub-grain boundary

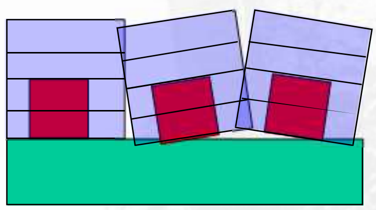


Hammond page 227 Cullity page 174

Mosaic spread

- Mosaicity is created by slight misorientations of different crystals as they nucleate and grow on the substrate. When the crystals join, they form boundaries.





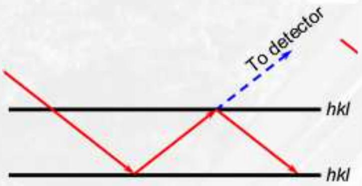
In an ideal case, each nuclei (red) is perfectly oriented.
When the crystals grow and meet, there is perfect bounding between the crystallites → no boundary.

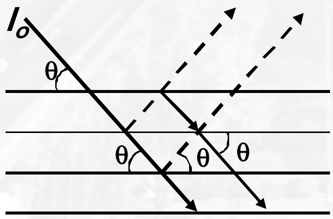
If the nuclei (red) are slightly misaligned, then boundaries will be formed.

Scott A Speakman

Ideally imperfect crystal

- Diffracted intensity; perfect xtal << ideally imperfect xtal
- Decrease in intensity as the crystal becomes more perfect (large mosaic blocks)
- Ideally imperfect crystal consists of very small mosaic blocks, uniformly disoriented. → no extinction
- Kinematical theory vs. dynamical theory
- Powder specimens should be ground as fine as possible.
- Grinding → reduce crystal size, increase # of diffraction cones, decrease mosaic block size, disorient mosaic blocks, strain the crystals non-uniformly





Primary Extinction

- Does not kill the reflection but lower intensity.
- How to avoid? – give some stress (increase mosaicity by e.g. LN2 quenching, heat & quenching, etc.)

Cullity page 177, 360

todos

- XRD-4 Read

Cullity Chapter 5-1, 5-2, 5-4

Jenkins & Snyder chap 3.9.2; 3.9.3 (p89~p94)

Hammond chap 9.3

Cullity Chapter 5-5, 5-6

Krawitz chap 11.6 (p343~p346)

Cullity Chapter 14-1. 14-2, 14-3, 14-4, 14-6

- XRD-4 Homework (due in 1 week)

Cullity 5-1; 5-3